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Surface Structures of Adsorbed Gases on Solid Surfaces. A Tabulation of Data Reported by Low-Energy Electron Diffraction Studies

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/. Introduction

Studies of ordered surface structures formed upon adsorption of gases on solid surfaces are carried out using low-energy electron diffraction (LEED). The technique of diffraction is sensitive to order, and low-energy electrons provide the needed surface sensitivity; therefore, LEED is well suited for determining the surface structures of monolayers of adsorbed gases. The number of investigations that report on ordered adsorbates has increased dramatically in the last four years. In 1974, a tabulation of over 300 surface structures observed in LEED experiments was published in a review article by Somoriai and Kesmodel.¹ Since that time over 500 surface structures have been reported in more than 300 papers. In this review we have updated and expanded the previous tabulation to include this large amount of new data.

Not only has the number of LEED studies increased, but the types of experiments now cover a wider range. In the past most of the reported surface structures were from experiments dealing with the adsorption of small inorganic molecules $(H_2, O_2, CO,$ $CO₂$, N₂, etc.) on low Miller index surfaces. At that time only three studies on stepped or high Miller index surfaces were reported. Since 1974 the number of LEED experiments on stepped surfaces has greatly increased. As a result we have prepared a separate table for listing the surface structures formed on stepped surfaces. There is also a large number of LEED investigations of organic molecule adsorption on surfaces, and we have listed these surface structures in a separate table. Ethylene and acetylene are the two most widely studied organic molecules, but there have also been several adsorption studies of large organic molecules both in the monolayer and multilayer regimes. We have excluded the LEED studies of ordered structures formed by condensed metallic monolayers from our tables and discussion.

The surface unit cells formed during the adsorption of mole-

cules and atoms on various substrates yield information on the size, shape, and orientation of the surface unit mesh. While they do not give the precise location or identification of the adsorbed surface species, this is possible by the analysis of the diffraction beam intensities. In surface crystallography the intensity profiles of the diffraction spots obtained as a function of the incident electron beam energy (l-V curves) are analyzed to determine the positions of atoms in the surface unit mesh. Out of the 300 LEED articles in the past four years, over 30 contained experimental /-V curves and/or the analysis of these curves.

The results of the surface crystallography analysis have been reviewed recently.² This analysis involves multiple scattering calculations and requires the availability of accurate $I-V$ data on a large number of diffraction beams. Most LEED studies report only the size and shape of the two-dimensional surface unit mesh and do not provide enough information to permit surface structure analysis.

//. The LEED Experiment

A typical LEED apparatus is shown in Figure 1. A monoenergetic beam of electrons (10 to 300 eV) is directed at the surface of a single crystal which backscatters a portion of the incoming electrons. A set of herispherical grids is used to remove the inelasticaily backscattered electrons while the elastically backscattered electrons are post-accelerated onto a phosphorus screen for viewing of the diffraction pattern. The crystal and the detection system are enclosed in an ultrahigh vacuum (UHV) chamber in order to attain and maintain a clean surface. As seen in Figure 1 there is a window present in the UHV chamber directly opposite the phosphorus screen which allows the diffraction pattern to be viewed and photographed from outside the UHV chamber. A polaroid camera is commonly used for photographing the diffraction pattern and the published LEED patterns are from such photographs. LEED is commonly combined with other techniques such as Auger electron spectroscopy (AES), thermal desorption spectroscopy (TDS), ultraviolet photoelectron spectroscopy (UPS), and electron loss spectroscopy (ELS). This allows correlation of the surface structure information with surface composition, binding energies, electronic properties, and vibrational modes for an adsorbate.

A well-ordered crystal surface will yield a diffraction pattern consisting of bright, well-defined spots with very low background intensity. The sharpness and overall intensity of the spots depend on the degree of order on the surface. Although the surface may

Figure 1. Schematic diagram of a typical LEED apparatus.

Figure 2. Real space vectors \vec{a} and b and reciprocal space vectors \vec{a}^* and b* of a two-dimensional hexagonal lattice.

be irregular on a microscopic and submicroscopic scale (e.g., consisting of atomic terraces and ledges), the presence of sharp diffraction features indicates that the surface is ordered on an atomic scale; that is, most of the surface atoms are located in a two-dimensional lattice structure. The size of these ordered domains determines the quality of the diffraction pattern. Because of the experimental limitations on the coherence width of the electron beam, the quality of the diffraction pattern does not improve when the ordered domains become larger than 500 A in diameter. However, if the ordered domains are significantly smaller than 500 A, the diffractions spots become broader and less intense.

///. **Conversion of the Diffraction Pattern to a Surf ace Structure**

LEED diffraction patterns represent the reciprocal lattice of the surface and the diffraction pattern must be converted to real space in order to obtain the surface structure. In this section we will show how this conversion is performed. First, the relationship between the reciprocal and real lattices of the substrate will be shown; then determination of adsorbate surface structures from the LEED patterns will be discussed.

The diffraction pattern or reciprocal lattice has translational periodicity which is given by the vector T^* which has the form

$$
\vec{\tau}^* = h\vec{a}^* + k\vec{b}^* \tag{1}
$$

where h and k are integers and \vec{a}^* and \vec{b}^* are the vectors of the primitive surface reciprocal mesh. The translational periodicity of the surface in real space is given by the vector T which has the form

$$
\vec{\tau}^* = n\vec{a} + m\vec{b} \tag{2}
$$

where n and m are integers and \ddot{a} and b are the vectors of the

Figure 3. LEED patterns of a clean Pt(111) surface and the same surface after exposure to C_2H_2 . In both diffraction patterns the incident electron beam energy is 68 eV.

primitive surface mesh. The reciprocal unit cell vectors \vec{a}^* and b^* are related to the real space unit cell vectors \vec{a} and b by the following equations:

$$
\vec{a}^* = (b \times \vec{z}) / (\vec{a} \cdot \vec{b} \times \vec{z}) \tag{3a}
$$

$$
b^* = (\vec{z} \times \vec{a})/(\vec{a} \cdot \vec{b} \times \vec{z}) \tag{3b}
$$

where \vec{z} is the surface normal. The relationship between the reciprocal and real space vectors is illustrated for a two-dimensional hexagonal lattice in Figure 2.

Adsorbing a gas on a surface usually results in a change in the diffraction pattern, corresponding to the appearance of a new surface mesh. This is illustrated in Figure 3 which shows a diffraction pattern of a clean Pt(111) surface and the diffraction pattern formed after the adsorption of acetylene. Figure 4 shows the unit mesh responsible for the diffraction patterns in Figure 3 superimposed on a model of the Pt(111) surface. No information concerning the location of the acetylene molecule within this unit mesh is indicated since this information can only be obtained from analysis of the diffraction spot intensities.

In order to make the transition from the diffraction pattern in Figure 3 to the surface structure in Figure 4, the adsorbate surface reciprocal mesh is referenced to the substrate reciprocal mesh. This is done by visual inspection of the diffraction pattern where the differences in spot intensities are neglected and only the positions of the diffraction beams are considered.

For the general case the relationship of adsorbate reciprocal mesh to the substrate reciprocal is given by the equations

$$
\vec{a}^{\,*}{}' = m_{11}^{\,*} \vec{a}^{\,*} + m_{12}^{\,*} \vec{b}^{\,*} \tag{4a}
$$

$$
b^{*'} = m_{21}^* \ddot{a}^* + m_{22}^* \ddot{b}^* \tag{4b}
$$

where $\vec{a}^{\,*}$ and $\vec{b}^{\,*}$ are the vectors of the primitive adsorbate reciprocal mesh and the coefficients m_{11}^* , m_{12}^* , m_{21}^* , and m_{22} ^{*} define the matrix

$$
M^* = \begin{pmatrix} m_{11}^* & m_{12}^* \\ m_{21}^* & m_{22}^* \end{pmatrix}
$$

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Figure 4. Real space unit cells of $Pt(111)–(1\times1)$ and $Pt(111)–(2\times2)–$ C_2H_2 .

In real space the adsorbate mesh is related to the substrate mesh by the equations

$$
\vec{a}' = m_{11}\vec{a} + m_{12}\vec{b} \tag{5a}
$$

$$
b' = m_{21}\vec{a} + m_{22}\vec{b}
$$
 (5b)

where \vec{a}' and \vec{b}' are the vectors of the primitive adsorbate mesh and the coefficients m_{11} , m_{12} , m_{21} , and m_{22} define the matrix

$$
M = \begin{pmatrix} m_{11} & m_{12} \\ m_{21} & m_{22} \end{pmatrix}
$$

The coefficients of the two matrices M and M^* are related by the following equations:

$$
m_{11} = \frac{m_{22}^{*}}{m_{11}^{*}m_{22}^{*} - m_{21}^{*}m_{12}^{*}}
$$
 (6a)

$$
m_{12} = \frac{-m_{21}^*}{m_{11}^*m_{22}^* - m_{21}^*m_{12}^*}
$$
 (6b)

$$
m_{21} = \frac{-m_{12}^*}{m_{11}^*m_{22}^* - m_{21}^*m_{12}^*}
$$
 (6c)

$$
m_{22} = \frac{m_{11}^*}{m_{11}^* m_{22}^* - m_{21}^* m_{12}^*}
$$
 (6d)

so that if either M or M^* is known the other may be readily calculated. In LEED experiments M^* is determined by visual inspection of the diffraction pattern and then transformed to give M which defines the surface structure in real space.

For the case of acetylene adsorption on Pt(111) visual inspection of the LEED patterns in Figure 3 gives $M^* = \binom{1}{0}^2 \binom{0}{1}$. From employing eq 6a through 6d the matrix M is found to be $\binom{20}{02}$, so $\vec{a}' = 2\vec{a}$ and $b' = 2b$ as depicted in Figure 4.

In addition to the matrix method of denoting surface structures, another system, originally proposed by Wood,³ is also used. While the matrix notation can be applied to any system, Wood's notation can only be used when the angle between the adsorbate vectors \vec{a}' and \vec{b}' is the same as the angle between the substrate vectors \vec{a} and \vec{b} . If this condition is met, then the surface structure is labeled using the general form $(n \times m)R\Phi^{\circ}$ or $\alpha(n \times m)R\Phi^{\circ}$, depending on whether the unit mesh is primitive or centered. In Wood's notation the adsorbate unit mesh is related to substrate

Figure 5. LEED patterns of (a) clean Rh(IOO) at 74 eV and (b) oxygencovered Rh(100) at 85 eV.

Figure 6. Real space units cells of the $(2X2)R45^{\circ}$ -O (solid lines) and $c(2X2)-O$ (dashed lines) on the Rh(100) surface.

unit mesh by the scale factors n and m where

$$
|\vec{a}'| = n|\vec{a}'| \tag{7a}
$$

$$
|\vec{b}'| = m|\vec{b}'| \tag{7b}
$$

 $R\Phi$ ^o indicates a rotation of the adsorbate unit mesh by Φ ^o from the substrate unit mesh. For $\Phi = 0$ the $R\Phi^{\circ}$ label is omitted, so the surface structure in Figure 4 is labeled as a (2×2) . The label for the total system refers to the type of substrate, the surface structure formed by the adsorbate and the adsorbate. The platinum-acetylene system shown in Figure 4 would be labeled as Pt(111)- $\binom{2}{0}$ -C₂H₂ in matrix notation and as Pt(111)-(2 \times 2)- C_2H_2 in Wood's notation. Wood's notation is more commonly used, and the matrix notation is usually only applied to systems where the angle between the adsorbate vectors differs from the angle between substrate vectors.

An example of an adsorbate which has a centered unit mesh is shown in Figures 5 and 6. In Figure 5 diffraction patterns from a clean Rh(100) surface and a Rh(100) surface after exposure to oxygen are shown. By visual inspection it can be seen that $M^* = (\frac{1}{6}, -\frac{1}{6})$, so using eq 6a through 6d yields $M = \begin{pmatrix} 1 & 1 \\ -1 & 1 \end{pmatrix}$. M defines the primitive unit mesh of the adsorbate, which is drawn in with solid lines in Figure 6. This unit mesh is labeled as $(\sqrt{2}X\sqrt{2})R45^{\circ}$ in Wood's notation. Since the centered unit mesh drawn in with dotted lines in Figure 6 also describes the adsorbate mesh, another way of labeling this structure would be $c(2\times2)$. The total system is labeled as Rh(100)–($\begin{bmatrix} 1 & 1 \\ 1 & -1 \end{bmatrix}$ –O, Rh(100)-($\sqrt{2X}\sqrt{2}$) $R45^{\circ}$ -O or Rh(100)-c(2X2)-O. Of these three labels only the first two refer to the primitive unit mesh of the oxygen surface structure.

IV. Types of Substrates Used in LEED Studies

Low Miller index surfaces of metallic single crystals are the most commonly used substrates in LEED investigations. The reasons for the widespread use of the low index surfaces are that they have the lowest surface free energy and therefore are the most stable, have the highest rotational symmetry, and are the most densely packed. Also, in the case of transition metals and semiconductors they are chemically less reactive than the

TABLE I. Miller Indices, Stepped Surface Designations, and Angles between the Macroscopic Surface and Terrace Planes for fee Crystals

higher Miller index crystal faces.

The metal substrates used in the LEED experiments have either face-centered cubic (fee), body-centered cubic (bcc), or hexagonal closed-packed (hep) crystal structures. For the cubic metals the (111), (100), and (110) planes are the low Miller index surfaces, and they have threefold, fourfold, and twofold rotational symmetry, respectively. The top layer of a (111) surface actually has sixfold symmetry, but the rotational symmetry of the top three layers together is threefold. Since the near surface region can influence where gases adsorb on the surface and the LEED l-V curves exhibit threefold rotational symmetry at normal incidence, the (111) surface will be considered to have threefold rotational symmetry. Although most of the adsorption studies have been carried out on fcc and bcc crystals, there have also been several studies reported on hep crystals. For hep metals the basal or (0001) plane is the surface most frequently studied by LEED investigations, and it is the most densely packed plane having threefold rotational symmetry. Again the top layer of the (0001) surface has sixfold rotational symmetry, but the rotational symmetry of the top two layers together is threefold.

In the last few years LEED studies of high Miller index or

C. Pt (S)-[7(III) $x(310)$] \leftrightarrow Pt(10,8,7)

Figure 8. LEED patterns and real space drawings of the (a) low index, (b) stepped, and (c) kinked platinum surfaces.

stepped surfaces have become more frequent. Some of the first LEED studies of stepped surfaces were carried out on $UO₂$ crystals by Ellis and Schwoebel⁴ in 1968, but only recently have there been many studies of gas adsorption on stepped surfaces. Almost all of the high Miller index surface studies have been on fcc metals, where the atomic structure of these surfaces consists of periodic arrays of terraces and steps. A nomenclature which is more descriptive of the actual surface configuration has been developed for these surfaces.⁵ Using this nomenclature a Pt(755) surface would be designated as a Pt(S)-[6(111)X(100)j surface which indicates it is a stepped surface consisting of six-atom-wide (111) terraces separated by one-atom-high (100) steps. In Table I the stepped surface nomenclature for several high Miller index surfaces of fee crystals has been tabulated. In Figure 7 the locations of these high Miller index surfaces are shown on the unit stereographic triangle. As can be seen from that figure all the stepped surfaces which have low Miller index type steps lie on the [100], [110], and [111] zone lines. For surfaces which lie inside the unit stereographic triangle the steps themselves have steps, and this type of surface is classified as a kinked surface. The only kinked surface for which surface structures have been reported is the $Pt(10,8,7)$ or $Pt(S)$ - $[7(111) \times (310)]$ surface. The real space drawings and LEED patterns of the platinum (111), (755), and (10,8,7) surfaces are shown in Figure 8.

In calculating the stepped surface designations that are listed in Table I, it was assumed that the surfaces were stable in a monatomic step configuration, which is generally the case for the clean surfaces.⁶ This can readily be verified by LEED. In LEED patterns of stepped surfaces the step periodicity is superimposed on the terrace periodicity resulting in the splitting of the terrace diffraction spots into doublets or triplets at certain beam voltages. This splitting can be seen in the LEED pattern in Figure 8. The direction of the splitting is perpendicular to the step edge and the magnitude of the splitting is inversely proportional to the terrace width, so the terrace width can be obtained by measuring the splitting observed in LEED pattern. The step height can be determined from the formula

$$
V_{00} \text{ (singlet max)} = (150/4d^2)s^2 \tag{8}
$$

where V_{00} are the voltages where a singlet intensity maximum of the $(0,0)$ beam is observed, d is the step height, and s is an

 \bar{z}

TABLE II. Surface Structures on Substrates **with** Threefold Rotational Symmetry

TABLE Il (Continued)

 $\mathcal{A}^{\mathcal{A}}$

 \sim

 $\mathcal{A}^{\frac{1}{2}}_{\mathbf{q}_k}$

TABLE Il (Continued)

J,

TABLE III. Surface Structures on Substrates with Fourfold Rotational Symmetry

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TABLE III (Continued)

(2X4)-CO

10

TABLE III (Continued)

 $\hat{\mathcal{A}}$

TABLE III (Continued)

TABLE IV. Surface Structures on Substrates with Twofold Rotational Symmetry

TABLE IV (Continued)

integer. By combining the terrace width and step height with the angle between the terrace and step planes, the macroscopic surface plane can be determined. This calculated macroscopic surface can then be compared with back reflection Laue X-ray diffraction results for the same surface.

The stepped surface nomenclature has mainly been used for fcc metals, but there are a few studies where it has been applied to bcc and hep metals. In these investigations the surfaces had fairly low step densities so the macroscopic surface was within a few degrees of a low index face.

TABLE V. Surface Structures on Stepped Substrates

The stability of stepped surfaces is an important consideration in LEED studies. Although these surfaces have higher surface free energies than the low index faces, most of the clean stepped surfaces are stable in a single step height configuration from room temperature to the melting point of the metals.⁶ When gases are adsorbed on these surfaces, however, their stability can noticeably change. Some surfaces reconstruct, forming multiple height steps and large terraces. Other high index surfaces form large low index facets while some retain the single step height configuration. In one study several high Miller index surfaces of platinum have been investigated to determine their stability when clean, oxygen covered, and carbon covered.⁶

There have been several LEED studies on semiconductor surfaces such as Ge and Si. For clean semiconductor surfaces reconstruction is the rule rather than the exception. In fact, on some surfaces such as Si(111) by changing the temperature different reconstructed surfaces are observed.⁷ A few studies have been carried out on metal oxide substrates. There have been studies on the epitaxial growth of metal oxides on metal single crystals. This is generally carried out by exposing the metal crystal to oxygen at a temperature sufficiently high enough to promote oxide growth. There has been a LEED study of Xe adsorbed on NaCI, but because of severe electron beam damage on the alkali halide substrates, very little work has been done using ionic crystal surfaces.

V. Surface Structures of Adsorbates

Tables H-Vl list the surface structures observed in LEED experiments when various gasses are adsorbed on surfaces. Because of a large increase in the number of studies of organic molecule adsorption and adsorption studies on stepped surfaces, separate tables for these two categories have been made. Tables H-Vl contain over 800 surface structures.

The surface structures of gases adsorbed on substrates of threefold, fourfold, and twofold rotational symmetry are listed in Tables II, III, and IV, respectively. Most of the substrates in these tables are low index faces and the gases adsorbed are, for the most part, small inorganic molecules such as H_2 , O_2 , N_2 , CO, and NO. Inspection of the tables permits one to propose two general rules that are usually obeyed during the adsorption of these small molecules: (1) the observed surface structures have the same rotational symmetry as the substrate, and (2) the unit cell of the surface structure is the smallest allowed by the molecular dimensions and adsorbate-adsorbate interactions. There is a definite trend for an adsorbate to form identical surface structures on the same crystal face of several different substrates. For example, oxygen forms a (2×2) structure on the hexagonal faces of Ag, Cu, Ir, Nb, Ni, Pd, Pt, Re, Rh, and Ru. CO initially forms a ($\sqrt{3}$ X $\sqrt{3}$)R30° surface structure on most hexagonal metal surfaces and then compresses into a close-

TABLE Vl. Surface Structures Formed by Adsorption of Organic Compounds

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TABLE Vl (Continued)

 $\mathcal{A}^{\mathcal{A}}$

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TABLE Vl (Continued)

packed hexagonal overlayer of CO at high coverage. This hexagonal overlayer of CO has also been observed on some surfaces having fourfold rotational symmetry. Several adsorption studies of the inert gases Xe and Kr have been reported. These studies were carried out at low temperatures since the interaction between these gases and surfaces is weak. In all cases a hexagonal overlayer of the inert gas was formed. This hexagonal overlayer has been analyzed for Xe on Ag(111) and found to correspond to the (111) plane of the fcc inert gas solid.^{8,9}

The surface structures observed for gas adsorption on stepped surfaces are listed in Table V. In this table the stepped surfaces are denoted by either their Miller index label or stepped surface designation, depending on which system was used by the original author. By using Table I one may convert back and forth between these two systems. Most of the studies in this table have been performed in the last few years. It is interesting to compare the surface structures formed on stepped surfaces with those formed on the low index faces given in Tables H-IV. For stepped surfaces with fairly large terrace widths (\sim six to eight atoms or larger), the surface structure that forms on the terrace is generally the same one that forms on the low index face. The surface structures on the low index surfaces tend to be more well ordered than those on the stepped surfaces. An example of this is the existence of several one-dimensional structures on stepped surfaces. The one-dimensional structures cause streaks to occur in the LEED patterns and are denoted as $n-(1d)$ structures in the tables, with $n - 1$ is the number of streaks between rows of the substrate diffraction spots. Also, the adsorption of gases may cause faceting of the substrate owing to the high surface free energy of stepped surfaces.

The surface structures formed upon adsorption of organic molecules on several substrates are listed in Table Vl. This table includes both chemisorption and physisorption structures since many of the organic molecules only adsorb at low temperatures. Ethylene and acetylene are the two most widely studied organic molecules. There has also been an l-V analysis of the (2 $X2$) surface structures formed by these two molecules on Pt(111).¹⁰⁻¹² As can be seen from Table VI, adsorption on platinum surfaces has been the most widely studied. For some larger organic molecules both monolayer and multilayer structures have been reported. The multilayer structures formed by phthalocyanines and amino acids on copper^{13,14} grow at room temperature, while the *n*-paraffins on $Pt(111)^{15}$ only form multilayer structures at low temperatures. As seen from Table Vl there have also been several adsorption studies of organic molecules on stepped surfaces. More LEED studies of organic adsorbates on a larger variety of substrates are needed before correlations or trends among their surface structures can be made.

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