

Hydrothermal synthesis and structural determinations of some new polyoxofluorovanadates $[V_{14}O_{36}F_4]^{8-}$ interleaved by water molecules and organic cations $NH_3(CH_2)_nNH_3$ ($n = 4-6$)

Cyril Ninclaus, Didier Riou*† and Gérard Férey

Institut Lavoisier, UMR-CNRS C 173, Université de Versailles-St Quentin en Yvelines, 45 Avenue des Etats-Unis, 78035 Versailles Cedex, France

$[V_{14}O_{36}F_4]^{8-}$ polyanions are built up from two trimers of three edge-sharing VO_5F octahedra, two dimers of two square pyramids related by their basal plane, two VO_4 tetrahedra and two VO_4F trigonal bipyramids; between them are intercalated water molecules and diprotonated amines which stabilize the structures by hydrogen bonds.

In the course of investigations on microporous materials, we first focused our attention on the hydrothermal behaviour of the quinary systems M_2O_3 ($M = Al, Ga$)- P_2O_5 -HF-organic template- H_2O . This led to a new series, denoted ULM- n , of fluorinated alumino- and gallo-phosphates of which ULM-5¹ and ULM-16² are the more remarkable members. The free apertures of their 16-ring tunnels are $12.2 \times 8.3 \text{ \AA}$ in ULM-5 and $10.4 \times 10.4 \text{ \AA}$ in ULM-16. Recently, we proposed a mechanism for the formation of these phases³ in which the chemical nature of the metallic element does not play a predominant role. The possibility to substitute completely aluminium or gallium by a 3d transition metal was thus explored and led to the synthesis and characterization of some microporous Fe^{III} compounds with interesting magnetic properties⁴⁻⁵ and vanadophosphates templated by organic cations.^{6,7} We also synthesized the fresnoïte related $(NH_4)_2VO(V_{2-x}P_xO_7)$ (x ca. 1), which shows that V^V and P atoms are statistically distributed on the same tetrahedral sites.⁸ If such a substitution is possible in the ULM series containing vanadium, then the existence of vanadium oxides with microporous frameworks could become a reality. The only results reported up to now concern vanadium oxides with lamellar structures $V_2O_5 \cdot 0.5$ diamine,⁹⁻¹¹ $NMe_4V_4O_{10}$ ¹² and $(C_6H_{14}N_2)V_6O_{14} \cdot H_2O$.¹³

The title compounds were synthesized hydrothermally using 1,4-diaminobutane (1,4-DAB), 1,5-diaminopentane (1,5-DAP) and 1,6-diaminohexane (1,6-DAH) as organic templates. The aqueous mixture with molar ratio 1 V_2O_5 :2 HF:0.8 diamine (1.4 in the case of 1,6-DAH):80 H_2O was placed in a stainless-steel autoclave lined with Teflon and heated under autogenous pressure at 180 °C for 24 h. The pH of the liquid phase was constant at ca. 4-5 during the reaction. The solid phase was filtered off, washed with deionized water and dried in air at room temperature. The prepared products were pure with yields of around 70%. The crystals appear black but are blue after grinding. The crystal morphology of the compound prepared with 1,5-DAP is very striking. A scanning electron micrograph (Fig. 1), performed on a Hitachi S2300, shows that they grow as big hollow needles. Suitable single crystals of both phases intercalating 1,5-DAP and 1,6-DAH were isolated for structural determination‡ by X-ray diffraction. The quality of the crystals prepared with 1,4-DAB was not sufficient to perform the same study. Nevertheless, the IR spectra of the compounds (Fig. 2), recorded on a Nicolet Magna-IR 550 in the range 2000-300 cm^{-1} , show unambiguously that the three polyanions are similar. $\nu(V=O_{terminal})$ and $\nu(V-O_{bridging})$ are observed at 970 and 750 cm^{-1} respectively, the bands observed around 1500-1600 cm^{-1} are characteristic of $-NH_3^+$. Chemically, we measured for each compound (i) the fluorine content by

potentiometric titration with a specific electrode, (ii) the vanadium content by redox titration using MnO_4^- and Fe^{2+} for V^{4+} and V^{5+} respectively, (iii) the amount of organic template and water by thermogravimetry performed on a SETARAM apparatus operating under air. These measurements give with a very good accuracy the general formula $[V^{4+}_2V^{5+}_{12}O_{36}F_4] \cdot 4$ diamine- nH_2O ($n = 6$ for 1,4-DAB, 8 for 1,5-DAP and 4 for 1,6-DAH). In addition, some density measurements were realized on a Micromeritic multipycnometer under He flow for the two compounds whose structures have been solved by single-crystal X-ray diffraction [$D_m = 2.01(3) \text{ g cm}^{-3}$, $D_c = 2.016$ for 1,5-DAP; $D_m = 2.01(4)$, $D_c = 1.982$ for 1,6-DAH].

$V_{14}O_{36}F_4 \cdot 4N_2C_6H_{18} \cdot 4H_2O$ **1** is triclinic and can be described as primitive pseudo-cubic stacking of $[V_{14}O_{36}F_4]^{8-}$ polyanions between which are intercalated the organic cations and the water molecules (Fig. 3). $V_{14}O_{36}F_4 \cdot 4N_2C_5H_{16} \cdot 8H_2O$ **2** presents a similar organization but its symmetry is monoclinic and the polyanions are oriented in two different ways. Consequently, two parameters are doubled and $V(2) = 4 V(1)$. The decrease of steric hindrance from 1,6-DAH to 1,5-DAP due to the shortening of the aliphatic chain (C_5 vs. C_6) is compensated by the presence of four additional water molecules in **2**.

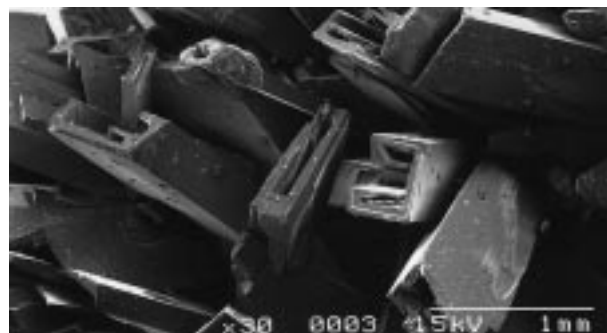


Fig. 1 SE micrograph of $V_{14}O_{36}F_4 \cdot 4N_2C_5H_{16} \cdot 8H_2O$ crystals

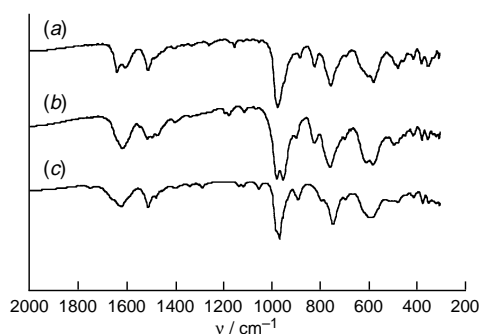


Fig. 2 IR spectra of the polyanionic phases interleaving organic cations $NH_3(CH_2)_nNH_3$ with $n = 6$ (a), 5 (b) and 4 (c)

The $[V_{14}O_{36}F_4]^{8-}$ anion contains four types of vanadium polyhedra. Two trimers of edge-sharing VO_5F octahedra form the heart of the polyanion. In each trimer, the vertex shared by the three octahedra is fluorine, F^- being characterized by bond valence analysis¹⁴ and thermal parameters in agreement with the previous chemical analysis. Within each trimer, V^{4+} occupies the octahedron which is not linked by edges to one of the four VVO_5 square pyramids relating the two trimers. The connection exists in such a way that each square pyramid shares an edge with one octahedron of one trimer and the two other O atoms of its basal plane with two octahedra of the opposite trimer (Fig. 4). On the external side of each trimer are grafted a VVO_4 tetrahedron and a VVO_4F trigonal bipyramid. Each trigonal bipyramid has three free apices, one of them is occupied by a F atom which gives rise to strong interactions with the H atoms of the amino groups. In **1** and **2**, the polyanions realize the same connectivities between the polyhedra. Nevertheless, in **1** the polyanions are built up around an inversion centre and possess 2×7 independent vanadium atoms whereas in **2**, each polyanion is built up from 14 vanadium atoms resulting from 14 independent crystallographic sites. The attribution of the V^{4+} on one particular site was possible on the basis of the bond valence analysis¹⁴ and on the detailed examination of the distribution of V–O bond lengths in the octahedra (supplementary material available from authors).

The presence of fluorine in a polyanionic phase is generally as encapsulating inside the polyanion where F^- plays the role of

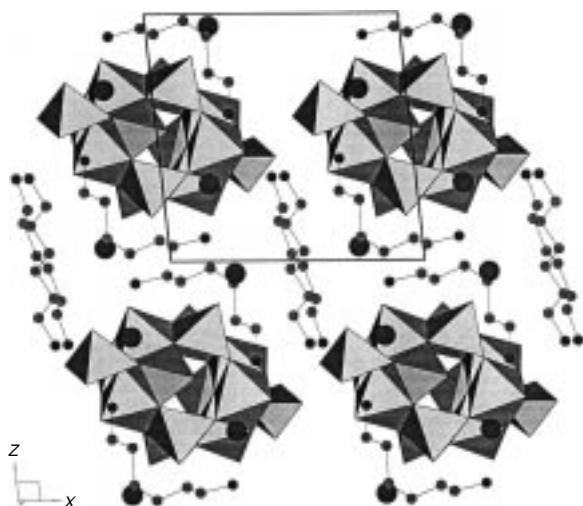


Fig. 3 Projection along [010] of the structure of $V_{14}O_{36}F_4 \cdot 4N_2C_6H_{18} \cdot 4H_2O$ with a polyhedral model for the $[V_{14}O_{36}F_4]^{8-}$ polyanions and a 'ball and stick' model for the diamines and the water molecules. Large circles refer to water and small circles to 1,6-DAH.

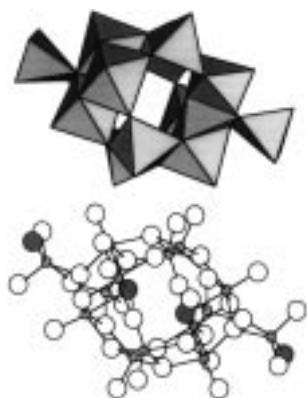


Fig. 4 Polyhedral and 'ball and stick' representations of the $[V_{14}O_{36}F_4]^{8-}$ polyanion. Vanadium atoms are indicated by small circles, oxygen and fluorine atoms with open and large grey circles, respectively.

template. Recently, we have shown in the $[V_4(HPO_4)(PO_4)_3O_6F]^{8-}$ polyanions⁷ a fluorine atom linked to the vanadium atoms and divided between four VO_5F octahedra building the core of the polyanion. This situation is almost comparable to the octahedral trimers described here and also encountered by Müller *et al.*¹⁵ in $Na_6[H_6V_{12}O_{30}F_2] \cdot 22H_2O$. The originality of **1** and **2** is the presence of a third type of fluorine, located at the outer shell of the polyanion and which stabilizes the whole structure by strong hydrogen bonds.

Footnotes

† E-mail: riou@chimie.uvsq.fr

‡ *Crystal data:* $V_{14}O_{36}F_4 \cdot 4N_2C_6H_{18} \cdot 8H_2O$: $M = 1296$, monoclinic, space group $P2_1/n$ (no. 14); $a = 11.296(4)$, $b = 24.53(2)$, $c = 23.22(1)$ Å, $\beta = 99.34(7)^\circ$, $U = 6349(6)$ Å³, $Z = 4$, $D_c = 2.016$, $D_m = 2.01(3)$ g cm⁻³, $\mu(Mo-K\alpha) = 20.73$ cm⁻¹, $\lambda = 0.71073$ Å, graphite monochromator, crystal dimensions: $0.200 \times 0.082 \times 0.536$ mm. The data were collected on an Enraf-Nonius CAD4 diffractometer up to $2\theta = 60^\circ$. A total of 19702 reflections was measured of which 6742 unique reflections ($R_{int} = 0.046$) with $I \geq 2\sigma(I)$ were considered. The data were corrected for Lorentz-polarization effects, an absorption correction based on the crystal morphology did not improve the refinement. The structure was solved by direct methods analysis of SHELXS-86 (option TREF): the vanadium atoms were first located and the all remaining atoms including those of the amine skeleton were revealed from difference Fourier maps. Refinement (812 parameters) was performed by full-matrix least-squares analysis (SHELXL-93). The reliability factors converged to $R_1 (F_o) = 0.052$ and $wR_2 (F_o^2) = 0.148$. $V_{14}O_{36}F_4 \cdot 4N_2C_6H_{18} \cdot 4H_2O$: $M = 1910$, triclinic, space group $P\bar{1}$ (no. 2); $a = 11.746(1)$, $b = 11.960(2)$, $c = 12.717(1)$ Å, $\alpha = 115.547(7)$, $\beta = 93.261(6)$, $\gamma = 94.055(9)^\circ$, $U = 1600.2(3)$ Å³, $Z = 1$, $D_c = 1.982$, $D_m = 2.01(4)$ g cm⁻³, $\mu(Mo-K\alpha) = 21.70$ cm⁻¹, $\lambda = 0.71073$ Å, graphite monochromator, crystal dimensions: $0.057 \times 0.179 \times 0.270$ mm. The data were collected on a Siemens AED2 four-circle diffractometer up to $2\theta = 60^\circ$. A total of 9316 reflections was measured of which 5707 with $I \geq 2\sigma(I)$ were considered. The data were corrected for Lorentz-polarization and absorption effects. The structure was solved by direct methods analysis of SHELXS-86 (option TREF): the vanadium atoms were first located then all the remaining atoms were deduced from difference Fourier maps. Refinement (400 variables) as above. Two carbon atoms whose thermal motion was rather high, were split over two crystallographic sites. The reliability factors converged to $R_1 (F_o) = 0.046$ and $wR_2 (F_o^2) = 0.126$. Atomic coordinates, bond lengths and angles, and thermal parameters have been deposited at the Cambridge Crystallographic Data Centre (CCDC). See Information for Authors, Issue No. 1. Any request to the CCDC for this material should quote the full literature citation and the reference number 182/388.

References

- 1 Th. Loiseau, D. Riou, F. Taulelle and G. Férey, *Zeolites and Related Microporous Materials: State of the Art 1994*, *Stud. Surf. Sci. Catal.*, 1994, **84**, 395.
- 2 Th. Loiseau and G. Férey, *J. Mater. Chem.*, 1996, **6**, 1073.
- 3 G. Férey, *J. Fluorine Chem.*, 1995, **72**, 187.
- 4 M. Cavellec, C. Ninclaus, D. Riou, J. M. Grenèche and G. Férey, *Zeolites*, 1996, **17**, 250.
- 5 M. Cavellec, J. M. Grenèche, D. Riou and G. Férey, *Microporous Mater.*, 1997, **8**, 3.
- 6 D. Riou and G. Férey, *J. Solid State Chem.*, 1994, **111**, 422.
- 7 D. Riou, F. Taulelle and G. Férey, *Inorg. Chem.*, 1996, **35**, 6392.
- 8 C. Ninclaus, R. Retoux, D. Riou and G. Férey, *J. Solid State Chem.*, 1996, **112**, 139.
- 9 D. Riou and G. Férey, *J. Solid State Chem.*, 1995, **120**, 137.
- 10 D. Riou and G. Férey, *Inorg. Chem.*, 1995, **34**, 6520.
- 11 Y. Zhang, R. C. Haushalter and A. Clearfield, *Inorg. Chem.*, 1996, **35**, 4950.
- 12 P. Y. Zavalij, M. S. Whittingham, E. A. Boylan, V. K. Pecharsky and R. A. Jacobson, *Z. Kristallogr.*, 1996, **211**, 464.
- 13 L. F. Nazar, B. E. Koene and J. F. Britten, *Chem. Mater.*, 1996, **8**, 327.
- 14 N. E. Brese and M. O'Keefe, *Acta Crystallogr., Sect. B*, 1991, **29**, 269.
- 15 A. Müller, R. Rohlfing, A. L. Barra and D. Gatteschi, *Adv. Mater.*, 1993, **5**, 915.

Received, in Basel, Switzerland, 20th November 1996; Com. 6/07863J