

# Thermodynamic Properties and Third-Law Cycle for Malononitrile

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The heat capacity of malononitrile,  $\text{CH}_2(\text{CN})_2$ , has been determined by adiabatic calorimetry from 5° to 320° K. A gradual transition occurs in the stable form at 260.3° K. with an entropy increment of 1.15 cal. per mole °K. The compound melts at 304.9° K. with an entropy of melting of 8.46 cal. per mole °K. Low-temperature heat capacities were also made on the metastable crystal obtained by quenching the stable form through the 260.3° K. anomaly. This phase has a gradual transformation which takes place over the 137° to 151° K. with an excess entropy of 0.05 cal. per mole °K. The heat capacity ( $C_p$ ), entropy ( $S^\circ$ ), and Gibbs energy function [ $-(G^\circ - H_0^\circ)/T$ ] at 298.15° K. are 26.36, 31.30, and 15.34 cal. per mole °K., respectively. The approximate accord of the experimental entropy of gaseous malononitrile at 298.15° K. ( $68.5 \pm 0.4$  cal. per mole °K.) with the value (69.02) calculated from spectroscopic data indicates the absence of disorder in the crystal at 0° K.

THE characteristic features of plastic crystals as defined by Timmermans (7) are typical in systems composed of molecules of high symmetry. The extent to which these properties pertain to systems of less symmetrical molecules is also of considerable interest. Of the first three dinitriles in the series  $\text{NC}(\text{CH}_2)_n\text{CN}$ , only succinonitrile ( $n = 2$ ) has been shown to possess a plastic crystalline phase (12); glutaronitrile ( $n = 3$ ) shows a normal entropy of melting (2). The first member of the series—i.e., malononitrile—is solid at room temperature, and its entropy of melting was determined by freezing-point depression as 7.9 cal. per mole °K. (8). Since such values are often not reliable, a deeper insight into the phase behavior and molecular freedom of malononitrile was sought by direct calorimetric measurements.

## EXPERIMENTAL

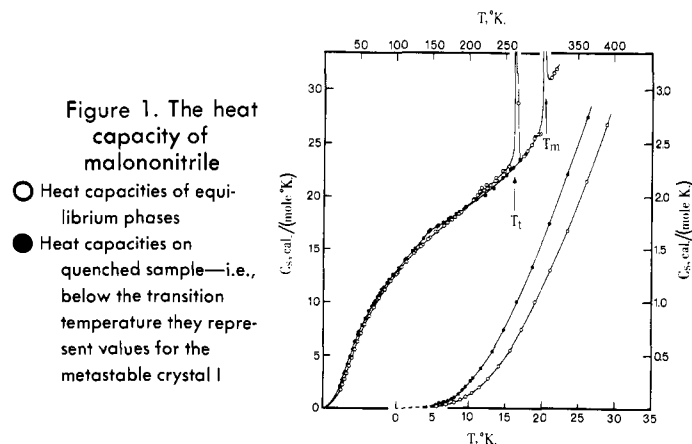
The commercial malononitrile sample was purified by 56 passes in an automatic multiple-zone apparatus. The 76.350 grams (in vacuo), degassed, completely colorless, transparent molten sample was transferred into the calorimeter under a helium atmosphere. Helium gas (81.5 torr) was added to facilitate thermal equilibration in the sample space. The purity of the sample as determined by fractional melting will be discussed later; a density of 1.045 grams per cc. was employed. Measurements were made in the gold-plated calorimeter (laboratory designation W-24A, fabricated from oxygen-free, high-conductivity copper) within the Mark III cryostat (9). The quasi-adiabatic technique was used (10) with manual shield control below 80° K. and with three channels of automatic electronic shield control to maintain adiabaticity to within a millidegree above that temperature. All measurements of time, temperature, potential, resistance, and mass were referred to standardizations or calibrations of the National Bureau of Standards. Between 70 and 80% of the total heat capacity was provided by the sample.

## RESULTS AND DISCUSSION

**Thermal Behavior.** Experimental heat capacities plotted in Figure 1 are also presented in chronological order in Table I so that the temperature increments employed can usually be deduced by differences in the mean temperature of adjacent determinations. The data are given in terms of the defined thermochemical calorie equal to 4.1840

joules and an ice point of 273.15° K. The probable error in the heat capacity data is considered to be about 2% up to 10° K., 0.3% at 30° K., and less than 0.2% above 100° K. Of six series of runs made through the transition region, only three corresponded to complete conversion. Three sets of heat capacity measurements were made below 90° K. The close agreement between the data of the first set (series II through V) with those on undercooled crystal I of the third set indicated that the former measurements were also made on the metastable (undercooled crystal I) form. The second set (series XIV through XVI) was preceded by complete transformation of crystal I to crystal II and therefore represents values for the stable form. The third set (series XVIII through XX) was on the undercooled form of crystal I obtained by quenching—i.e., reducing the calorimeter temperature to 140° K. within an hour. This was achieved by the expedient of temporarily breaking the vacuum in the cryostat with 1 torr helium gas.

Because the conversion of crystal I to crystal II ( $T_c = 260.3^\circ \text{K.}$ ) is very slow under ordinary conditions, rapid cooling ( $>14^\circ \text{K. per hour}$ ) of crystal I from 300° to 140° K. permits undercooling to still lower temperatures and consequently allows heat capacity determinations on the undercooled (metastable) crystal I phase to be made. Even upon relatively slow cooling of the sample ( $5^\circ \text{K. per hour}$ ) to 140° K. and holding it at this temperature for 2 days, crystal II does not appear. However, the conversion is initiated upon heating the sample to 190° K.; holding it at this temperature nearly 4 days provides complete conversion. The heat evolved during the first 3 hours of the transformation of undercooled crystal I to crystal II (about 95%



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Table II. Enthalpy of Phase Changes for Malononitrile<sup>a</sup>

| Designation                 | TRANSITION     |                     | MELTING                     |                |                     |
|-----------------------------|----------------|---------------------|-----------------------------|----------------|---------------------|
|                             | Number of Runs | $H_{280} - H_{240}$ | Designation                 | Number of Runs | $H_{315} - H_{295}$ |
| Series VII                  | 6              | (1204) <sup>b</sup> | Run A                       | 1              | 3156                |
| Series X                    | 9              | 1218                | Run B                       | 1              | 3157                |
| Run C                       | 2              | 1216                | Run C                       | 4              | (3109) <sup>c</sup> |
| Run D                       | 3              | 1220                |                             |                |                     |
| Run E                       | 3              | (1165) <sup>c</sup> |                             |                |                     |
| Av.                         |                | 1218 <sup>d</sup>   | Av.                         |                | 3156 <sup>d</sup>   |
| Lattice $H_{240} - H_{280}$ |                | 916.5               | Lattice $H_{295} - H_{315}$ |                | 576                 |
| $\Delta H_t$                |                | 302                 | $\Delta H_m$                |                | 2580                |

<sup>a</sup> Units: cal., mole, °K. <sup>b</sup> Incomplete conversion. <sup>c</sup> Runs taken after partial decomposition of the sample. <sup>d</sup> Runs in parentheses are excluded from the average.

8) and 304.65°K. (5). The mole fraction of the liquid-soluble, solid-insoluble impurity was estimated with the expression:

$$N_2 = \Delta H_m(T_0 - T_1) / RT_1^2$$

The premelting enthalpy has been included in  $\Delta H_m$  in the evaluation of impurity.

An initial set of fractional melting runs (series I, Table I; in which true thermal equilibration was not waited for) indicated 99.9% mole % purity. During the course of the study, enthalpies of transition and melting were investigated a number of times, and all these values are in agreement within experimental error. Observation of larger than normal positive drifts in the liquid region suggested the onset of thermal decomposition at these temperatures. For this reason, the time that the sample was in the liquid state was carefully minimized and only after completion of all other measurements was the sample subjected to customary fractional melting procedures. However, true constancy of temperature was not attained even after a week of equilibration. The gradual but continuous decrease in the apparent melting temperature presumably was a consequence of impurities formed by decomposition or polymerization of the sample. The triple point of the sample and the purity computed from this data set (series XX) were 303.26°K. and about 99.5 mole %, respectively. Enthalpy runs through transition and melting subsequently made for this partially decomposed sample gave values significantly smaller than previously determined values. Hence, the measurements prior to those of series XX reported here pertain to a sample of 99.9% purity.

**Thermodynamic Functions.** Smoothed values of the heat capacity at selected temperatures obtained by a digital computer fit of the experimental data and carefully compared with large scale plots are presented in Table IV. Molal values of the entropy, enthalpy increment, and Gibbs energy function are also listed in Table IV at selected temperatures. These have been integrated by a high-speed digital computer using a least-squares polynomial fit through the data points. Anomalous regions were analyzed by numerical quadrature. Below 5°K., the heat capacity data were extrapolated using the Debye  $T^3$  limiting law. Nuclear spin and isotopic mixing contributions have not been included in the entropy and Gibbs energy function.

**Mechanism of Transition and Fusion.** The high value of  $\Delta S_m$ , 8.46 cal. per mole °K., indicates that malononitrile is not a plastic crystal. This value is in reasonable accord with van de Vloed's (8) reported value of  $\Delta S_m$  determined from cryoscopic data. Since no plastic crystalline state occurs, the interpretation and mechanism of the apparently  $\lambda$ -type transformation with a maximum at 260.3°K. remain uncertain. Although the entropy of transition is 1.15 cal.

Table III. Fractional Melting Data for Malononitrile<sup>a</sup>

| $T$                             | $\Delta T$ | $\Sigma \Delta H_i$ | $T_{\text{final}}$ | $1/F$   |
|---------------------------------|------------|---------------------|--------------------|---------|
| 304.65                          | 0.186      | 937                 | 304.74             | 3.144   |
| 304.78                          | 0.073      | 1293                | 304.82             | 2.279   |
| 304.83                          | 0.039      | 1650                | 304.85             | 1.785   |
| 304.87                          | 0.029      | 2007                | 304.88             | 1.468   |
| 304.89                          | 0.017      | 2364                | 304.90             | 1.246   |
| 304.90                          | 0.003      | 2721                | 304.90             | 1.083   |
| Triple point of sample =        |            |                     |                    | 304.911 |
| Triple point of pure compound = |            |                     |                    | 304.987 |
| Mole fraction of impurity =     |            |                     |                    | 0.001   |

<sup>a</sup> Data from series I. Units: cal., mole, °K.

per mole °K. and approximates  $R \ln 2$ —i.e., 1.38 cal. per mole °K.—the suggestion that the transition involves an order-disorder type of reorientation into indistinguishable configurations is unwarranted in the absence of crystal structure data. Some evidence for a phase transition in the range 233° to 265°K. was also detected by nuclear quadrupole resonance techniques (13). This transition may be accompanied by a considerable volume change, as the calorimeter was distended by about 5% upon completion of the measurements. Borosilicate glass tubes containing sample shatter on cooling through the transition.

**Third-Law Comparison.** Figure 1 shows that the heat capacity curve of undercooled crystal I merges with that of crystal II at 268°K. The entropy increments,  $S_{270} - S_{0^\circ \text{K.}}$ , via undercooled crystal I and via the stable forms (crystals I and II) were calculated as 23.84 and 23.85 cal. per mole °K., respectively. The excellent agreement between these two entropy increments indicates a lack of (or the same extent of) residual disorder at 0°K. in both malononitrile crystalline phases.

**Comparison with Spectroscopic Entropy.** The entropy of gaseous malononitrile has been calculated from spectral data (4). This has been recalculated from new data and recently revised assignments for structural parameters of Fujiyama and Shimanouchi (3) to determine the vibrational, rotational, and translational entropy contributions. A total (spectroscopic) ideal gas entropy at 298.15°K. of 69.02 cal. per mole °K. was obtained. The entropy from calorimetric and vapor pressure data is summarized in Table V and discussed briefly here. Recently, Boyd (1) has made determinations of the vapor—i.e., sublimation—pressure of malononitrile by the Knudsen effusion technique above the transition temperature. The computed values of the sublimation pressure and the enthalpy of sublimation ( $\Delta H_s$ ) at 298.15°K. are 0.09 torr and ~18 kcal. per mole. The total entropy of the gas at 298.15°K. would be ~74 cal. per mole °K. or about 5 cal. per mole °K. higher than the spectroscopic entropy if Boyd's values are employed. Alternatively, if the vapor pressures and  $\Delta H_v$  relations,  $\log P = 0.222n + 11.936 - (173n + 2996)/T$  ( $P$  in microns) and  $\Delta H_v^{298.15} = (0.79 \pm 0.02)n + 13.71 \pm 0.11$  ( $\Delta H_v$  in kcal. per mole), for the dinitriles  $[\text{NC}(\text{CH}_2)_n\text{CN}]$  for  $n = 2$  to 8] determined by Woodman, Murbach, and Kaufman (11) are extrapolated to  $n = 1$ , values of 0.0338 torr and  $14.5 \pm 0.1$  kcal. per mole would be obtained. Admittedly, extrapolation of the foregoing generalization back to the first member of a homologous series is a suspect procedure. However, the resultant entropy of the gas,  $68.5 \pm 0.4$ , accords better with the spectroscopic value and is within the limits of reliability of the data employed.

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Table IV. Thermodynamic Functions of Malononitrile<sup>a</sup>

| $T$   | $C_p$ | $S^\circ$ | $H^\circ - H_0^\circ$ | $-(G^\circ - H_0^\circ)/T$ | $T$                   | $C_p$                                   | $S^\circ$ | $H^\circ - H_0^\circ$ | $-(G^\circ - H_0^\circ)/T$ |
|---|-------|-----------|-----------------------|----------------------------|-----------------------|---|-----------|-----------------------|----------------------------|
| CRYSTAL II-STABLE FORM                          |       |           |                       |                            | CRYSTAL I-STABLE FORM |   |           |                       |                            |
| 5   | 0.008 | 0.004     | 0.016                 | 0.001                      | 298.15                | 26.36                                   | 31.30     | 4758                  | 15.34                      |
| 10  | 0.122 | 0.037     | 0.028                 | 0.009                      | 300                   | 31.50                                   | 31.49     | 4816                  | 15.44                      |
| 15  | 0.477 | 0.143     | 1.654                 | 0.033                      | 304.91                | 27.09 <sup>b</sup>                      | 31.91     | 4939                  | 15.71                      |
| 20  | 1.123 | 0.363     | 5.542                 | 0.086                      |                       | LIQUID                                  |           |                       |                            |
| 25  | 1.985 | 0.703     | 13.25                 | 0.173                      |                       |   |           |                       |                            |
| 30  | 2.956 | 1.150     | 25.57                 | 0.297                      | 304.91                | 30.48 <sup>b</sup>                      | 40.37     | 7519                  | 15.71                      |
| 35  | 3.965 | 1.681     | 42.87                 | 0.456                      | 310                   | 30.95                                   | 40.87     | 7676                  | 16.11                      |
| 40  | 4.940 | 2.275     | 65.16                 | 0.646                      | 320                   | 31.90                                   | 41.97     | 7991                  | 17.00                      |
| 45  | 5.857 | 2.910     | 92.19                 | 0.862                      |                       | CRYSTAL I-METASTABLE FORM (UNDERCOOLED) |           |                       |                            |
| 50  | 6.710 | 3.572     | 123.6                 | 1.100                      |                       |   |           |                       |                            |
| 60  | 8.229 | 4.933     | 198.5                 | 1.625                      | 5                     | 0.025                                   | 0.007     | 0.026                 | 0.002                      |
| 70  | 9.534 | 6.302     | 287.5                 | 2.195                      | 10                    | 0.229                                   | 0.065     | 0.503                 | 0.015                      |
| 80  | 10.67 | 7.652     | 388.7                 | 2.793                      | 15                    | 0.771                                   | 0.251     | 2.882                 | 0.059                      |
| 90  | 11.69 | 8.968     | 500.5                 | 3.407                      | 20                    | 1.569                                   | 0.577     | 8.646                 | 0.145                      |
| 100   | 12.60 | 10.248    | 622.1                 | 4.027                      | 25                    | 2.523                                   | 1.028     | 18.83                 | 0.274                      |
| 110   | 13.45 | 11.49     | 752.4                 | 4.649                      | 30                    | 3.546                                   | 1.578     | 33.99                 | 0.445                      |
| 120   | 14.25 | 12.69     | 890.9                 | 5.270                      | 35                    | 4.578                                   | 2.202     | 54.30                 | 0.650                      |
| 130   | 15.01 | 13.86     | 1037                  | 5.886                      | 40                    | 5.553                                   | 2.878     | 79.66                 | 0.886                      |
| 140   | 15.73 | 15.00     | 1190                  | 6.497                      | 45                    | 6.458                                   | 3.584     | 109.7                 | 1.146                      |
| 150   | 16.41 | 16.11     | 1352                  | 7.101                      | 50                    | 7.291                                   | 4.309     | 144.1                 | 1.426                      |
| 160   | 17.07 | 17.19     | 1519                  | 7.698                      | 60                    | 8.762                                   | 5.772     | 224.6                 | 2.029                      |
| 170   | 17.70 | 18.25     | 1693                  | 8.288                      | 70                    | 10.01                                   | 7.219     | 318.6                 | 2.667                      |
| 180   | 18.31 | 19.28     | 1873                  | 8.870                      | 80                    | 11.10                                   | 8.628     | 424.3                 | 3.325                      |
| 190   | 18.91 | 20.28     | 2059                  | 9.444                      | 90                    | 12.06                                   | 9.992     | 540.2                 | 3.990                      |
| 200   | 19.50 | 21.27     | 2251                  | 9.728                      | 100                   | 12.93                                   | 11.31     | 665.2                 | 4.656                      |
| 210   | 20.07 | 22.23     | 2449                  | 10.57                      | 110                   | 13.74                                   | 12.58     | 798.6                 | 5.319                      |
| 220   | 20.61 | 23.18     | 2652                  | 11.12                      | 120                   | 14.53                                   | 13.81     | 939.8                 | 5.976                      |
| 230   | 21.09 | 24.10     | 2861                  | 11.67                      | 130                   | 15.33                                   | 15.00     | 1089                  | 6.620                      |
| 240   | 21.68 | 25.01     | 3075                  | 12.20                      | 140                   | 16.38                                   | 16.21     | 1247                  | 7.298                      |
| 250   | 22.28 | 25.91     | 3295                  | 12.73                      | 150                   | 17.03                                   | 17.37     | 1416                  | 7.928                      |
|   |       |           |                       |                            | 160                   | 17.34                                   | 18.47     | 1587                  | 8.548                      |
| CRYSTALS I + II (STABLE FORM)-TRANSITION REGION |       |           |                       |                            | 170                   | 17.82                                   | 19.53     | 1764                  | 9.160                      |
| 255   | 23.50 | 26.36     | 3409                  | 12.99                      | 180                   | 18.31                                   | 20.57     | 1944                  | 9.763                      |
| 260   | 85.90 | 27.40     | 3677                  | 13.26                      | 190                   | 18.82                                   | 21.57     | 2130                  | 10.36                      |
| 260.3 (max.)                                    | 90.30 | 27.50     | 3704                  | 13.27                      | 200                   | 19.30                                   | 22.54     | 2321                  | 10.93                      |
| 265   | 25.45 | 28.40     | 3939                  | 13.53                      | 210                   | 19.73                                   | 23.49     | 2517                  | 11.51                      |
| 270   | 23.52 | 28.84     | 40.57                 | 13.81                      | 220                   | 20.06                                   | 24.42     | 2716                  | 12.08                      |
|   |       |           |                       |                            | 230                   | 20.54                                   | 25.32     | 2919                  | 12.63                      |
|   |       |           |                       |                            | 240                   | 21.27                                   | 26.21     | 3128                  | 13.18                      |
|   |       |           |                       |                            | 250                   | 22.10                                   | 27.09     | 3345                  | 13.71                      |
|   |       |           |                       |                            | 260                   | 22.78                                   | 27.97     | 3570                  | 14.24                      |
|   |       |           |                       |                            | 270                   | 23.52                                   | 28.85     | 3802                  | 14.77                      |

<sup>a</sup> Units: cal., mole, °K. <sup>b</sup> Assuming melting to be truly isothermal.

Table V. Entropy of Malononitrile (Ideal Gas) at 298.15°K.<sup>a</sup>

| Temp., °K.    |                                   | $\Delta S$          | $\Delta S$                |
|---------------|-----------------------------------|---------------------|---------------------------|
| 0-298.15      | Crystal I                         | 31.30               | 31.30                     |
| 298.15-304.91 | Crystal I                         | 0.61                | ...                       |
| 304.91        | $\Delta H_m/T_m$ melting          | ...                 | 8.46                      |
| 304.91-298.15 | Liquid                            | -0.68               | ...                       |
|               | $\Delta H_v/298.15$               |                     |                           |
|               | vaporization                      | ...                 | 48.63 ± 0.44 <sup>c</sup> |
|               | $\Delta H_s/298.15$               |                     |                           |
|               | sublimation                       | ~60 <sup>b</sup>    | ...                       |
| 298.15        | $R \ln P$ compression to 1 atm.   | ~-17.3 <sup>b</sup> | -19.8 <sup>c</sup>        |
|               | Ideal gas correction <sup>d</sup> | 0                   | 0.0                       |
|               | Entropy (ideal gas)               | ~74                 | 68.5 <sub>0</sub> ± 0.4   |
| 298.15        | Spectroscopic Entropy             |                     | 69.02                     |

<sup>a</sup> Units: cal., mole, °K. <sup>b</sup> Using Boyd (1) data for enthalpy of sublimation and vapor pressure. <sup>c</sup> Enthalpy of vaporization and vapor pressure computed from an empirical relationship (11). <sup>d</sup> Assumed negligible for vapor pressure of malononitrile ( $<10^{-1}$  torr at 298.15°K.).

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