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## A Metalloid Al<sub>14</sub> Cluster with the Structure of a "Nano-Wheel"\*\*

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Dedicated to Professor Heinrich Vahrenkamp on the occasion of his 60th birthday

Solutions of metastable aluminum(i) halides<sup>[1]</sup> can be prepared by condensation of the high-temperature molecules AlX (X=Cl, Br, I) with suitable donor-containing solvents. The thermodynamically favored disproportionation of these compounds to aluminum metal and the trihalide can be controlled kinetically by the choice of halide, donor, and temperature. Thus, with NEt<sub>3</sub> as donor a planar Al<sub>4</sub> species (for example,  $Al_4Br_4 \cdot 4NEt_3^{[2]}$ ) is obtained, whereas with THF the polyhedral subhalide  $Al_{12}(AlBr_2 \cdot THF)_{10} \cdot 2THF$  forms as a result of "internal" disproportionation.<sup>[3]</sup>

An additional variation of the disproportionation is acheived by replacement of the halide with suitable bulky substituents. Use of the N(SiMe<sub>3</sub>)<sub>2</sub> group proved to be especially successful: Reaction of LiN(SiMe<sub>3</sub>)<sub>2</sub> with a solution of AlI provided a Al<sub>77</sub>R<sub>20</sub><sup>2-</sup> compound with the largest *metalloid* cluster yet characterized by diffraction methods. This compound can be viewed as an intermediate on the way

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to aluminum metal. [4] Recently, utilizing the more reactive AlCl solutions, two further *metalloid* clusters with the same substituents— $Al_7R_6^{-[5]}$  and  $Al_{12}R_8^{-[6]}$ —could be isolated as intermediates in the formation of the  $Al_{77}$  unit.

In these clusters (Al<sub>7</sub>, Al<sub>12</sub>, Al<sub>77</sub>) the number of direct metal – metal contacts exceeds the number of metal – ligand interactions. We classify such species as metalloid clusters, <sup>[5]</sup> to distinguish them from the variety of "metal clusters" included in Cotton's original definition. <sup>[7]</sup> In the search for further intermediates on the way to the metal, the title compound  $[Al_{14}\{N(SiMe_3)_2\}_6I_6Li(OEt_2)_2]^-[Li(OEt_2)_4]^+ \cdot Tol (1; Tol = toluene)$ , a mixed (I,  $N(SiMe_3)_2$ ) metalloid cluster, could be isolated and structurally characterized for the first time.

Compound 1 was synthesized by slight variation of the reaction conditions under which the  $Al_{77}$  cluster was formed. After addition of solid LiN(SiMe<sub>3</sub>)<sub>2</sub> to a pretreated AlI · Et<sub>2</sub>O solution, the mixture was warmed to 55 °C several times and slowly cooled. After a few weeks 1 crystallized at +7 °C as dark red-brown platelets. The novel formation of the mixed substituted compound 1 appears to be plausible, as only the less reactive AlI reacts in the way described here. Treatment of the more reactive AlCl with the same starting material LiN(SiMe<sub>3</sub>)<sub>2</sub> leads to  $Al_7R_6^-$  and  $Al_{12}R_8^-$  clusters with complete substitution of the halide. It is possible that a  $Al_7R_3X_3^-$  species is a common precursor that is rapidly substituted in the case of X = Cl. For X = I the substitution is so slow that dimerization to form 1 is kinetically preferred.

Figure 1 presents the results of the X-ray structure analysis<sup>[8]</sup> of **1**. The  $[\text{Li}(\text{OEt}_2)_4]^+$  ion, which fills the voids resulting from packing of  $\text{Al}_{14}^{2-}$  cluster units, is not shown.<sup>[9]</sup>

The main structural unit is represented by two staggered, approximately Al-centered  $Al_6$  rings. The central Al atoms deviate somewhat from the planes of the rings and are separated by 2.728 Å. The other Al–Al distances range from 2.570 Å (between Al atoms with iodine ligands) to 2.910 Å

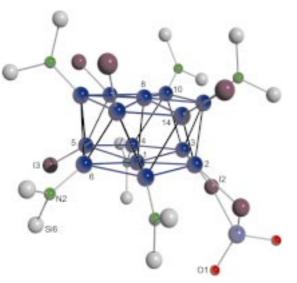


Figure 1. Molecular structure of **1**; for reasons of clarity the C and H atoms have been omitted. Blue: Al atoms. Selected distances [Å]: Al1–Al5 2.623(2), Al1–Al4 2.758(2), Al2–Al3 2.597(2), Al4–Al5 2.763(2), Al1–Al8 2.728(2), Al4–Al10 2.910(3), Al2–Al14 2.570(2), Al6–N2 1.855(5), Al5–I3 2.550(2), Al3–I2 2.636(2).

(between Al atoms with  $N(SiMe_3)_2$  ligands); that is, the values are in the range expected from other polyhedral or metalloid Al clusters.<sup>[10]</sup>

In the case of the  $Al_7R_6^-$  cluster **2** as well as **1** the average oxidation state for the Al atoms is 0.71, which is in contrast to 0.23 for the  $Al_{77}$  cluster. The disproportionation of the initially monovalent Al species to aluminum metal is therefore equally far advanced for **2** and **1**. To be able to discuss the bonding situation in the two species with aluminum in the same average oxidation state, we carried out density functional theory (DFT) calculations<sup>[11]</sup> on the model compounds  $Al_7R_6^-$  (**2a**;  $R = NH_2$ ) and  $Al_{14}R_{12}^{2-}$  (**1a**;  $R = NH_2$ ). These calculations indicated that the dimerization **2a**  $\rightarrow$  **1a** is exothermic ( $-275 \text{ kJ mol}^{-1}$ ). The geometrical parameters of **2a** and **1a**, which are shown in Figure 2, correspond at a first glance to the

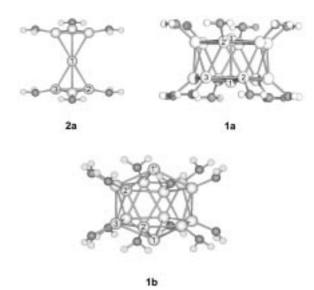


Figure 2. The model compounds  $Al_7(NH_2)_6^-$  (**2a**),  $Al_{14}(NH_2)_{12}^2^-$  (**1a**), and polyhedral  $Al_{14}(NH_2)_{12}^{2-}$  (**1b**). The following characteristic Al–Al distances [Å] were obtained by DFT calculations; [10] **2a**: Al1–Al2 2.760, Al2–Al3 2.543; **1a**: Al1–Al2 2.721, Al2–Al3 2.713, Al2–Al2′ 2.690, Al1–Al1′ 2.693; **1b**: Al1–Al2 2.908, Al2–Al3 2.638, Al2–Al2′ 2.808, Al1–Al1′ 4.903.

experimentally determined values for  $\bf 2$  and  $\bf 1$ . Additionally, calculations were carried out on the isomer  $\bf 1b$ , which is also shown in Figure 2. In this case the central Al–Al distance is highly elongated and an approximately polyhedral structure results, as found for  $M_{14}$  Frank–Kasper polyhedra, for example  $Mg_{23}(Al,Zn)_{49}$ . However, the central atom is missing in  $\bf 1b$ .

Although the polyhedral structure of **1b** leads one to expect a bonding situation as described by Wade for a *precloso* or *nido* structure,<sup>[13]</sup> the calculations showed that **1b** is clearly destabilized with respect to **1a** (+123 kJmol<sup>-1</sup>); that is, a metalloid cluster is energetically favored over a polyhedral cluster.<sup>[14]</sup> As already discussed for other metalloid clusters (Al<sub>7</sub>,<sup>[5]</sup> Al<sub>12</sub>,<sup>[6]</sup> Al<sub>77</sub>,<sup>[4]</sup> Ga<sub>22</sub>,<sup>[15]</sup>), the experimental and theoretical findings presented here confirm that Wade's rules are not appropriate for describing bonding situations of this type.

To obtain a deeper understanding of the exceptional bonding situation in **2** and **1**, we calculated the <sup>27</sup>Al NMR shifts for the central (**2a**, **1a**) and apical Al atoms (**1b**). <sup>[16]</sup> The

results for  ${\bf 2a}$  ( $\delta=652$ ),  ${\bf 1a}$  ( $\delta=358$ ), and  ${\bf 1b}$  ( $\delta=-313$ ) show—as already discussed for  ${\bf 2}^{[5]}$ —that for  ${\bf 2a}$  and  ${\bf 1a}$ , but not for polyhedral  ${\bf 1b}$ , the central Al atoms approach the bonding situation in aluminum metal ( $\delta=1640$ ). [17]

The geometry of the partially preformed metal structure present in metalloid clusters is rationalized for  ${\bf 1}$  in Figure 3. Closest packing as in the metal structure results from rotation by  $30^\circ$  and subsequent translation.

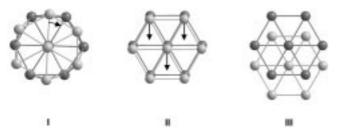


Figure 3. Rotation of the upper ring by  $30^{\circ}$  about the z axis—which incorporates the two central Al atoms of the two six-membered rings in the Al<sub>14</sub> cluster (I)—leads to AA stacking, as found in hexagonal-primitive packing (II). If the upper ring is shifted in the layer of the page by  $(\frac{1}{4}, \frac{1}{4}, 0)$ —based on the basis vectors of the aluminum metal lattice—one obtains AB stacking (III), which represents one section of the closest packing of aluminum metal.

The results presented here have again demonstrated the large synthetic potential of AlX/AlR solutions for the experimentally challenging preparation of metalloid clusters in a well defined form. These clusters should probably play a key role in understanding the mechanism of metal formation.

## Experimental Section

Gaseous AlI (40 mmol) was condensed at -196 °C with toluene (64 mL) and diethyl ether (16 mL) according to the method described in ref. [1]. An aliquot (6 mL) of the ca. 0.30 m dark red-brown All · Et<sub>2</sub>O solution was concentrated at -78 °C under vacuum and dried. The residue was dissolved in toluene (12 mL), and the solution obtained was again concentrated to 6 mL and added to donor-free LiN(SiMe<sub>3</sub>)<sub>2</sub> (150 mg). The reaction mixture was allowed to warm to -25 °C within 2 d. After a further 2 d at 7 °C, the amide had dissolved and LiI precipitated, which was subsequently redissolved in Et<sub>2</sub>O (1 mL). The solution was warmed for 1.5 h at 55 °C and then kept at room temperature for 1 d. This temperature cycle was repeated several times, and then the warm solution was allowed to cool slowly within 3 d to room temperature. During this process disproportionation of Al<sup>I</sup> species led to the formation of aluminum metal. The filtered solution was again warmed to 55 °C and allowed to cool slowly. After 3 d at  $7^{\circ}$ C, 1 h at  $-80^{\circ}$ C, and several weeks at  $7^{\circ}$ C, 1 crystallized as dark redbrown platelets that are chemically and mechanically very labile (yield: ca.

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- [8] Crystal structure analysis of  $[Al_{14}[N(SiMe_3)_2]_6I_6Li(OEt_2)_2]^{-1}$  $[Li(OEt_2)_4]^+ \cdot Tol \ \ \textbf{(1)} : \ STOE\text{-}IPDS \ \ \text{``Image Plate''} \ \ diffractometer,$ 55 kV, 50 mA, Mo<sub>K $\alpha$ </sub> radiation ( $\lambda = 0.71073$ ), measurement temperature 200(2) K. Crystal dimensions  $0.8 \times 0.1 \times 0.1$  mm, monoclinic I, space group  $P2_1/n$ , a = 14.5638(5), b = 37.546(5), c = 23.266(4) Å,  $\beta =$ 94.775(7)°,  $V = 12678(1) \text{ Å}^3$ , Z = 4;  $\rho_{\text{calcd}} = 1.390 \text{ Mg m}^{-3}$ ,  $\mu_{\text{Mo}} =$ 1.720 mm  $^{-1},\,\theta_{\rm min}\,{=}\,1.93^{\circ},\,\theta_{\rm max}\,{=}\,25.99^{\circ};$  of 78 033 reflections measured, 24071 were independent,  $16967 > 2\sigma(I)$ ,  $5127 > 3\sigma(I)$ ; 788 parameters, empirical absorption correction,  $R_1 = 0.0623$ ,  $wR_2 = 0.1595$ ; max./min. residul electron density 1.805/-0.968 e Å<sup>-3</sup>. The structure was solved by direct methods and refined against  $F^2$  for all observed reflections. Software programs used: Shelxs-86, Shelxl-93 (G. M. Sheldrick, Universität Göttingen). Crystallographic data (excluding structure factors) for the structure reported in this paper have been deposited with the Cambridge Crystallographic Data Centre as supplementary publication no. CCDC-136893. Copies of the data can be obtained free of charge on application to CCDC, 12 Union Road, Cambridge CB21EZ, UK (fax: (+44)1223-336-033; e-mail: deposit@ccdc.cam. ac.uk). The asymmetric unit contains an  $[Al_{14}[N(SiMe_3)_2]_6I_6]^{2-}$  ion which is coordinated to a [Li(OEt<sub>2</sub>)<sub>2</sub>]<sup>+</sup> ion through two of the I ligands. Several of the methyl groups are disordered over two positions. Furthermore, a Li+ ion coordinated to four Et2O molecules was found along with a toluene molecule whose atoms could be refined with common isotropic temperature factors. All hydrogen atoms were placed in calculated positions and refined according to a riding model.
- [9] As a supplement to the structure analysis, the presence of an even number of electrons could be confirmed by the lack of an ESR signal. The ESR measurement on crystalline 1 was carried out by Dr. B. Pilawa, Physikalisches Institut, Universität Karlsruhe.
- [10] For purposes of comparison the Al–Al distances [Å] of selected substances are listed:  $[Al_7\{N(SiMe_3)_2\}_6]^-$  2.540 (in the  $Al_3\{N(SiMe_3)_2\}_8$  ring; cf. Figure 2) and 2.737 (Al–Al contacts to the central Al atom); [5]  $[Al_{12}\{N(SiMe_3)_2\}_8]^{2-}$  2.542–2.759; [6]  $Al_{12}(AlBr_2 \cdot THF)_{10} \cdot 2THF$  2.526–2.762; [3]  $Al_{metal}$  2.860.
- [11] All quantum-chemical calculations were carried out with the RIDFT module (B-P86 functional) of the program TURBOMOLE with SV(P) basis sets. a) TURBOMOLE: O. Treutler, R. Ahlrichs, J. Chem. Phys. 1995, 102, 346; b) B-P86 functional: A. D. Becke, Phys. Rev. A 1998, 38, 3098; J. P. Perdew, Phys. Rev. B 1996, 33, 8822; c) RIDFT: K. Eichkorn, O. Treutler, H. Öhm, M. Häser, R. Ahlrichs, Chem. Phys. Lett. 1995, 242, 652; K. Eichkorn, F. Weigend, O. Treutler, R. Ahlrichs, Theor. Chem. Acc. 1997, 97, 119.
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- [14] The MO analysis for 1a showed that this stabilization with respect to 1b results from bonding interactions of the 3s electrons of the central and apical Al atoms with the Al<sub>14</sub> framework.
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