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Characterization and Structure of $\text{La}_4\text{Re}_6\text{O}_{19}$, a New Metal Cluster Compound

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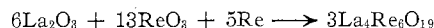
Crystals of $\text{La}_4\text{Re}_6\text{O}_{19}$ have been prepared hydrothermally, and their structure has been solved and refined by least squares to an R of 0.027. The crystals are cubic with $a = 9.0308$ (2) Å at 25°, $d_o = 8.92$ (6) and $d_c = 8.91$ for $Z = 2$ and space group I23. The data consisted of 56 reflections collected from the powder using a diffractometer. Rhenium-oxygen octahedra share edges in pairs to form binuclear clusters; in each cluster there is a short Re-Re distance of 2.42 Å. These clusters are joined, by sharing corners of the octahedra, to form a three-dimensional network. Within voids of this network there are La_4O tetrahedra. Magnetic and electrical transport properties of $\text{La}_4\text{Re}_6\text{O}_{19}$ have also been studied.

Introduction

A lanthanum rhenium oxide with body-centered-cubic symmetry ($a \sim 9.0$ Å) was discovered by one of us several years ago.³ At that time a single-crystal X-ray study established that the space group was I23, I23, or Im3, but the lack of a sufficiently large single-phase sample prevented characterization and structure determination. Now, a new method of preparation has led to a product which is readily purified. A characterization, including a structure determination, has been carried out. The composition of this new lanthanum rhenium oxide has been found to be $\text{La}_4\text{Re}_6\text{O}_{19}$.

Preparation and Characterization

Although the preparation of $\text{La}_4\text{Re}_6\text{O}_{19}$ can be carried out in evacuated silica ampoules at 1000°,³ a hydrothermal preparation has been found to offer the advantage of a product which is readily purified. Single-phase $\text{La}_4\text{Re}_6\text{O}_{19}$ has never been obtained in one step by any method, even when the reaction below was attempted.



A typical hydrothermal charge was 1.955 g of La_2O_3 (99.999%), 4.216 g of ReO_3 (from Alfa Inorganics), and 3 ml of water. The container was a $3/8$ -in. diameter gold tube, about 5 in. long and sealed by welding at both ends. Reaction conditions were generally 700° for 12 hr with 3000 atm of external pressure applied to the gold tube. The product was multiphase, but only crystals of $\text{La}_4\text{Re}_6\text{O}_{19}$ remained after washing with 6 N HCl. Although some of these crystals had well-developed faces, most were irregular with roughly a spherical shape. The mean diameter of a crystal often was as large as ~ 0.5 mm.

Although the composition of the purified product was derived primarily from the structure determination, supporting analytical data were obtained. Chemical analysis for lanthanum with EDTA gave 28.0% (28.10% calcd). Thermogravimetric analysis in an

H_2 -Ar (15%) atmosphere showed that 10.5% of the sample was associated with the reducible rhenium; a weight loss of 10.52% is calculated. The density of the $\text{La}_4\text{Re}_6\text{O}_{19}$ crystals was determined by the standard pycnometer method using benzene at 25° (obsd, 8.92 ± 0.06 g/cm³; calcd, 8.91 g/cm³, for $Z = 2$).

Resistivity measurements by the four-probe method were carried out on two crystals from two separate preparations. Metallic behavior was observed from 4.2° ($\rho \sim 10^{-5}$ ohm-cm) to room temperature ($\rho \sim 10^{-4}$ ohm-cm). Thermoelectric power measurements at 298°K showed a Seebeck coefficient *vs.* copper of +1.6 $\mu\text{V}/\text{deg}$.

Magnetic susceptibility measurements were performed using a Faraday balance at 298, 77, and 4.2°K. Between fields of 1×10^3 and 8×10^3 oersteds the expression $\chi_m = \chi_m^0 + (a/H)$ was followed very well. The values of $\chi_m^0 \times 10^3$ and a were: at 298°K, 0.25 and 0.52; at 77°K, 0.43 and 2.15; and at 4.2°K, 2.34 and 0.65, respectively. The field dependence observed could easily be due to a very small amount of an unknown ferromagnetic contamination.

X-Ray Data

Integrated X-ray powder intensity data were gathered with a Norelco diffractometer using Cu $K\alpha$ radiation and an LiF bent-crystal monochromator. The intensity of each peak was determined by scanning at 0.25°/min and accumulating both time and counts. The background subtracted from a peak was determined from a plot of background *vs.* 2θ . No extraneous peaks were observed, and, since the symmetry is cubic, resolution problems were not encountered.

The sample was prepared by repeatedly grinding it and passing it through a 325-mesh sieve. The powder was then packed in a tray by several different methods and X-ray data were taken to determine if preferred orientation was a serious problem. Comparison of the several sets of data led to the conclusion that preferred orientation was not significant. The depth of the flat sample tray was about 1 mm, which effectively prohibits any X-rays from passing through the sample. The dimensions of the tray and beam slits were such that the entire beam was intercepted by the sample for all

(1) Operated with support from the U. S. Air Force.

(2) Contribution No. 1375.

(3) J. M. Longo, Ph.D. Dissertation, University of Connecticut, 1964.

peaks except the 110 (which was rejected from the refinement for this reason).

The intensity of a group of equivalent reflections was taken to be $I = F^2SLpM$, where F has its usual definition, S is the scale factor, L is the Lorentz factor of $1/(\sin^2 \theta \cos \theta)$, p is the polarization factor of $(1 + \cos^2 2\theta_M \cos^2 2\theta)/2$, and M is the multiplicity factor. θ_M equals 22.5° for the LiF monochromator used. The observed intensity is compared to the sum of non-equivalent reflections which fall at the same 2θ value. The scattering factor for O^{2-} was taken from Tokonami.⁴ The scattering factors for neutral La and Re are those of Cromer and Waber,⁵ and the anomalous dispersion corrections for La, Re, and O, were taken from Cromer.⁶

The cell edge of $\text{La}_4\text{Re}_6\text{O}_{19}$ was determined by use of a Guinier camera using strictly monochromatic $\text{Cu K}\alpha_1$ radiation. A potassium chloride ($a = 6.29310 \text{ \AA}$) internal standard was used, and a least-squares refinement was carried out. At 25° the unit cell edge is $9.0308 (2) \text{ \AA}$.

Structure Determination and Refinement

The preliminary single-crystal investigation showed no reflections in violation of the body-centered requirement of $h + k + l = 2n$. There were no other systematic absences observed. This information and the Laue group limited the space group to the three mentioned below.

Initial computations for structure determination and refinement were made using a program⁷ developed at Lincoln Laboratory. Trial atomic positions for La and Re were obtained from compositional and geometrical considerations within the possible space groups $I2_3$, $I23$, and $Im3$. Using only the first 14 lines, good results were obtained for La in 8(c) and Re in 12(e) of space group $I23$. Assuming octahedral coordination of rhenium by oxygen, 36 oxygens were placed in 12(d) and 24(f). Refinement of this model led to an R value of about 6%.

It was now apparent that this structure could be viewed as a three-dimensional network in which rhenium-oxygen octahedra are connected by sharing edges and corners as shown in Figure 1. The lanthanums are located within voids of this network, and the formula at this point was $\text{La}_4\text{Re}_6\text{O}_{19}$.

The final refinements were performed using a least-squares program written by C. T. Prewitt. All possible reflections, except the 110, which occurred from 0 to $160^\circ 2\theta$ were used. Real and imaginary anomalous dispersion corrections were made to A and B for all atoms. Since there still appeared to be space available within the voids of the network, additional oxygens were placed at 2(a) and at 8(c) with $x = 0.34$. Several cycles of least squares indicated that there was an oxygen at 2(a) but that there was no significant scattering power at 8(c) with $x \sim 0.34$. Therefore, the

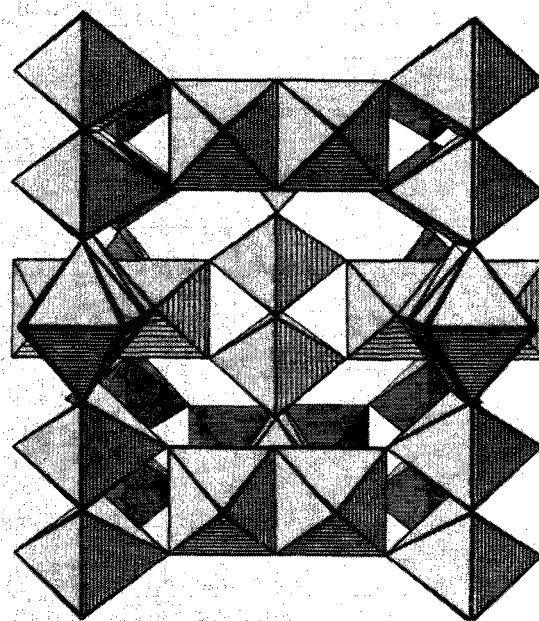


Figure 1.—The rhenium-oxygen octahedra network.

formula was now $\text{La}_4\text{Re}_6\text{O}_{19}$. The R which was minimized was $R = \sqrt{\sum(I_o - I_c)^2} / \sqrt{\sum I_o^2}$. No weighting scheme was used since data collected by counting and refined on intensities are weighed naturally (*i.e.*, the strongest peak is assumed to have the best accuracy). Based on 56 reflections the final R was 0.027. The corresponding R' was 0.038 where $R' = \sum |I_o - I_c| / \sum I_o$.

The final values of the atomic parameters are given in Table I. The population factors of the atoms

TABLE I
FINAL ATOMIC POSITIONAL AND
ISOTROPIC THERMAL PARAMETERS^a

Atom	Position	x	y	z	B
La	8(c)	0.1613 (4)	0.1613	0.1613	0.01
		0.1599 ^b			
Re	12(e)	0.3659 (4)	0.5	0.0	0.01
		0.3663			
O_1	12(d)	0.331 (5)	0.0	0.0	0.97 (99)
		0.322			
O_2	24(f)	0.340 (4)	0.272 (4)	0.979 (5)	1.24 (83)
		0.341			
O_3	2(a)	0.0	0.0	0.0	0.01

^a Parameters are expressed as fractions of the cell edge. Standard deviations are in parentheses. ^b Single-crystal work of N. Morrow and L. Katz, American Crystallographic Association Meeting, 1967, Paper R-6.

within the voids of the network (O at 2(a) and La) were allowed to vary, but these sites remained fully occupied within their standard deviations. The temperature factors of La, Re, and O_3 tended to become negative: $-0.04 (11)$, $-0.02 (3)$, and $-4.85 (279)$, respectively. This is apparently caused by the very high absorption of $\text{La}_4\text{Re}_6\text{O}_{19}$ with Cu radiation ($\mu = 1768.7$). Although absorption is generally taken to be constant with θ for the powder diffractometer method,⁸ this is

(8) B. D. Cullity, "Elements of X-Ray Diffraction," Addison-Wesley Publishing Co., Inc., Reading, Mass., 1956.

(4) M. Tokonami, *Acta Cryst.*, **19**, 486 (1965).

(5) D. T. Cromer and J. T. Waber, *ibid.*, **18**, 1041 (1965).

(6) D. T. Cromer, *ibid.*, **18**, 20 (1965).

(7) P. M. Raccach and J. B. Goodenough, *Phys. Rev.*, **155**, 932 (1967).

TABLE III

Bond Distances (Å)

Re-Re	1×2.422 (7)	$\text{O}_{1a}\text{-O}_{1b}$	1×3.053 (95)
	4×3.519 (2)	O_{2a}	2×2.464 (34)
La-Re	3×3.397 (2)	O_{2b}	2×2.849 (41)
	3×3.859 (4)	O_{2d}	2×3.042 (41)
Re-O _{1a}	2×1.948 (37)	O_{2c}	2×3.126 (36)
O_{2a}	2×1.913 (36)	$\text{O}_{2a}\text{-O}_{2b}$	2×2.766 (23)
O_{2b}	2×2.081 (34)	O_{2c}	2×2.907 (72)
La-O _{1a}	3×2.567 (28)	O_{2d}	2×2.681 (15)
O_{2a}	3×2.513 (46)	$\text{O}_3\text{-O}_{1a}$	6×2.989 (48)
O_{2b}	3×2.934 (46)	O_{2a}	12×3.939 (36)
O_3	1×2.523 (7)		

Bond Angles (Deg)

$\text{O}_{1a}\text{-Re-O}_{2c}$	2×174 (12)	$\text{O}_{1a}\text{-Re-O}_{2b}$	2×90 (1)
$\text{O}_{2b}\text{-Re-O}_{2d}$	1×167 (9)	$\text{O}_{1a}\text{-Re-O}_{2d}$	2×98 (1)
		$\text{O}_{2a}\text{-Re-O}_{2b}$	2×88 (1)
$\text{O}_{1a}\text{-Re-O}_{1b}$	1×103 (1)	$\text{O}_{2a}\text{-Re-O}_{2d}$	2×84 (1)
$\text{O}_{1a}\text{-Re-O}_{2a}$	2×79 (1)	$\text{O}_{2a}\text{-Re-O}_{2c}$	1×99 (2)

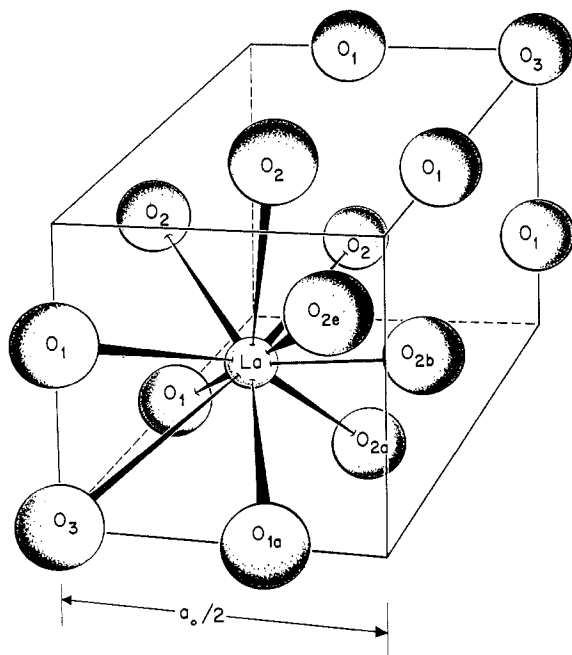


Figure 2.—The lanthanum-oxygen coordination.

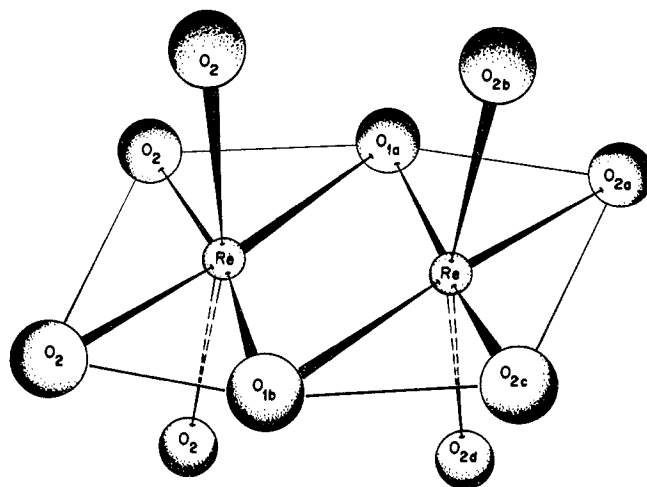


Figure 3.—A rhenium-oxygen octahedra pair.

all typical of metals and clearly indicate that electron delocalization has occurred. The results of all three measurements are inconsistent with a hopping model which would require the rhenium atoms to be only statistically equivalent. If three-dimensional delocalization is to occur only by metal-metal interaction, then direct overlap of the rhenium 5d orbitals must occur not only at the 2.42-Å distance but also at a distance of 3.52 Å. It is, however, likely that the delocalization involves orbitals of oxygen in a manner similar to that which has been proposed for ReO_3 .¹⁴

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(14) A. Ferretti, D. B. Rogers, and J. B. Goodenough, *J. Phys. Chem. Solids*, **26**, 2007 (1965).