Effect of Temperature on Silica and Silica Xerogels as Measured by Low-Temperature (-196°C.) Nitrogen Adsorption

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Low-temperature $(-196^{\circ} C.)$ nitrogen adsorption-desorption isotherms have been measured on pure silica and a range of silica xerogels thermally modified from 240° to $1100^{\circ} C.$ B.E.T. surface areas and pore distribution characteristics are reported for each material. Results indicate that thermal modification has a significant effect on the surface areas and distribution data, especially above $700^{\circ} C.$ In the majority of cases, B.E.T. areas are greater than the equivalent cumulative data. A range of noncylindrical pores is indicated in each material. With the exception of Macroporous silica gel, both the adsorption and desorption branches of the isotherms can be used for pore volume and surface area distribution determinations. In the former case, the desorption branch of the isotherm seems to lead to more apt results.

SILICA gels and silicas are widely used in industry, and a close understanding of the surface and bulk properties of such materials is of practical importance in the technology of cracking catalysts, catalyst supports, mineral processing, ceramics, and adsorbents. Such an understanding is, in addition, directly involved in the manufacture and use of siliceous fillers and thickening agents in organic systems, including paints, inks, elastomers, and lubricating greases.

Recent studies have been reported of the sorption of ammonia (6), ethylamine (23, 24), and sulfur dioxide (14) on silicas and silica xerogels of high purity. Surface and structural modifications were achieved by thermal treatments, largely in the range 240° to 1100° C. To characterize the materials, a large number of low-temperature (-196° C.) nitrogen adsorption-desorption isotherms were determined.

Low-temperature gas adsorption isotherms have been widely used in the past to evaluate the effects of heat treatment on silica gel and gel-like structures (1, 2, 3, 3)5, 10, 18, 21, 22), particularly by the application of the B.E.T. equations (5). The method was used here to characterize materials, with special regard to surface area. Additional detailed information on pore-size distributions has been obtained by applying the Cranston and Inkley method (7) to the isotherms. This method provides independent comparative surface area data (to the B.E.T. values) and can be applied to both the adsorption and desorption branches of the isotherms. The results were calculated on the assumption of a molecular cross-sectional area for nitrogen of 16.2 A.² (the most frequently used figure). However, there have been suggestions that the quadrupole moment of nitrogen can interact with hydroxyl or other polar groups, leading to a change in cross-sectional area (19). In spite of this, nitrogen adsorption generally gives the most meaningful results in low-temperature adsorption studies.

Results from the adsorption measurements and from the isotherms were analyzed with the aid of an Elliott 803 computer.

EXPERIMENTAL

Materials. The major portion of the study was carried out with Davison 923 chromatographic grade silica gel (>99.98% pure). Its surface area was 630 meters² per gram (B.E.T.: N_2 , -196°C.) and lump density, 1.145 grams per cc. The particle size range was 75 to 150 microns. Before

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use, it was heated in air at $240^\circ\,C.$ for three hours, cooled in a desiccator $(P_2O_5),$ weighed, and transferred quickly to the sorption apparatus.

Additional materials with the following properties were presented by Joseph Crosfield and Sons, Ltd.

Microporous Sorbsil silica gel. Surface area (B.E.T.: N_2 , -196° C.), 730 meters² per gram; lump density, 1.217 grams per cc.; particle size range, 75 to 150 microns.

Macroporous silica gel. 215 meters² per gram, 0.336 gram per cc., 75 to 150 microns.

High Surface Area silica gel. 840 meters² per gram, 1.120 grams per cc., 500 to 2000 microns.

Neosyl F.X. pure silica. 690 meters^2 per gram, 0.258 gram per cc., 40 to 100 microns.

The first three materials were shown by chemical analysis to contain iron up to 25 p.p.m. as the principal impurity.

Apparatus. Lump densities were measured by immersion in mercury (13) after heat treatment at 240°C. for three hours in air.

Mercury has a high surface tension and a large contact angle, and is thus unable to enter pores of radius below about 50,000 A., when a 1-atm. pressure exists across the meniscus. To measure this mercury density, an all-glass vacuum apparatus similar to that used by Jepson (13) was employed. Data were obtained on the out-gassed sample (90° C. for one hour) by measuring the volume of the sample container and the weight of mercury required to fill the container to a preset mark in the presence of the material under investigation. Measurements were made at 20° C. The mercury density, D, was then calculated from the equation .

$$D = \frac{W_1}{V - \frac{W_2}{D_T}} \tag{1}$$

where

 W_1 = weight of material

 W_2 = weight of mercury in sample container

V = volume of sample container

 D_{τ} = density of mercury at 20° C.

Duplicate determinations were within an experimental error of 1% .

Important surface parameters such as surface area and pore-size distribution were measured here using the lowtemperature gas adsorption technique (N_2 : -196° C.) developed by Brunauer *et al.* (5). A brief review of the theory behind such measurements is given in a subsequent section.

A constant-volume apparatus similar to that of Joyner (15) was used for the measurement of the low-temperature nitrogen adsorption isotherms. The essential features are described below.

The pumping system consisted of a two-stage mercury diffusion pump backed by a two-stage rotary oil pump. Mercury vapor was prevented from entering the lowpressure side of the apparatus by a liquid nitrogen trap. Vacuum was measured with a Pirani-type gage (Model 8/2) of range 2 × 10⁻¹ to 10⁻⁴ mm. of Hg and a McLeod gage $(10^{-6} \text{ mm. of Hg})$. The apparatus easily held a vacuum of 10^{-3} mm. of Hg over seven days.

Purified "white spot" cylinder nitrogen (99.9%) was supplied by the British Oxygen Company, Ltd. This was stored in pre-evacuated apparatus over two columns (35 cm. by 4 cm.) packed with alternate layers of self-indicating silica gel and calcium chloride. The gas admittance section consisted of the nitrogen purification and storage system and, in addition, a system of taps. These taps were arranged so that small quantities of nitrogen could be admitted to the measuring section as required. This was necessary to prevent sudden upsurges of gas forcing the large quantities of mercury enclosed in the apparatus into the adsorption system, and to obtain an even distribution of points on the isotherms.

Essential features of the measuring system included a small "dead space" (10.3 cc.) and a small plug of 0.001inch gold foil (23.75 carat purity) to prevent mercury vapor contacting the adsorbate. The dead space was made as small as possible with the aid of capillary tubing and smallbore vacuum taps. It was calculated from the ideal gas law for room and liquid nitrogen temperatures. The average temperature of the sample flask when immersed in liquid nitrogen and the saturation pressure of the adsorbate were determined with the aid of an incorporated vapor pressure thermometer (20). To assist in the reduction of the dead space, a combined gas buret and manometer were employed. The buret was thermostatted ($20^{\circ} \pm 0.1^{\circ}$ C.) and was precalibrated with mercury before use. The volumes of gas adsorbed and the corresponding pressure readings were taken with a cathetometer reading to ± 0.001 cm. A difference of 1 cc. on the burette was equivalent to a differential reading of 1.105 cm. on the cathetometer. This relationship was obtained over the whole burette span. The zero level on the burette was kept constant with the aid of a mercury reservoir.

Isotherms measured with this equipment were reproducible to better than $\pm 5\%$.

Hydroxyl group concentrations were determined gravimetrically by direct calcination of the samples in air (24).

THEORY

The most widely used method at present to characterize materials with regard to surface area and pore size distribution (up to 300-A, pore diameter) is that of low-temperature (usually -196° C.) gas adsorption. Brunauer *et al.* (5) developed an expression for the adsorption of a gas on a solid. This was based on Langmuir's original concept (11) of monolayer adsorption and was extended to take account of multilayer adsorption on the solid. Constant heats of adsorption in the first and subsequent layers were assumed, and were considered equal to the heat of liquefaction of the adsorbate. The most widely used form of their expression is

$$p/V(p_o - p) = 1/VmC + (C - 1)p/VmCp_o$$
 (2)
where

V = volume of gas adsorbed at equilibrium pressure p (cc.)

- p_{o} = saturation vapor pressure of adsorbate (mm. of Hg)
- Vm = the volume of gas contained in the monolayer (cc.)
- constant which is related to the heats of adsorption and C= liquefaction of the gas

Normally, a plot of $p/V(p_o - p)$ against p/p_o yields a straight line of slope $(\hat{C} - 1) / Vm\hat{C}$ and intercept 1 / VmC. Hence, values for Vm and C can be obtained. The equation usually holds over the range $0.05 < p/p_{\circ} < 0.35$. These limits were not valid in the present study. The specific surface area is evaluated from Vm by means of the equation

$$S = Vm \frac{N}{V} A \tag{3}$$

where

V = molar volume (22,400 cc. at N.T.P.)

 $N = \text{Avogadro's number} (N = 6.023 \times 10^{23} \text{ molecules mole}^{-1})$

A = area occupied per molecule in the adsorbed phase

The generally accepted value for nitrogen is A = 16.2A.², as derived from the density of liquid nitrogen. Hence, for nitrogen adsorption at -196° C., the relationship becomes

$$S = 4.36 Vm$$
 (4)

In spite of the generally accepted invalidity of many of the original assumptions in the B.E.T. theory, it still remains the most widely used method for surface area determination and gives absolute values which are generally accurate to within $\pm 10\%$.

In addition to surface area, low-temperature gas adsorption can provide data on pore-size distributions. In this study, the method of Cranston and Inkley (7) has been used to derive such data. This is a development of the method of Barrett et al. (2) and has three novel features:

It is more exact than that of Barrett et al., and provides an estimate of total specific surface which is almost independent of the BET value and may therefore be compared with it. A comparison of pore volumes calculated in two ways is also available.

The method may be applied to either the adsorption or desorption branch of the isotherm.

Such differences as exist between the cumulative and the B.E.T. surface areas, and also between the two estimates of pore volume, are not to be regarded purely as experimental errors. The differences provide a measure of the validity of the physical assumptions made, and therefore give an indication of the character of the pores in the materials.

In terms of pore diameters, the working equation is

$$V_{12} = R_{12} \left(v_{12} - k_{12} \sum_{d_2 - 1/2 \Delta d}^{d_{\max}} \frac{d - 2t_{12}}{d^2} V_d \Delta d \right)$$
(5)

where

 Δd = an increment of pore diameter (A.)

 $V_d \Delta d =$ the volume of pores having diameters between $(d - \frac{1}{2}\Delta d)$ and $(d + \frac{1}{2}\Delta d)$

diameter of the largest pore

- $d_{\max} = V_{12} =$ volume of pores having diameters between d_1 and d_2 (cc.) $t_{12} = \frac{1}{2}(t_1 + t_2)$
- $k_{12} = 4(t_2 t_1)$

 t_1 and t_2 are the adsorbed layer thicknesses corresponding to pressures p_1 and p_2 . R_{12} includes terms such as r_1 , r_2 , and t.

Tables of values for R_{12} , k_{12} and the factor (d - 2t)/(d - 2t) d^2 are provided in Cranston and Inkley's original paper (7).

The method is widely used at present to obtain poresize distribution data on porous materials. Complications arise from the variety of pore shapes found in these materials, since the original assumption of cylindrical pores is invalidated. Also, the angle of contact is taken to be zero, and the density and surface tension of the adsorbate in the pores are both assumed to have the same values as the bulk liquid. In spite of this, the method is still one

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of the best means of obtaining pore distribution data in the range 20- to 300-A. pore diameters. The average pore diameter in the material under investigation may also be calculated from the relevant data.

RESULTS

Davison Silica Xerogel. The adsorption-desorption isotherms on Davison silica gel are shown in Figure 1 for samples treated in air at 240° C. for three hours, and for 20 hours at 500°, 600°, 700°, 800°, and 900° C. After treatment, each sample was placed in the adsorption apparatus and carefully evacuated to 10^{-4} mm. of Hg. An additional out-gassing at 90° C. for another hour under a vacuum of 10^{-5} to 10^{-6} mm. of Hg was then carried out.

Each isotherm at -196° C. shows a relatively large initial adsorbed quantity of nitrogen. The monolayers appear to be complete at $p/p_{\circ} = 0.085$, 0.091, 0.091, 0.102, 0.103, and 0.109 for the samples treated at 240°, 500°, 600°, 700°, 800°, and 900° C., respectively. B.E.T. plots of the adsorption (Figure 2) are all straight lines holding over the relative pressure ranges noted in Table I.



Figure 1. Nitrogen adsorption-desorption isotherms (-196° C.) on Davison silica xerogel dehydrated from 240° to 900° C: a) 240° C.; b) 500° C.; c) 600° C.; d) 700° C.; e) 800° C.; f) 900° C.

The decrease in the total adsorbed quantity of nitrogen $(p/p_o = 1.0)$, with increase in dehydration temperature, is most pronounced above 700°C. (Figure 1). A steady compression of the hysteresis loop with increasing dehydration temperature is also apparent. Up to $p/p_o = 0.5$, adsorption equilibrium times of 30 minutes were required. From $p/p_o = 0.5$ to 1.0, times of 60 minutes were necessary. On desorption, equilibrium was attained within 60 minutes from $p/p_o = 1.0$ to 0.4. Below this point, 30 minutes sufficed.

Pore-size distribution data (both surface area and pore volume) were obtained from these isotherms using the method of Cranston and Inkley (7) (Figures 3 to 5). Surface physical data calculated from this and the B.E.T. method are summarized in Tables I and II.

Figure 3 shows that, as the temperature of heat treatment is increased from 240° to 900° C., there is no marked change in the distribution of pores. However, the number of pores having the same diameters decreases slowly up to 700° C.,



Figure 2. B.E.T. plots of nitrogen adsorption on Davison silica xerogel dehydrated from 240° to 900° C.

	1	able I. B.E.T. Data for Silica Gels and Silica							
		B.E.T. Data							
	Material	Linearity, p/p_o	Slope	Intercept	B.E.T. ''C'' value	S, specific surface, meter ² /g.			
а	Microporous Gel	0.05 - 0.15	5.934×10^{-3}	3.5×10^{-5}	170.6	730.0			
b	Neosyl F.X. pure silica low iron silica	0.05 - 0.15	6.320×10^{-3}	4.5×10^{-5}	141.5	685.0			
с	Macroporous silica gel	0.05 - 0.125	2.001×10^{-2}	2.75×10^{-4}	73.78	215.0			
d	High Surface Area gel	0.05 - 0.15	5.177×10^{-3}	3.6×10^{-5}	144.8	830.0			
e	Davison 923 chromatographic grade silica gel 100-200 mesh pre-treated at 240° C.	0.05-0.175	6.914×10^{-3}	5.5 $\times 10^{-5}$	126.7	630.0			
f	Davison 923 gel pre-treated at 500° C.	0.05 - 0.175	7.020×10^{-3}	5.2 $\times 10^{-5}$	136.0	615.0			
g	Davison 923 gel pre-treated at 600° C.	0.05-0.20	$7.929 \times 10^{+3}$	7.8×10^{-5}	102.6	545.0			
h	Davison 923 gel pre-treated at 700° C.	0.05-020	$8.909 \times 10^{+3}$	1.06×10^{-4}	85.04	485.0			
i	Davison 923 gel pre-treated at 800° C.	0.05-0.20	1.191×10^{-2}	1.53×10^{-4}	78.81	360.0			
j	Davison 923 gel pre-treated at 900° C.	0.05-0.20	2.694×10^{-2}	3.7×10^{-4}	73.82	160.0			

Table II. Pore Size Distribution Data for Silica Gels and Silica

	Pore Size Distribution Data									Average Pore Diameter, A. $T = \frac{4V_{0.931}^a \times 15.84}{4V_{0.931}^a \times 15.84}$		
	S'_{a} .		Ra	tio	Adsorpti	on Branch	Ratio	Desorpti	on Branch	Ratio	u – "	\overline{S}
Material	$(m.^{2}/g.)$	$(m.^{2}/g.)$	S_{a}^{\prime}/S	S_{d}^{\prime}/S	$V^{a}_{0.931}$	$V'_{\mathfrak{a}}$	$V_{a}^{\prime}/V_{0.931}^{a}$	$V^{d}_{\scriptscriptstyle 0.931}$	V_{d}^{\prime}	$V_{d}^{\prime}/V_{0.931}^{d}$	Ads.	Des.
а	515.0	515.0	0.704	0.704	230.0	163.85	0.712	230.0	163.89	0.712	20.0	20.0
ь	545.0	545.0	0.797	0.797	255.2	213.60	0.837	255.2	213.60	0.837	23.6	23.6
с	168.45	180.79	0.784	0.846	290.0	280.62	0.968	587.5	585.39	0.996	85.5	173.2
d	649.47	647.75	0.777	0.775	319.0	258.97	0.812	319.0	259.16	0.812	24.2	24.2
e	514.40	523.94	0.822	0.838	273.7	239.57	0.875	273.7	241.65	0.883	27.7	27.7
f	519.30	518.70	0.842	0.841	261.8	232.20	0.887	261.8	232.39	0.888	26.9	26.9
g	501.83	501.44	0.921	0.921	236.0	218.34	0.925	236.0	218.53	0.926	27.5	27.5
ĥ	457.92	457.61	0.947	0.947	203.8	192.07	0.942	203.8	192.18	0.943	26.7	26.7
i	362.55	362.26	1.003	1.002	155.8	151.20	0.971	155.8	151.32	0.971	27.3	27.3
j	150.28	154.12	0.942	0.966	71.6	68.96	0.963	71.6	69.82	0.975	28.4	28.4



Figure 3. Pore volume distribution data for Davison silica xerogel obtained from the adsorption isotherms (240° to 900° C.)

and much more rapidly above this temperature. The average pore diameter in the gels, as calculated from the adsorption and desorption isotherms, remains virtually constant at 27 ± 1 A. The figure shows that the majority of the pores have diameters in the range <20 to 80 A.

Initially, at 240° C. there is a maximum in the distribution curve at ~ 23 A., with a secondary maximum just below 20 A. In the region of 800° C., the former disappears. Treatment at 900° C. flattens the distribution curve further, but leaves the original general pore distribution and the maximum just below 20 A.

Equivalent data for these materials calculated from the desorption isotherms are shown in Figure 4. The general information here is similar to that just described. Up to a diameter of ~ 32 A., distributions calculated from both branches of the isotherms are very similar. Above ~ 32 A., a third peak is found in the desorption data at 38 A., which becomes depressed as the dehydration temperature increases (especially above 800° C.). The curves fall off rapidly to near zero at pore diameters of ~ 50 A. (Figure 3).

Typical surface area distributions for Davison xerogel pretreated at 240° and 900° C. are shown in Figure 5. This illustrates results obtained from both branches of the isotherms. Here again, there are striking similarities between both distributions and with the equivalent pore-volume distribution data.



Figure 4. Pore volume distribution data for Davison silica xerogel obtained from the desorption isotherms (240° to 900° C.)



Figure 5. Surface area distribution data for Davison silica xerogel from adsorption and desorption isotherms at 240° and 900° C.

The variation of B.E.T. surface area and lump density of the Davison gel with dehydration temperature is shown in Figure 6. From 240° to 500° C., the surface area remains virtually constant. Above 500° C., it begins to fall steadily, up to about 750° C., where the rate of loss of area greatly increases with increasing temperature. A point of inflection, apparent at 600° C., was almost exactly reproducible in three separate experiments.



Figure 6. Variation of surface area and mercury density with temperature for Davison silica xerogel treated from 240° to 1100° C.

Concurrent with this, the density increases steadily up to 700° C. and more rapidly at temperatures $>700^{\circ}$ C. Between 1000° and 1100° C., a slowing down in the rate of densification of the gel is apparent.

Relevant hydroxyl group concentrations for this material are 9.6, 4.0, 3.0, 2.1, 2.0, and 1.9 μ moles per meter² after treatment at 240° C. for three hours and 500°, 600°, 700°, 800°, and 900° C. for 20 hours each, respectively.

Crosfield Silica Gels and Silica. Before use, each sample was heated at 240° C. for three hours. On transferring to the low-temperature adsorption apparatus, a further activation at 90° C. was carried out under a vacuum of 10^{-5} mm. of Hg for one hour on the High Surface Area gel, Microporous gel, and Neosyl F.X. pure silica. The Macroporous gel had to be carefully evacuated at room temperature for about one hour, before slowly raising the temperature to 90° C. over another hour, while still evacuating the sample. This was necessary because of the large amount of adsorbed gases evolved during evacuation.

Nitrogen adsorption-desorption isotherms for the High Surface Area, Microporous, and Neosyl F.X. materials are shown in Figure 7. For the High Surface Area gel, a large quantity of nitrogen is adsorbed at a low relative pressure. A B.E.T. plot (Figure 7) of the adsorption indicates a monolayer capacity of 191.8 cc. per gram $(p/p_o = 0.077)$. Equilibrium adsorption times were the same as for the Davison gel.

The isotherm for the Microporous gel also shows significant uptake of nitrogen at low pressures. A B.E.T. plot of the adsorption (Figure 7) indicates a monolayer capacity for the gel of 167.5 cc. per gram $(p/p_o = 0.071)$. The isotherm exhibits a very compressed hysteresis loop over the range $p/p_o = 0.35$ to 0.75. Equilibrium was attained in 30 minutes over the entire relative pressure range for this gel.

Adsorption on the Neosyl F.X. pure silica is again quite large in the low-pressure region. From the B.E.T. plot (Figure 7), a monolayer capacity of 157.1 cc. per gram is indicated $(p/p_o = 0.093)$. Coincidence of the adsorption and desorption isotherms is obtained over the whole relative pressure range. Adsorption and desorption equilibrium times were 30 minutes.

Pore-volume distributions for these three materials, obtained from both branches of the isotherms, are shown in Figure 8. The distribution for Microporous silica gel is shown in Figure 8,a. Data from adsorption and desorption isotherms coincide almost exactly, and hence only the



Figure 7. Nitrogen adsorption-desorption isotherms (–196° C.) on Crosfield materials (240° C.): a) high surface area gel; b) Neosyl F.X. silica; c) microporous gel. Insets: B.E.T. plots of the adsorption



Figure 8. Pore volume distribution data for Crosfield materials obtained from the adsorption and desorption isotherms: a) microporous gel; b) Neosyl F.X. silica; c) high surface area gel.

adsorption data are illustrated. The average pore diameter in this material is 20 A.

Figure 8,b shows the pore-volume distribution data for Neosyl F.X. pure silica. Here again, data from both branches of the isotherm coincide exactly.

Pore volume data for the High Surface Area material are shown in Figure 8,c. Adsorption and desorption distributions agree quite well up to 40 A. Above 40 A., the adsorption distribution tails off to near zero at ~100 A. In contrast to this, the desorption data give a further sharp maximum at 38 A. and rapidly decrease to small values of $\Delta V / \Delta D$ at 42 A., tailing off to near zero at 80 A.

Equivalent surface area distributions for these materials are shown in Figure 9 (a, b, c). These are all very similar to the corresponding volume distributions for each material. Generally, the surface area distributions (from adsorption and desorption isotherms) tend to become narrower and larger in the regions of smaller pore diameters. Hence, they tend towards zero at smaller pore diameters than indicated by the volume distribution data. Maxima in the higher diameter ranges—e.g., at 38 A. in Figure 9,c—become relatively much smaller. This is clearly seen from the relative heights of the maxima in the adsorption and desorption distribution data for the High Surface Area gel (Figures



Figure 9. Surface area distribution data for Crosfield materials obtained from the adsorption and desorption isotherms: a) microporous gel; b) Neosyl F.X. silica; c) high surface area gel.

8,c and 9,c). As with the volume distribution results, adsorption and desorption area distributions for Microporous silica gel and Neosyl F.X. silica coincide almost exactly.

The final nitrogen adsorption-desorption isotherm on Crosfield Macroporous silica gel is shown in Figure 10. The adsorption is small at relative pressures below 0.70, but rapidly increases in the range $p/p_o \sim 0.80$ to 0.95. From a B.E.T. plot (inset, Figure 10) of the adsorption, a monolayer coverage of 49.3 cc. per gram is indicated $(p/p_o = 0.105)$. The measurement of this isotherm, especially in the higher relative pressure regions $(p/p_o > 0.7)$, proved to be tedious, with very long equilibrium adsorption and desorption times (over 24 hours) required. Great difficulty was encountered in obtaining reproducibility above p/p_o = 0.7 until a technique was used involving a saturation of the gel surface $(p/p_o > 0.9 \text{ at } -196^\circ \text{C.})$ with nitrogen. Subsequently, this nitrogen was all desorbed at -196°C., followed by evacuation for one hour at 90°C. Only after a treatment of this nature could isotherm reproducibility be obtained on subsequent adsorption-desorption experiments. The figure shows a saturation adsorbed amount $(p/p_{o} \sim 1.05)$ of 760 cc. per gram.

Pore volume and surface area distributions from both branches of the isotherms are shown in Figure 11. Volume



Figure 10. Nitrogen adsorption-desorption isotherm (–196° C.) on Crosfield macroporous silica gel (240° C.). Inset: B.E.T. plot.



AREA ADSORPTION DESORPTION VOLUME O ADSORPTION OESORPT ON (01×) ΔS/ΔD cc/G/A M²/G/A 30 A V/AD 20 0-0-0-0 160 200 280 240 320 PORE DIAMETER (A)

Figure 11. Pore volume and surface area distribution data obtained from adsorption and desorption isotherms on macroporous silica gel.

distributions from the adsorption and desorption isotherms are very different in character. Hence, the adsorption data indicate a range of pores from <20 to 300 A., with maxima at 20 and 300 A. Desorption results, however, point to a pore range from 48 to 300 A., with two maxima at 256 and 275 A. The curves differ markedly, especially above 200 A., where much larger $\Delta V / \Delta D$ values are indicated in the desorption data. Average pore diameters calculated from the isotherms are 85.5 A. from the adsorption and 173.2 A. from the desorption isotherm, respectively (Table II). Equivalent surface area distributions also differ from each other and from the corresponding volume distribution data. Hence, the adsorption results show a sharp maximum at 20 A., which decreases rapidly to $\Delta S/\Delta D = 6$ meters² per gram per A. $(\times 10)$ at 80 A. and remains in this region up to 300 A. Desorption data again start at 48 A. and remain fairly constant at $\Delta S/\Delta D = 5$ meters² per gram per A. $(\times 10)$ up to 200 A. Above this point, there is an increase in the $\Delta S/\Delta D$ values, with maxima at 265 and 285 A. While the over-all character of the volume and area distributions is the same, the tendency is again towards sharper and larger maxima in the small pore diameter regions (<80 A.) and depression of the maxima in the higher diameter regions (>200 A.).

For all these materials, comparative information between the B.E.T. data and the pore distribution data was calculated and is summarized in Tables I and II.

DISCUSSION

Surface Area and Related Properties. The results on Davison xerogel heat-treated from 240° to 900°C. show an initial quantity of adsorption at low pressures. Monolayer adsorbed quantities of 143.5, 141.4, 124.9, 110.9, 82.9 and 36.6 cc. of nitrogen per gram of sample are obtained for the gel treated at 240°, 500°, 600°, 700°, 800°, and 900° C., respectively. The corresponding relative pressures are 0.085, 0.091, 0.091, 0.102, 0.103, and 0.109. Hydroxyl group concentrations at these temperatures are 9.6, 4.0, 3.0, 2.1, 2.0 and 1.9 μ moles per meter². Although these latter figures refer to measurements made gravimetrically and are not strictly representations of the surface hydroxyl group concentrations (8, 16, 17, 25, 26), nevertheless, these results seem to indicate the absence of specific adsorption, in that there is no apparent correlation between hydroxyl group concentration and monolayer capacity. Possible reactions of the quadrupole moment of N_2 with the hydroxyl groups (19) may occur, but the measurements are not considered sensitive enough to allow for discussion of such effects. Rather, the uptake of nitrogen and the monolayer capacity

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are a function of the decreasing surface area of the gel. This is in direct contrast to the ethylamine adsorption data on the same samples (23-25) and to results obtained for benzene adsorption on silica gels (16). Here there is a direct correlation between hydroxyl group concentrations and monolayer capacities of ethylamine and benzene (owing to specific hydrogen-bonding interactions). B.E.T. plots of the adsorption shown in Figure 2 exhibit an increase in slope as the dehydration temperature is increased, and Table I indicates a range of linearity in the plots from a minimum of $p/p_o = 0.05$ to a maximum of $p/p_o = 0.20$. Normally, a range of linearity in p/p_o from 0.05 to 0.35 is obtained for nitrogen adsorption on catalysts and catalyst supports (11). Deviations below $p/p_o = 0.05$ are often attributed to surface heterogeneity (11), while capillary condensation accounts for deviations above 0.20 (11).

The isotherms illustrated in Figure 1 all exhibit sharp "knees" in the adsorption branch in the region of p/p_o = 0.1. Intercepts in the B.E.T. equation (Table I) are very small ($\sim 5 \times 10^{-5}$ to $\sim 4 \times 10^{-4}$) and increase in value as the dehydration temperature increases. Concurrent with this, the value of the B.E.T. constant, *C*, decreases. Small intercepts are characteristic of nitrogen adsorption on oxides and hydrated oxides (11), and point to relatively large values of *C*.

C is related to the heat of adsorption (E_1) in the first layer and the heat of liquefaction (E_L) of the adsorbate (N_2) by the relationship (5)

$$C = \frac{a_1 b_2}{b_1 a_2} \cdot \exp \frac{E_1 - E_L}{RT}$$
(6)

where a_1 , b_2 , b_1 , and a_2 are constants such that the factor a_1b_2/b_1a_2 is not much different from unity. Hence, the equation reduces to the approximate form

$$E_1 - E_L = RT \ln C \tag{7}$$

Thus, C is directly related to the net heat of adsorption $(E_1 - E_L)$. The shape of the isotherms then depends on the value of C—which normally varies widely from 1 to 10,000 according to adsorbate-adsorbate and adsorbate-adsorbate interactions (11).

Sharp knees in the isotherms are indicative of a relatively strong adsorbate-adsorbent interaction with consequent large values of C (>50) (11). Table I shows that as the dehydration temperature changes from 240° to 500°C., C increases from 126.7 to 136.0. Above 500°C., the value drops steadily to 73.8 at 900°C. Though the value of Ccan only be taken as a rough guide (11) to the magnitude of E_{i} , these data indicate that the surface of Davison xerogel is reasonably "active," particularly after treatment at 500°C. for 20 hours. Differences in activity between 240° and 500°C. might be explained in terms of removal of surface hydroxyl groups leading to exposure of line and point imperfections in the gel structure. Alternatively, strongly bonded water may be removed by the increasing temperature from a more active type of surface site. Mutually hydrogen-bonded hydroxyl groups are present at 240° C. on surface silicon atoms which are separated by <3A. (8). The breaking of these bonds may also expose some active sites at temperatures up to 500° C. Above 500° C., the gradual fall in C is possibly an indication of a gradual "annealing" of the surface structure due to dehydration and sintering, which leads to a more homogeneous and less active surface at 900° C.

The isotherms on the Crosfield materials can not be discussed in such detail, since less data were obtained. However, Figures 7 and 10 illustrate sharp "knees" in the isotherms below $p/p_o = 0.1$. Small intercepts from 4.5 \times 10⁻⁵ to 2.75 \times 10⁻⁴ and high values of C from 73.8 to 170.6 (Table I) are indicated. B.E.T. plots hold from $p/p_o = 0.05$ up to 0.15. These conditions have been reported by Gregg and Sing (11) as being prime factors in the

determination of monolayer capacities on nonporous solids. The only drawback is that high values of C point to the possibility of extensive localized adsorption which, in turn, may affect the cross-sectional area of the adsorbate molecule. Fortunately, nitrogen is unusual, in that it normally gives isotherms with a high C value—and hence a sharp knee—but not sufficiently high to cause excessive localization.

Brunauer et al. (4) classified isotherm shapes into five empirical types corresponding to various solid materials, both nonporous and porous. Data for the Davison xerogel and Crosfield High Surface Area gel resemble the Type IV isotherm. Microporosity (d < 20 A.) also appears to be present in the materials. The hysteresis loops can be readily accounted for in a manner similar to that described by Gregg and Sing (11) for Type IV isotherms (4) and by de Boer (9) for his Type E isotherm. Hence, hysteresis can be attributed to the presence of tubular pores which contain a narrow constriction, or to closed pores of the "ink-bottle" type. Such pores begin to fill when the pressure, as calculated by the Kelvin equation, reaches the value corresponding to r_{w} (radius of the "wide" portion of the pore). They then continue to fill up, with the meniscus gradually increasing in radius and the pressure gradually rising correspondingly. On desorption, these pores do not empty until the pressure falls to the value corresponding to r_c (radius of the constriction), when they empty at once. Figures 1 and 7.a show such a pattern, with a sharp drop in the desorption isotherms at $p/p_{\circ} \sim 0.5$. According to this picture, the adsorption branch of the isotherms corresponds to equilibrium conditions and therefore should be used for calculations of pore-size distributions.

Isotherms for Crosfield Microporous silica, particularly, and Neosyl F.X. pure silica seem most closely related to the Type I classification of Brunauer *et al.* (4). This type is characteristic of microporous materials and gives rise to uncertainties in data interpretation. Hence, the attractive forces between molecules in small pores leads to a distortion of the isotherms in the direction of increased adsorption. Monolayer capacities, and thus B.E.T. surface areas for these materials, are probably too large (11). For most microporous materials, very small or no hysteresis loops are found, as in the case here.

The isotherm for Crosfield Macroporous silica gel does not conform closely to any of the classifications of de Boer (9) or Brunauer *et al.* (4), though it is slightly similar to that denoted Type A by the former author. Pore shapes associated with such hysteresis are discussed in detail by de Boer (9). Two facts emerge here. First, from the relative pressure at which significant adsorption occurs $(p/p_o > 0.8)$, results indicate a very wide-pored material. Secondly, the effective radii at which adsorption and desorption occur into and out of these pores are not too far removed from each other, as shown by the narrowness of the loop. Such a situation might be attained in the group of pores denoted Type A (3, 4, and 5) by de Boer. Again, the B.E.T. surface area of this gel probably approaches a more significant value.

Pore Distributions. As Davison xerogel is treated from 240° to 900° C., the pore distribution remains essentially the same, with average pore diameters of 28, 27, 28, 27, 27, and 28 A. at 240° , 500° , 600° , 700° , 800° , and 900° C., respectively. However, the number of pores having the same diameters is reduced by this treatment. Examination of the pore volume distribution data (Figures 3 and 4) reveals that up to 600° C. there is no very marked change from that at 240° C. From 600° to 700° C., there is a more apparent reduction in the number of pores in the gel, which becomes very marked between 700° and 800° C., and then continues rapidly from 800° to 900° C. Figure 5 illustrates the drop in area distributions, as measured from both branches of the isotherms, at 240° and 900° C. The data

of Figure 6 illustrate this effect in more detail. Thus, only a very small change in surface area is indicated between 240° and 500° C. Some significant loss in area occurs between 500° and 600° C., and above 700° C. the rate of loss of area is vastly accelerated. This is as expected, since reduction in the number of pores, and thus pore volume, will consequently be followed by loss of area. The figure also shows a rapid increase in the rate of densification of the gel above 700° C., which begins to flatten out at temperatures > 1000° C.

Henry and Ross (12) found very similar patterns on dehydration of silica xerogel spheroids. A marked decrease in surface area was observed around 800° C., together with a sharp decrease in density at the same temperature. At 900° C. and above, they found a rapid increase to a high density, which became almost constant at about 1.83 grams per cc. from 950° to 1200° C. In the region 900° to 1000° C., a very sharp drop in surface area was obtained.

Further confirmation of the present results is to be found in the work of other authors (10, 18, 21). The observations are consistent with sintering of the material in the region of its Tammann temperature; for silica, this is about 790° C.

An explanation for this loss of area and pore volume with no marked change in pore distribution would seem to be fairly straightforward. The authors suggest that the fine pores in the gel collapse first following heat treatment. Concurrent with this, some slightly larger pores are reduced in diameter to that of the collapsed pores. This process is visualized as extending over the entire range of pores in the gel. The result would be an over-all reduction in the number of pores, but a retention of the same distribution characteristics.

Figures 3 and 4 show clearly that slightly different distributions are indicated, according to which branch of the isotherm is chosen for distribution determinations. Hence, a sharp maximum at 38 A. is found in the desorption data which is not indicated in equivalent adsorption distributions. Pores of this order of diameter are just emptied at $p/p_o \sim 0.425$, which corresponds in each case to the low-pressure end of the hysteresis loop in the isotherms (Figure 1). This feature is a manifestation of the hysteresis phenomena discussed earlier. The diameter, 38 A., would then correspond to the diameter of the constriction or narrow region of the ink-bottle pores. Obviously, pores of this shape persist to high dehydration temperatures.

Surface area distributions for Davison Gel treated at 240° and 900° C. are illustrated in Figure 5. Again, the same differences between adsorption and desorption distributions occur. It emerges that the largest peaks in the distribution curves occur at regions of lower pore diameter than indicated by corresponding volume distribution data. This is consistent with the observation that the surface area of porous materials tends to be a function of the smaller pores.

Figure 8 shows the adsorption and desorption volume distribution data on Crosfield Microporous gel, Neosyl F.X. silica, and High Surface Area gel. Results indicate that for the former two materials, adsorption and desorption data give the same distribution patterns, which concurs with the absence of significant hysteresis phenomena, characteristic of very fine-pored materials (11). Neosyl F.X. silica is not a gel, but a pure, very finely divided silica. A significant contribution to surface area will come from the small-particle character of the material. Results on the High Surface Area gel follow the pattern on Davison xerogel. Hence, the peak in the desorption distribution at 38 A.—corresponding to $p/p_o \sim 0.425$ —comes from the emptying of pores which have some form of constriction or narrowed portion, as described earlier.

Surface area distributions follow exactly the same patterns as the volume data. Adsorption and desorption distributions coincide for Microporous and Neosyl F.X. materials, but not for the High Surface Area gel. Again, as with Davison xerogel, the largest maxima occur at the lowest pore diameters, indicating that the small pores make the largest contribution to the over-all area. All these materials have some micropores ($d < \sim 20$ A.), which can largely affect the accuracy of the B.E.T. surface area.

The volume and area adsorption and desorption distributions for Crosfield Macroporous gel (Figure 11) illustrate the discrepancies which can occur, depending on which isotherm branch is used for the calculation. A much larger average pore diameter (173 A.) is indicated by the desorption data (86 A. from the adsorption isotherm). The figures illustrate most clearly that surface area is mainly a function of small pores, with relatively minor area contributions from larger pores, >200-A. diameter.

Following Cranston and Inkley's (7) views, it is interesting to compare results obtained from the adsorption and desorption isotherms for nitrogen on the Davison and Crosfield materials, with the corresponding B.E.T. data (Tables I and II). Cranston and Inkley's original assumptions included the view that the pores were cylindrical with one end closed. Hence, if this and the other assumptions are valid for the present materials, a comparison of the B.E.T. specific surface area (S) and the cumulative surface area should show good agreement.

Tables I and II show that for the higher surface area materials, namely, the Davison gel at 240°, 500°, and 600° C., both S'_a/S (the cumulative surface area obtained from the adsorption isotherm, divided by the B.E.T. surface area) and S'_d/S (the cumulative surface area from the desorption isotherm divided by the B.E.T. value) lie between 0.822 and 0.921. As the temperature of heat treatment of this material is increased to 900°C., the surface area falls and these fractions move closer to unity. If we assume that the B.E.T. surface areas can be taken as norms for Davison xerogel, as discussed, then these deviations point to more complicated pore shapes. This is as expected and indeed indicated by the hysteresis loop in the isotherms (Figure 1). Above 600° C., the convergence of the fractions $S'_a/$ S and S'_d/S towards unity indicates that the number of constricted pores is being steadily reduced. This is seen also in the steady compression of the hysteresis loop.

A further comparison in the tables between the volume adsorbed at $p/p_o = 0.931$, as measured on both branches of the isotherms ($V_{0.931}^a$ and $V_{0.931}^d$), and the cumulative volumes measured by the Cranston and Inkley method (V'_a and V'_d) shows the same trend. Hence, the fractions $V'_a/V_{0.931}^a$ and $V'_d/V_{0.931}^d$ tend towards unity for the gel pretreated at 700°, 800°, and 900° C. Results also show that the adsorption and desorption isotherms lead to almost equal values for the above fractions.

In spite of the fact that true equilibrium is attained only on the adsorption isotherm, it is apparent in this case that either branch can be used to obtain general information from pore distribution plots. Hence, comparison of both data yields useful information on pore shapes, with particular regard to constrictions in the pores.

With reference to the Crosfield adsorbents, Tables I and II show that for the Microporous, High Surface Area, and Neosyl F.X. materials, either the adsorption or desorption isotherms can be used for evaluation of distribution data. The fractions S'_a/S , S'_a/S , $V'_a/V^a_{0.331}$, and $V'_d/V^d_{0.331}$ tend to be generally <0.8, as was found for the lower temperatures of heat-treatment on Davison gels. This is expected for Microporous and Neosyl F.X. materials, where fine pores can cause large errors in monolayer capacity and would tend to decrease the values of S'_a/S and S'_a/S from unity, as observed. Similar considerations apply to the latter two factors. Here, capillary condensation in pores too small to be included in the Cranston and Inkley calculation— <10 A. diameter—would also give rise to deviations from unity.

For the Macroporous gel (Figure 10), the desorption isotherm gives a much more significant picture of the pore structure. Hence, comparison between the B.E.T. surface area and cumulative area as determined from the adsorption isotherm leads to a figure of 0.784 for S'_{a}/S . On the other hand, $S'_a/S = 0.846$. Similarly, $V'_a/V^a_{0.931} = 0.968$, whereas $V'_{d}/V'_{0.931} = 0.996$. Calculations also reveal an average pore diameter for this material of 85.5 A., as measured from the adsorption data. Desorption data lead to a figure of 173.2 A. Analysis of the quantity of nitrogen adsorbed on this Macroporous material above and below $p/p_a \sim$ 0.9 shows that the latter figure is more likely to be correct, and substantiates the above view regarding the desorption isotherm.

Thus, both branches of the isotherms lead to similar results, with the exception of the Crosfield Macroporous gel, where the desorption data seem most apt. In most cases, comparison between volume distributions from both branches leads to useful insight into pore shapes. Cumulative surface areas (S'_{a}) S'_{a} and volumes (V'_{a}, V'_{d}) are smaller than equivalent B.E.T. data (S) and saturation adsorbed volumes $(V_{0.931}^a, V_{0.931}^d)$ in all cases. This is particularly true for the Microporous materials, where deviations are greatest (Table II). As stated, all the present materials possess a degree of microporosity. Differences between the B.E.T. and cumulative results can then be attributed to two main factors. First, microporosity in porous solids leads to deviations in the B.E.T. monolayer capacities. Thus, in most cases, and especially with Microporous silica gel and Neosyl F.X. pure silica, the surface area (S) is undoubtedly too large. The second important factor is that differences in cumulative volumes and saturation adsorbed volumes can be attributed to capillary condensation in pores too small to be included in the Cranston and Inkley calculation (<10 A.). Thus, a certain fraction of the surface areas of these materials is not included in the cumulative area calculations, leading to the deviations noted in Table II.

In addition to the factors outlined above, deviations from Cranston and Inkley's original assumptions must be considered. All the pores are not cylindrical with one end closed. This certainly changes the shapes of the pore distributions, but apparently does not affect the cumulative information obtained. Table II shows that similar values for all the fractions noted are found in calculations from both the adsorption and desorption isotherms. The exception to this is Crosfield Macroporous gel. Here, it seems more feasible to choose the desorption data for subsequent calculations. This material exhibits an isotherm characteristic of a porous solid having roughly spherical particles and large-diameter pores (<150 A.) approximately cylindrical in shape. Again, deviations between B.E.T. and cumulative data can only be attributed to the presence of microporosity in this gel.

NOMENCLATURE

- = specific surface areas from Table I
- S_a' surface area obtained from the adsorption branch of the = isotherm (cumulative S.A.)

- S'_d = surface area obtained from the desorption branch of the isotherm (cumulative S.A.)
- volume of nitrogen adsorbed (cc. N.T.P. per gram) at $V_{0.931}^{a}$ a relative pressure of 0.931 on the adsorption branch of the isotherm
- $V_{0.931}^{d}$ = volume of nitrogen adsorbed (cc. N.T.P. per gram) at a relative pressure of 0.931 on the desorption branch of the isotherm
 - V'_a = cumulative pore volume (cc. N.T.P. per gram) obtained from the adsorption branch of the isotherm
 - cumulative pore volume (cc. N.T.P. per gram) obtained $V'_{d} =$ from the desorption branch of the isotherm
- an increment in pore volume (cc. N.T.P. per gram) $\Delta V =$
- ΔS = an increment in surface area (meters² per gram)
- an increment in pore diameter (A.) Δd =

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