Oxygen Vacancy Ordering in Superlattices of the Two Novel Oxides, $La_2Ni_2O_5$ and $La_2Co_2O_5$, prepared by Low Temperature Reduction of the Parent Perovskites

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Controlled reduction of LaBO₃ (B = Co or Ni) perovskites gives rise to ordered structures of composition La₂B₂O₅, belonging to the homologous series La_nB_nO_{3n-1}, which retain the original features of the perovskite; the co-ordination of the transition metal ions differ, with Ni²⁺ ions showing octahedral and square-planar co-ordination and the Co²⁺ ions showing octahedral and tetrahedral co-ordination.

Superstructures resulting from the ordering of anion vacancies in metal oxides,¹ particularly those of perovskite structure,² have attracted considerable attention in the last few years. The best known example of such an oxide is Ca₂Fe₂O₅ possessing the brownmillerite structure with alternate sheets of octahedra and tetrahedra³ along the *b*-axis of an orthorhombic unit cell; Ca₂Mn₂O₅ and Ca₂Co₂O₅, on the other hand, possess sheets of square-pyramids^{4,5} along the *c*-axis of an orthorhombic unit cell (Figure 1). We have investigated two novel oxides of the formula $La_2B_2O_5$ (B = Co or Ni), prepared by low temperature reduction⁶ of the parent perovskites, LaBO₃, and representing the n = 2 members of the homologous series $La_n B_n O_{3n-1}$. This homologous series was first proposed by Gai and Rao7 on the basis of a thermogravimetric study of LaNiO₃ in air which showed the presence of members with $n \ge 7$. While La₂Ni₂O₅ can be prepared by the reduction of LaNiO₃ at 600 K in pure or dilute (10% H₂-90% N_2 by volume) hydrogen, La₂Co₂O₅ can only be prepared by the reduction of LaCoO₃ in dilute hydrogen at 670 K. Neither La₂Co₂O₅ nor La₂Ni₂O₅ can be made by the solid state reaction of La₂O₃ and the transition metal oxide. Reduction of LaBO₃ gives rise to these oxides because of the topotactic nature of the reaction; La₂B₂O₅ so formed can be oxidized back to the parent LaBO₃ at low temperature as seen from the typical thermogravimetric curves shown in Figure 2.

X-Ray powder diffraction data of La₂Co₂O₅ could be indexed on an orthorhombic unit cell (a = 5.460, b = 15.908, and c = 5.700 Å) with a brownmillerite structure involving alternate sheets of octahedra and tetrahedra along the *b*-axis (Figure 1). The electron diffraction pattern also confirms this structure, although some streaking is noticed owing to some disorder. The topotactic relationship between the orthorhombic La₂Co₂O₅ and the cubic perovskite can be expressed as $(100)_0 \parallel (110)_c$ and $(040)_0 \parallel (001)_c$ and $(001)_0 \parallel (1\overline{10})_c$.



Figure 1. Schematic representations of the structures of (a) $Ca_2Fe_2O_5$ with alternate sheets of octahedra and tetrahedra along the *b*-axis and (b) $Ca_2Mn_2O_5$ with sheets of square pyramids along the *c*-axis (\bullet : oxygen; \circ : oxygen vacancy).

The X-ray powder diffraction pattern of La₂Ni₂O₅ is quite different from that of brownmillerite or that of the parent LaNiO₃ and could be indexed on a tetragonal cell (a = 7.816and c = 7.468 Å), the relationship with the cubic perovskite being $a \approx c \approx 2a_c$. Electron diffraction studies confirm these unit cell dimensions. In Figure 3, the electron diffraction pattern of La₂Ni₂O₅ in the [001] zone-axis is shown, along with the projection of the proposed structure and the expected



Figure 2. Thermogravimetric curves showing reduction of LaNiO₃ to $La_2Ni_2O_5$ in pure hydrogen (•) and reoxidation of $La_2Ni_2O_5$ to $LaNiO_3$ in oxygen (°). A similar behaviour was found in the $LaCoO_3-La_2Co_2O_5$ system.



Figure 3. (a) [001] Zone-axis electron diffraction pattern of $La_2Ni_2O_5$. (b) Projection along [001] of the proposed structure of $La_2Ni_2O_5$. (c) Expected electron diffraction pattern in the same direction. (d) Proposed 3-D structure for $La_2Ni_2O_5$ (\bigcirc : oxygen; \bigcirc : oxygen vacancy; \bigcirc : nickel; lanthanum ions are not shown).

electron diffraction pattern. The most likely structure of this oxide is one involving the ordering of oxygen vacancies along the [110] direction in the (001) LaO planes of the perovskite lattice as depicted in Figure 3. Such an arrangement of oxygen vacancies results in planes of $LaO_{0.5}\Box_{0.5}$ and NiO₂ stacked alternately parallel to the c-axis giving rise to square planar and octahedral co-ordination for Ni²⁺. The topotactic relationship between the tetragonal La₂Ni₂O₅ and the cubic perovskite may be expressed as: $(200)_t \parallel (100)_c$ and $(002)_{\ell} \parallel (001)_{\ell}$. This tetragonal cell is different from that of La₂NiO₄ which bears the relationship with the cubic perovskite cell, $a \simeq a_c$ and $c \simeq 3a_c$. The difference between the structures of La₂Co₂O₅ and La₂Ni₂O₅ can also be understood in terms of the site preference energies of Co²⁺ and Ni²⁺ ions. It appears that Ni2+, which has lower tetrahedral site preference energy than Co2+, prefers a square planar coordination; both Ni²⁺ and Co²⁺ however have relatively high octahedral vs. tetrahedral site preference energies.

Both $La_2Co_2O_5$ and $La_2Ni_2O_5$ exhibit high electrical resistivity and show i.r. bands in the 1000–200 cm⁻¹ region, characteristic of perovskite related structures; metallic LaNiO₃ does not show any i.r. bands. Both these oxides exhibit a large magnetic susceptibility, expecially at low

temperatures. The origin of this magnetic behaviour is not clear.

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