

strained by hydrogen bonding and packing requirements; however, this may be in part a reflection of absorption errors.

The dihedral angles that completely specify the orientation of the thiourea groups relative to the planes defined by Ni and S atoms are given in Table III. Three factors contribute to these orientations: (a) hydrogen bonding to the bromide ion, (b) steric requirements that restrict the thiourea groups, and (c) electrostatic interactions between electrons in the S-C $p\pi$ molecular orbitals and the metal d_{zz} , d_{yz} , d_{xy} orbitals, all of which are filled. Obviously all three factors contribute to the specific orientation of a

thiourea group and studies are currently underway in this laboratory to determine which factor(s) predominate(s). Preliminary photographic data indicate that $\text{Ni}(\text{tu})_6\text{Br}_2$ and $\text{Ni}(\text{tu})_6\text{I}_2$ are isomorphous and a comparison of the specific orientations of thiourea groups in this series to the orientation of thiourea groups in $\text{Pt}(\text{tu})_4\text{Cl}_2$,²⁰ $\text{Pd}(\text{tu})_4\text{Cl}_2$,¹⁹ and the isomorphous series $\text{Mn}(\text{tu})_4\text{Cl}_2$, $\text{Fe}(\text{tu})_4\text{Cl}_2$, $\text{Cd}(\text{tu})_4\text{Cl}_2$, and $\text{Co}(\text{tu})_4\text{Cl}_2$ ^{18,42} is being made.

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(42) J. E. O'Connor and E. L. Amma, to be submitted for publication.

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The Crystallography and Structure of α -Zirconium Bis(monohydrogen orthophosphate) Monohydrate¹

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The crystal structure of α -zirconium bis(monohydrogen orthophosphate) monohydrate, $\text{Zr}(\text{HPO}_4)_2 \cdot \text{H}_2\text{O}$, has been determined from integrated precession data (568 nonzero reflections). The crystals are monoclinic, space group $P2_1/c$, with cell dimensions $a = 9.076 \pm 0.003 \text{ \AA}$, $b = 5.298 \pm 0.006 \text{ \AA}$, $c = 16.22 \pm 0.02 \text{ \AA}$, and $\beta = 111.5 \pm 0.1^\circ$. The calculated density with $Z = 4$ is 2.76 g/cm^3 compared to an observed density of $2.72 \pm 0.04 \text{ g/cm}^3$. The structure was refined to an R factor of 8.4% by least-squares methods. The structure is a layered one, each layer consisting of planes of zirconium atoms bridged through phosphate groups which alternate above and below the metal atom planes. Each phosphate group bonds to three different zirconium atoms producing octahedral coordination about the zirconium atoms. The Zr-O bond distances range from 2.04 to 2.11 \AA . The fourth phosphate oxygen bears the hydrogen and points toward an adjacent layer. The layers are arranged relative to each other in such a way as to form zeolitic-type cavities. A water molecule resides in the center of each cavity and is hydrogen bonded to phosphate groups. The relation between structure and ion-exchange properties of the crystals is discussed in this and a forthcoming paper.

Introduction

Preparations commonly called zirconium phosphate are of interest because they function as ion exchangers. These phosphates have been prepared in both amorphous and crystalline forms. When solutions containing phosphate ion and Zr(IV) salts are mixed, amorphous gels of variable composition result.² These gels are transformed into microcrystals by refluxing them in various phosphate-containing solutions. If the refluxing medium is phosphoric acid, the resultant crystals are the normal or α phase.³ However, refluxing in a mixture of phosphoric acid and sodium dihydrogen phosphate produces the β and γ phases.⁴

The α phase (α -ZrP) has been shown to be zirconium bis(monohydrogen orthophosphate) monohydrate, $\text{Zr}(\text{HPO}_4)_2 \cdot \text{H}_2\text{O}$, on the basis of its composition, dehydration, and ion-exchange behavior.³ Crystals large enough for a single-crystal X-ray diffraction study were prepared from the microcrystals as described below. An attempt to solve the structure by two-dimensional methods was only partially successful.⁵

Severe overlap of several oxygen atom peaks in the Fourier projections prevented a complete solution of the structure. A three-dimensional study has now been carried out, the details of which are reported here.

Experimental Section

Preparation.—Single crystals were prepared by heating a mixture of gel and 12 M phosphoric acid in a sealed quartz tube for several weeks at 170–180°. Very thin hexagonal or triangular platelets were obtained.

(1) Portions of this paper were taken from the Ph.D. thesis of G. D. Smith presented to the Chemistry Department, Ohio University, June 1968. This work was supported by the National Science Foundation under Grant No. GP-6433.

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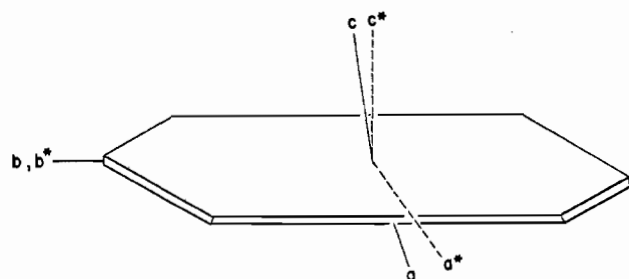


Figure 1.—Schematic drawing on an α -ZrP hexagonal platelet. The real axes are indicated by solid lines and the reciprocal axes by dashed lines.

X-Ray Data.—A single crystal of α -ZrP (hexagonal platelet) is shown schematically in Figure 1. The unit cell was found to be monoclinic. The real and reciprocal axes are indicated in the figure by solid and dashed lines, respectively.

Accurate cell dimensions were obtained from precession photographs using Mo $K\alpha$ radiation (0.7107 Å). The films were corrected for shrinkage by exposing pairs of fiducial spots along both the horizontal and the vertical directions of the film. The cell constants were found to be $a = 9.076 \pm 0.003$ Å, $b = 5.298 \pm 0.006$ Å, $c = 16.22 \pm 0.02$ Å, and $\beta = 111.5 \pm 0.1^\circ$.

The unit cell volume of 725.7 Å³ with four formula units per unit cell gives a density of 2.76 g/cm³. This compares with an observed density, determined pycnometrically in cyclohexane, of 2.72 ± 0.04 g/cm³ (average of four determinations).

A survey of $h0l$, $h1l$, $h2l$, $0kl$, $1kl$, $2kl$, $hk0$, $hk1$, and $hk2$ precession photographs (taken with zirconium-filtered Mo $K\alpha$ radiation) revealed the systematic absences $h0l$ for $l = 2n + 1$ and $0k0$ for $k = 2n + 1$. These absences are consistent with the space group $P2_1/c$. Since there are four formula units per unit cell, no symmetry restrictions need be imposed.

Intensity data were gathered from integrated precession photographs using Mo $K\alpha$ radiation. Two crystals approximately 0.4 mm long by 0.2 mm wide by 0.025 mm thick were used to gather the intensity data. One was mounted so that its b^* axis coincided with the dial axis and $nk1$, $n = 0-4$, and hkn , $n = 0-10$, photographs were obtained. The other was mounted so that the dial axis coincided with the a^* axis and the precession axis was b . This setting was used to obtain $h0l$, $h1l$, and $h2l$ photographs. Long (24-48 hr) and short (2 hr) exposures of each layer were taken to ensure that all reflections were within the proportional response range of the film. The intensities of the integrated spots were read with a Joyce-Loebl Mark IIIC double-beam automatic recording microdensitometer (Joyce-Loebl and Co., Ltd.). The intensities were corrected for Lorentz and polarization factors and then placed on the same scale following the least-squares procedure of Monahan, Schiffer, and Schiffer.⁶ No correction for absorption was applied ($\mu = 19.0$ cm⁻¹). Since all of the intensity photographs were taken with the X-ray beam traversing the thin section of the crystal, the absorption correction varies only slightly from photograph to photograph. In all, 568 nonzero intensities were recorded.

Solution and Refinement of the Structure

The zirconium and phosphorus positional parameters were known fairly well from the previous two-dimensional study.⁶ These were further refined by least-squares and then a three-dimensional electron density map was prepared using signs calculated for these positions. Sections were taken perpendicular to the b

TABLE I
X-RAY POSITIONAL AND THERMAL PARAMETERS
WITH ESTIMATED STANDARD DEVIATIONS^a

Atom	X	Y	Z	B, Å ²
Zr ₁	0.7606 (2)	0.250 (2)	0.5144 (1)	0.67 (2)
P ₂	0.0008 (6)	0.752 (5)	0.6129 (4)	1.13 (8)
P ₃	0.4711 (6)	0.255 (5)	0.1044 (4)	0.98 (8)
O ₄	0.105 (2)	0.806 (5)	0.560 (1)	1.7 (3)
O ₅	0.934 (3)	0.486 (6)	0.601 (2)	2.6 (4)
O ₆	0.869 (2)	0.941 (5)	0.589 (4)	1.6 (4)
O ₇	0.104 (2)	0.760 (9)	0.716 (1)	2.2 (3)
O ₈	0.344 (3)	0.057 (5)	0.062 (1)	2.0 (4)
O ₉	0.418 (2)	0.514 (5)	0.069 (1)	1.6 (3)
O ₁₀	0.512 (2)	0.257 (9)	0.205 (1)	2.3 (3)
O ₁₁	0.375 (2)	0.814 (5)	0.910 (1)	1.9 (4)
O ₁₂	0.254 (2)	0.265 (8)	0.259 (1)	3.1 (4)

^a Numbers in parentheses here and in succeeding tables are estimated standard deviations in the least significant digit.

axis every 0.25 Å. The atomic parameters obtained from this map were in good agreement with those derived from the two projections. However, the calculated structure factors for all reflections with l odd were in poor agreement with the observed data. This resulted from an ambiguity in location of the origin which could not be resolved in projection because of the small number of $(h0l)$ and $(0kl)$ reflections with l odd. On shifting the origin by $z = 1/4$, a marked improvement in agreement was observed.

The structural parameters obtained above were refined by the block-diagonal least-squares method.⁷ The function minimized was $\sum w(|F_o| - |F_c|)^2$. The atomic scattering factors for the neutral atoms were those tabulated in ref 8. Only the scattering factor for zirconium was corrected (real part only) for dispersion.⁹ For the initial cycles of refinement a weighting scheme similar to that suggested by Hughes was employed.¹⁰ The weights were $w = 1$ when $|F_o|/K \leq 4.8$ and $w = 4.8K/|F_o|$ when $|F_o|/K > 4.8$. K is the over-all scale factor which places $|F_o|$ on the absolute scale. Individual isotropic temperature factors were assigned for each atom.

The relaxation factors suggested by Hodgson and Rollet¹¹ were found to produce oscillations of the positional parameters. However, when a constant relaxation factor of 0.5 was employed, rapid convergence was obtained. In five cycles the residual, $R = \sum (||F_o| - |F_c||) / \sum |F_o|$, was reduced from 0.184 to 0.109. At this point a new weighting scheme was introduced.¹² The new weights were $w = 1/(a +$

(7) The programs used in this work were all locally modified for use on the Ohio University IBM 360 Model 44 computer: (a) C. Calvo, "Data Process for Precession Photographs," McMaster University, Hamilton, Ont., Canada; (b) F. R. Ahmed, "NRC-8 Fourier Program; NRC-10 Structure Factor Least Squares," National Research Council, Ottawa, Ont., Canada; (c) M. E. Pippy and F. R. Ahmed, "NRC-12 Scan of Interatomic Distances and Angles," National Research Council, Ottawa, Ont., Canada.

(8) "International Tables for X-Ray Crystallography," Vol. III, The Kynoch Press, Birmingham, England, 1962, p 202.

(9) See ref 8, p 213.

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(6) J. E. Monahan, M. Schiffer, and J. P. Schiffer, *Acta Cryst.*, **22**, 322 (1967).

TABLE II

OBSERVED AND CALCULATED STRUCTURE FACTORS X 10 (IN ELECTRONS) FOR α-ZrP₂O₇

Table with multiple columns for L, FO, PC, and various structure factor values. Includes headers for observed and calculated factors and a grid of numerical data points.

* Asterisks denote unobserved reflections.

TABLE III
 BOND DISTANCES IN α -ZrP (ÅNGSTRÖM UNITS)

Zr ₁ -O ₄	2.04 (2)	P ₂ -O ₄	1.51 (2)	P ₃ -O ₈	1.52 (3)
O ₅	2.10 (3)	O ₆	1.52 (4)	O ₉	1.50 (3)
O ₆	2.06 (3)	O ₆	1.50 (3)	O ₁₀	1.54 (2)
O ₈	2.06 (3)	O ₇	1.60 (2)	O ₁₁	1.54 (2)
O ₉	2.11 (2)	Av	1.53	Av	1.52
O ₁₁	2.06 (2)				
Av	2.07				

 TABLE IV
 BOND ANGLES IN α -ZrP (DEGREES)

O ₄ -Zr ₁ -O ₅	90.9 (1.0)	O ₄ -P ₂ -O ₅	113.7 (1.7)	O ₈ -P ₃ -O ₉	112.1 (1.6)
O ₄ -Zr ₁ -O ₆	88.6 (0.9)	O ₄ -P ₂ -O ₆	109.7 (1.6)	O ₈ -P ₃ -O ₁₀	109.4 (1.8)
O ₄ -Zr ₁ -O ₈	90.1 (0.9)	O ₄ -P ₂ -O ₇	109.8 (1.7)	O ₈ -P ₃ -O ₁₁	110.3 (1.6)
O ₄ -Zr ₁ -O ₉	91.1 (0.9)	O ₅ -P ₂ -O ₆	110.1 (1.7)	O ₈ -P ₃ -O ₁₀	108.4 (1.8)
O ₄ -Zr ₁ -O ₁₁	178.8 (1.0)	O ₅ -P ₂ -O ₇	102.3 (1.8)	O ₈ -P ₃ -O ₁₁	110.0 (1.5)
O ₅ -Zr ₁ -O ₆	90.1 (1.0)	O ₆ -P ₂ -O ₇	111.0 (1.8)	O ₁₀ -P ₃ -O ₁₁	106.3 (1.8)
O ₅ -Zr ₁ -O ₈	89.8 (1.1)				
O ₅ -Zr ₁ -O ₉	89.9 (0.9)				
O ₅ -Zr ₁ -O ₁₁	90.2 (1.0)				

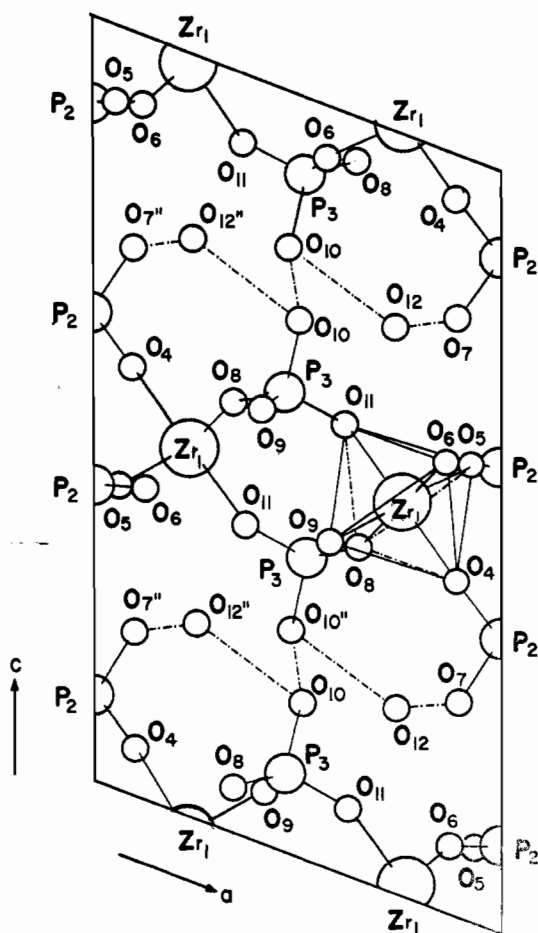


Figure 2.—Projection of the α -ZrP structure on (010). The glide plane is at $y = 1/4, 3/4$, and the 2_1 axes are at $x = 0, 1/2$; $z = 1/4, 3/4$. The doubly primed atoms are related to similarly numbered unprimed atoms by the 2_1 axes. Broken lines indicate possible hydrogen bonding (Scheme I). The water molecular is O₁₂.

$|F_o|K + b|F_o|^2K^2$, where $a = 2|F_{\min}|K$ and $b = 2|F_{\max}|K$. The final value of R was 0.084 and the weighted R was 0.089.

The final parameters together with their standard deviations are listed in Table I. Table II lists the observed and calculated structure amplitudes. The final difference map showed several positive regions the highest of which was $1.25 \text{ e}/\text{\AA}^3$ in the vicinity of a zirconium atom. These peaks could not be assigned to hydrogen atoms and must arise from other factors, primarily absorption, thermal motion errors, and termination-of-series effects.

Description of the Structure

The structure of α -ZrP projected onto the ac plane is shown in Figure 2. The zirconium atoms are situated alternately slightly above and below the ab planes at $z = 0$ and $1/2$. These are bridged to phosphate groups located above and below the metal atom planes to form layers. One of the idealized layers is shown in more detail in Figure 3. Any three adjacent zirconium atoms in the same layer form a slightly distorted equilateral triangle. The phosphate groups are located such that the phosphorus atoms are in the centers of the equilateral triangles above and below the metal atom planes in alternate triangles. Three of the phosphate oxygens are bonded to the three metal atoms forming the triangle. The fourth oxygen bears the hydrogen and is directed nearly perpendicular to the ab plane.

The bond distances and bond angles are given in Tables III and IV. The coordination about the zirconium atoms is octahedral, the largest deviation from 90° angles being 2.3° . The phosphate tetrahedra are slightly distorted. In one phosphate group the P-O bond of the P-OH group is 1.60 \AA while the other three P-O bonds average 1.51 \AA . This same phosphate group has one bond angle (O_5 -P₂-O₇) which deviates

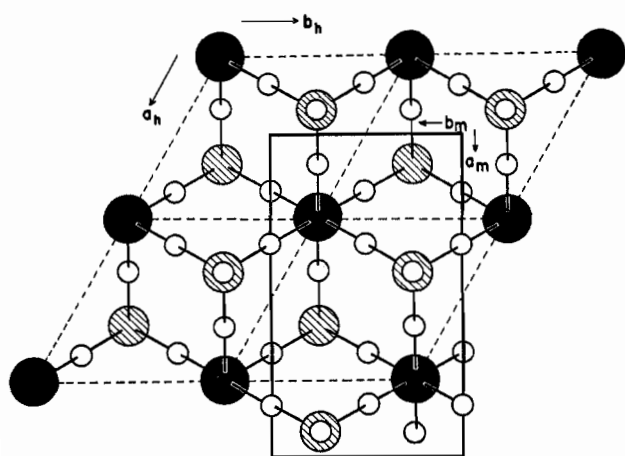


Figure 3.—Idealized layer in the α -ZrP structure showing relationship of the pseudo-hexagonal cell (broken lines) to the true, monoclinic unit cell.

significantly from tetrahedral. The other phosphate group is more regular, the bond lengths varying from 1.50 to 1.54 Å and the angles from 106 to 112°. These bond distances and angles are rather similar to those found in potassium dizirconium triphosphate.¹³

The most interesting features of the α -ZrP structure are those which make it possible for the crystals to exchange ions. The description of these features is facilitated by making use of a pseudo-hexagonal unit cell which is related to the true unit cell by

$$a_h = \frac{1}{2}a_m / \cos 30^\circ \cong 5.24 \text{ \AA}$$

$$b_h = b_m = 5.298 \text{ \AA}$$

$$c_h = 1.5c_m \cos 21.5^\circ = 22.62 \text{ \AA}$$

The hexagonal c axis is coincident with the c^* axis of the monoclinic unit cell and hence is perpendicular to the hexagonal face (001) of the crystal. The origin of the new cell is placed at a zirconium atom as shown in Figure 3. The positions of the metal atoms in the pseudo-cell are very close to $0, 0, 0; \frac{1}{3}, \frac{2}{3}, \frac{1}{3}; \frac{2}{3}, \frac{1}{3}, \frac{2}{3}$. The phosphorus atoms are (ideally) at $0, 0, \pm \frac{1}{4}; \frac{1}{3}, \frac{2}{3}, \frac{1}{12}; \frac{1}{3}, \frac{2}{3}, \frac{7}{12}; \frac{2}{3}, \frac{1}{3}, \frac{5}{12}; \frac{2}{3}, \frac{1}{3}, \frac{11}{12}$. Thus, any two adjacent layers are displaced relative to one another by $\frac{1}{3}, \frac{2}{3}, \frac{1}{3}$. This places the phosphorus atoms directly above the zirconium atom (and *vice versa*) in the layer below.

The arrangement of atoms described above results in the presence of cavities between layers of the type shown in Figure 4. The phosphorus and zirconium atoms in each layer form a six-membered ring (oxygen neglected) with a chair shape. However, one of the rings is rotated 60° relative to the other. The cavities are closed at the top and bottom by phosphate groups, respectively, above and below the centers of the six-membered rings. There is exactly one such cavity per

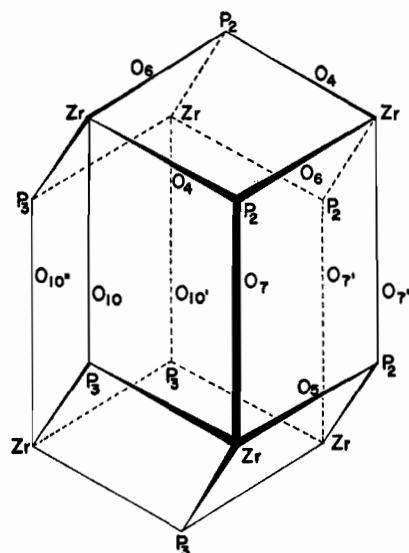


Figure 4.—Idealized picture of a cavity formed by two adjacent layers in α -ZrP. The symmetry of the cavity is D_{3d-32m} with the water located in the center of the cavity and the $\bar{3}$ axis parallel to c^* . The singly primed atoms are related to unprimed atoms of the same number by one unit cell translation. To preserve clarity, not all of the oxygen atoms have been included.

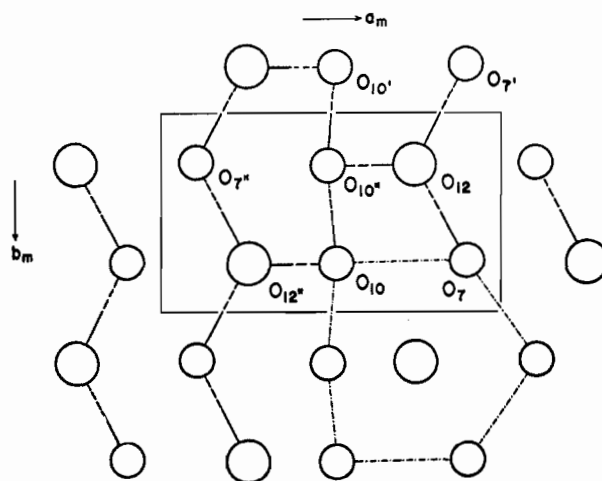


Figure 5.—Projection of oxygen atoms involved in hydrogen bonding by Scheme I onto (001). The water molecule is O_{12} . The dot-dashed line outlines one cavity and the solid line one unit cell. The half-solid, half-broken lines indicate hydrogen bonds with the solid portion representing the O-H bond.

formula unit of α -ZrP and the water molecule sits in the center of the cavity.

The water molecule (O_{12}) is held in the cavity by hydrogen bonds. This is inferred from the oxygen-oxygen distances in Table V. There are three short O-O distances involving the water molecule (O_{12} - O_{10}'' , O_{12} - O_7 , and O_{12} - O_7'). One possible arrangement of hydrogen bonds is that in which one hydrogen atom of a phosphate group is directed toward the water molecule (P_2 - O_7 -H---O< $\frac{H}{H}$) and the water hydrogens are directed at O_{10}'' and O_7' . The disposition of these

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TABLE V
POSSIBLE HYDROGEN BOND DISTANCES IN α -ZrP
(ÅNGSTRÖM UNITS)

O ₁₂ -O ₇	3.06 (6)
O ₁₀ ''	2.78 (3)
O ₇ '	2.82 (6)
O ₁₀ ''-O ₁₀	3.07 (6)
O ₁₀ '	3.07 (6) ^a

^a Next shortest distance is O₁₀-O₇ which is 3.24 (3) Å.

bonds in projection is shown in Figure 5. All three of these hydrogen bonds are intralayer bonds. The remaining hydrogen bonds form very weak interlayer bonds of the type P₃-O₁₀-H---O₁₀'' and P₃-O₁₀''-H---O₁₀'. These hydrogen bonds form a zigzag array running parallel to the *b* axis. All of the interlayer hydrogen bonds involve P₃-type phosphate groups whereas the P₂ type are all involved in intralayer hydrogen bonding.

An alternative hydrogen-bonding scheme involves the assumption that the shortest hydrogen bonds originate from the two P-OH groups with water as the acceptor atom for both these bonds. Thus, the bonding

scheme would be P₂-O₇-H---O₁₂ and P₃-O₁₀''-H---O₁₂. One hydrogen atom of the water forms a very long bond to O₇ and the other hydrogen forms no hydrogen bond in this scheme. There are no interlayer hydrogen bonds so that the layers are held together by van der Waals forces.

Both hydrogen-bonding schemes are speculative at present and the correct choice must await a neutron diffraction study. However, in either case the forces between layers are quite weak and one might expect cleavage parallel to the layers (001). This is in fact observed.

The zeolitic nature of the crystals and the weak forces between layers provide a basis for explaining their sieving behavior and the ready expansion of the lattice during exchange.³ These aspects of the exchange mechanism will be discussed in a forthcoming paper.

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The Crystal Structure of Rhenium(VII) Oxide

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The crystal structure of rhenium(VII) oxide, Re₂O₇, has been determined from three-dimensional single-crystal X-ray data collected by film techniques. The least-squares refinement converges to a conventional *R* factor of 0.053 for 1126 nonzero reflections. The compound crystallizes in the orthorhombic system, space group P2₁2₁2₁, with *a* = 12.508, *b* = 15.196, *c* = 5.448 Å, and *V* = 1035.5 Å³. The unit cell contains 8 formula units. Measured and calculated densities are 6.14 and 6.214 g/cm³, respectively. The structure consists of strongly distorted ReO₆ octahedra and fairly regular ReO₄ tetrahedra which are connected through corners to form polymeric double layers in the *ac* plane. The double layers have only van der Waals contacts to neighboring ones. Parallel zigzag chains of octahedra, every other chain being part of the upper or lower half of the double layer, are linked through corners of the tetrahedra in such a way that rings (o-t-o-t) (o, octahedron, t, tetrahedron) within the layers are formed. The octahedra contain three short (1.65–1.75 Å) and three long (2.06–2.16 Å) Re–O bonds, while in the tetrahedra the bond distances range from 1.68 to 1.80 Å. The structure is one of the relatively few known examples where metal atoms of the same oxidation number occur with coordination numbers 4 and 6 in the same structure. From the crystal structure the mechanisms of evaporation (to form O₃ReOReO₃ molecules) and of hydrolysis (where Re₂O₇(OH₂)₂ is formed) can be derived. In its structural and bond properties, rhenium(VII) oxide represents an intermediate between the polymeric oxides MoO₃ and WO₃ and the more covalently bonded OsO₄, which forms a molecular structure.

Introduction

The very simple crystal structure of rhenium(VI) oxide¹ has been known for a long time and has become the model for a basic structure type. Structure and bonding of solid rhenium(VII) oxide, on the other hand, have been completely unknown up to now and have been the object of speculation and controversy.

A mass spectroscopic investigation² and vapor density measurements³ have shown that the oxide in the gas phase consists of Re₂O₇ molecules which have to be formulated as O₃Re–O–ReO₃ with tetrahedrally coordinated rhenium. From the physical and spectroscopic properties of the compound there were

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