# **Rb2Au6Sb4S10: a novel sulfosalt with two different interpenetrating anionic frameworks:**  $[Au_3Sb_4S_8]$ <sup>-</sup> and  $[Au_3S_2]$ <sup>-</sup>

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## The layered compound  $Rb_2Au_6Sb_4S_{10}$  consisting of two **interpenetrating**  $[Au_3Sb_4S_8]$  and  $[Au_3S_2]$  frameworks was **prepared from the reaction of Au with a polythioantimonate flux.**

Polychalcoantimonate fluxes can be used for the synthesis of new ternary and quaternary thioantimonate and selenoantimonate compounds.<sup>1–4</sup> This method is complementary to conventional direct combination of the binary sulfides<sup>5</sup> or hydro(solvento)thermal synthesis.<sup>6</sup> The polychalcoantimonate fluxes are formed by the *in situ* fusion of  $A_2\dot{Q}/Sb/Q$  and contain  $[Sb<sub>x</sub>Q<sub>y</sub>]$ <sup>*n*-</sup> ligands (A = Na, K, Rb, Cs; Q = S, Se) as well as polychalcogenide ligands. The key feature of this method is that the polychalcoantimonate units form and coordinate to metal ions to build up extended lattices. Examples include  $A_2AgSbS_4$  $(A = K, Rb, Cs),$ <sup>3,4</sup> Cs<sub>3</sub>Ag<sub>2</sub>Sb<sub>3</sub>Q<sub>8</sub> (Q = S, Se),<sup>3,4</sup> KThSb<sub>2</sub>Se<sub>6</sub>,<sup>2</sup>  $A_2$ AuSbS<sub>4</sub> (A = Rb, Cs)<sup>3b</sup> and KHgSbS<sub>3</sub>.<sup>7</sup> Continuing our investigations of the coinage metals, particularly Au, we report here, the synthesis, structural characterization, and physical properties of a novel quaternary gold thioantimonate compound,  $Rb_2Au_6Sb_4S_{10}$ : The novelty in this two-dimensional compound derives from the fact that its layers are comprised of two different and independent interwoven frameworks. The only other structurally characterized example of two interpenetrating frameworks is  $K_2PdSe_{10}$ .<sup>8</sup>

The strikingly complex structure of  $Rb_2Au_6Sb_4S_{10}$ § is composed of two different interpenetrating layered frameworks,  $[Au_3Sb_4S_8]$ <sup>-</sup> and  $[Au_3S_2]$ <sup>-</sup>, Fig. 1. As a result,  $Rb_2[Au_3Sb_4$ - $S_8$ [ $Au_3S_2$ ] is a more descriptive formula, and to the best of our knowledge, represents the first reported example of a compound in which a binary framework is interpenetrating with a ternary one. The  $[Au_3Sb_4S_8]$ <sup>-</sup> layer is strongly undulating and consists of infinite  $[Sb_4S_7]^2$ <sup>-</sup> one-dimensional chains bound to  $[Au_3S]^+$ units. The  $[Sb_4S_7]^{2-}$  chain [Fig. 2(*a*)] is comprised of four condensed  $SbS<sub>3</sub>$  pyramids forming a twelve membered  $Sb-S$ ring. Two of the  $SbS<sub>3</sub>$  units share two corners leaving one terminal sulfide while the other two  $SbS<sub>3</sub>$  units share all three corners. The dimensions of the ring are  $6.49(3)$  Å  $[Sb(4)-Sb(2)]$ by 7.83(3)  $\AA$  [Sb(1)–Sb(3)]. The chains alternate above and below the layer in a staggered fashion [Fig. 2(*b*)]. The Sb atoms are in pyramidal coordination with Sb–S distances in the range from  $2.21(6)$  to  $2.65(6)$  Å [mean  $2.46(3)$  Å] and compare well with those reported for  $Cs_2Sb_4S_8^1$  and  $Cs_3Ag_2Sb_3S_8^3$ .<sup>3,4</sup> The discrete  $[Au_3\hat{S}]^+$  unit has a pyramidal sulfide linked to three linear Au<sup>+</sup> cations. The Au–S distances range from 2.28(4) to 2.46(5) Å and compare well with those found in  $CsAu<sub>3</sub>S<sub>2</sub>,<sup>9a</sup>$ AAuS<sup>9b</sup> (A = Na, K, Rb, Cs), KAuS<sub>5</sub><sup>10</sup> and AAuSbS<sub>4</sub><sup>3</sup>  $(A = Rb, Cs)$ . The S-Au-S angles range from 170 to 174°.

The second framework, which is interwoven with the one described above is a  $[Au_3S_2]$ <sup>-</sup> layer. The  $[Au_3S_2]$ <sup>-</sup> layer is puckered with twelve-membered Au–S rings in an  $anti-B<sub>2</sub>O<sub>3</sub>$ motif {ring dimensions:  $6.91(2)$  Å [Au(4) $\cdots$ Au(4)] by 7.19(3) Å  $[Au(3)\cdots Au(3)]$ . Fig. 3(*a*) highlights the pyramidal sulfide and the puckered nature of the layer and Fig. 3(*b*) shows a perpendicular view. The Au–S distances are in the range from  $2.25(4)$  to 2.46(5) Å and the S–Au–S angles range from 165 to 178°. The  $[Au_3S_2]$ <sup>-</sup> layered structure is similar but not identical to that observed in  $CsAu<sub>3</sub>S<sub>2</sub>.<sup>9a</sup>$ 

Upon further inspection, it was observed that the Au+ centers in  $Rb_2Au_6Sb_4S_{10}$  aggregate to form a column that runs along the *c*-axis. There are two types of Au···Au interactions: those at  $\leq 3.25$  Å and those between 3.25 and 3.6 Å. Fig. 3(*c*) shows a view perpendicular to these Au-based columns. In Fig. 3(*c*) the interactions  $\leq 3.25$  Å are represented as solid lines while the Au $\cdots$ Au interactions between 3.25 and 3.60 Å are represented as dashed lines. The layers are separated by ten-coordinate  $Rb(1)$ <sup>+</sup> [Rb(1)–S (mean) 3.59(3) Å] and eight-coordinate Rb(2)  $[Rb(2)$ + [Rb(2)–S (mean) 3.50(3) Å].

The optical spectrum of  $Rb_2Au_6Sb_4S_{10}$  reveals the presence of a sharp optical gap of 1.37 eV, suggesting the material is a semiconductor.

The far-IR spectrum of  $Rb_2Au_6Sb_4S_{10}$  displays absorptions at *ca*. 377 and 350 cm<sup> $-1$ </sup> which can be tentatively assigned to Sb–S stretching modes in the 'Sb<sub>2</sub>S<sub>4</sub>'-like backbone of the  $[Au_3Sb_4S_8]$ <sup>-</sup> framework.<sup>1,3</sup> Absorptions in the range 381–347  $cm<sup>-1</sup>$  are tentatively assigned to the Sb–S vibrational stretching modes by analogy with the  $Cs_2Sb_4S_8^1$  and  $Cs_3Ag_2Sb_3S_8^3$ Absorptions below 347  $cm<sup>-1</sup>$  are assigned to Au–S vibrations as compared to  $Rb_2AuSbS_4$ .<sup>3</sup> By comparison with  $KAuS_5$  the absorption at *ca*. 323 cm<sup> $-1$ </sup> is assigned as an Au–S stretching vibration.<sup>10</sup> The Raman spectrum¶ of  $Rb_2Au_6Sb_4S_{10}$  displays absorptions in the range  $377-350$  cm<sup>-1</sup> which are assigned to



Fig. 1 Structure of Rb<sub>2</sub>Au<sub>6</sub>Sb<sub>4</sub>S<sub>10</sub> viewed down the *c*-axis. Selected distances ( $\AA$ ) and angles ( $\degree$ ) with esds in parentheses: Au(1)–S(3) 2.32(1), Sb(1)–S(6) 2.43(2), Au(1)–S(5) 2.30(1), Sb(1)–S(8) 2.47(1) ( $\times$ 2), Au(2)– S(2) 2.28(2), Sb(2)–S(5) 2.54(2), Au(2)–S(4) 2.30(2), Sb(2)–S(1) 2.48(1)  $(\times 2)$ , Au(3)–S(3) 2.32(1), Sb(3)–S(6) 2.58(2), Au(3)–S(7) 2.31(1), Sb(3)– S(1) 2.48(1) ( $\times$ 2), Au(4)–S(3) 2.29(2), Sb(4)–S(4) 2.40(2), Au(4)–S(7) 2.30(2), Sb(4)–S(8) 2.47(1)  $(\times 2)$ , Au(1)–Au(3) 3.192(3), Au(1)–Sb(1) 3.393(5), Au(1)–Au(4) 3.060(4), Au(1)–Sb(3) 3.233(5), Au(2)–Au(4)  $3.593(1)$ , Au(2)–Sb(2)  $3.091(6)$ , Au(1)–Au(1')  $3.588(4)$ , Au(2)–Au(3)  $3.389(4)$ , Au(3)–Au(3')  $3.461(5)$ , Au(3)–Au(4)  $3.558(3)$ ; S(2)–Au(1)–S(5)  $170.9(7)$ ,  $S(1)$ –Sb(3)–S(1') 89.4(7),  $S(2)$ –Au(2)–S(4) 173.9(8),  $S(1)$ –Sb(3)–S(6) 95.6(5), S(3)–Au(3)–S(7) 178.4(7), S(1')–Sb(3)–S(6) 95.6(5), S(3)–Au(4)–S(7) 170.0(8).



**Fig. 2** (*a*) ORTEP view of the  $[Sb_4S_7]^{2-}$  chain with labelling; (*b*) ORTEP view of the complete  $[Au_3Sb_4S_8]$ <sup>-</sup> framework with labelling



Fig. 3 (*a*) ORTEP view of the  $[Au_3S_2]$ <sup>-</sup> layer highlighting the pyramidal sulfides in the undulating layer; (b) perpendicular view of the  $[Au_3S_2]$ <sup>-</sup> layer with labelling; (*c*) Au···Au interactions of the Au column

Sb–S modes and the absorptions below  $350 \text{ cm}^{-1}$  are assigned to Au–S stretching vibrations.

DTA data, followed by careful XRD analysis of the residues, show that  $Rb_2Au_6Sb_4S_{10}$  melts incongruently at *ca*. 442 °C. Examination of the residue by powder XRD revealed that the compound decomposes to an amorphous material and Au metal.

In conclusion,  $Rb_2Au_6Sb_4S_{10}$  represents the first example of a sulfosalt with two different interpenetrating anionic frame-

works. Although one of the frameworks  $[Au_3S_2]$ <sup>-</sup> can exist by itself, efforts to isolate the second framework alone are in progress.

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## **Notes and References**

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 $\ddagger$  Rb<sub>2</sub>Au<sub>6</sub>Sb<sub>4</sub>S<sub>10</sub> was synthesized from a mixture of Rb<sub>2</sub>S (0.102 g, 0.5) mmol), Au (0.098 g, 0.5 mmol), Sb (0.031 g, 0.25 mmol) and S (0.064 g, 2 mmol) sealed under vacuum in a Pyrex tube and heated to 350 °C for 4 days followed by cooling to 150 °C at  $4^{\circ}$ C h<sup>-1</sup>. The excess  $Rb_x$ [Sb<sub>y</sub>S<sub>z</sub>] flux was removed by washing with DMF to reveal analytically pure black needles in low yield (20% yield based on Sb). The crystals are air- and water-stable. Microprobe analysis carried out on several randomly selected crystals gave an average composition of  $RbAu_{4.6}Sb_{2.7}S_9$ . This technique tends to underestimate the amount of Rb.

§ *Crystallography*: A Siemens SMART Platform CCD diffractometer equipped with Mo-K $\alpha$  was used for a crystal of  $0.400 \times 0.010 \times 0.005$  mm. An empirical radiation ( $\lambda = 0.71073$  Å) absorption correction was applied to the data during data processing. *Crystal data* at  $-120$  $a = 12.4402(2)$ ,  $b = 26.0790(4)$ ,  $c = 6.9614(1)$  Å,  $U = 2258.3(7)$  Å<sup>3</sup>,  $Z = 4, D_c = 4.043$  g cm<sup>-3</sup>, space group *Pnnm* (no. 63),  $\mu = 48.83$  cm<sup>-1</sup>, index ranges  $-14 \le h \le 14$ ,  $-31 \le k \le 28$ ,  $-8 \le l \le 8$ ; total data 15609; unique data 2340 ( $R_{\text{int}} = 0.146$ ), data with  $F_o^2 > 3\sigma(F_o^2)$  1181; no. of variables, 124; final *R*,  $R_w = 0.080$ , 0.096; GOF 2.66; max. peak in difference electron density map = 7.97 e<sup>-</sup> Å<sup>-3</sup>. The structure was solved with SHELXS-86 and refined with the TEXSAN Structure Analysis Package (Molecular Structure Corporation, 1985) of crystallographic programs. CCDC 182/760.

¶ Far-IR (CsI matrix) gave absorptions at *ca*. 381w, *ca.* 359w (sh), 347m,  $325m$ ,  $306w$ ,  $295w$ ,  $270s$  and  $233w$  cm<sup>-1</sup>. Raman spectra (ground crystals) gave absorptions at *ca*. 377w, 350s, 323m, 285w, 266m and 250s cm<sup>-1</sup>.

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