Synthesis of Poly(pyrrole)-Silica-Magnetite Nanocomposite Particles

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Pyrrole is polymerised in the presence of ultrafine silica-coated magnetite particles; the resulting colloidal poly(pyrrole)-silica-magnetite nanocomposites have a conducting polymer content of 75 mass% and exhibit superparamagnetism.

Recently the synthesis of conducting polymer nanocomposites has received increasing attention. $^{1-5}$

We have reported the synthesis of stable colloidal dispersions of both poly(aniline)-silica and poly(pyrrole)-silica nanocomposites using ultrafine silica sols.⁶⁻⁸ With the exception of tin(iv) oxide sols,9 our attempts to extend this work to include other inorganic oxides have been disappointing. For example, magnetite sols prepared according to Massart's method¹⁰ simply dissolve under the relatively acidic conditions (1.2 mol dm⁻³ HCl) required for aniline polymerisation and are flocculated by the relatively high ionic strength of the $FeCl_3$ [or $(NH_4)_2S_2O_8$] oxidant solutions required for pyrrole polymerisation.¹¹ Philipse et al.¹² have recently reported the synthesis of stable dispersions of silica-coated magnetite particles. In principle these coated particles should exhibit the colioid stability and surface chemistry of normal silica particles. In the present study we have prepared similar silica-coated magnetite particles and utilised these dispersions to prepare novel colloidal poly(pyrrole)-silica-magnetite nanocomposites (Fig. 1).

Magnetite sols were prepared as aqueous dispersions according to the method of Massart.¹⁰ Thus, aqueous solutions of FeCl₃·6H₂O (160 cm³, 1 mol dm⁻³) and FeCl₂·4H₂O (40 cm³, 2 mol dm⁻³, dissolved in 2 mol dm⁻³ HCl) were mixed prior to their addition to NH₄OH (2000 cm³, 0.7 mol dm⁻³). The resulting precipitate was washed three times by decantation of the aqueous phase followed by addition of H₂O (first two washes) or NEt₄OH (third wash) to give a total volume of 1250 cm³ and pH of *ca*. 12.5. The addition of NEt₄OH resulted in the peptization of the magnetite to form a stable sol. A small fraction of non-peptized aggregates were removed by magnetic sedimentation followed by decantation.

Silica-coated magnetite was prepared by the aqueous deposition of silica from supersaturated silicic acid by a method similar to that described by Philipse et al. 12 A solution of 66.4 g sodium silicate (BDH, containing 27.0 ± 1.5 m/v % SiO_2) in 1600 cm³ distilled water was passed through a column containing 120 g 'Duolite C255 (H)' ion exchange resin (BDH) previously regenerated by successive flushing with hot distilled water, 3.0 mol dm⁻³ HCl and cold distilled water. The pH of the eluant was 7.5, and this was raised immediately to pH 9.5 by the addition of a few drops of sodium silicate stock solution in order to prevent homogeneous silica nucleation. This eluant was added to a 3000 cm³ three-necked flask containing 1200 cm³ of the magnetite sol. The pH of this stirred mixture (initially pH 12) was carefully lowered to pH 10 by titration with 0.5 mol dm⁻³ HCl over a period of approximately 1 h. The dispersion was stirred for a further 2 h then purified by dialysis in visking tubing for four days against, on successive days, three 101 aq. NEt₄OH solutions at pH 10, 9.5 and 8, respectively, and finally distilled water. The



Fig. 1 Schematic representation of the synthesis of poly(pyrrole)-silica-magnetite nanocomposites

resulting silica-coated magnetite sol had a solids content of 0.88 \pm 0.02 m/v %.

Pyrrole was polymerised in the presence of the silica-coated magnetite particles according to a procedure first described by Maeda and Armes⁷ for 20 nm silica particles. Pyrrole (0.50 cm³) was added to 150 cm³ of an aqueous dispersion containing 1.0 g (dry mass) silica-coated magnetite particles and $1.92 \text{ g} (\text{NH}_4)_2 \text{S}_2 \text{O}_8$ (or 4.55 g FeCl₃·6H₂O). This reaction mixture was left stirring at room temp. for at least 16 h. The resulting poly(pyrrole)-silica-magnetite particles were centrifuged (5000 rpm for 30 min) and redispersed in distilled water four times in order to remove ionic impurities and excess silica-magnetite particles. There was a significant excess of silica-magnetite particles left in the decanted supernatant solution after the first centrifugation/redispersal cycle. After the fourth cycle a colourless supernatant was obtained.

Transmission electron microscopy (TEM) studies indicated that the magnetite sol was polydisperse, with particle diameters in the size range 5-30 nm. Once coated with silica, these particles were much more resistant to both solution acidity and electrolyte-induced flocculation. This improved stability is critical for successful nanocomposite syntheses. Our initial attempts to synthesise colloidal poly(pyrrole)silica-magnetite nanocomposites using the FeCl₃ oxidant to polymerise pyrrole were unsuccessful (only macroscopic precipitates were obtained). However, use of the $(NH_4)_2S_2O_8$ oxidant resulted in a stable colloidal dispersion. Particle size analysis of this dispersion using a Brookhaven disc centrifuge gave a number-average particle diameter of 180 ± 50 nm, with some evidence for larger sub-micronic aggregates. This value is in reasonable agreement with our TEM studies. Elemental microanalyses of this dried nanocomposite indicated a poly-(pyrrole) content of ca. 75 mass% [based on its reduced



Fig. 2 Magnetisation curves for (a) the dried pristine magnetite sol; (b) the silica-coated magnetite particles; (c) the poly(pyrrole)-silica-magnetite nanocomposite

carbon content relative to bulk poly(pyrrole)]. Its compressed pellet room-temperature conductivity (four-point probe measurements) was approximately 10^{-3} S cm⁻¹. This value is comparable with those obtained on poly(pyrrole)–silica colloids prepared using the same oxidant^{7,8} and has been attributed to partial over-oxidation of the conducting polymer.⁸ FTIR spectroscopy studies (KBr disc) on the poly-(pyrrole)–silica–magnetite nanocomposite support this hypothesis since a small carbonyl peak at *ca*. 1700 cm⁻¹ is clearly resolved.

Magnetisation measurements were carried out as described previously;¹³ our results are depicted in Fig. 2. Superparamagnetism is exhibited by all three samples. The saturation magnetisation of the magnetite sol was approximately 64 emu g⁻¹; this plateau value is somewhat lower than the literature value of 92 emu g⁻¹ for pure magnetite.[†] The saturation magnetisation values for the silica-coated magnetite particles and the poly(pyrrole)-silica-magnetite nanocomposite were 23 and 6.5 emu g⁻¹, respectively, which correspond to magnetite contents of 36 and 9 mass%. Thus we estimate the silica content of the poly(pyrrole)-silica-magnetite nanocomposite to be approximately 16 mass% by difference.

In summary, our initial 'proof-of-concept' experiments have demonstrated the feasibility of synthesising conducting polymer colloids which exhibit superparamagnetic properties. Our future work will focus on increasing the magnetite content and improving the long-term colloid stability‡ of these novel dispersions. If these two problems can be overcome we anticipate that such particles may have applications in immunodiagnostic assays.

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Footnotes

 \dagger Other workers have attributed similar differences in magnetisation between colloidal and bulk materials (*e.g.* for maghemite) to 'surface anisotropy and the presence of a dead magnetic layer'. See ref. 5 and references therein.

[‡] The poly(pyrrole)-silica-magnetite dispersions flocculated irreversibly after storage for one week at room temp. This stability is markedly inferior to our poly(pyrrole)-silica particles which are indefinitely stable under the same conditions (see refs. 7 and 8).

References

- 1 K. Kawai, N. Mihara, S. Kuwabata and H. Yoneyama, J. Electrochem. Soc., 1990, 137, 1793.
- 2 M. G. Kanatzidis, L. M. Tonge, T. J. Marks, H. O. Marcy and C. R. Kannewurf, J. Am. Chem. Soc., 1989, 111, 4139.
- 3 V. Mehrotra, J. L. Keddie, J. M. Miller and E. P. Giannelis, J. Non-Cryst. Solids, 1991, 136, 97.
- 4 R. Partch, S. Gangolli, E. Matijevic, W. Cai and S. Arajs, J. Colloid Interface Sci., 1991, 144, 27.
- 5 G. Bidan, O. Jarjayes, J. M. Fruchart and E. Hannecart, *Adv. Mater.*, 1994, **6**, 152.
- 6 M. Gill, J. Mykytiuk, S. P. Armes, J. L. Edwards, T. Yeates, P. J. Moreland and C. Mollett, J. Chem. Soc. Chem. Commun., 1992, 108.
- 7 S. Maeda and S. P. Armes, J. Colloid Interface Sci., 1993, 159, 257.
- 8 S. Maeda and S. P. Armes, J. Mater. Chem., 1994, 4, 935.
- 9 S. Maeda and S. P. Armes, submitted to Chem. Mater.
- 10 R. Massart, IEEE Trans. Magnetics, 1981, 17, 1247.
- 11 M. D. Butterworth, C. DeArmitt, C. Forder and S. P. Armes, unpublished results.
- 12 A. P. Philipse, M. P. B. van Bruggen and C. Pathmamanoharan, Langmuir, 1994, 10, 92.
- 13 C. Forder, S. P. Armes, A. W. Simpson, C. Maggiore and M. Hawley, J. Mater. Chem., 1993, 3, 563.