

Oligo(fluorenyleneethynylene)germylene)s and their metallopolymers†

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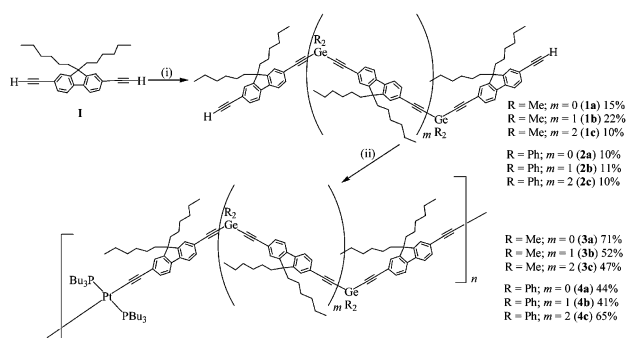
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Oligo(fluorenyleneethynylene)germylene)s and their polyplatinyne)s are synthesized and photophysically characterized; inclusion of heavy germylene bridges greatly boosts the phosphorescence decay rate in metallopolymers.

The challenge of developing molecular systems composed of main group heteroatoms and acetylene units has had aesthetic appeal in recent years.¹ These acetylene-based scaffolds and their metal-capped congeners play a vital role in molecular optoelectronics.² While the study of oligo- and poly(aryleneethynylene)s represents an active research area,³ little is understood on related systems with the heavier group 14 element germanium, and no oligo(aryleneethynylene)germylene)s of precise length and constitution have ever been described. We report here the unprecedented synthesis and photophysics of stable blue-emitting oligo(fluorenyleneethynylene)germylene)s (OFEG)s with alternating Ge and diethynylfluorene units. The first examples of Ge-bridged Pt(II) metallopolymers, derived from these oligomers, are also presented which display very fast phosphorescence decay rates. The triplet states in light-emitting diodes can be utilized in metal-organics through light-harvesting techniques.⁴ To optimise this, work should focus on polymers with high energy triplets to avoid competition with nonradiative decay. Up till now, no reports on polymetalynes with comparable radiative and nonradiative decay rate constants are known and it is highly desirable to achieve this goal for practical applications. The key feature of this study is the inclusion of a conjugation-interrupting sp³-Ge linker that can limit the effective conjugation length (ECL) and trigger the triplet light emission by taking advantage of the heavy-atom effect of Ge atoms.

Lithiation of **1** by ⁿBuLi in THF at -78 °C followed by germylation with R₂GeCl₂ (R = Me, Ph) gave a separable mixture of monodisperse OFEG **1a-c** and **2a-c** (Scheme 1). In both series, the yields of the dimers (*m* = 1) and trimers (*m* = 2) were increased at the expense of the monomers (*m* = 0) when more ⁿBuLi was used. The merit of these short-chain oligomers over their polymeric counterparts lies in their precise molecular architectures, better solubility and their predetermined chemical composition.⁵ These



Scheme 1 Reagents and conditions: (i) ⁿBuLi, -78 °C, R₂GeCl₂, THF; (ii) *trans*-[PtCl₂(PBu₃)₂], CuI, NEt₃, CH₂Cl₂.

† Electronic supplementary information (ESI) available: experimental procedures and spectral data of all new compounds. See <http://www.rsc.org/suppdata/cc/b4/b409580d/>

OFEGs are very organic-soluble and are fully characterized by NMR, IR and MS. The structures of **1a** and **2a** were confirmed by X-ray analyses (Fig. 1).⁶ Stable polyplatinyne)s **3a-c** and **4a-c** were obtained by the dehydrohalogenative polymerization between *trans*-[PtCl₂(PBu₃)₂] and the respective OFEG and their excellent film-forming properties and GPC data (Table 2) suggest a macromolecular nature for these materials. All the new compounds display excellent thermal stability in the TGA curves (Tables 1 and 2). The onset decomposition temperatures (*T*_{dec}) are almost invariant of the chain length and the R group. While *trans*-[Pt(PBu₃)₂C≡C(9,9-dihexylfluorene-2,7-diyl)C≡C]_n, **5** commences decomposition at 349 °C,⁷ addition of a GeR₂ unit into the aryl-acetylene segment significantly increases the thermal stability of these Pt polyynes.

The absorption spectra of OFEG consist of organic ¹(ππ*) transitions in the near-UV region (Table 1), and a slight bathochromic shift in λ_{max} with increased extinction coefficients being observed for both series as *m* increases. But, the magnitude of red shifts rapidly decreases with increasing *m* value and there is no further significant shift in going from **2c** to the longer-chain polymer [-C≡C(9,9-dihexylfluorene-2,7-diyl)C≡CGePh₂]_n (λ_{max} = 346 nm). So, there would be little benefit for *m* > 3 in the control of spectral properties. The R groups on Ge exert minimal effect on the absorption energies. The transition energies of the Pt polymer are lowered relative to those of the OFEG, suggesting a well extended singlet excited state in the polymers. Examination of the absorption behavior of **1-4** in CH₂Cl₂ and in the solid state precludes any solid-state aggregates in thin films.

Our OFEGs are good light emitters and provide the best finite models for the Ge-containing poly(fluorenyleneethynylene)s. They emit remarkably strong purple-blue fluorescence at room temperature. The fluorescence quantum yields (Φ_F) of these oligomers are found to be chain length dependent and they increase down each series from **a** to **c** (*i.e.* *m* = 2 > 1 > 0). We also note that the Me compounds exhibit higher Φ_F than the Ph congeners and the emissions from **1b** and **1c** are highly efficient with Φ_F up to 0.92. These observations suggest that intrachain aggregation of luminophores may exist in these oligomers, and this interaction seemed to be more important as the chain length increases.^{3a,8} While the presence of Ge can make the molecules more flexible, the opportunity for one chromophore unit to be located proximal to the other in space would increase with the molecular weight. The fluorescence radiative decay rate, (*k*_r)_F, is found to increase down each series or when passing from Ph to Me substituents. However, such fluorescence enhancement by increasing the *m* value tends to reach saturation up to *m* = 3 and Φ_F converges to a limiting value in their polymeric homologue (*n*_{ECL} = 3). For the Pt polymers,

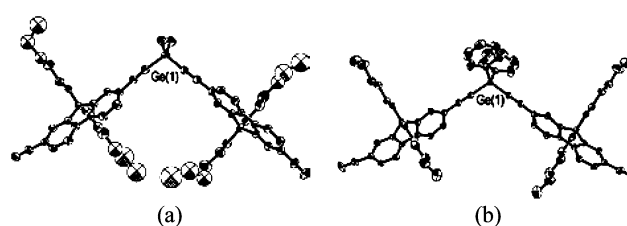


Fig. 1 ORTEP drawings of (a) **1a** and (b) **2a**.

Table 1 Photophysical data (at 293 K) and thermal analyses of OFEG

OFEG	$\lambda_{\text{abs}}/\text{nm}^a$ (CH ₂ Cl ₂)	$\lambda_{\text{abs}}/\text{nm}$ (film)	E_{gap}/eV	$\lambda_{\text{em}}/\text{nm}$ (CH ₂ Cl ₂)	Φ_{F}^b	$\tau_{\text{F}}/\text{ns}$	$(k_{\text{r}})_{\text{F}}/\text{s}^{-1}$	$T_{\text{dec}}/^{\circ}\text{C}$
1a	337 (12.8)	339	3.57	362, 379	0.78	1.18	6.6×10^8	447
1b	343 (15.6)	343	3.51	368, 384	0.90	1.18	7.6×10^8	442
1c	344 (20.7)	345	3.49	369, 386	0.92	1.16	7.9×10^8	442
2a	339 (13.0)	341	3.54	364, 379	0.35	1.09	3.2×10^8	442
2b	345 (16.9)	347	3.49	371, 386	0.66	1.08	6.1×10^8	446
2c	346 (22.3)	348	3.47	371, 386	0.75	1.05	7.1×10^8	443

^a ϵ ($10^4 \text{ dm}^3 \text{ mol}^{-1} \text{ cm}^{-1}$) values are shown in parentheses. ^b Measured in CH₂Cl₂ relative to quinine sulfate in 0.1 N H₂SO₄ ($\Phi_{\text{F}} = 0.54$), $\lambda_{\text{ex}} = 313 \text{ nm}$.

Table 2 Structural, thermal stability and emission data (at 20 K) of metallopolymers **3a–c** and **4a–c**

Polymer	M_{w}^a	M_{n}^a	$T_{\text{dec}}/^{\circ}\text{C}$	$\lambda_{\text{em}}/\text{nm}$ (film)	$\tau_{\text{P}}/\mu\text{s}$	Φ_{P}	$(k_{\text{nr}})_{\text{P}}/\text{s}^{-1}$	$(k_{\text{r}})_{\text{P}}/\text{s}^{-1}$
3a	42970	13580	410	412, 547	1.27	0.45	4.3×10^5	3.5×10^5
3b	22950	11150	414	411, 544	2.08	0.43	2.7×10^5	2.1×10^5
3c	22790	16150	407	410, 544	1.41	0.45	3.9×10^5	3.2×10^5
4a	23970	10400	404	414, 548	1.32	0.17	6.3×10^5	1.3×10^5
4b	28350	12830	407	411, 545	1.21	0.18	6.8×10^5	1.5×10^5
4c	22600	17550	418	411, 545	1.16	0.20	6.9×10^5	1.7×10^5

^a Calibration against polystyrene standards.

both fluorescent and phosphorescent emissions arise from ligand-centred ($\pi\pi^*$) transitions. In dilute fluid solutions, we observe an intense ($\pi\pi^*$) fluorescence peak near 400 nm for **3a–c** and **4a–c**, which do not display a large shift at high concentrations, excluding an excimer origin. At low temperatures, lower-lying spin-forbidden phosphorescence bands emerge at around 545 nm and the substantial Stokes shifts of these peaks from the dipole-allowed absorptions, plus the long emission lifetimes in the microsecond range are suggestive of their triplet parentage, and they are assigned to the ³($\pi\pi^*$) states of the organic chromophores. The triplet energy does not vary much with oligomer chain length, *i.e.* the lowest T_1 state is confined to a single repeat unit. Variation of the R group does not seem to alter this strong confinement. Insertion of a conjugation hindered GeR₂ group in **3** and **4** shifts the phosphorescence bands to the blue relative to **5**.⁷ Values of $\Delta E(S_0-T_1)$ (energy gap between S_0 and T_1) are found to be *ca.* 2.27–2.28 eV for both series. This corresponds to S_1-S_0 gaps of ~ 3.0 eV. The $\Delta E(S_1-T_1)$ values lie within the narrow range of 0.73–0.75 eV, in line with the S_1-T_1 energy gap of 0.7 ± 0.1 eV for metal polyynes of group 10–12 elements.⁹

The phosphorescence lifetimes (τ_{P}), quantum yields (Φ_{P}), and radiative ($(k_{\text{r}})_{\text{P}}$) and nonradiative ($(k_{\text{nr}})_{\text{P}}$) decay rates at 20 K are listed in Table 2. Although the measured Φ_{P} values are relatively insensitive to the value of *m*, they are found to vary with the ER₂ group (E = Si, Ge). The GeMe₂ systems give more efficient phosphorescence than the GePh₂ congeners by over 2 times. But, replacement of GePh₂ by SiPh₂ reduces Φ_{P} by almost half ($\Phi_{\text{P}} \sim 10$ –13% for a SiPh₂ system) that can be correlated to the heavy-atom effect associated with the Ge atoms in the former case. The markedly different properties exhibited by these germylene polymers compared with their silylene analogues are mirrored by differences in other pairs of Ge and Si systems.¹⁰ The $(k_{\text{r}})_{\text{P}}$ values at 20 K are $(2.1\text{--}3.5) \times 10^5 \text{ s}^{-1}$ for **3a–c** and $(1.3\text{--}1.7) \times 10^5 \text{ s}^{-1}$ for **4a–c**. Relative to **5** ($(k_{\text{r}})_{\text{P}} \sim 4.4 \times 10^4 \text{ s}^{-1}$), insertion of the germylene component can increase $(k_{\text{r}})_{\text{P}}$ by about 1 order of magnitude. Now, we are able to get comparable orders of magnitude for $(k_{\text{nr}})_{\text{P}}$ and $(k_{\text{r}})_{\text{P}}$ which have never been observed in polymetalloynes reported so far. For phosphorescence in aromatic hydrocarbons, $(k_{\text{r}})_{\text{P}}$ lies typically between 0.1 and 1 s^{-1} .¹¹ So, heavy-atom derivatization using Pt and Ge atoms together with conjugation interruption by the latter can greatly boost $(k_{\text{r}})_{\text{P}}$ values by *ca.* 5 orders of magnitude. It is likely that the high energy benzene stretching modes of the Ph group is efficient at promoting $(k_{\text{nr}})_{\text{P}}$ for **4a–c**, making Φ_{P} and $(k_{\text{r}})_{\text{P}}$ values smaller than those in **3a–c**.

In summary, we have developed a novel approach based on the heavy Ge conjugation-interrupter in metallopolymers that can limit ECL and result in much larger $(k_{\text{r}})_{\text{P}}$ values. The present work has great potential to excel in optoelectronics that demand light energy

harvesting from the T_1 state and one can take benefit from the large yield and radiative decay rate of triplet excitons.

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- Crystal data for **1a**: C₆₀H₇₂Ge, $M_{\text{w}} = 865.82$, monoclinic, space group $C2/c$, $a = 33.747(2)$, $b = 10.2041(5)$, $c = 31.427(1) \text{ \AA}$, $\beta = 91.242(1)^{\circ}$, $V = 10819.8(9) \text{ \AA}^3$, $Z = 8$, $\mu(\text{Mo-K}\alpha) = 0.600 \text{ mm}^{-1}$, 65722 reflections measured, 24369 unique, $R(\text{int}) = 0.106$. Final $R = 0.0860$ and $R_{\text{w}} = 0.1050$ for 4866 observed reflections with $I > 1.5\sigma(I)$. For **2a**: C₇₀H₇₆Ge, $M_{\text{w}} = 989.96$, monoclinic, space group $P2/c$, $a = 9.1566(7)$, $b = 12.0396(9)$, $c = 26.852(2) \text{ \AA}$, $\beta = 97.021(2)^{\circ}$, $V = 2938.0(4) \text{ \AA}^3$, $Z = 2$, $\mu(\text{Mo-K}\alpha) = 0.560 \text{ mm}^{-1}$, 17699 reflections measured, 7122 unique, $R(\text{int}) = 0.038$. Final $R = 0.0510$ and $R_{\text{w}} = 0.0530$ for 4172 observed reflections with $I > 1.5\sigma(I)$. CCDC 237762–237763. See <http://www.rsc.org/suppdata/cc/b4/b409580d/> for crystallographic data in .cif or other electronic format.
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