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## The Crystal Structure of Cs<sub>4</sub>Mg<sub>3</sub>F<sub>10</sub><sup>1</sup>

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The compound Cs<sub>4</sub>Mg<sub>3</sub>F<sub>10</sub> crystallizes in space group Cmca with  $a_0 = 6.1333$  (3),  $b_0 = 14.561$  (2), and  $c_0 = 13.653$  (1) Å. The observed density is  $4.0 \pm 0.4$  g/cm<sup>3</sup> and the calculated density is 4.3276 g/cm<sup>3</sup>;  $Z = 4$ . The structure consists of corrugated sheets of linked MgF<sub>6</sub> octahedra parallel to (010) with the sheets held together by the Cs<sup>+</sup> ions. The basic unit within a corrugated sheet consists of three octahedra in which the central octahedron shares two opposing faces with the other two octahedra. The basic units are linked by the outside octahedra of each unit sharing corners with the outside octahedra of other basic units. The Mg<sup>2+</sup>-F<sup>-</sup> distances vary from 1.890 to 2.104 Å. The Cs<sup>+</sup>-F<sup>-</sup> distances vary from 2.940 to 3.471 Å.

### Introduction

The use of molten salts, in particular fluorides of the heavy metals in conjunction with alkali and alkaline earth fluorides, as working fluids in the molten salt breeder reactor program has led to an extensive investigation of the phase diagrams of such systems. As part of this program, the binary system CsF-MgF<sub>2</sub> is being investigated and it was observed that well-formed, euhedral crystals grew in an ingot of the composition 50 mol % CsF-50 mol % MgF<sub>2</sub>. X-Ray diffraction powder patterns of this mixture and optical petrographic examination of the ingot products showed the presence of MgF<sub>2</sub> and of an additional phase which was not the known tetragonal compound CsF·MgF<sub>2</sub>.<sup>2</sup> Single crystals of this unknown phase were selected from the ingot for crystal structure analysis in order to elucidate its stoichiometry and structural parameters.

### Experimental Section

Optical examination of a single crystal showed it to be biaxial with the indices  $n_x = 1.454$ ,  $n_z = 1.462$ , and  $2V = 50^\circ$ . Weissenberg and precession photographs were obtained from a well-formed, acicular crystal; the diagrams had diffraction symmetry mmm and extinctions were observed for  $hkl$ ,  $h + k = 2n + 1$ ;  $h0l$ ,  $l = 2n + 1$ ;  $h\bar{k}0$ ,  $h = 2n + 1$ . The space groups C2ca (41) and Cmca (64) are consistent with the diffraction data, and the preliminary lattice constants obtained from the photographs were  $a = 6.11$  Å,  $b = 14.58$  Å, and  $c = 13.64$  Å.

The density of the material was determined using a Berman balance. Several well-formed crystals with a combined weight of approximately 3.5 mg were used and from the weight loss in benzene the density was found to be  $4.0 \pm 0.4$  g/cm<sup>3</sup>. The number of formula weights consistent with this density for the compositions  $x\text{CsF} \cdot y\text{MgF}_2$  ( $x:y = 2:1, 3:2, 4:3, \text{ and } 5:4$ ) were calculated and it was apparent that the 2:1 and 4:3 compositions were likely stoichiometries consistent with the crystal system and space groups. The 4:3 composition was especially probable because it is close to the 50-50 mol % mixture. For either possibility the unit cell contains 16 Cs<sup>+</sup> which primarily would determine the scattered intensities. It was also observed that the  $h0l$  diffraction diagram had very strong diffraction spectra when  $l = 4n$  and the spectra were either absent or extremely faint for  $l \neq 4n$ . The decrease in the  $h0l$  intensities was clearly due to the decrease of the scattering factors of the atoms with  $(\sin \theta)/\lambda$ , indicating that at least the Cs<sup>+</sup> positions and probably several other atoms

have  $x$  parameters near 0 and that the  $z$  parameters have a relationship of  $z$  and  $1/4 - z$  for some of the crystallographically independent atoms.

A single crystal was selected, ground in a Bond grinder to an ellipsoidal shape, with axial dimensions  $0.136 \times 0.158 \times 0.104$  mm, and placed on a Picker four-circle automatic single-crystal diffractometer for the collection of three-dimensional X-ray diffraction intensities. Lattice constants were determined from a least-squares refinement of precise  $2\theta$  measurements from 32 Mo K $\alpha$  reflections having values between  $43$  and  $72^\circ$ , yielding  $a = 6.1333$  (3),  $b = 14.561$  (2), and  $c = 13.653$  (1) Å ( $\lambda$  0.70926 Å).<sup>3</sup> Diffraction intensities were collected to a value of  $2\theta = 80^\circ$ ; 2040 independent reflections were measured of which 1797 were greater than  $\sigma(F_o)$ . Integrated intensities were collected using unfiltered Mo K $\alpha$  radiation measured with a scintillation-counter detector. An angular scan range of  $1.35^\circ$  was step scanned in increments of  $0.01^\circ$  with 0.5-sec counts at each step, and the background was measured on each side of the peak for 20 sec.<sup>3</sup> The data were corrected for Lorentz-polarization effects and for absorption ( $\mu = 12.36$  mm<sup>-1</sup>). The equations

$$\sigma^2(F_o^2) = [T + B + 0.10(T - B)]^2 / A(Lp)^2$$

$$\sigma(F_o) = \sigma(F_o^2) / 2F_o$$

were used to estimate the variance for the structure factors where  $T$  is the total count,  $B$  is the background count,  $A$  is the absorption correction, and  $Lp$  is the Lorentz-polarization expression. A standard reflection, (008), was used throughout the data gathering to check on intensity drift. One measurement of the standard intensity was assigned unit value and all others were expressed as fractions of this value. The intensities obtained between two successive measurements of the standard reflection were multiplied by the appropriate fraction so that all measured intensities were referred to a single value of the (008) intensity.

### Structure Determination

The direct method for the determination of phases was used in the solution of the structure. A Wilson plot was constructed which provided an approximate scale factor and also indicated a centric distribution of intensities so that the space group Cmca was chosen. A program written by Dewar<sup>4</sup> making use of the symbolic addition method yielded a sufficient number of signed  $E(hkl)$ 's so that the first calculated  $E$  map displayed peaks which could be identified as Cs<sup>+</sup>, and they were in the 8f positions of the space group with the  $z$

(1) Research sponsored by the U. S. Atomic Energy Commission under contract with the Union Carbide Corp.

(2) G. D. Brunton, H. Insley, T. N. McVay, and R. E. Thoma, ORNL Report 3761, Oak Ridge National Laboratory, Oak Ridge, Tenn., 1965.

(3) W. R. Busing, R. D. Ellison, H. A. Levy, S. P. King, and R. T. Roseberry, Report ORNL 4143, Oak Ridge National Laboratory, Oak Ridge, Tenn., 1968.

(4) R. Dewar and A. Stone, University of Chicago, personal communication, 1966.

TABLE I  
 ATOMIC PARAMETERS AND THEIR STANDARD DEVIATIONS IN PARENTHESES ( $\times 10^4$ ) FOR  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$ <sup>a</sup>

Atom	<i>x</i>	<i>y</i>	<i>z</i>	$\beta_{11}$	$\beta_{22}$	$\beta_{33}$	$\beta_{12}$	$\beta_{13}$	$\beta_{23}$
Cs <sub>1</sub>	0	2542 (0.4)	3939 (0.4)	86 (1)	15 (0.2)	28 (0.3)	0	0	-3 (0.2)
Cs <sub>2</sub>	0	4482 (0.3)	1389 (0.3)	72 (0.1)	15 (0.2)	16 (0.2)	0	0	-0.8 (0.1)
Mg <sub>1</sub>	0	1170 (2)	1575 (2)	57 (5)	8 (0.8)	12 (1)	0	0	0 (0.7)
Mg <sub>2</sub>	0	0	0	48 (7)	10 (1)	11 (1)	0	0	-0.3 (1)
F <sub>1</sub>	2100 (6)	958 (2)	403 (2)	61 (7)	19 (1)	17 (1)	-6 (2)	9 (3)	-2 (1)
F <sub>2</sub>	2500	1098 (4)	2500	74 (10)	19 (2)	19 (2)	0	-15 (4)	0
F <sub>3</sub>	0	4732 (4)	3575 (3)	107 (12)	15 (2)	16 (2)	0	0	3 (2)
F <sub>4</sub>	0	2465 (4)	1482 (5)	126 (15)	14 (2)	38 (4)	0	0	7 (2)

<sup>a</sup> The temperature factor is  $\exp[-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl)]$ .

parameter relationships as expected on the basis of the  $h0l$  diffraction intensities. Additional peaks consistent with  $\text{Mg}^{2+}$  and  $\text{F}^-$  were present but it was difficult to decide the particular ionic species from the peak heights. From the  $\text{Cs}^+$  positions of the  $E$  map, a set of structure factors was calculated, and the signs from this calculation were used in the construction of a three-dimensional electron density map which then yielded peaks that could be unequivocally assigned to  $\text{Mg}^{2+}$  and  $\text{F}^-$  and fixed the stoichiometry at the 4:3 composition. The structure was refined by a least-squares procedure using the XFLS program, a modification of ORFLS,<sup>5</sup> the structure factors were weighted by  $1/\sigma^2$ , and the final  $R$  with anisotropic temperature factors is 0.075 and 0.0785 for weighted  $R$ . The value of  $R$  for the 1797 reflections greater than  $\sigma(F_o)$  is 0.0617, and weighted  $R$  is 0.0782. The standard deviation of an observation of unit weight is 0.988. The scattering factors used for all atoms are those published by Cromer and Waber;<sup>6</sup> the  $\text{Cs}^+$  scattering was corrected for the real ( $\Delta f'$ ) and imaginary contributions ( $\Delta f''$ ).<sup>7</sup> The final atomic parameters and anisotropic temperature factors are shown in Table I, and the set of calculated and observed structure factors scaled to absolute values is listed in Table II. A three-dimensional difference electron density function was calculated with this last set of parameters, and no physically significant peaks were observed. The largest height of the random peaks was  $\pm 4 \text{ e}^-/\text{\AA}^3$ .

After the stoichiometry had been determined from the structure, ingots with the  $\text{CsF}:\text{MgF}_2$  compositions of 1:1, 2:1, and 4:3 were prepared. The 1:1 ingot again was a mixture of  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$  and  $\text{MgF}_2$ . The 2:1 ingot was a mixture of  $\text{CsF}$  and  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$ . The 4:3 ingot was almost all  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$  but not free enough from other phases to warrant chemical analysis. The exact melting behavior of  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$  has not been determined and the studies of the  $\text{CsF}-\text{MgF}_2$  phase equilibria are incomplete.

### Discussion

The structure of  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$  consists of corrugated sheets of linked  $\text{MgF}_6$  octahedra parallel to (010) and is illustrated in Figure 1. The basic unit within a corrugated sheet consists of three octahedra where the central

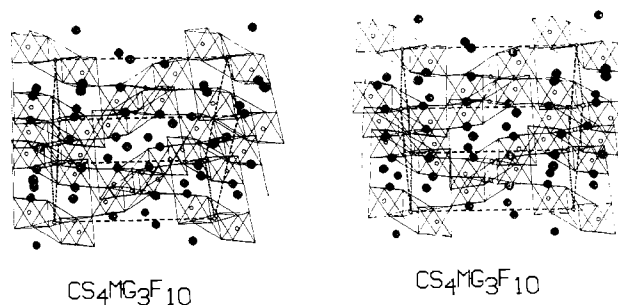


Figure 1.—A stereographic drawing of the linkages of octahedra in  $\text{Cs}_4\text{Mg}_3\text{F}_{10}$ . The unit cell is outlined by the dashed lines; the vertical axis is  $b$  and the horizontal axis is  $c$ . The structure is seen after a  $50^\circ$  rotation around the  $c$  axis.

octahedron shares two opposing faces with two other octahedra. Each end octahedron within this unit of three thus shares one face with the central octahedron and corners with two octahedra which in turn are part of two separate units of three. The sheets are held together by the  $\text{Cs}^+$ , and the pronounced cleavage perpendicular to the (010) reflects the weaker bonding. The  $\text{Mg}^{2+}$  ion at the origin is at the center of the octahedron which shares two faces, and, because of the symmetrical sharing of these faces, the six  $\text{Mg}-\text{F}$  bond distances are the same, Table III. The  $\text{F}^--\text{F}^-$  contacts forming the edges of this octahedron are also equal, and the  $\text{F}^--\text{Mg}^{2+}-\text{F}^-$  angles deviate only  $\pm 8^\circ$  from a right angle. (The crystallographically independent ions are labeled with subscripts, e.g.,  $\text{Mg}_1$ ,  $\text{Mg}_2$ ,  $\text{F}_1$ ,  $\text{F}_2$ ,  $\text{F}_3$ , etc., Table I.) The  $\text{Mg}_1^{2+}$  ion occupies the octahedron which shares one face and two corners. This octahedron is more irregular as can be seen from the  $\text{Mg}_1-\text{F}$  distances which vary from 1.89 to 2.10  $\text{\AA}$ , Table III. The long distances  $\text{Mg}_1-\text{F}_1$  and  $\text{Mg}_1-\text{F}_3$  are to  $\text{F}^-$  in the shared face, the shorter distances  $\text{Mg}_1-\text{F}_2$  are to  $\text{F}^-$  ions which participate in corner sharing between two octahedra, and the shortest distance is to the  $\text{F}_4^-$  which is unshared. The  $\text{F}-\text{Mg}_1-\text{F}$  angles are also quite distorted with values ranging between  $76.5$  and  $101^\circ$ . The octahedron in  $\alpha\text{-Al}_2\text{O}_3$ , where one octahedron shares one face, one edge, and one corner, shows similar irregularities.<sup>8</sup>

The sum of the ionic radii for  $\text{Cs}^+$  and  $\text{F}^-$  is 3.05  $\text{\AA}$ , and it is seen that the  $\text{Cs}^+-\text{F}^-$  distances vary from 3.05 to almost 3.5  $\text{\AA}$ . The environment around  $\text{Cs}_1$  is shown in Figure 2a. The 11  $\text{F}^-$  ions form a nearly

(5) W. R. Busing, K. O. Martin, and H. A. Levy, Report ORNL-TM-305, Oak Ridge National Laboratory, Oak Ridge, Tenn., 1962.

(6) D. T. Cromer and J. T. Waber, *Acta Cryst.*, **18**, 104 (1965).

(7) "International Tables for X-Ray Crystallography," Vol. III, The Kynoch Press, Birmingham, England, 1962.

(8) R. E. Newnham and Y. M. De Haan, *Z. Krist.*, **117**, 235 (1962).



TABLE III<sup>a</sup>  
 BOND DISTANCES AND ANGLES IN Cs<sub>4</sub>Mg<sub>3</sub>F<sub>10</sub>

Distances, Å			
Cs <sub>1</sub> -F <sub>1</sub> (2)	3.048 (3)	Cs <sub>2</sub> -F <sub>3</sub>	3.007 (5)
Cs <sub>1</sub> -F <sub>4</sub> (2)	3.122 (1)	Cs <sub>2</sub> -F <sub>3</sub> (2)	3.089 (1)
Cs <sub>1</sub> -F <sub>3</sub>	3.228 (5)	Cs <sub>2</sub> -F <sub>1</sub> (2)	3.092 (3)
Cs <sub>1</sub> -F <sub>1</sub> (2)	3.229 (4)	Cs <sub>2</sub> -F <sub>1</sub> (2)	3.098 (4)
Cs <sub>1</sub> -F <sub>2</sub> (2)	3.261 (3)	Cs <sub>2</sub> -F <sub>2</sub> (2)	3.192 (4)
Cs <sub>1</sub> -F <sub>4</sub>	3.357 (6)	F <sub>1</sub> -F <sub>1</sub>	2.575 (7)
Cs <sub>1</sub> -F <sub>4</sub>	3.471 (6)	F <sub>1</sub> -F <sub>3</sub>	2.606 (6)
Mg <sub>1</sub> -F <sub>4</sub>	1.890 (6)	F <sub>1</sub> -F <sub>2</sub>	2.881 (3)
Mg <sub>1</sub> -F <sub>2</sub> (2)	1.989 (2)	F <sub>1</sub> -F <sub>4</sub>	2.940 (6)
Mg <sub>1</sub> -F <sub>1</sub> (2)	2.077 (4)	F <sub>1</sub> -F <sub>3</sub>	2.982 (5)
Mg <sub>1</sub> -F <sub>3</sub>	2.104 (6)	F <sub>1</sub> -F <sub>1</sub>	2.999 (7)
Mg <sub>2</sub> -F <sub>1</sub> (4)	1.977 (3)	F <sub>2</sub> -F <sub>4</sub>	2.872 (6)
Mg <sub>2</sub> -F <sub>3</sub> (2)	1.984 (5)	F <sub>2</sub> -F <sub>3</sub>	2.909 (6)
Cs <sub>2</sub> -F <sub>4</sub>	2.940 (6)	F <sub>2</sub> -F <sub>2</sub>	3.067 (2)

Angles, Deg			
F <sub>4</sub> -Mg <sub>1</sub> -F <sub>2</sub> (2)	95.9	F <sub>2</sub> -Mg <sub>1</sub> -F <sub>1</sub>	163.7
F <sub>4</sub> -Mg <sub>1</sub> -F <sub>1</sub> (2)	95.2	F <sub>1</sub> -Mg <sub>2</sub> -F <sub>1</sub> (2)	97.8
F <sub>2</sub> -Mg <sub>1</sub> -F <sub>2</sub>	101.0	F <sub>1</sub> -Mg <sub>2</sub> -F <sub>1</sub> (2)	82.2
F <sub>2</sub> -Mg <sub>1</sub> -F <sub>1</sub> (2)	89.8	F <sub>1</sub> -Mg <sub>2</sub> -F <sub>3</sub> (4)	97.8
F <sub>2</sub> -Mg <sub>1</sub> -F <sub>3</sub> (2)	90.9	F <sub>1</sub> -Mg <sub>2</sub> -F <sub>3</sub> (4)	82.2
F <sub>1</sub> -Mg <sub>1</sub> -F <sub>1</sub>	77.3	F <sub>3</sub> -Mg <sub>2</sub> -F <sub>3</sub>	180.0
F <sub>1</sub> -Mg <sub>1</sub> -F <sub>3</sub> (2)	76.5	F <sub>1</sub> -Mg <sub>2</sub> -F <sub>1</sub>	180.0
F <sub>4</sub> -Mg <sub>1</sub> -F <sub>3</sub>	169.2		

<sup>a</sup> The number in parentheses following the atom designations denotes the number of bonds with such a value around a cation. The number in parentheses following the distances is the standard deviation,  $10^3\sigma$ . Symmetry-equivalent atoms are not distinguished separately in the table.

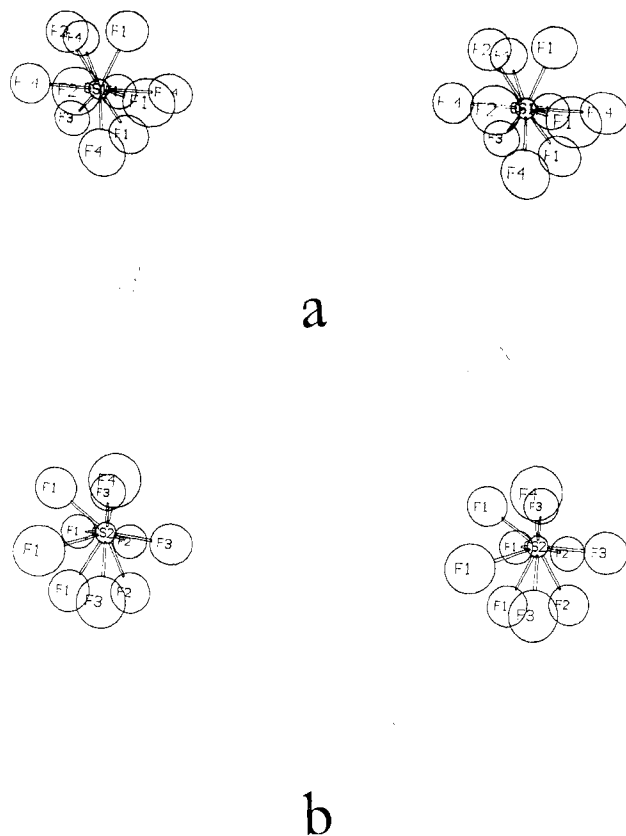


Figure 2.—(a) The coordination of F<sup>-</sup> around Cs<sub>1</sub><sup>+</sup>. (b) The coordination of F<sup>-</sup> around Cs<sub>2</sub><sup>+</sup>. The orientation is the same as in Figure 1.

atoms, are oriented with one of the principal axes along one of the crystallographic axes, Table IV. The Cs<sub>2</sub>, Mg<sub>1</sub>, and Mg<sub>2</sub> ions are vibrating isotropically, and since

 TABLE IV  
 THE PRINCIPAL AXES, RMS THERMAL  
 DISPLACEMENTS, AND ORIENTATIONS<sup>a</sup>

Atom	<i>r</i>	$\mu(r)$ , Å	$c_{r,a}$	$c_{r,b}$	$c_{r,c}$
Cs <sub>1</sub>	1	0.124 (1)	0	-0.966	-0.258
	2	0.128 (1)	-1	0	0
	3	0.164 (1)	0	0.258	-0.966
Cs <sub>2</sub>	1	0.117 (1)	-1	0	0
	2	0.120 (1)	0	-0.438	-0.899
	3	0.129 (1)	0	-0.899	0.438
Mg <sub>1</sub>	1	0.093 (5)	0	-1.000	-0.004
	2	0.104 (5)	-1	0	0
	3	0.106 (5)	0	0.004	-1.000
Mg <sub>2</sub>	1	0.096 (7)	1	0	0
	2	0.102 (7)	0	0.752	0.659
	3	0.104 (7)	0	-0.659	0.752
F <sub>1</sub>	1	0.096 (7)	0.882	0.133	-0.453
	2	0.129 (5)	-0.301	-0.581	-0.757
	3	0.151 (5)	0.363	-0.803	0.472
F <sub>2</sub>	1	0.097 (9)	0.805	0	0.593
	2	0.141 (7)	0	1	0
	3	0.151 (7)	-0.593	0	0.80E
F <sub>3</sub>	1	0.111 (8)	0	-0.675	0.73E
	2	0.139 (8)	0	0.738	0.675
	3	0.143 (8)	1	0	0
F <sub>4</sub>	1	0.111 (9)	0	0.957	-0.288
	2	0.155 (9)	1	0	0
	3	0.195 (9)	0	0.288	0.957

<sup>a</sup>  $\mu(r)$  is the rms displacement along axis *r*, and  $c_{r,j}$  is the direction cosine between axis *r* and the *j*th cartesian axis.

the latter two are confined to octahedral interstices, such isotropy can be expected. The Cs<sub>1</sub>, F<sub>2</sub>, F<sub>3</sub>, and F<sub>4</sub> vibrations can be described by ellipsoids of revolution. The largest vibration direction of F<sub>4</sub> is perpendicular to the Mg<sub>1</sub>-F<sub>4</sub> bond direction, which is physically realistic since this ion is not shared between octahedra.