Synthesis of Boron Nitride via a Polymeric Vinylpentaborane Precursor

Sir:

In a recent communication,¹ we showed that 2-vinylpentaborane could undergo a thermally induced polymerization reaction to form either soluble or cross-linked poly(2-vinylpentaborane) oligomers. Furthermore, we demonstrated that these materials, when heated under an argon atmosphere, produced pure boron carbide (B_4C) in high ceramic yield. We have now found that when poly(alkenylpentaboranes) are pyrolyzed in the presence of a suitable reactant, they can serve as precursors to other important nonoxide ceramics. As an illustration, we report here the high-yield synthesis of boron nitride from the soluble oligomer poly(2-vinylpentaborane) (I).

There have been several previous reports of the use of polymer precursor routes as alternatives for the preparation of boron nitride (BN),²⁻⁷ and these studies have resulted in the production of ceramic materials in a range of yields and purities. Recently, several studies have shown that the pyrolysis of certain boron polymers in the presence of ammonia can lead to enhanced yields and purity of boron nitride products. For example, Seyferth and Rees⁸ recently reported that ammonia pyrolysis of diamminelinked decaborane polymers produced crystalline BN with reduced carbon contamination. Likewise, Ultrasystems⁹ has described the synthesis of BN from ammonia pyrolysis of soluble polyborazine compounds.

Pentaborane(9) is known to react readily with ammonia and alkylated amines to form a variety of products.¹⁰ Thus, depending on reaction conditions, simple deprotonation $[(NH_4^+)(B_5H_8^-)]$,¹¹ formation of a diammoniate $[(H_2B(NH_3)_2^+)(B_4H_7^-)]$,¹¹ or production of borazines may occur.¹² We have now found that I also readily reacts with ammonia at moderate temperatures to produce an intermediate that, when heated at higher temperatures, gives boron nitride in high yields.

The conversion of polymer I to boron nitride was accomplished by slowly heating (2 °C/min) a 470-mg sample of I from 25 to 350 °C under a flow of gaseous ammonia (~100 mL/min), followed by faster heating at the rate of 10 °C/min until a maximum temperature of 1000 °C was achieved. This temperature was then maintained for an additional 2 h, after which it was found that 472 mg of a light yellow solid was produced. Elemental analysis¹³ of the material showed a B/N ratio of 1.05 with a measured level of carbon of 0.81%. The diffuse-reflectance IR spectrum¹⁴ is consistent with spectra previously reported for

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- (13)
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 (13) Anal. Calcd for BN: B, 43.55; N, 56.45; Found BN prepared by pyrolysis of I: B, 45.24; N, 55.88; C, 0.81; H, 0.46.
- pyrolysis of 1: B, 45.24; N, 55.05, C, 0.65, A, 67.05 IR (diffuse): 3640 m, 3420 s, 2800 w, 2560 m, 2320 w, 1440 vs, vbr, (14) 1100 s, 940 w, 800 vs, 680 s, 620 s cm⁻¹.

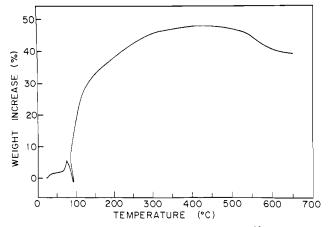


Figure 1. TGA analysis of polymer I under ammonia.¹⁶

boron nitride.¹⁵ The material produced under these conditions was largely amorphous; however, crystalline BN (as determined by X-ray powder diffraction) was obtained when heating was continued for longer periods at 1450 °C.

Boron nitride prepared from I in the manner described above was obtained in 100.4% ceramic yield (based on a theoretical ceramic yield²¹ of 139.0%) and a chemical yield of 72.2% based on the following equation:

$$\begin{array}{c|c} & H & H \\ \hline 1 & 1 \\ \hline 1 & 1 \\ \hline 1 & 1 \\ \hline 3 & H_{2} \\ \hline 1 \\ \hline 3 & H_{3} \\ \hline \end{array} + 5 \text{ NH}_{3} \xrightarrow{\text{ NH}_{3}} 5 \text{ BN } + 2 \text{ CH}_{4} + 9 \text{ H}_{2}$$
(1)

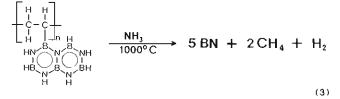
Thus, the addition of 5 equiv of NH₃ to I, followed by formation of boron nitride at higher temperatures, should result in a theoretical weight increase of 39%. This prediction is in excellent agreement with the thermogravimetric analysis¹⁶ of I under an ammonia atmosphere (Figure 1), which showed a sharp weight increase in the range 75-400 °C (\sim 48%) followed by a gradual weight loss from 475 to 650 °C ($\sim 10\%$).

In an effort to isolate the intermediate compound at the point of maximum weight gain, a 480-mg sample of I was placed in a tube furnace under a flow of gaseous ammonia ($\sim 100 \text{ mL/min}$). The sample was initially heated to 75 °C and then slowly heated (2 °C/min) through the temperature range 75-350 °C. Under these conditions a white, brittle material (640 mg) II was produced that was insoluble in a variety of polar and nonpolar organic solvents. A diffuse-reflectance IR spectrum¹⁷ revealed the presence of N-H, C-H, B-H, and B-N bonds, while elemental analysis¹⁸ is consistent with an empirical formula of $C_{2.0}B_{5.1}N_{4.8}H_{8.0}.\,$ Since the B/C ratio in this material is identical with that of starting polymer I, the polymeric hydrocarbon backbone has probably remained intact. A UV-visible spectrum of the material showed a broad absorption with a maximum at 212 nm, similar to those that have been shown to arise from the $\pi - \pi^*$ electronic transition in borazine and fused-borazine¹⁹ polymers. This observation, coupled with the measured B/N ratio of ~ 1 in II, suggests that the pendant pentaborane cages in I have reacted with ammonia to form borazine-type ring systems in II. Indeed, the analytical data are in excellent agreement with the formation of a borazanaphthalene unit,¹⁹ as shown in eq 2.

These results are thus consistent with the formation of BN from I in a two-step process. The initial step involves the formation

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- (16) TGA results were obtained by heating a sample of polymer I under ammonia from room temperature to 700 °C with a heating rate of 10 °C/min.
- (17) IR (diffuse): 3420 vs, 3160 s, 2940 s, 2900 s, 2850 s, 2500 m, 1600 m, 1400 s, 1110 m, 700 m, 610 m cm⁻
- Elemental analysis of polymer II: B, 33.15; N, 41.03; C, 14.70; H, 4.93. (18)
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of a borazine polymer intermediate (eq 2), followed by a second decomposition reaction at higher temperatures to form BN as shown in eq 3. Indeed, it was also found that further treatment



of a pure sample of II under ammonia at higher temperatures resulted in formation of BN

In summary, the results described herein demonstrate that poly(2-vinylpentaborane) oligomers are useful soluble ceramic precursors that can be readily converted to boron nitride in high yields. The potential applications for the processing of BN in this manner are numerous and include synthesis of composite materials²⁰ and the generation of thin films or coatings.²¹ Furthermore, owing to the highly reactive nature of the pentaborane cage, these polymers may well serve as precursors to a wide variety of other important boron-containing ceramics and work exploring these possibilities is currently in progress.

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Synthesis of Thiomolybdenyl Complexes with $[Mo_2(S)_2(O)_2]^{2+}$ Cores and Substitutionally Labile Ligands. Crystal and Molecular Structure of the $[Mo_2O_2S_4(DMF)_3]$ Complex

Sir

The reactivity characteristics of the Mo-coordinated S²⁻, S₂²⁻. and S_4^{2-} terminal ligands (L²⁻) in the $[L_2Mo_2(\mu-S)_2S_nO_{2-n}]^{2-}$ complexes (n = 0-2) are greatly affected by the nature of the neighboring terminal ligands on the Mo atom.¹ A detailed study of neighbor-ligand effects in the relative nucleophilic properties of coordinated S^{2-} , S_2^{2-} , or S_4^{2-} ligands is expected to be useful in the evaluation of specific features that may be important in the catalysis of the hydrodesulfurization (HDS) reaction.² Among

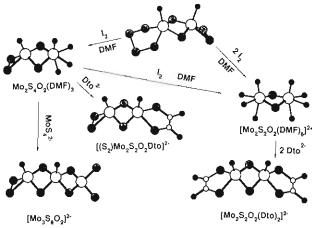


Figure 1. Synthesis and derivatives of the [Mo₂O₂S₄(DMF)₃] and $[Mo_2O_2S_2(DMF)_6]^{2+}$ complexes.

these features are included (a) the presence of proximal terminal oxo groups on the Mo atoms and (b) the interactions of the Mo atoms with oxides or hydroxides on the surface of the supporting y-Al₂O₃ matrix.

The systematic synthesis of thiomolybdates with specific functional groups, and reactivity studies of such groups depend on the facile synthesis of simple derivatives of the [Mo₂(µ- $S_{2}S_{n}O_{2-n}]^{2+}$ cores, with substitution-labile terminal ligands. In the past, oxidative desulfurization reactions have been effective in the removal of coordinated S_x^{2-} ligands and their replacement by substitutionally labile ligands. Such reactions are exemplified in the synthesis of the [Cp₂Fe₂(CH₃CN)₂(SEt)₂]²⁺ and Os(I)₂- $(CO)_2(PPh_3)_2$ complexes. The former was obtained³ by air oxidation of $[Cp_2Fe_2(S_2)(SEt)_2]^{0,+}$ in CH₃CN solution and the latter⁴ by the I₂ oxidation of the $Os(\eta^2-S_2)(CO)_2(PPh_3)_2$ complex in CH₂Cl₂.

The reactions of $(Et_4N)_2[Mo_2O_2S_8]^5$ (I) with I₂ in dimethylformamide (DMF) are shown in Figure 1. The reaction of I with an equimolar amount of I2 proceeds readily at ambient temperature and following unexceptional workup affords the orange crystalline $[Mo_2O_2S_4(DMF)_3]$ product⁶ (II) in 70% yield. A similar reaction of I with 2 equiv of I_2 or the reaction of II with an equimolar amount of I₂ results in the formation of the $[Mo_2O_2S_2(DMF)_6]^{2+}$ cation in high yields. The latter can be isolated in crystalline form as the diiodide salt III.6 The infrared spectra of the complexes show the Mo-O vibrations at 948 and 954 cm⁻¹ for II and at 928 and 947 cm⁻¹ for III. The C-O stretching vibrations of the coordinated DMF molecules appear as doublets at 1641, 1657 and 1647, 1664 cm⁻¹ for II and III, respectively. The Mo- η^2 -S₂ and Mo-S_b vibrations in II are found at 527 and 469 cm⁻¹, and the Mo–S_b stretching vibration in III occurs at 474 cm⁻¹. The conductivity of III in DMF solution (Λ

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- crystalline product was 9.6 g or 45%.
 (6) Anal. Calcd for Mo₂S₄O₃N₃C₉H₂₁: C, 18.92; N, 7.35; H, 3.70; S, 22.44; Mo, 33.58. Found: C, 19.55; N, 7.43; H, 3.70; S, 21.70; Mo, 32.37. Calcd for I₂Mo₂S₂O₈N₆C₁₈H₄₂: C, 22.00; N, 8.57; H, 4.28; S, 6.53; Mo, 19.59; I, 25.92. Found: C, 22.00; N, 8.14; H, 4.53; S, 7.22; Mo, 17.75; I, 24.53. Severe twinning of the well-shaped orange crystals of this compound has prevented us from finding a single crystal suitable for an X-ray determination.

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