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Polytypism in Columnar Group 14 Halide Salts: Structures of $(Et_2NH_2)_3Pb_3X_9 \cdot nH_2O$ (X = CI, Br) and (β -alaninium)₂Snl₄

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The crystal structures of three hybrid organoammonium metal halide salts composed of edge-sharing MX₆ octahedra have been determined. The genesis of these structures can be traced to the parent hexagonal MX₂ structure via dimensional reduction and recombination arguments. The structures of $(Et_2NH_2)_3Pb_3X_9 \cdot nH_2O$ (X = Br, I) contain unique columnar $(Pb_3X_9)_n^{3n-}$ structures, built up of edge-shared PbX₆ octahedra. The interaction of the $Et_2NH_2^+$ cations with the parent PbX₂ structures leads to a rearrangement of the lattice into the observed columnar structure. Groups of six $Et_2NH_2^+$ cations are hydrogen bonded to these columns, girdling them at their narrowest points. These hydrogen bonds contribute to the formation of the zigzag nature of the columnar inorganic framework. The resultant structures are recombinate analogues (polytypes) of the $(Pb_3X_9)_n^{3n-}$ stacks that would be obtained by the dimensional reduction process of the parent layer PbX₂ structure into simple edge-shared ribbons of PbX₆ octahedra. These structures can be described in terms of the stacking of planar bibridged Pb₃X₈²⁻ units decorated with a single halide ion at a terminal lead ion site. In a similar fashion, (β -alaH)₂Sn₂I₆ contains corrugated (Sn₂I₆)_n²ⁿ⁻ columns (β -ala = β -alanine), with the cations sitting in the clefts of the columns.

Introduction

One of the crystal engineering strategies for the design of low-dimensional inorganic materials involves the dimensional reduction of higher dimensional parent structures. Tulsky and Long have applied this concept to describe the types of low-dimensional systems that can be obtained from MX_n structures via reaction with anionic AX_m reagents.¹ However, this strategy is not restricted to these basic systems. For example, from the parent AMX_3 perovskite system,² this concept can be used to rationalize the synthesis of the layer structures found in the high- T_c copper oxide based superconductors.³ This concept can be extended to hybrid organic/ inorganic materials when A^{m+} is an organic cation. Here again, the dimensional reduction of the AMX_3 perovskite structure, where X^- is a halide ion, has yielded many interesting layered (RNH₃)₂MX₄ magnetic and semiconducting systems.⁴ The resultant two-dimensional framework of corner-shared MX_6 octahedra are stabilized by hydrogen bonding between the $-NH_3^+$ moieties of the organoammonium ions and the nonbridging halide ions of the inorganic sheets. This produces a bilayer type structure in which the central inorganic framework is sheathed by the organic R groups.⁵ In general, unfortunately, the multiple challenges of diversity of hydrogen-bonding capabilities, ligand-bridging capabilities, and organic steric effects make the prediction of such dimensional reduction schemes uncertain.

The layered hexagonal MX₂ structures, consisting of layers of edge-shared octahedra, can also be utilized as a parent template for the design of novel low-dimensional systems.⁶ Tulsky and Long have also documented the possible subclasses that are observed from reaction with inorganic AX species.¹ These include perforated layers in which M²⁺

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Chart 1



cations are removed from the layers to compensate for the introduction of A⁺ cations⁷ between the layers as well as single and double chains of corner-shared octahedra.⁸ We have utilized this concept extensively in our studies of structural and magnetic properties of hybrid organic/inorganic salts of copper(II) halide salts.⁹ With the Cu(II) ion, the Jahn–Teller effect leads to a ferrodistortive version of the layered MX₂ structure that can be envisioned as stacks of planar bibridged (CuX₂)_nⁿ linear chains (Chart 1).¹⁰ This resultant structure is particularly susceptible to crystal engineering manipulation. Reaction with organic AX salts can be used to further reduce these chains into smaller planar bibridged Cu_nX_{2n+2}^{2–} oligomers (n = 1-7)¹¹ as well as to perforate the layers.¹²

Recombination¹³ of the stacking arrangements of these planar oligomers (via the longer semicoordinate Cu···X bonds induced by the Jahn-Teller effect) leads to a wide variety of polytypes. The structure of two such polytypes for planar $Cu_4X_{10}^{2-}$ anions are shown in Figure 1. The stacking shown in Figure 1a for (Me₄N)₂Cu₄Cl₁₀ corresponds to a simple ribbon cut from the parent CuX₂ structure.^{11a} In contrast, the stacking of the tetramers shown in Figure 1b for (Me₃NH)₂Cu₄Br₁₀ represents a polytype in which recombination of the planar oligomers has occurred.^{11b} A convenient diagrammatic representation of the stacking patterns observed for these polytypes has been developed, as well as a shorthand notation to specify the repeat pattern in the stacks. Chart 2 illustrates these stacking diagrams for the two polytypes shown in Figure 1. Also given is their designated Geiser notation, $n(t_{\parallel}, t_{\perp})(t'_{\parallel}, t'_{\perp})$.^{11c-g} Here n

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Figure 1. Examples of polytypic stacking patterns observed in copper(II) halides: (a) 4(3/2, 1/2) stack in (Me₄N)₂Cu₄Cl₁₀; (b) 4(3/2, 1/2)(1/2, -1/2) stack in (Me₄P)₂Cu₄Br₁₀.

Chart 2





denotes the length of the oligomers, t_{\parallel} denotes the translation parallel to the long axis, and t_{\perp} denotes the relative translation of adjacent anions perpendicular to the long axis, with the $(t_{\parallel}, t_{\perp})$ symbol repeated as often as necessary to describe the

 $(t_{||}, t_{\perp})$ symbol repeated as often as necessary to describe the repeat pattern in the stack. Stacking patterns with different values of $t_{||}$ and t_{\perp} represent different polytypes. The simple 4(3/2, 1/2) stack in Chart 2a represents a simple dimensional reduction of the layers in the parent CuX₂ structures. The more complex 4(3/2, 1/2)(1/2, -1/2) stack in Chart 2b involves further reduction into oligomeric units followed by reconstruction of the semicoordinate interoligomer Cu···X bonds.

The study of recombination in columnar stacks for hybrid organic/inorganic systems based on the edge-sharing arrangement of octahedra of the CdX₂ type has not been as extensive. In the absence of the Jahn–Teller distortion there is no strong directional preference for the dimensional reduction of the parent structure. As documented by Tulsky and Long,¹ the vast majority of the 1-D systems obtained for pure inorganic systems are single or double chains of edge-shared octahedra (Chart 3). For the double chains, planar bibridged M_2X_6 units can be identified and the Geiser

Chart 3



notation can be applied to this type of system, yielding a 2(1/2,1/2) stack designation.¹⁴ Moreover, a search of the Cambridge Database for hybrid organic/organic structures with extended edge-shared octahedral networks revealed only a limited number examples and none involved recombinant structures. However, no examples of polytypes of this structure appear to exist in the simple inorganic systems.

In this paper, three new and novel hybrid organic/inorganic columnar structures are reported. The first of these are for the isostructural (DEA)₃Pb₃Br₉ and (DEA)₃Pb₃I₉.¹/₂H₂O compounds, where DEA is diethylammonium. For several years, we have had interest in the structural properties of the metal halide salts of the diethylammonium cation. Many of these salts contain MX₄²⁻ anions and exhibit interesting structural phase transitions.^{15a-c} A few compounds have previously been found to contain extended inorganic frameworks. The compound (DEA)₂Cu₄Br₁₀·EtOH salt contains chains of edge-shared tetrahedra in addition to stacks of $Cu_3Br_8^{2-}$ anions with a 3(1/2, 1/2)(-1/2, -1/2) stacking pattern.^{15d} In contrast, the structure of (DEA)₃Cd₂Br₇ contains chains of face-shared CdBr₆ octahedra plus isolated CdBr₄²⁻ ions.^{15e} In addition, we report on the structure of $(\beta$ -alaH)₂Sn₂I₆, which also contains a recombinatorial columnar structure. This compound arose as part of our efforts to understand the range of organoammonium cations that can be incorporated in the Sn(II) and Pb(II) halide layer perovskite structures.¹⁶ These represent the first examples of polytypic structures based on dimensional reduction and recombination of the hexagonal MX₂ structure type.

Experimental Section

Synthesis. The lead(II) bromide salt was prepared by reacting (DEA)Br and PbBr₂ in a 2:1 ratio in a deficiency of H₂O at \sim 70 °C in a sealed test tube. The solution was allowed to cool, and at

(16) For a general review, see: Mitzi, D. B. Prog. Inorg. Chem. 1999, 48, 1.

a later date, crystals were removed from the tube. A small colorless crystal ($0.3 \times 0.3 \times 0.1$ mm) was selected for the X-ray diffraction study.

The corresponding lead(II) iodide salt was prepared by the reaction of (DEA)Cl and PbI₂ in a 1:1 ratio in a concentrated (6 M) HI solution. The resulting solution was slowly evaporated until crystals formed. The crystals were separated from the mother liquor, and a crystal of dimensions $0.14 \times 0.07 \times 0.03$ mm was selected for the diffraction study.

Crystals of (β -alaninium)₂SnI₄ [(β -alaH)₂SnI₄] were grown from a slowly cooled 2-butanol/hydriodic acid solution containing 0.323 g (3.6 × 10⁻³ mol) of β -alanine and 0.671 g (1.8 × 10⁻³ mol) of SnI₂ (Aldrich, anhydrous beads, 99.999%). These were placed in a test tube under an inert atmosphere, and 10 mL of concentrated (57 wt %) aqueous hydriodic acid was added with a syringe. The components in the tube were thoroughly mixed and then heated to 100 °C. The solution was cooled at 4.4 °C/h to -10 °C, yielding 0.86 g (86% theoretical yield) of yellow/orange, rod-shaped crystals. The crystals were filtered in an inert atmosphere. Chemical analysis of (β -alaH)₂SnI₄ yielded the following. Calcd: C, 16.99; H, 3.4; N, 3.47. Found: C, 16.67; H, 2.34; N, 3.18.

X-ray Diffraction. Data for all three compounds were collected on a Bruker 3-circle platform diffractometer equipped with a SMART 1K CCD detector. The frame data were acquired with the SMART¹⁷ software at 295 K using Mo K α radiation ($\lambda = 0.71073$ Å). The frames were then processed using the SAINT software¹⁸ to give the hkl file corrected for Lp/decay. The absorption correction was performed using the SADABS¹⁹ program. The structures were solved by the direct method using the SHELX-90²⁰ program and refined by least-squares method on F^2 , SHELXL-93,²¹ incorporated in SHELXTL V 5.03.22 Hydrogen atoms were included in calculated positions. All non-hydrogen atoms were refined anisotropically. The lead(II) bromide and iodide salts were isostructural; however, the room-temperature data set for the iodide salt clearly showed disorder of the ethyl groups on the cations. Hence, a second data set was collected at -70 °C. This also showed disorder of the ethyl groups. Consequently, the cations were thus refined with the C-N and C-C distances loosely constrained. In addition, the difference electron density maps revealed the presence of a partially occupied water molecule site nearly equidistant from I(4), I(7), I(9), and I(11). This was included with an assumed occupancy of 0.5. Data collection and refinement parameters are given in Table 1, while Tables 2-4give selected bond distances and angles.

Results

Lead(II) Halide Salts. Figure 2 shows the contents of the asymmetric unit of the bromide salt, with the addition of symmetry-related Br^- ions to complete the octahedral coordination sphere of the three Pb^{2+} ions. The iodide salt has an identical lead/halide core. The coordination spheres for both Pb(1) and Pb(3) show significant variations in the Pb-X bond lengths, with those to X(7) and X(11) being

⁽¹⁴⁾ Because of the lack of the presence of a ferrodistortive distortion in these structures, two possible orientations for the planar bibridged M_2X_6 units can be identified. Nevertheless, in more complex structures, it is frequently possible to identify an optimal orientation to describe the stacking.

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Table 1. Crystal Data and Structure Refinement Parameters

empirical formula	C12H36N3Pb3Br9	C ₁₂ H ₃₇ N ₃ O _{.5} Pb ₃ I ₉	C ₆ H ₁₆ N ₂ O ₄ Sn ₂ I ₆
fw	1563.20	1986.11	1178.99
$T(\mathbf{K})$	293(2)	183(2)	81(2)
space group	$P2_{1}/c$	$P2_{1}/c$	$P2_1/c$
a (Å)	10.753(6)	11.281(7)	20.604(3)
b (Å)	23.976(11)	24.917(15)	13.146(2)
c (Å)	14.239(6)	14.371(8)	8.482(1)
β (deg)	111.50(5)	110.92(1)	95.75(3)
$V(Å^3)$	3416(3)	3774(4)	2286.0(5)
Z	4	4	4
λ (Å)	0.71073	0.71073	0.71073
D_{calc} (Mg/m ³)	3.040	3.496	3.426
$\mu ({\rm mm}^{-1})$	25.284	20.722	10.297
\mathbf{R}_1^a	0.0458	0.0354	0.0334
WR_2^b	0.1085	0.0622	0.0730

 ${}^{a} \mathbf{R}_{1} = \Sigma ||F_{o}| - |F_{c}|| / \Sigma |F_{o}|^{2}. {}^{b} \mathbf{w} \mathbf{R}_{2} = \{ \Sigma [w(F_{o}^{2} - F_{c}^{2})^{2} / \Sigma [w(F_{c}^{2})^{2}] \}^{1/2}.$

Table 2. Bond Distances (Å) and Angles (deg) for (DEA)₃Pb₃Br₉^a

Distances						
Pb(1)-Br(9)	2.807(2)	Pb(2)-Br(7)#2	3.038(2)			
Pb(1)-Br(6)	2.912(2)	Pb(2)-Br(7)	3.090(2)			
Pb(1)-Br(10)	3.020(2)	Pb(2)-Br(11)	3.178(2)			
Pb(1)-Br(5)	3.085(2)	Pb(3)-Br(8)	2.787(2)			
Pb(1)-Br(11)#1	3.178(2)	Pb(3)-Br(12)	2.822(2)			
Pb(1)-Br(11)	3.229(2)	Pb(3)-Br(6)#2	3.049(2)			
Pb(2)-Br(4)	2.877()	Pb(3)-Br(4)	3.054(2)			
Pb(2)-Br(10)#1	2.958(2)	Pb(3)-Br(7)#2	3.272(2)			
Pb(2)-Br(5)	3.033(2)	Pb(3)-Br(11)#2	3.302			
	1.00	100				
$D_{n}(0) D_{n}(1) D_{n}(6)$	Allg	$B_{\pi}(10) \# 1 = B_{\pi}(2) = B_{\pi}(7)$	174 21(4)			
BI(9) = PD(1) = BI(0) Br(0) = Pb(1) = Br(10)	92.70(7)	Br(10)#1=PO(2)=Br(7) Br(5)=Br(2)=Br(7)	1/4.51(4)			
BI(9) = PD(1) = BI(10) Br(6) = Pb(1) = Br(10)	93.73(0)	BI(3) = PU(2) = BI(7)	90.03(7)			
BI(0) = PD(1) = BI(10) Br(0) = Pb(1) = Br(5)	95.15(0)	Br(4) = Pb(2) = Br(11)	03.70(7)			
BI(9) = PD(1) = BI(3) Br(6) = Pb(1) = Br(5)	91.71(0)	DI(4) = PU(2) = DI(11) Dr(10) #1 = Dh(2) = Dr(11)	1/0.10(3)			
BI(0) = PD(1) = BI(3) Br(10) = Bh(1) = Br(5)	91.03(0)	Br(10)#1=PO(2)=Br(11) Br(5)=Br(2)=Br(11)	80.13(3)			
BI(10) = PU(1) = BI(3) Br(0) = Pb(1) = Br(11) # 1	170.95(3)	BI(3) = PU(2) = BI(11) $B_{\pi}(7) = B_{\pi}(11)$	00.14(<i>J</i>)			
Br(9) - Pb(1) - Br(11) # 1	93.51(0)	Br(7) = PD(2) = Br(11)	87.05(5)			
Br(0) = PD(1) = Br(11)#1	1/3.09(4)	Br(7) = PD(2) = Br(11) Dr(8) = Dr(2) = Dr(12)	88.01(5)			
Br(10) - Pb(1) - Br(11)#1	85.10(5)	Br(8) - Pb(3) - Br(12)	93.90(6)			
Br(5) - Pb(1) - Br(11) # 1	89.31(5)	Br(8) - Pb(3) - Br(6) # 2	88.84(7)			
Br(9) - Pb(1) - Br(11)	1/6.39(5)	Br(12) - Pb(3) - Br(6) # 2	92.68(7)			
Br(6) - Pb(1) - Br(11)	90.39(6)	Br(8) - Pb(3) - Br(4)	93.33(7)			
Br(10) - Pb(1) - Br(11)	85.95(6)	Br(12) - Pb(3) - Br(4)	93.83(7)			
Br(5) - Pb(1) - Br(11)	86.32(5)	Br(6)#2 - Pb(3) - Br(4)	1/2.97(5)			
Br(11)#1-Pb(1)-Br(11)	83.45(6)	Pb(2) - Br(4) - Pb(3)	97.58(6)			
Br(4) - Pb(2) - Br(10)#1	91.77(6)	Pb(2) - Br(5) - Pb(1)	95.54(5)			
Br(4) - Pb(2) - Br(5)	95.16(6)	Pb(1) - Br(6) - Pb(3)#2	96.84(7)			
Br(10)#1-Pb(2)-Br(5)	92.00(7)	Pb(2)#2-Br(7)-Pb(2)	96.22(7)			
Br(4) - Pb(2) - Br(7) #2	89.87(6)	Pb(2)#1-Br(10)-Pb(1)	98.04(6)			
Br(10)#1-Pb(2)-Br(7)#	2 93.74(7)	Pb(2) - Br(11) - Pb(1)#1	90.47(5)			
Br(5) - Pb(2) - Br(7)#2	172.25(5)	Pb(2) - Br(11) - Pb(1)	89.99(5)			
Br(4) - Pb(2) - Br(7)	93.33(6)	Pb(1)#1-Br(11)-Pb(1)	96.55(6)			

^{*a*} Symmetry transformations used to generate equivalent atoms: #1, -x, -y, -z + 2; #2, -x + 1, -y, -z + 2.

the longest. This is probably due to some degree of localization of the lone pairs on the Pb(II) ions into that region of space. The central feature of this core is the existence of groups of three edge-shared octahedra that defines a linear bibridged $(Pb_3X_8)^{2-}$ unit that can be used to describe the construction of the columnar structure in these two compounds. The framework of the two isostructural compounds consist of corrugated columns of $(Pb_3X_9)_n^{3-}$ stacks that run parallel to the *a* axis. This is illustrated in Figure 3 for $X = I^-$. The stacks are built up of edge-shared PbX₆ octahedra linked together in a kinked zigzag fashion. The complex structure of the stacks is intimately associated with the role played by the hydrogen-bonding interactions between the DEA⁺ cations and the halide ions in the stacks.

Table 3. Bond Distances (Å) and Angles (deg) for DEA)₃Pb3I₉· $^{1/2}$ H₂O^{*a*}

	Dista	nces	
Pb(1) - I(9)	3.033(1)	Pb(2)-I(5)	3.226(2)
Pb(1) - I(6)	3.127(2)	Pb(2) - I(7)	3.291(2)
Pb(1)-I(10)	3.229(2)	Pb(2)-I(11)	3.391(2)
Pb(1) - I(5)	3.238(2)	Pb(3)-I(8)	3.024(2)
Pb(1)-I(11)#1	3.337(2)	Pb(3)-I(12)	3.027(2)
Pb(1)-I(11)	3.372(2)	Pb(3)-I(6)#2	3.239(2)
Pb(2) - I(4)	3.087(2)	Pb(3)-I(4)	3.269(2)
Pb(2)-I(10)#1	3.165(2)	Pb(3)-I(7)#2	3.463(2)
Pb(2)-I(7)#2	3.207(2)	Pb(3)-I(11)#2	3.462(2)
	Ang	les	
I(9) - Pb(1) - I(6)	92.06(5)	I(10)#1-Pb(2)-I(7)	174.69(3)
I(9) - Pb(1) - I(10)	94.34(4)	I(7)#2-Pb(2)-I(7)	87.91(5)
I(6) - Pb(1) - I(10)	92.14(4)	I(5) - Pb(2) - I(7)	86.36(5)
I(9) - Pb(1) - I(5)	89.26(4)	I(4) - Pb(2) - I(11)	178.57(3)
I(6) - Pb(1) - I(5)	89.17(4)	I(10)#1-Pb(2)-I(11)	87.95(4)
I(10) - Pb(1) - I(5)	176.12(3)	I(7)#2-Pb(2)-I(11)	87.23(3)
I(9)-Pb(1)-I(11)#1	87.96(5)	I(5) - Pb(2) - I(11)	88.48(3)
I(6)-Pb(1)-I(11)#1	179.98(4)	I(7) - Pb(2) - I(11)	86.89(4)
I(10)-Pb(1)-I(11)#1	87.86(4)	I(8)-Pb(3)-I(12)	92.82(4)
I(5)-Pb(1)-I(11)#1	90.84(4)	I(8)-Pb(3)-I(6)#2	93.46(5)
I(9) - Pb(1) - I(11)	175.88(2)	I(12)-Pb(3)-I(6)#2	89.94(4)
I(6) - Pb(1) - I(11)	91.45(5)	I(8) - Pb(3) - I(4)	91.03(5)
I(10) - Pb(1) - I(11)	87.69(3)	I(12) - Pb(3) - I(4)	98.37(4)
I(5) - Pb(1) - I(11)	88.62(3)	I(6)#2-Pb(3)-I(4)	170.35(3)
I(11)#1 - Pb(1) - I(11)	88.54(5)	Pb(2)-I(4)-Pb(3)	95.16(3)
I(4)-Pb(2)-I(10)#1	91.75(4)	Pb(2) - I(5) - Pb(1)	94.08(3)
I(4)-Pb(2)-I(7)#2	91.39(4)	Pb(1)-I(6)-Pb(3)#2	94.28(5)
I(10)#1-Pb(2)-I(7)#2	93.07(5)	Pb(2)#2-I(7)-Pb(2)	92.09(5)
I(4) - Pb(2) - I(5)	92.93(3)	Pb(2)#1-I(10)-Pb(1)	95.11(4)
I(10)#1 - Pb(2) - I(5)	92.28(5)	Pb(1)#1-I(11)-Pb(1)	91.46(5)
I(7)#2-Pb(2)-I(5)	173.02(3)	Pb(1)#1-I(11)-Pb(2)	89.07(4)
I(4) - Pb(2) - I(7)	93.44(4)	Pb(1)-I(11)-Pb(2)	88.75(3)

^{*a*} Symmetry transformations used to generate equivalent atoms: #1, -x, -y, -z + 2; #2, -x + 1, -y, -z + 2.

Table 4. Bond Distances (Å) and Angles (deg) for $(\beta$ -alaH)₂Sn₂Cl₆^a

	Dist	ances	
Sn(1) - I(1)	3.202(1)	Sn(2) - I(3)	3.645(1)
Sn(1) - I(2)	2.923(1)	Sn(2)-I(3)#1	3.285(1)
Sn(1)-I(1)#1	3.114(1)	Sn(2)-I(4)	3.046(1)
Sn(1) - I(3)	3.615(1)	Sn(2) - I(5)	3.286(1)
Sn(1)-I(3)#1	3.239(1)	Sn(2)-I(5)#2	3.063(1)
Sn(1)-I(4)	3.128(1)	Sn(2) - I(6)	2.917(1)
	Ar	igles	
I(2)-Sn(1)-I(1)#1	91.18(2)	I(4) - Sn(1) - I(3)	82.44(2)
I(2) - Sn(1) - I(4)	85.28(2)	I(1) - Sn(1) - I(3)	103.53(2)
I(1)#1-Sn(1)-I(4)	100.30(3)	I(6) - Sn(2) - I(4)	86.79(2)
I(2) - Sn(1) - I(1)	89.91(2)	I(6)-Sn(2)-I(5)#2	89.12(2)
I(1)#1-Sn(1)-I(1)	86.12(2)	I(4)-Sn(2)-I(5)#2	101.32(2)
I(4) - Sn(1) - I(1)	172.03(2)	Sn(1)#2-I(1)-Sn(1)	100.88(2)
I(2) - Sn(1) - I(3)	162.82(2)	Sn(2) - I(4) - Sn(1)	107.79(2)
I(1)#1-Sn(1)-I(3)	79.30(1)		

^{*a*} Symmetry transformations used to generate equivalent atoms: #1, *x*, -y + 3/2, z - f; #2, x, -y + 3/2, z + f.

As can be seen from Figure 4 for the bromide salt, groups of six DEA⁺ cations girdle the columnar stacks at their narrowest points. It is this slippage and kinking of the stacks at these points that leads to the recombinatorial structure of these compounds. The main difference in the two structures is the inclusion of a lattice water molecule (at partial occupancy) in the iodide structure, as shown in Figure 3. This lattice water of hydration can form hydrogen bonds to any of the four iodide ions in the cleft in the stack produced by the polytypic nature of the stack. The O····I distances range from 3.20 to 3.50 Å.



Figure 2. Illustration of the asymmetric unit for (DEA)₃Pb₃Br₉ (augmented with symmetry-related Br⁻ ions to complete the Pb coordination spheres).



Figure 3. Illustration of the $(Pb_3I_9)_n^{3n-}$ columnar structure in $(DEA)_3Pb_3I_9$. $^{1/2}H_2O$, showing the site for the lattice water molecule.



Figure 4. Illustration of the $(Pb_3Br_9)n^{3\eta-}$ columnar structure in $(DEA)_3Pb_3$ -Br₉ and the girdling of the columns by sets of six DEA⁺ cations.



Figure 5. Illustration of the packing of the $(Pb_3Br_9)_n^{3n-}$ columns in $(DEA)_3Pb_3Br_9$, showing the hydrogen bonding linking the columns. The view is parallel to the *a* axis, and the *b* axis is vertical.

As illustrated in Figure 5 for $X = Br^{-}$, these columnar stacks are arranged in a pseudohexagonal fashion, with the



Figure 6. Illustration of the asymmetric unit in $(\beta$ -alaH)₂Sn₂Cl₆, augmented by additional iodide ions to complete the Sn octahedra.

Chart 4



3(1/2, 1/2)(3/2, -1/2)[X]

stacks running parallel to the *a* axis. Two of the three independent DEA⁺ cations form linking X····H–N–H···X hydrogen bonding interactions in the [110] directions to provide three-dimensional stability to the lattice. These interactions lead to short X···X contacts between 8 and 9 in adjacent stacks. For the bromide salt, these contact distances are 3.794(3) Å, while the contact distances are 4.002(2) Å in the iodide salt. The third DEA⁺ cation forms only intrastack hydrogen bonds. The inclusion of the lattice water in the iodide salt does not dramatically affect the N–H···X hydrogen bonding scheme that was observed for the bromide salt. However, it would appear to be associated with the observed reorientation and disorder of the ethyl groups in the DEA⁺ cations.

The columnar structure illustrated in Figure 3 may be viewed as being built up of stacks of the planar $Pb_3X_8^{2-}$ moieties, with each moiety augmented by a X⁻ ion to complete the octahedral coordination geometry of the terminal Pb ion (Pb(3)) in the moiety. The columns can then be represented by the pattern diagrammed in Chart 4. An extended Geiser notation can be developed to designate this pattern, 3(1/2, 1/2)(3/2, 1/2)[X], where the quantity in the brackets specifies the ligand(s) that decorate the planar moiety to complete the octahedral coordination of the metal ion. In the diagrammatic representation in Chart 4, the "o" indicates the position of the additional ligand that decorates the planar moiety.

 $(\beta$ -alaH)₂Sn₂Cl₆ Structure. The asymmetric unit of this structure, augmented to complete the very distorted octahedral coordination of the Sn(II) ions, is shown in Figure 6. The bond distances from both Sn atoms to I(3) are substantially longer than the others (3.645, 3.615 Å). Concomitantly, the distances to the iodide ions trans to I(3) are the shortest. This is presumably due to the localization



Figure 7. Illustration of the $(Sn_2I_6)2^{2n-}$ columnar stack in(β -alaH)₂Sn₂-Cl₆ illustrating the hydrogen bonding of the $(\beta$ -alaH)₂²⁺ dimers in the clefts of the stacks.

Chart 5



$$2(1/2, 1/2)(-1/2, -1/2)$$

of the lone pair of electrons on the Sn(II) ions in the region of the Sn–I(3) bonds. This distortion is significantly larger than in the Pb(II) salts, in accord with the smaller radius of the Sn(II) ion. The β -alaH⁺ cations form hydrogen-bonded pairs (O(3)–H(3C)···O(1) = 2.629(7) and O(2)–H(2C)– O(4) = 2.646(7) Å).

The tin/iodide species aggregate into a columnar structure that run parallel to the c axis, as seen in Figure 7. In addition, the $-NH_3^+$ moieties of the β -alaH⁺ cations hydrogen bond to the I^- ions in a bidendate fashion, as seen in Figure 7. The tin/iodide framework takes on a washboard structure to accommodate this mode of hydrogen bonding. Sitting in the clefts in this framework, the $-NH_3^+$ species hydrogen bond to iodide ions in three Sn_2I_6 layers. The stacking diagram for this stack, denoted by the augmented Geiser symbol of 2(1/2, 1/2)(-1/2, -1/2), is shown in Chart 5. This arrangement again corresponds to a recombinatorial structure of the dimer chain illustrated in Chart 3b. The -CO₂H····HO₂Cportions of the stacks shown in Figure 7 interdigitate to form layers lying parallel to the (100) planes, as shown in Figure 8, with weak N-H···I hydrogen bonds tying the layers together.

Discussion

Examination of the Cambridge Database²³ for hybrid organic/inorganic salts with extended edge-shared octahedra



Figure 8. Illustration of packing of the $(Sn_2I_6)_2^{2n-1/(\beta-alaH)_2^{2+1}}$ columns in $(\beta-alaH)_2Sn_2Cl_6$.

structures revealed only a limited number of examples of networks based on dimensional reduction of the parent hexagonal MX₂ lattice. No examples of recombinant structures were found. Many of these include coordinated neutral ligands in addition to the halide ions. These can introduce additional steric and/or electronic factors that can lead to variations of the basic structural patterns expected by dimensional reduction arguments. This is amply illustrated in the compounds that contain bibridged chains. Thus, (Nmethylethylenediammonium)CdCl₄·1/2H₂O contains linear bibridged chains (Chart 3a) in which the edge sharing is always in a trans configuration.²⁴ However, in the PbX₂-(2-methylpyridine) structures,²⁵ the coordinated pyridine causes the bridges to assume a cis conformation, while, in $PbI_2(py)_2$, the cis configuration occurs only every sixth metal ion.²⁶ ($C_3H_7N_2S$)CdCl₃, along with several other compounds, contains double chains of the type shown in Chart 3b.²⁷ They thus have the simple stacking notation 2(1/2, 1/2) like those noted by Tulsky and Long¹ in the double-chain inorganic salts. A similar situation occurs in $(\varphi_4 P)_4 [Pb_{15}I_{34}(dmf)_6]$ (φ = C_6H_5 ; dmf = dimethylformamide).²⁸ In this compound, planar bibridged $Pb_3I_8^{2-}$ and $Pb_3I_6(dmf)_2$ units can be identified and the stacking pattern (neglecting the differences in the stoichiometry of the two types of planar moieties) is given by the notation 3(1/2, 1/2) as illustrated in Chart 6a.²⁹ These n(1/2, 1/2) patterns can be viewed as simple ribbons obtained by dimensional reduction of the parent CdX₂

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- (29) With labeling of the two moiety units as a (for ionic) and s (for solvated), the repeat pattern for the moieties in the stack is (*sassa*).

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structure. A second type of ribbon obtained by the dimensional reduction process has stacking patterns denoted by the notation $n(3/2, 1/2)X_2$. Bonomartini-Corradi et al. reported two structures with this type of pattern. This includes the structure of (N-ethylethylenediammonium)₂Cd₄Cl₁₂•H₂O and the structure of $(C_7H_{16}N)_2[Cd_5Cl_{12}(H_2O)_2]•H_2O.^{30}$ The former contains planar bibridged Cd₄Cl₁₀²⁻ oligomers decorated with pairs of chloride ions, while the latter contains planar bibridged Cd₅Cl₁₂²⁻ oligomers decorated with pairs of water molecules. The latter form stacks shown in Chart 6b. An even more complex ribbon is found in Cd₃Br₆(H₂O)-(DMSO)₂•H₂O, where a 3(1/2, 1/2)(3/2, 1/2)[X] stacking pattern is observed.³² Compare this stacking pattern (Chart 7) with that observed in the lead halide salts reported in this paper (Chart 4). All of these stacks may be viewed as ribbons



obtained via dimensional reduction of the parent layered MX_2 structure.

Finally, it should be noted that the compound $(C_5H_{12}-ClN_2)Hg_2Cl_5$ contains a perforated MX₂ layer in which Hg₂- Cl_2^{2+} units are removed from the layer,³⁰ similar to the (MeEt₃N)Cu₃Cl₇ structure where Cu₂Cl₂²⁺ units are excised.¹² This type of perforation was not found by Tulsky and Long¹ in their summary of purely inorganic layered structures of this type.

In summary, this study further demonstrates the potential of using hybrid organic/inorganic systems to produce metal halide structures with novel polymeric structures. This capability has been demonstrated effectively, for example, in the development of semiconducting and conducting Pb-(II) and Sn(II) layer perovskite systems.⁴ In general, the combination of diverse hydrogen-bonding capabilities and steric effects causes subtle changes in the ways that the metal halide coordination polyhedra can self-assemble in the solid state. Given the unusual results from this study, continued exploration of the structural chemistry of hybrid organic/inorganic metal halide salts is needed to delineate the structural diversity of such systems.

Supporting Information Available: Crystal data for all three compounds in CIF format. This material is available free of charge via the Internet at http://pubs.acs.org.

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Chart 7

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