# **Optical Transitions of Ho3+ in a Lead Silicate Glass**

### FERNANDO FERMI

*Dipartimento di Fisica, Universitd di Parma, Viale delle Scienze, 43100 Parma (Italy)* 

### **GIANLUIGI INGLETTO\***

*Istituto di Biochimica, Facoltà di Medicina Veterinaria, Universitd di Parma, Vti de1 Taglio, 43100 Parma (Italy)* 

### **CARLA ASCHIERI**

*MONTEDIPE, Via deN'Elettricitd 41, 30175 Porto Marghera ( VEl (Italy)* 

#### and **MARCO BETTINELLI**

*Dipartimento di Chimica Inorganica, Metallorganica ed*  Analitica, Università di Padova, Via Loredan 4, *35131 Padua (Italy)* 

(Received April 7.1989)

The presence of a high concentration of the heavy and easily polarized  $\overline{Pb}^{2+}$  ion has been found to perturb in a significant way the optical properties of glasses doped with trivalent lanthanide ions  $(Ln^{3+})$ . Lead metaphosphate glasses containing  $Nd^{3+}$  exhibit extremes in several of their spectroscopic properties [l], and silicate glasses of molar composition 38% PbO-62%  $SiO<sub>2</sub>$  doped with Eu<sup>3+</sup> show a strong dependence of the intensities of the  $4f^6 \rightarrow 4f^6$  transitions on the energy of the upper levels and an unusual low luminescence quantum yield [2]. In this note we report the optical properties of the same 38% PbO-62%  $SiO<sub>2</sub>$  glass doped with Ho<sup>3+</sup>. The stimulated emission cross sections of the potential laser transitions  $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{8}$  and  ${}^{5}F_{5} \rightarrow {}^{5}I_{8}$  are relatively high due to a narrow effective linewidth, caused by the presence of  $Pb^{2+}$  as network modifier.

# **Experimental**

The molar composition of the lead silicate glass under investigation was  $37.5\%$  PbO,  $61.5\%$  SiO<sub>2</sub> and  $1.0\%$  Ho<sub>2</sub>O<sub>3</sub>. Appropriate quantities of PbO and  $SiO<sub>2</sub>$  (both Carlo Erba RPE) and  $Ho<sub>2</sub>O<sub>3</sub>$  (Riedel De Haen reagent grade) were mixed in Pt crucibles. Several batches were melted and annealed following a procedure previously reported [2].

The samples were then cut to a 1.5 mm thickness and polished for optical measurements. Absorption and luminescence spectra were measured at 293 and 80 K with the equipment previously described [2]. Additional luminescence spectra were recorded at 293 K using a Perkin-Elmer 65040 spectrofluori**TABLE** 1. Oscillator strengths P in the absorption spectrum of the 37.5% PbO-61.5%  $SiO_2-1.0%$  Ho<sub>2</sub>O<sub>3</sub> glass at 293 K and intensity parameters  $\Omega_{\lambda}$ . The ground level of Ho<sup>3+</sup> is  $5I<sub>8</sub>$ 



 $\Omega_2 = (5.2 \pm 0.2) \times 10^{-20} \text{ cm}^2$ ;  $\Omega_4 = (1.8 \pm 0.3) \times 10^{-20} \text{ cm}^2$ ;  $\Omega_6 = (1.2 \pm 0.5) \times 10^{-20}$  cm<sup>2</sup>; r.m.s. = 6.01  $\times 10^{-7}$ .

meter. The excitation wavelength was always  $\lambda = 446$ nm. The luminescence spectra were not corrected for the response of the photomultiplier for  $\lambda > 700$ nm.

The refractive index  $n_d = 1.76$  of the glass was measured using standard techniques.

# **Results and Discussion**

The absorption spectra at 293 K of the lead silicate glass doped with  $Ho^{3+}$  are composed of ten  $4f^{10} \rightarrow 4f^{10}$  transitions in the infrared and the visible region, and a steeply rising edge starting at about 330 nm. The bands are clearly inhomogeneously broadened; their positions, experimental oscillator strengths and assignments are reported in Table 1. At 80 K no further resolution of the bands is obtained, and the oscillator strengths of the various transitions show differences up to  $\pm 25\%$  with respect to 293 K. The Judd-Ofelt phenomenological parameters  $\Omega_{\lambda}$  ( $\lambda = 2$ , 4 and 6) were calculated with a least-squares fitting procedure from the oscillator strengths at 293 K, using the reduced matrix elements calculated by Weber *et al.* [3]. The results are reported in Table 1, together with the calculated oscillator strengths. The root mean square deviation (r.m.s.) is 15.3% of the average oscillator strength.

The values of the intensity parameters appear to be the lowest found to date for  $Ho^{3+}$  in oxide glasses [4-6]. Weber *et al.* [1] have found exceptionally low  $\Omega_{\lambda}$  intensity parameters for Nd<sup>3+</sup> in phosphate

0020-1693/89/\$3.50 **Delet Sequoia/Printed in Switzerland** CO20-1693/89/\$3.50

<sup>\*</sup>Author to whom correspondence should be addressed.





glasses containing high concentrations of PbO (corresponding to the metaphosphate composition or higher in PbO) and attributed this behaviour to the high polarizability of  $Pb^{2+}$  and the directional nature of the Pb-0 bond. This is clearly possible also for the glass under investigation even if these considerations are not consistent with our results for  $Eu^{3+}$  in the same lead silicate matrix [2]. In this last case the intensity parameters depend on the energy of the upper levels involved in the transitions and, when expressed as  $T_{\lambda}$ , are higher than the corresponding values for a lead-free silicate glass [7]. This peculiar behaviour of  $Eu<sup>3+</sup>$  in the lead silicate matrix has been attributed to a strong mixing of the  $4f^6$  states with low-lying states of opposite parity, probably of charge-transfer nature [2].

The luminescence spectrum at 293 K of the lead silicate glass doped with  $Ho^{3+}$ , after excitation in the  ${}^5G_6$  level, is reported in Fig. 1. The spectrum does not change at 80 K. The three bands are assigned to the transitions  $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{8}, {}^{5}F_{5} \rightarrow {}^{5}I_{8}$ and  $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{7}$  in increasing wavelength order, if the thermal population of  ${}^{5}F_{4}$  is taken into account [4].

The radiative transition probabilities  $A$  and the branching ratios  $\beta$  for emission from the excited levels  ${}^{5}F_{4}$ ,  ${}^{5}S_{2}$  and  ${}^{5}F_{5}$ , calculated from the  $\Omega_{\lambda}$ intensity parameters, are reported in Table 2 for the lead silicate glass doped with  $Ho^{3+}$ . The branching ratios are very similar to the values determined for other oxide glasses  $[4-6]$  and reasonably similar to those for fluoride glasses  $[8-10]$  and are not strongly influenced by the different values of the intensity parameters.

From the calculated radiative emission probabilities and the luminescence spectrum, the peak stimulated emission cross sections  $\sigma_p$  [11, 12] have been evaluated for the potential laser transition  $({}^{5}F_{4}, {}^{5}S_{2}) \rightarrow {}^{5}I_{8}$  and  ${}^{5}F_{5} \rightarrow {}^{5}I_{8}$ . The level  ${}^{5}F_{4}$  lies roughly 100 cm<sup>-1</sup> above  $5S_2$  [13] and laser emission is possible from both levels [14]. In Table 3 the values of  $\sigma_p$  are reported together with the peak

TABLE 2. Calculated radiative transition probabilities *A,*  branching ratios  $\beta$  and radiative lifetimes  $\tau_R$  for emission from excited levels of  $Ho^{3+}$  in the 37.5% PbO-61.5% SiO<sub>2</sub>- $1.0\%$  Ho<sub>2</sub>O<sub>3</sub> glass

Initial state	Final state	$\boldsymbol{A}$ $(s^{-1})$	β	$\tau_{\bf R}$ (ms)
${}^{5}F_4$	$\mathbf{^{5}I_{8}}$	3383.0	0.801	
	$\mathbf{^{5}I}_{7}$	390.4	0.092	
	$^5{\rm I}_6$	279.3	0.066	
	$\boldsymbol{^{5}\text{I}_5}$	137.7	0.033	
	$^{5}{\rm I}_4$	23.2	0.005	
	$^{5}{\rm F}_{5}$	13.7	0.003	
				0.24
$^{5}\mathrm{S}_2$	$^5\mathrm{I}_8$	1311.0	0.531	
	$5_{I_7}$	901.6	0.365	
	$^{5}{\rm I}_6$	169.3	0.068	
	$\mathbf{^{5}I_{5}}$	42.1	0.017	
	$\mathbf{^{5}I_{4}}$	46.0	0.019	
	${}^{5}F_5$	0.5	0.001	
				0.40
${}^{5}F_5$	$\boldsymbol{^{5}\text{I}}_{8}$	1816.0	0.763	
	5I <sub>7</sub>	455.7	0.191	
	$\mathbf{^{5}I_{6}}$	101.3	0.043	
	$\boldsymbol{^{5}\text{I}_5}$	8.1	0.003	
	$\mathbf{^{5}I_{4}}$	0.1	0.001	
				0.42

TABLE 3. Peak wavelength  $\lambda_p$ , effective linewidth  $\Delta \lambda_{eff}$ and stimulated emission cross section  $\sigma_p$  of some laser transitions of Ho<sup>3+</sup> in the 37.5% PbO-61.5% SiO<sub>2</sub>-1.0% Ho<sub>2</sub>O<sub>3</sub> glass at 293 K



wavelengths  $\lambda_p$  and the effective linewidths  $\Delta \lambda_{eff}$ . The values of  $\sigma_p$  are relatively high, despite the low intensity parameters, and are comparable with the highest values for the glasses in the 48%  $ZrF_4-23%$ BaF<sub>2</sub>-8% AlF<sub>3</sub>-20% RF-1% HoF<sub>3</sub> series (RF = LiF-NaF or NaF-KF pair) [10]. The effective linewidth appears in the denominator of the expression for  $\sigma_{\rm p}$  and the low values of  $\Delta\lambda_{\rm eff}$ , due to the reduced ligand field around the  $Ho<sup>3+</sup>$  ion caused by the strong and covalent  $Pb - O$  bonds [1], are important in determining the cross sections.

As a conclusion, the 37.5% PbO-61.5%  $SiO<sub>2</sub>$ - $1.0\%$  Ho<sub>2</sub>O<sub>3</sub> glass shows some optical properties similar to those of fluoride glasses and is characterized by low values of the Judd-Ofelt parameters, by narrow emission linewidths and by relatively high stimulated emission cross sections of the  $({}^{5}F_{4}$ ,  ${}^{5}S_{2}) \rightarrow {}^{5}I_{8}$  and  ${}^{5}F_{5} \rightarrow {}^{5}I_{8}$  potential laser transitions.

The high value of the refractive index is also effective in increasing the peak cross sections for the aforementioned transitions.

# Acknowledgements

The authors thank Professor N. Sabbatini (Dip. di Chimica 'G. Ciamician', Universita di Bologna) for the use of the spectrofluorimeter and Mr F. De Zuane (ICTR-Area di Ricerca di Padova de1 CNR) for technical assistance.

### References

1 M. J. Weber, L. A. Boatner and B. C. Sales, *J. Non-Cryst. Solids, 74* (1985) 167.

- 2 F. Fermi, L. Tellini, G. Ingletto, A. Vinattieri and M. Bettinelli, *Inorg. Chim. Acta.* 150 (1988) 141.
- *3 M.* J. Weber, B. H. Matsinger, V. L. Donlan and G. T. Surratt, *J. Chem. Phys., 57* (1972) *562.*
- *4* R. Reisfeld and J. Hormadaly, *J. Chem. Phys., 64* (1976) *3207.*
- *5* R. Reisfeld, J. Hormadaly and A. Muranevich, *Chem. Phys. Lett., 38 (1976) 188.*
- *6* R. Reisfeld, J. Hormadaly and A. Muranevich, *J. Non-Cryst. Solids, 29 (1978) 323.*
- *7* R. Reisfeld,Struct. *Bonding (Berlin), 13 (1973) 53.*
- *8* K. Tanimura, M. D. Shin& W. A: Sibley, M. G. Drexhage and R. N. Brown, *Phys. Rev. B, 30* (1984) *2429.*
- 9 R. Reisfeld, M. Eyal, E. Greenberg and C. K. Jørgensen, *Chem. Phys. Lett., 118 (1985) 25.*
- 10 S. Buddhudu and F. J. Brvant. *Suectrochim. Acta. Part*  A, 44 (1988) 1381.
- 11 M. J. Weber, D. C. Ziegler and C. A. Angell, *J. Appl. Phys., 53* (1982) 4344.
- 12 R. Reisfeld, *Inorg. Chim. Acta, 140* (1987) 345.
- 13 C. K. Jdrgensen, R. Reisfeld and M. Eyal, *J. Electrochem. Sot., 133* (1986) 1961.
- 14 R. Reisfeld and Y. Kalisky, Chem. *Phys. Lett., 75 (1980) 443.*