# Vapor-Liquid Equilibria for Solutions of Dendritic Polymers

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Vapor—liquid equilibrium data were obtained for dendritic polymer solutions using a classic isothermal gravimetric-sorption method; the amount of solvent absorbed by the dendrimer was measured at increasing solvent activity. The polymers were polyamidoamine (PAMAM) dendrimers of generations 1, 2, and 4 and benzyl ether dendrimers with different end groups (aromatic rings, dodecyl chains, methyl ester groups, perfluoroalkyl chains) of generations 2 to 6, and two series of benzyl ether linear polymers that are analogues of the dendrimers. Solvents were acetone, acetonitrile, chloroform, cyclohexane, methanol, *n*-pentane, *n*-propylamine, tetrahydrofuran, and toluene. The temperature range was 35 to 89 °C. The amount of solvent absorbed by the dendrimers depends, sometimes strongly, on the kind of dendrimer end groups. The relation between solvent absorption and dendrimer generation number, or molecular weight, depends on the solvent–dendrimer system and on temperature. Solvent absorption in linear polymers is below that for corresponding dendrimers, all or in part owing to crystallinity in the linear polymers.

### Introduction

Dendrimers (also called cascade polymers or starburst molecules) are highly branched, treelike macromolecules with a branch point at each monomer unit. They consist of a central core, concentric "shells", and an external surface. Each family of dendrimers, i.e., dendrimers made with the same repeat unit, consists of different generations, each corresponding to a different number of shells around the core. In the past 10 years many kinds of dendrimers have been synthesized. Detailed reviews are given by Tomalia and Durst (1993), Newkome (1994), Newkome et al. (1996), and Frechet and Hawker (1996).

For dendrimers, the variety of molecular structure, size, shape, topology, flexibility, and surface chemistry offers a wide variety of possible applications for these new materials. Organometallic dendrimers may provide a new class of advanced catalysts that combine the advantages of heterogeneous and homogeneous catalysis. Since the catalytic sites can be located at the surface of the dendrimer, the activity of each individual site is maximized. Recovery of the dendrimer catalyst is easily achieved by ultrafiltration owing to the well-defined size of the macromolecules (Knapen, 1994).

Balzani (1994) suggests using transition metal-based dendrimers as devices for harvesting solar energy. The peripheral groups of the molecule collect sunlight, which is then channeled through the supramolecular dendrimer array and concentrated in a specific central site.

Potential dendrimer applications are also in medicine and biotechnology. For example, Wiener et al. (1994) studied the use of polyamidoamine (PAMAM) dendrimers as magnetic-resonance imaging contrast agents. Haensler and Szoka (1993) showed that PAMAM dendrimers are suitable for gene-transfer vehicles, and Barth et al. (1994) studied the use of boronated dendrimers delivered by monoclonal antibodies in cancer therapy.

Dendrimers with an interior hydrophobic core surrounded by a hydrophilic surface layer behave as unimolecular micelles capable of solubilizing hydrophobic molecules in aqueous solutions without a critical micelle concentration (Hawker et al., 1993). These characteristics suggest application of dendrimers in a recyclable solubilization and extraction procedure that can be used in the recovery of organic substances from water. Jansen et al. (1994) demonstrated that it is possible to encapsulate guest molecules in a dendritic polymer, the so-called "dendriticbox", by constructing a dense shell in the presence of the guest molecules. In a successive paper (Jansen and Meijer, 1995), shape-selective liberation was studied for molecules trapped in the dendrimer.

The capability of starburst molecules to host small molecules makes them attractive as drug-delivery agents (Tomalia, 1991). Further, if dendrimers are connected in a network, they could be used as filters, membranes, chromatographic materials and adsorbents. The ability to control their size, shape, and surface chemistry offers ideas for applications as molecular electronic devices, biosensors, chemical sensors. For example, Wells and Crooks (1996) showed that dendrimer monolayers provide suitable interfaces for chemical sensing applications and that the film reactivity and selectivity toward vapor-phase dosants depend on the generation of PAMAM used.

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Table 1.	Properties	of the	Solvents
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solvent	supplier	lot number	purity (%)
acetone	Fisher Scientific <sup>a</sup>	952818	99.6
acetonitrile	Fisher Scientific <sup>a</sup>	932673	99.9
chloroform	Fisher Scientific <sup>a</sup>	952629	99.9
cyclohexane	Fisher Scientific <sup>a</sup>	952398	99.9
methanol	Fisher Scientific <sup>a</sup>	952430	99.9
<i>n</i> -pentane	Fisher Scientific <sup>a</sup>	920827	99+
<i>n</i> -propylamine	Aldrich Chemical	24095-8	99+
	Company, Inc. <sup>b</sup>		
tetrahydrofuran	Fisher Scientific <sup>a</sup>	902693	99.9
toluene	Fisher Scientific <sup>a</sup>	933715	99.9

<sup>*a*</sup> Pittsburgh, PA. <sup>*b*</sup> Milwaukee, WI.

Table 2. PAMAM Dendrimers' Properties

generation	$M_{ m W}{}^a$	surface groups
1	1430	8
2	3256	16
4	14215	64

<sup>*a*</sup>  $M_{\rm W}$  = molecular weight.

Grinthal (1993) affirms that polypropyleneamine dendrimers are easier to process than the analogous linear polymers, since they have lower viscosities. This property allows injection molding of some dendrimer thermoplastics even if the same is not possible for corresponding linear polymers.

The purpose of this work is to determine how polymer– solvent vapor–liquid equilibria (VLE) change with dendrimer generation number and with surface groups and to what extent the solubility of a solvent in a dendrimer differs from that in a corresponding linear polymer of the same chemical structure.

This work reports binary VLE data for several polar and nonpolar solvents in polyamidoamine (PAMAM) dendrimers (generations 1, 2, 4), in benzyl ether dendrimers with different end groups (generations 2 to 6), and in linear benzyl ether polymers.

### Materials, Experimental Method, and Results

Table 1 shows characteristics of the solvents. The solvents were degassed with a standard freeze-thaw procedure and used without further purification.

The PAMAM (polyamidoamine) dendrimers (G1, G2, G4, G = generation) were bought from Dendritech through Aldrich (Milwaukee, WI, catalog numbers: 41,238-4, 41,-240-6, and 41,244-9 for G1, G2, and G4, respectively). They had a 4-functional diamine core and primary amine (NH<sub>2</sub>) end groups. They are sequenced copolymers of ethylenediamine and methyl acrylate. The repeat unit is (NCH<sub>2</sub>-CH<sub>2</sub>CONHCH<sub>2</sub>CH<sub>2</sub>). Their properties are shown in Table 2. Their glass-transition temperatures are between 0 and 15 °C, increasing with generation number. Because PAM-AM dendrimers are sold in methanol solution (20 wt % of polymer), they must be dried before performing the VLE experiment. We used high vacuum (26.7-53.3 Pa) until the mass of the dendrimer solution was constant; we assumed then that no solvent was present. Owing to strong hydrogen bonding between PAMAM and methanol, the removal of traces of methanol was slow; several days (from 5 to 7) were required to dry the dendrimers.

Keeping pure PAMAM dendrimers at 35–40 °C for several days should not affect their structure (Spindler, 1996). However, Dendritech warns that prolonged heating of dendrimer solutions at temperatures over 45 °C can lead to formation of defect structures; further, the presence of other components could affect stability. For our experimental apparatus, we could not use temperatures lower than 35 °C. Also, below about 50 °C, the vapor pressures of the solvents would be too low to perform a meaningful experiment.

Benzyl ether dendrimers are synthesized with the convergent method: the construction of the macromolecule is started at what finally becomes its "periphery". 3,5-Dihydroxyl benzyl alcohol is the dendrimer building block (Hawker and Frechet, 1990). Because the core of the molecules is bifunctional, two main branches emanate from the interior.

We studied several types of benzyl ether dendrimers, all with the same interior core but different end groups (aromatic rings (AR), dodecyl chains (C12), aromatic rings with a methyl ester group in para-position (ME), perfluoroalkyl chains ( $CF_3-(CF_2)_3-CH_2-O-$ ) (F)). Figure 1 shows the chemical structure of a dendrimer of generation 3. The dendrimers' characteristics are given in Table 3. D224 indicates an AR dendrimer formed by three generation-4 wedges linked to a central 3-functional core.

Benzyl ether linear polymers were studied to compare the solvent absorption of polymers with the same chemical structure but different geometry.

There are two possible series of linear polymers analogous to the AR dendrimers (Hawker et al., 1997). The first, derived from the polymerization of 3-hydroxybenzyl alcohol, can be considered the linear analogue of the polyether dendrimer minus the numerous chain-end groups. The second, derived from 3-hydroxy-5-benzyloxibenzyl alcohol, can be considered the "exact" linear analogue of the dendrimers. With ELA-Gn we indicate the exact linear analogue of the AR-Gn dendrimer. ELA and AR have exactly the same molecular formulas.

Table 4 gives characteristics of the linear and ELA polymers. L[7]2 is obtained by condensing two L[7] polymers, thus obtaining a linear polymer with two aromatic rings as chain ends.

The polydispersity of the benzyl ether dendrimers and linear polymers is very low (1.002-1.01) owing to the stepwise method used for their synthesis. The benzyl ether dendrimers and linear polymers were synthesized at IBM Almaden Research Center, San Jose, CA.

Before starting the VLE experiment, the polymers were dried overnight under vacuum at 50–70  $^\circ\mathrm{C}$  to remove any traces of solvents or monomers remaining after polymerization.

The experimental method for VLE of polymer solutions is based on the classical gravimetric sorption technique [see, for example, Panayiotou and Vera (1984)]. The apparatus and experimental procedure are described in detail by Gupta and Prausnitz (1995).

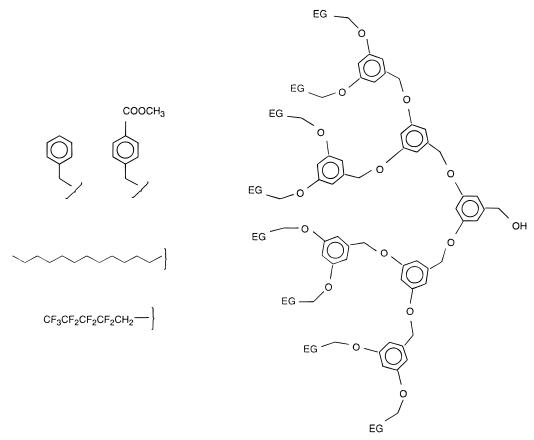
Binary systems studied are given in Table 5. Experimental VLE are given in Table 6.

We plot the solvent liquid mass fraction ( $w_1$ ) versus solvent activity ( $a_1$ ), defined as the ratio of the pressure to the saturation pressure of the pure solvent ( $P/P^{\circ}$ ). We used the equations suggested by Daubert and Danner (1989) to calculate the saturation pressure.

The temperature was controlled within  $\pm 0.3$  °C. The uncertainty in the pressure reading is 0.13 kPa. The uncertainty in  $w_1$  is 5–10% below  $w_1 = 0.1$  and 2–5% above  $w_1 = 0.1$ .

### Solvent-Induced Crystallization (SINC)

Solvent-induced crystallization was observed during absorption of acetone at 50 °C; chloroform at 50 °C, and toluene at 70 °C in AR-G3 (Figure 2); chloroform at 50 °C



**Figure 1.** Chemical structure of a benzyl ether dendrimer of generation 3. "EG" stands for end group. On the left, four different end groups are shown: aromatic rings, aromatic rings with a methyl ester group in the para-position, dodecyl chains, and perfluoroalkyl chains.

Table 3. Properties of the Benzyl Ether Dendrimers

polymer	termination	generation	$M_{ m W}$	surface groups	<i>T</i> g <sup><i>a</i></sup> /°C
AR-G3	aromatic ring	3	1592	8	40
AR-G4	aromatic ring	4	3288	16	40
AR-G5	aromatic ring	5	6680	32	40
AR-G6	aromatic ring	6	13464	64	40
D224	aromatic ring		10116	48	40
F	perfluoroalkyl chain	4	5560	16	65
C12-G3	dodecyl alkane	3	2216	8	20
C12-G4	dodecyl alkane	4	4536	16	20
C12-G5	dodecyl alkane	5	9176	32	20
ME-G2	methyl ester	2	976	4	29
ME-G3	methyl ester	3	2056	8	52
ME-G4	methyl ester	4	4216	16	59

<sup>*a*</sup>  $T_{\rm g}$  = glass-transition temperature.

# Table 4. Properties of the Benzyl Ether Linear Polymers

polymer	$M_{ m W}$	total no. of aromatic rings	no. of terminal aromatic rings	$T_{\rm m}{}^{a\!/\! m o}{ m C}$	$T_{\rm g}/^{\circ}{ m C}$
L[7]	850	8	1	70	2
L[15]	1698	16	1	92	23
L[7]2	1850	18	2	101	28
ELA G2	747	7	4	110	
ELA G3	1592	15	8	130	40 - 43
ELA G4	3288	31	16	140	40 - 43
ELA G5	6680	63	32	150	40 - 43

<sup>*a*</sup>  $T_{\rm m}$  = melting temperature.

in ME-G2, G3, G4; toluene at 70 °C and acetone at 50 °C in ME-G4; THF at 70 °C in ELA-G4; chloroform at 50 °C in ELA-G3, G4; and toluene at 70 °C in ELA-G3, G4, G5.

Table 5. Systems Studied

polymer	solvent	t/°C
AR-G3, AR-G4, AR-G5	acetone	50
AR-G3, AR-G4, AR-G5	chloroform	50
AR-G3, AR-G4, AR-G5, AR-G6	toluene	70
C12-G3, C12-G4, C12-G5	acetone	50
C12-G3, C12-G4, C12-G5	chloroform	50
C12-G3, C12-G4, C12-G5	cyclohexane	60
C12-G3, C12-G4, C12-G5	<i>n</i> -pentane	40
C12-G3, C12-G4, C12-G5	toluene	70
ME-G2, ME-G3, ME-G4	chloroform	50
ME-G4, F, D224	acetone	50
ME-G4, F, D224	toluene	70
AR-G4, L[7]	acetone	70
L[7], L[15], L[7]2	chloroform	50
AR-G4, AR-G5, L[7]	chloroform	70
AR-G4, AR-G6, L[7]	toluene	70
AR-G4, AR-G6, L[7]	tetrahydrofuran	70
AR-G4, L[15], L[7]2	toluene	89
AR-G5, ELA-G5, L[15]	tetrahydrofuran	70
ELA-G2, ELA-G3, ELA-G4	chloroform	50
ELA-G2, ELA-G4	tetrahydrofuran	70
ELA-G3, ELA-G4, ELA-G5	toluene	70
PAMAM-G1, G2, G4	acetone	35
PAMAM-G1, G2, G4	acetonitrile	40
PAMAM-G1, G2, G4	chloroform	35
PAMAM-G1, G2, G4	methanol	35
PAMAM-G1, G2, G4	<i>n</i> -propylamine	35

The initial solvent-free polymer was in the amorphous state. At low solvent absorption, the polymer behaved in the usual way: absorption of the solvent in the liquid polymer increased with rising solvent vapor pressure, but when the solvent mass fraction had reached about 0.04, the polymer started to reject the solvent (Figure 2). This rejection corresponds to the onset of crystallization in the polymer; during this process, the solvent is rejected from

	•	-	-	•								
		(1			cetone; $t = 5$			(		(1		
<i>w</i> <sub>1</sub> (ARG3)	<i>w</i> <sub>1</sub> (ARG4)	<i>w</i> <sub>1</sub> (ARG5)	P/kPa	<i>w</i> <sub>1</sub> (ARG3)	<i>w</i> <sub>1</sub> (ARG4)	<i>w</i> <sub>1</sub> (ARG5)	P/kPa	w1(ARG3)	<i>w</i> <sub>1</sub> (ARG4)	<i>w</i> <sub>1</sub> (ARG5)	P/kPa	
$0.013 \\ 0.022$	$0.009 \\ 0.015$	0.011 0.012	$15.9 \\ 21.8$	0.031 0.023	0.100	0.089	$40.6 \\ 47.2$	0.045 0.072	0.176 0.220	$0.159 \\ 0.205$	60.7 67.6	
0.029	0.020	0.019	27.2	0.031	0.124	0.114	53.1	0.093	0.255	0.238	71.4	
0.039	0.039 0.039 0.040 34.6 Solvent: Chloroform; $t = 50$ °C; $P = 69.2$ kPa											
W (ADC 2)	$w(\Lambda \mathbf{PC}\mathbf{A})$	we (APC 5)		Solvent: Ch $w_1(ARG3)$	$\frac{10roform; t =}{w_1(ARG4)}$		69.2 кРа <i>Р</i> /kРа	<i>w</i> <sub>1</sub> (ARG3)	$u_{\ell}(\Lambda \mathbf{PC}\mathbf{A})$	W(ADC5)	<i>P</i> /kPa	
<i>w</i> <sub>1</sub> (ARG3) 0.024	<i>w</i> <sub>1</sub> (ARG4) 0.038	<i>w</i> <sub>1</sub> (ARG5) 0.038	9.1	0.040	0.205	<i>w</i> <sub>1</sub> (ARG5) 0.172	33.1	0.090	0.383	<i>w</i> <sub>1</sub> (ARG5) 0.356	49.0	
0.051	0.057	0.046	15.0	0.040	0.265	0.172	38.8	0.030	0.385	0.330	53.7	
0.029 0.030	$0.109 \\ 0.159$	$0.086 \\ 0.121$	21.6 28.1	0.073	0.328	0.299	44.0	0.234	0.540	0.484	58.5	
0.050	0.155	0.121	20.1	Solvente T	aluona $t = 7$	$0 \circ C \cdot D = 2$	) 1 kDo					
<i>w</i> <sub>1</sub> (ARG3)	<i>P</i> /kPa	<i>w</i> <sub>1</sub> (ARG3)	<i>P</i> /kPa	$w_1(ARG3)$	oluene; $t = 7$ P/kPa	$\frac{0}{w_1}$ (ARG3)	$\frac{D.1 \text{ kPa}}{P/\text{kPa}}$	w <sub>1</sub> (ARG3)	<i>P</i> /kPa	<i>w</i> <sub>1</sub> (ARG3)	<i>P</i> /kPa	
0.020	5.4	0.011	9.8	0.012	11.1	0.014	15.5	0.019	20.1	0.030	24.3	
0.020	8.7	0.011	10.4	0.012	11.1	0.014	15.5	0.015	20.1	0.050	24.5	
				Solvent: T	oluene; $t = 7$	$^{\circ}$ C; $P^{s} = 30$	).1 kPa					
$W_1(ARG4)$	$w_1(ARG5)$	$W_1(ARG6)$	P/kPa	$w_1(ARG4)$	$w_1(ARG5)$	$W_1(ARG6)$	₽⁄kPa	$w_1(ARG4)$	$W_1(ARG5)$	$W_1(ARG6)$	P/kPa	
0.012 0.026	$0.006 \\ 0.021$	0.012 0.028	4.7 8.0	0.076 0.113	0.070	0.090 0.133	$\begin{array}{c} 14.4 \\ 17.5 \end{array}$	0.158 0.209	0.170 0.236	0.195 0.266	$\begin{array}{c} 21.3\\ 24.3\end{array}$	
0.020	0.021	0.028	11.1	0.115		0.155	17.5	0.209	0.230	0.200	24.3	
				Solvent: A	cetone; $t = 5$	$0 ^{\circ}\mathrm{C}; P^{\mathrm{s}} = 81$	l.9 kPa					
w1(D224)	<i>w</i> <sub>1</sub> (MEG4)	<i>w</i> <sub>1</sub> (FG4)	P/kPa	w1(D224)	<i>w</i> <sub>1</sub> (MEG4)	<i>w</i> <sub>1</sub> (FG4)	P/kPa	w1(D224)	<i>w</i> <sub>1</sub> (MEG4)	<i>w</i> <sub>1</sub> (FG4)	P/kPa	
0.010	0.021	0.026	16.3	0.062	0.058	0.071	37.4	0.138	0.126	0.144	58.3	
0.020 0.038	$0.030 \\ 0.033$	$0.036 \\ 0.046$	23.1 28.7	$0.089 \\ 0.114$	$0.089 \\ 0.110$	$0.094 \\ 0.119$	45.7 52.8	0.193	0.170	0.242	68.7	
				Solvent: T	oluene; $t = 7$	$0 ^{\circ}\text{C}: P^{\text{s}} = 30$	).1 kPa					
w1(D224)	<i>w</i> <sub>1</sub> (MEG4)	w1(FG4)	<i>P</i> /kPa	w <sub>1</sub> (D224)	w1(MEG4)	W1(FG4)	<i>P</i> /kPa	w1(D224)	<i>w</i> <sub>1</sub> (MEG4)	<i>w</i> <sub>1</sub> (FG4)	<i>P</i> /kPa	
0.018	0.023	0.019	7.1	0.060	0.034	0.054	14.8	0.118	0.099	0.107	19.8	
0.027 0.043	$0.023 \\ 0.022$	0.020 0.036	8.8 12.0	0.088	0.045	0.082	17.5	0.140	0.130	0.129	21.3	
0.010	0.022	0.000	12.0	Salvanti A	cetone; $t = 5$	$0 \circ C \cdot D = 0$						
$W_1(C12G3)$	W1(C12G4)	W1(C12G5)	<i>P</i> /kPa					W1(C12G3)	w <sub>1</sub> (C12G4)	w1(C12G5)	<i>P</i> /kPa	
0.011	0.014	0.011	13.3	0.053	0.050	0.043	38.2	0.124	0.116	0.095	61.7	
0.025	0.026	0.022	22.1	0.070	0.067	0.056	45.9	0.159	0.148	0.120	68.9	
0.035	0.034	0.029	30.0	0.094	0.087	0.073	53.3	0.200	0.185	0.147	74.3	
		(		Solvent: Ch						(*******		
<i>w</i> <sub>1</sub> (C12G3)										<i>w</i> <sub>1</sub> (C12G5)		
0.083 0.130	0.080 0.124	$0.079 \\ 0.122$	11.5 18.1	0.270 0.358	$0.265 \\ 0.360$	$0.269 \\ 0.364$	$\begin{array}{c} 34.6\\ 42.9\end{array}$	$0.600 \\ 0.653$	$0.622 \\ 0.690$	0.645 0.718	60.7 62.7	
0.182	0.178	0.179	24.8	0.458	0.466	0.476	52.7					
			2	Solvent: Cyc	lohexane; <i>t</i> =	= 60 °C; <i>P</i> <sup>s</sup> =	52.0 kPa	a				
$W_1(C12G3)$	$W_1(C12G4)$	<i>w</i> <sub>1</sub> (C12G5)	₽⁄kPa	$W_1(C12G3)$	$W_1(C12G4)$	<i>w</i> <sub>1</sub> (C12G5)	₽⁄kPa	$W_1(C12G3)$	<i>w</i> <sub>1</sub> (C12G4)	$W_1(C12G5)$	<i>P</i> /kPa	
0.034	0.033	0.030	9.5	0.154	0.153	0.136	28.5	0.294	0.296	0.261	42.1	
$0.069 \\ 0.099$	0.067 0.098	$0.058 \\ 0.089$	$\begin{array}{c} 15.9 \\ 21.1 \end{array}$	0.211	0.211	0.184	35.3	0.367	0.371	0.313	45.4	
				Solvent: <i>n</i> -F	Pentane; $t = $	40 °C; ₽ <sup>s</sup> =1	16.3 kPa					
w1(C12G3)	w1(C12G4)	w1(C12G5)	P/kPa	<i>w</i> <sub>1</sub> (C12G3)	w1(C12G4)	w1(C12G5)	<i>P</i> /kPa	w1(C12G3)	w1(C12G4)	w1(C12G5)	P/kPa	
0.025	0.021	0.024	21.2	0.071	0.074	0.075	54.3	0.156	0.156	0.169	86.4	
0.033 0.054	$0.032 \\ 0.055$	$0.034 \\ 0.059$	$\begin{array}{c} 31.6\\ 43.5\end{array}$	0.082 0.106	$0.077 \\ 0.104$	$0.084 \\ 0.110$	58.2 70.9	0.221	0.225	0.250	100.6	
0.034	0.055	0.033	45.5									
w <sub>1</sub> (C12G3)	w1(C12G4)	W4(C12C5)	<i>P</i> /kPa	Solvent: 1 $w_1(C12G3)$	oluene; $t = 7$			W4(C12C3)	w <sub>1</sub> (C12G4)	w <sub>1</sub> (C12G5)	<i>P</i> /kPa	
0.064	0.053	0.058	5.2	0.205	0.169	0.178	14.3	0.323	0.262	0.278	19.9	
0.106	0.090	0.093	8.5	0.262	0.218	0.226	17.0	0.435	0.338	0.365	23.3	
0.157	0.129	0.138	11.7									
	0	0		Solvent: Ch					(1) (2) (2)	(1)	D/1 5	
$\frac{W_1(\text{MEG2})}{0.018}$	W1(MEG3)	w <sub>1</sub> (MEG4)		<i>w</i> <sub>1</sub> (MEG2)	,	w <sub>1</sub> (MEG4)		<i>w</i> <sub>1</sub> (MEG2)	<i>w</i> <sub>1</sub> (MEG3)	-, ,	<i>P</i> /kPa	
0.018 0.033	0.022 0.039	0.054 0.082	9.3 15.7	0.051 0.084	0.044 0.065	0.118 0.203	$\begin{array}{c} 28.3\\ 34.3 \end{array}$	0.121 0.177	$0.091 \\ 0.116$	$0.302 \\ 0.346$	43.9 49.6	
0.045	0.035	0.089	22.1	0.100	0.084	0.260	39.3	0.271	0.159	0.403	54.8	

 Table 6. Experimental Vapor-Liquid Equilibria<sup>a</sup>

Table 6 (C	ontinued)				~		50 5 L D				
(1 [7])		D/I D	(1.[7])		cetone; $t = 7$			D/I D	(1.[7])		D/1 D
<i>w</i> <sub>1</sub> (L[7])	<i>w</i> <sub>1</sub> (ARG4)		$w_1(L[7])$	w1(ARG4)		<i>w</i> <sub>1</sub> (L[7])	w1(ARG4)		<i>w</i> <sub>1</sub> (L[7])	w1(ARG4)	P/kPa
0.018 0.030	0.011 0.015 0.022	22.9 29.1 35.2	$0.047 \\ 0.074 \\ 0.096$	$\begin{array}{c} 0.030 \\ 0.054 \\ 0.071 \end{array}$	49.4 65.5 78.5	0.128 0.162	0.097 0.121	95.3 109.1	0.181 0.187	0.131 0.139	113.5 118.1
				Solvent: Ch	loroform; t =	= 50 °C; <i>P</i> <sup>s</sup> =	= 69.2 kPa				
$w_1(L[7])$	$W_1(L[15])$	$W_1(L[7]2)$	<i>P</i> /kPa	$W_1(L[7])$	$W_1(L[15])$	$W_1(L[7]2)$	₽⁄kPa	$W_1(L[7])$	$W_1(L[15])$	$W_1(L[7]2)$	<i>P</i> /kPa
0.007	0.003	0.004	7.9	0.168	0.082	0.030	39.3	0.355	0.140	0.064	51.9
0.026 0.036	0.021 0.032	0.007 0.012	16.4 23.8	$0.185 \\ 0.314$	0.094 0.122	0.037 0.048	41.3 48.6	0.412 0.463	0.166 0.189	0.087 0.161	$\begin{array}{c} 57.0\\ 61.5\end{array}$
0.052	0.048	0.012	31.3						0.100	0.101	01.0
					loroform; t =						
	<i>w</i> <sub>1</sub> (ARG4)	<i>w</i> <sub>1</sub> (ARG5)	<i>P</i> /kPa	$W_1(L[7])$	<i>w</i> <sub>1</sub> (ARG4)	$W_1(ARG5)$	P/kPa	$W_1(L[7])$	<i>w</i> <sub>1</sub> (ARG4)	$W_1(ARG5)$	P/kPa
0.012	0.008	0.016	7.1	0.123	0.107	0.132	46.5		0.287	0.320	90.6
0.024 0.051	0.022 0.043	0.034 0.056	$17.4 \\ 23.7$	0.166 0.207	0.152 0.190	$0.176 \\ 0.215$	57.9 68.9		0.353 0.441	$0.389 \\ 0.486$	102.7 113.3
0.082	0.072	0.092	34.7	0.258	0.230	0.262	78.9		0.111	0.100	110.0
				Solvent: 7	Coluene; $t = 7$	70 °C; $P^{s} = 3$	30.1 kPa				
$W_1(L[7])$	w <sub>1</sub> (ARG4)	$W_1(ARG5)$	<i>P</i> /kPa	$W_1(L[7])$	<i>w</i> <sub>1</sub> (ARG4)	w <sub>1</sub> (ARG5)	P/kPa	$W_1(L[7])$	w <sub>1</sub> (ARG4)	$W_1(ARG5)$	<i>P</i> /kPa
0.038	0.032	0.032	6.3	0.119	0.110	0.111	15.7	0.211	0.192	0.205	22.1
0.091	0.082	0.087	13.3	0.151	0.134	0.145	18.7	0.245	0.226	0.242	23.1
			Sol	vent: Tetral	hydrofuran;	$t = 70 ^{\circ}\text{C}; P$	s = 115.2  k	Pa			
$W_1(ARG4)$	$W_1(ARG6)$	$W_1(L[7])$	<i>P</i> /kPa	$W_1(ARG4)$	$W_1(ARG6)$	$W_1(L[7])$	<i>P</i> /kPa	$W_1(ARG4)$	$W_1(ARG6)$	$W_1(L[7])$	<i>P</i> /kPa
0.047	0.035	0.071	23.3	0.119	0.120	0.172	52.5	0.209	0.217	0.297	78.9
0.071 0.098	$0.063 \\ 0.092$	0.102 0.137	$\begin{array}{c} 34.1 \\ 44.1 \end{array}$	$0.150 \\ 0.184$	$\begin{array}{c} 0.148 \\ 0.188 \end{array}$	0.213 0.262		0.252 0.300	$0.267 \\ 0.322$	0.375 0.452	87.7 93.3
0.038	0.032	0.157	44.1					0.300	0.322	0.452	33.5
···· (I [15])	(I [7]9)	w (ADC 4)	D/l-Do		Coluene; $t = 8$			(I [15])	(I [7]9)	w (ADC 4)	D/l-Do
<i>w</i> <sub>1</sub> (L[15])	<i>w</i> <sub>1</sub> (L[7]2)	<i>w</i> <sub>1</sub> (ARG4)	P/kPa	<i>w</i> <sub>1</sub> (L[15])	$w_1(L[7]2)$	<i>w</i> <sub>1</sub> (ARG4)	P/kPa	<i>w</i> <sub>1</sub> (L[15])	$w_1(L[7]2)$	<i>w</i> <sub>1</sub> (ARG4)	P/kPa
$0.010 \\ 0.022$	0.005 0.023	0.029 0.053	7.9 15.7	0.033 0.046	0.043 0.069	$0.102 \\ 0.138$	28.8 36.7	0.053 0.058	0.081 0.081	$0.166 \\ 0.176$	$\begin{array}{c} 43.1\\ 44.5\end{array}$
0.026	0.029	0.058	21.5	01010	01000	01100	0011	01000	01001	01210	1110
			Sol	vent: Tetral	hydrofuran;	$t = 70 ^{\circ}\text{C}; P$	s = 115.2  k	Pa			
$W_1(ARG5)$	$W_1$ (ELAG5)	$W_1(L[15])$	<i>P</i> /kPa	$W_1(ARG5)$	w1(ELAG5)	$W_1(L[15])$	) <i>P</i> /kPa	$W_1(ARG5)$	$W_1$ (ELAG5)	$W_1(L[15])$	<i>P</i> /kPa
	0.013	0.012	13.7	0.103	0.046	0.053	49.5	0.196	0.099	0.098	78.3
0.042 0.066	0.019 0.036	$0.032 \\ 0.045$	$\begin{array}{c} 23.2\\ 35.3 \end{array}$	$0.133 \\ 0.165$	$0.071 \\ 0.085$	$0.079 \\ 0.090$	60.3 68.3	0.285	0.154	0.147	98.4
0.000	0.000	0.045									
(EL A C 9)	(EL A C 2)	W (ELACA			loroform; $t =$				W (ELAC2)	w <sub>1</sub> (ELAG4	) <i>D</i> /l <sub>2</sub> D <sub>0</sub>
	0.008	0.025	-		0.101				0.177	0.206	49.5
$\begin{array}{c} 0.010\\ 0.014\end{array}$	0.008	0.025	$11.5 \\ 27.9$	0.015	0.101	0.130	34.3 42.5	0.028	0.177	0.200	49.5 56.3
			Sol	vent: Tetral	hydrofuran; a	t = 70 °C: P	s = 115.2  k	Pa			
W1(ELAG	2) <i>P</i> /kPa	w1(EL			$w_1$ (ELAG2)	<i>P</i> /kPa	w <sub>1</sub> (ELA		kPa w <sub>1</sub> (	ELAG2)	<i>P</i> /kPa
0.006	22.7	0.01		65.9	0.284	78.7	0.30			0.334	87.0
01000		0101			hydrofuran;						0110
W1(ELAG4)	<i>P</i> /kPa	w1(ELAG4)	<i>P</i> /kPa	w <sub>1</sub> (ELAG4	<i>,</i>	$\frac{u = 70^{\circ} \text{ C}, P}{w_1(\text{ELAG4})}$		$\frac{W_1(ELAG4)}{W_1(ELAG4)}$	) <i>P</i> /kPa	w1(ELAG4)	<i>P</i> /kPa
					,	- , ,		1. ,		- , ,	
$0.012 \\ 0.027$	$13.1 \\ 23.1$	0.032	34.8	0.055	48.1	0.068	60.7	0.089	72.1	0.129	92.3
				Solvent: 7	Coluene; $t = 7$	70 °C · <i>D</i> s — '	20.1 kPa				
W1(ELAG3)	Wa (FI AG4)	W4(FLAC5	) <i>P</i> /kPa					W4(FLAC3)	Wa (FI AC4)	W1(ELAG5	) <i>P</i> /kPa
0.019	WI(LLAU4)	0.010	6.5	0.052	0.053	0.040	15.8	0.095	0.092	0.078	24.0
0.019	0.041	0.010	11.5	0.032	0.068	0.040	19.9	0.035	0.032	0.078	24.0
				Solvent: A	Acetone; $t = 3$	35 °C; $P^{s} = A$	46.4 kPa				
W1(PAMA)	MG1) w <sub>1</sub> (	PAMAMG2)	) w <sub>1</sub> (P	AMAMG4)	<i>P</i> /kPa	w1(PAMAN		(PAMAMG	2) w <sub>1</sub> (PA	MAMG4)	<i>P</i> /kPa
0.010	, .	0.012		0.001	9.5	0.025	,	0.021	, ,	.019	22.1
0.013		0.013		0.008	12.0	0.056		0.047	0	.048	27.5
0.014		0.017		0.012	16.3	0.141		0.124	0	.113	30.0
				Solvent: Ac	etonitrile; <i>t</i> =	= 40 °C; <i>P</i> <sup>s</sup> =	= 23.5 kPa				
W1(PAMAN	MG1) w <sub>1</sub> (	PAMAMG2)	) w <sub>1</sub> (P	AMAMG4)	P/kPa	W1(PAMAN	/IG1) w	1(PAMAMG	2) w <sub>1</sub> (PA	MAMG4)	<i>P</i> /kPa
0.012		0.011		0.007	12.5	0.046		0.046	0	.037	20.1
0.024		0.021		0.019	16.8						

Table 6 (Continued)

Table	6	(Continue	ed)
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Solvent: Chloroform; $t = 35$ °C; $P^s = 39.9$ kPa									
w1(PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	<i>P</i> /kPa	$W_1$ (PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	₽⁄kPa		
0.002	0.001		8.5	0.279	0.223	0.189	26.2		
0.011	0.007	0.002	12.2	0.316	0.254	0.221	29.5		
0.018	0.015	0.005	15.7	0.361	0.298	0.273	32.4		
0.194	0.163	0.124	22.3	0.411	0.354	0.322	34.9		
0.246	0.191	0.158	23.2						
		Solvent: Me	thanol; <i>t</i> =	= 35 °C; $P^{\rm s}$ = 27.9 k	Pa				
w1(PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	<i>P</i> /kPa	w1(PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	₽⁄kPa		
0.048	0.037	0.029	6.7	0.236	0.223	0.190	19.2		
0.104	0.088	0.081	11.7	0.297	0.282	0.240	21.8		
0.161	0.141	0.128	15.9	0.378	0.375	0.328	23.9		
		Solvent: <i>n</i> -Pro	pylamine;	$t = 35 \text{ °C}; P^{s} = 62.7$	7 kPa				
w1(PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	<i>P</i> /kPa	$W_1$ (PAMAMG1)	w1(PAMAMG2)	w1(PAMAMG4)	₽⁄kPa		
0.004	0.003	0.002	18.1	0.025	0.026	0.019	34.8		
0.007	0.008	0.004	24.2	0.041	0.041	0.035	40.2		
0.014	0.017	0.007	29.7	0.071	0.071	0.065	47.2		

<sup>a</sup>  $w_1$ (POLY) = solvent weight fraction in the liquid phase in solution with the polymer "POLY". P(kPa) = vapor pressure in kPa. ARGn= aromatic-ring-terminated benzyl ether dendrimers of generation Gn. C12Gn = dodecyl-terminated benzyl ether dendrimers of generation Gn. MEGn = methyl-ester-terminated benzyl ether dendrimers of generation n. FGn = perfluoroalkyl-chain-terminated benzyl ether dendrimers of generation n. L[n] = benzyl ether linear polymer n. ELAGn = exact linear analog of the ARGn.

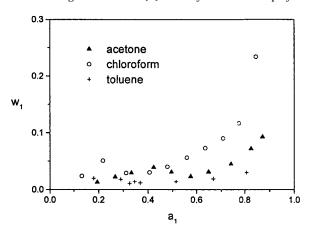


Figure 2. Activity of chloroform at 50 °C, acetone at 50 °C, and toluene at 70 °C in AR-G3. Solvent-induced crystallization phenomenon.

the crystallizing regions that are not able to dissolve solvents, and the extent of absorption falls. At higher solvent pressure, the degree of crystallinity remains constant and further absorption of the solvent takes place only in the noncrystalline regions.

Solvent-induced crystallization (SINC) phenomena have been reported previously, for example, for polycarbonate films (Kambour et al., 1966), isotactic polystyrene (Overbergh et al., 1975), and isotactic polypropylene (Vittoria, 1991; Vittoria et al., 1989).

Crystallization of polymers is facilitated by absorption of organic vapor because the low-molecular-weight plasticizer significantly increases the rate of crystallization. The dissolved solvent increases the nucleation probability and facilitates a higher rate of chain-segment diffusion. Crystallization rates are maximum at the temperature where the effect of the thermodynamic driving force, which increases with subcooling  $(t_m - t)$ , is balanced by the decrease in polymer mobility with cooling. The ability of the polymer to diffuse is proportional to  $(t - t_g)$ . Therefore, crystallization rates are maximum somewhere between melting temperature,  $t_m$ , and glass-transition temperature, tg.

The presence of a solvent lowers a polymer's  $t_{g}$  and  $t_{m}$ . When the amount of solvent in the polymer is sufficient to decrease  $t_g$  such that the polymer chains have enough mobility at *t*, and the degree of undercooling  $(t_m - t)$  is high enough, crystallization starts in the polymer. For AR-G3,  $t_{\rm g}$  is near 40 °C and  $t_{\rm m}$  is 125 °C. Absorption experiments were performed at 50 °C for acetone and chloroform and at 70 °C for toluene. At 70 °C, the amount of solvent required to induce crystallization was much less than that at 50 °C.

Following solvent-induced crystallization, crystallinity in the polymer samples was confirmed by a DSC (differential scanning calorimetry) experiment. For AR-G3, t<sub>m</sub> was 125 °C. Generations 1 and 2 of these dendrimers are obtained as crystalline solids with  $t_m = 85$  and 110 °C, respectively, while generations 3, 4, and 5 do not crystallize when synthesized (Hawker and Frechet, 1990).

For the ME-dendrimers, the higher the generation, the higher the amount of solvent required to crystallize the polymer. This trend follows because it is more difficult for larger molecules to arrange in an ordered way. Melting temperatures of the ME-dendrimers are 120, 122, and 94 °C for G2, G3, and G4, respectively.

DSC was performed twice for each polymer sample that experienced SINC. The first DSC run was for the sample that had been used in the VLE experiment; that sample showed a clear peak indicating crystallinity in the polymer. After the first DSC run, the sample (now totally liquefied) was cooled to room temperature and the DSC run was repeated. The second thermogram did not show any crystalline peak. The polymer did not crystallize during the cooling process, whereas the sample from the VLE experiment did show crystallinity because plasticizing solvent had been present during the VLE experiment.

### Discussion

PAMAM dendrimers are strongly hydrophilic polymers, completely miscible in water, lower alcohols, glycols, and ethylenediamine. They are insoluble in nonpolar solvents as confirmed by an experiment with carbon tetrachloride at 40 °C, when no absorption of the solvent was observed even at  $P/P^{s}$  equal to 0.6. PAMAM dendrimers are stable in methanol at room temperature, but a side reaction,

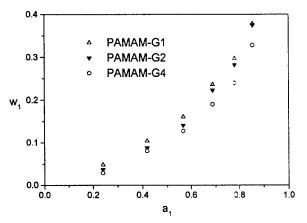
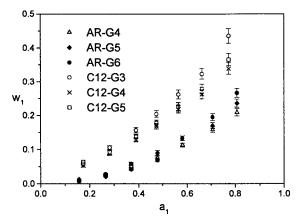


Figure 3. Activity of methanol at 35  $^\circ C$  in PAMAM-G1, G2, and G4.



**Figure 4.** Activity of toluene at 70 °C in AR-G4, AR-G5, AR-G6, C12-G3, C12-G4, and C12-G5.

known as retro-Michael reaction, can occur. This reaction accelerates at higher temperature and in protic solvents. Although our experiments with methanol were not longer than 1 week, we cannot be certain that no reaction occurred. However, we were able to reproduce our data.

Low absorption was observed with *n*-propylamine, probably because this amine is not sufficiently polar for the dendrimers. There is some adsorption of acetone in the PAMAM, but this may be caused by the Schiff base reaction: amines react strongly with molecules that contain an aldehyde or ketone group. The system PAMAM– chloroform at 35 °C shows a strange shape for the VLE curve: at the beginning the polymer does not absorb much chloroform, but, above  $w_1 = 0.15$ , it appears that chloroform is a good solvent for these polymers. For all solvents studied here, the amount of solvent absorbed was higher for lower generations (Figure 3). This trend is the one we would expect from the molecular weight of the dendrimers: the lower the polymer and solvent.

For the AR and C12 dendrimers there is not much difference in solvent absorption for different generations. For the AR dendrimers, the aromatic-ring/oxygen ratio is constant for every generation. Therefore, no significant change in chemical composition of the dendrimer occurs when changing generation.

For the AR dendrimers, the data with acetone and chloroform at 50 °C show that G4 absorbs more than G5. However, at 70 °C, this trend is inverted: G5 or G6 absorb more chloroform, toluene (Figure 4), or THF (Figure 5) than G4. However, the difference in absorption is small and not significantly higher than the experimental uncertainty, as

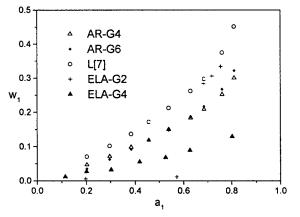


Figure 5. Activity of THF at 70 °C in AR-G4, AR-G6, L[7], ELA-G2, and ELA-G4.

shown in Figure 4. We studied only generations lower than G6. Hawker et al. (1993), Naylor et al. (1989), and Hawker et al. (1997) reported that for dendritic macromolecules, the transition from an open to a more globular structure occurs near G4. Generations higher than 5 might show differences in thermodynamic behavior larger than those reported here.

As expected, the VLE behavior of dendrimers varies with the chemistry of the surface groups. Comparison between the amount of toluene absorbed by C12 and AR dendrimers shows that C12 dendrimers absorb more toluene than the AR dendrimers (Figure 4). This result is surprising because the chemical affinity between the aromatic solvent and the dendrimer surface groups is expected to be stronger for the aromatic surface groups than for the paraffinic surface groups. Figure 4 shows that toluene is not a good solvent for AR dendrimers or, at least, not as good as suggested by the aromatic rings in the dendrimer. On the other hand, toluene and alkyl chains are compatible: toluene and octane mix well, and toluene is a good solvent for polyethylene above 60–70 °C (see Polymer Handbook, 1989). Therefore, a possible explanation for the fact that toluene is a better solvent for the C12 dendrimers than for the AR dendrimers is that the affinity between toluene and the aromatic rings of the dendrimer is decreased by oxygen atoms in the polymer.

Absorption of cyclohexane in the AR dendrimers (G4, G5, G6) at 60 °C is very small: at solvent activity 0.71 the solvent weight fraction in G4 is only 0.013. Therefore, it appears that the absorption of alkanes in the C12 dendrimers is due to the favorable presence of the C12 alkyl terminal groups, despite the unfavorable aromatic groups.

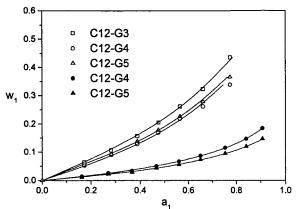
Acetone is a better solvent for AR than for C12 dendrimers, at least above  $w_1 = 0.05$ . Again, a likely explanation is that polar acetone interacts more favorably with the oxygen-containing groups of the AR dendrimers than with nonpolar alkyl chains in the C12 dendrimers.

The magnitude of the change in polymer–solvent compatibility depends strongly on the chemistry of the end groups. D224 and F dendrimers show similar VLE behavior in acetone and toluene. Probably the surface groups (aromatic rings and perfluoroalkyl chains) are too similar with respect to these two solvents to show a significant difference in the VLE data.

The VLE data show that absorption and solubility properties of dendrimers can be altered dramatically by changing the dendrimer chain-end functional groups. The solution properties of dendrimers that are completely insoluble in some solvents (for example, AR dendrimers in alkanes) or only partially compatible with other solvents

Table 7.	Parameters	Used y	with	Lattice-Cluster	Theory
(LCT)					

system	spacers/ arm	<i>r</i> (no. of polymer segments)	$\chi$ (interaction parameter)
AR-G4 in chloroform 50 °C	1	31	0.353
AR-G5 in chloroform 50 °C	1	127	0.476
AR-G4 in toluene 70 °C	1	31	0.875
AR-G6 in toluene 70 °C	1	127	0.692
C12-G4 in acetone 50 °C	2	61	1.144
C12-G5 in acetone 50 °C	2	125	1.280
C12-G3 in toluene 70 °C	1	15	-0.229
C12-G4 in toluene 70 °C	1	31	-0.013
C12-G5 in toluene 70 °C	1	63	-0.139



**Figure 6.** Fit (solid lines) with LCT of VLE data for C12dendrimers in toluene at 70  $^{\circ}$ C (open symbols) and in acetone at 50  $^{\circ}$ C (solid symbols).

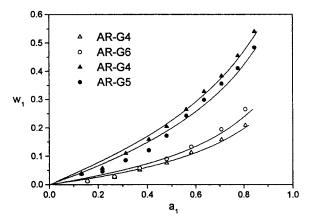
(for example, AR dendrimers in toluene) are modified significantly by simply adding different end groups to the dendritic molecules (for example, the dodecyl chains of the C12 dendrimers).

We fitted our VLE data to Freed's lattice-cluster theory (LCT) (Freed and Bawendi, 1989; Nemirovsky et al., 1992) as applied by Lue and Prausnitz (1997) to dendritic polymers. LCT represent the properties of Flory's lattice much better than does Flory–Huggins theory; it fits polymer–solvent VLE data better (Mio et al., 1997). LCT theory has only one adjustable parameter, interaction parameter:  $\chi = (\epsilon z)/2kT$ , where *k* is Boltzmann's constant, *T* is the temperature, *z* is the lattice coordination number (set equal to 6), and  $\epsilon$  is the energy parameter.

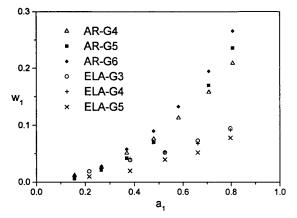
The other parameters used in LCT, called indices, are calculated from the geometry of the dendrimer molecules. Following Lue and Prausnitz (1997), who reported the indices for 3-functional-core dendrimers, it is straightforward to calculate the indices for our case, bifunctional-core dendrimers, given the generation number and the number of spacers per arm. Table 7 gives the geometric parameters of the dendrimer (spacer/dendrimer arm and total number of dendrimer segments, *r*) together with the  $\chi$  parameter obtained from the fit. The negative values of  $\chi$  for the system C12 dendrimer in toluene cannot be due to highly favorable interactions (e.g., hydrogen-bonding). However, the negative  $\chi$  indicates that toluene is a very good solvent for C12 dendrimers.

Figure 6 shows that LCT gives a good fit for the C12 dendrimers, but LCT is not able to reproduce the VLE curve of the AR dendrimers with the same accuracy (Figure 7).

In general, the initial slopes of sorption curves for the AR dendrimers are larger than those for C12 dendrimers (Figure 4), especially at lower temperatures (50 °C). Larger



**Figure 7.** Fit (solid lines) with LCT of VLE data for ARdendrimers in toluene at 70  $^{\circ}$ C (open symbols) and in chloroform at 50  $^{\circ}$ C (solid symbols).



**Figure 8.** Activity of toluene at 70 °C in AR-G4, AR-G5, AR-G6, ELA-G3, ELA-G4, and ELA-G5.

slopes may be due to the stiffness of the AR-dendrimers; further, at 50 °C, the polymers are only 10 °C above their glass-transition temperature. Therefore, the polymers cannot be easily permeated by the solvent. Only after some solvent has been absorbed does the polymer become sufficiently flexible and permeable. If a desorption experiment follows an absorption experiment, then, at the same solvent activity, the solvent mass fraction is higher for the desorption curve. Similar hysteresis phenomena have been observed previously (e.g., Bonner and Prausnitz, 1974); hysteresis is explained by the different bulk structure that the polymer assumes on desorption.

Because the linear polymers were in the semicrystalline state at the temperature of the experiment, or because they experienced SINC, it was not possible to compare data for liquid dendrimers and liquid linear polymers. Figures 5 and 8 show that, in general, linear polymers absorb less solvent than the corresponding dendrimers (L[7] data are discussed later). However, because linear polymers are semicrystalline while dendrimers are liquid with a lower  $t_{\rm g}$ , solubility of a solvent in the linear polymers is lower. It was not possible to raise the experimental temperature above the  $t_{\rm m}$  of the polymers owing to limitations of the apparatus.

Because of crystallinity, it is not surprising that linear polymers absorb less solvent than dendrimers. Similarly, linear analogues are less soluble in solvents than dendrimers (Hawker et al., 1997).

The data presented in this work show that, at the same temperature, there is a significant difference in the absorption behavior of two chemically identical polymers, one dendritic (amorphous) and the other linear (partially crystalline). This difference in solution properties follows from the branched structure of the dendrimer. Branching makes crystallization more difficult; therefore, the permeability of the highly branched polymer to the solvent exceeds that of the linear polymer.

In Figure 5 the sudden increase in THF absorption by ELA-G2 at 70 °C is caused by the dissolution of the polymer's crystalline regions by the solvent (Allen et al., 1965). At the beginning, the polymer is partially in the crystalline state, but at solvent activity around 0.6, enough solvent is present to dissolve all of the polymer, crystalline or amorphous, thus forming a liquid solution of polymer and solvent. Dissolution of the polymer's crystalline regions is also observed for L[7] with chloroform at 50 °C.

Since L[7] is the only sample with low enough  $T_{\rm m}$  (70 °C), we can only compare VLE behavior of L[7] and ARdendrimers at 70 °C or higher. At this temperature, there is no crystallinity. With toluene at 70 °C and chloroform at 70 °C, L[7] absorbs almost the same amount of solvent as AR dendrimers. L[7] absorbs more acetone than the dendrimers; with THF (Figure 5) this difference is larger (around 30% solvent weight fraction). Solvent molecules probably reach all segments of the L[7] molecules but not the internal groups of the dendrimers. This effect is more evident with THF and acetone probably because these solvents have a higher affinity with the ether links of the polymers. For toluene, there are two opposite effects: one is the steric hindrance and the other the different chemical affinity of toluene with the aromatic end groups and the ether-linked internal groups. The segments of L[7] are more accessible to toluene than are the internal dendrimer groups, but toluene has higher affinity to the dendrimersurface aromatic groups that to the ether-linked groups. While L[7] has only one aromatic chain end, the dendrimer has several. Because these two effects work in opposite directions, the absorption of toluene in the dendrimer and in L[7] is roughly the same.

### Conclusions

VLE of solutions containing PAMAM dendrimers (G1, G2, G4), benzyl ether dendrimers (from G2 to G6) or linear polymers were studied with several polar and nonpolar solvents at 35-89 °C.

The generation number does not significantly change the absorption behavior of dendrimers. Dendrimer surfacegroup chemistry is an important factor in determining the solubility of solvents in the dendrimers. The linear analogues absorbed less solvent than the dendrimers because they were partially in the crystalline state at the temperature of the experiment, while the dendrimers were liquid.

It is likely that there are larger differences in VLE of solutions containing linear or dendritic polymers occurring for generation numbers higher than 5 for more dilute polymer solutions.

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#### **Literature Cited**

- Allen, G.; Booth C.; Gee, G.; Jones, M. N. Effect of Incipient Crystallinity on the Swelling of Polypropylene in Diethyl Ketone. Part IV. *Polymer* **1965**, *5*, 367–376.
- Balzani, V. Greener Way to Solar Power. New Sci. 1994, 144, 31–34.
- Barth, R. F.; Adams, D. M.; Soloway, A. H.; Alam, F.; Darby, M. V. Boronated Starburst Dendrimer-Monoclonal Antibody Immunoconjugates: Evaluaion as a Potentail Delivery System for Neutron Capture Therapy. *Bioconjugate Chem.* **1994**, *5*, 58–66.

- Bonner, D. C.; Prausnitz, J. M. Thermodynamic Properties of Some Concentrated Polymer Solutions. J. Polym. Sci., Polm. Phys. Ed. 1974, 12, 51–73.
- Brandrup, J.; Immergut, E. H. *Polymer Handbook*, 3rd ed.; Wiley: New York, 1989.
- Daubert, T. E.; Danner, R. P. Physical and Thermodynamic Properties of Pure Chemicals: Data Compilation; Hemisphere Publishing Corp.: New York, 1989.
- Frechet, J. M. J.; Hawker, C. J. Synthesis and Properties of Dendrimers and Hyperbranched Polymers; Comprehensive Polymer Science, 2nd suppl.; Aggarwal, S., Russo, S., Eds.; Pergamon Press: Oxford, 1996; pp 234–298.
- Freed, K. F.; Bawendi M. G. Lattice Theories of Polymeric Fluids. J. Phys. Chem. **1989**, 93, 2194–2203.
- Grinthal, W. Dendrimers Branched out into the Marketplace. *Chem. Eng. News* **1993**, Nov, p 51.
- Gupta, R. B.; Prausnitz, J. M. Vapor–Liquid Equilibria of Copolymer + Solvent and Homopolymer + Solvent Binaries: New Experimental Data and Their Correlation. J. Chem. Eng. Data 1995, 40, 784– 791.
- Haensler J.; Szoka, F. Polyamidoamine Cascade Polymers Mediate Efficient Transfection of Cells in Culture. *Bioconjugate Chem.* 1993, 4, 372–379.
- Hawker, C. J.; Frechet, J. M. J. A New Convergent Approach to Monodisperse Dendritic Macromolecules. J. Chem. Soc., Chem. Commun. 1990, 15, 1010–1013.
- Hawker, C. J.; Frechet, J. M. J. Preparation of Polymers with Controlled Molecular Architecture. A New Convergent Approach to Dendritic Macromolecules. J. Am. Chem. Soc. 1990, 112, 7638– 7647.
- Hawker, C. J.; Wooley, K. L.; Frechet, J. M. J. Unimolecular Micelles and Globular Amphipiles: Dendritic Macromolecules as Novel Recyclable Solubilization Agents. *J. Chem. Soc., Perkin Trans.* 1993, *1*, 1287–1297.
- Hawker, C. J.; Wooley, K. L.; Frechet, J. M. J. Solvatochromism as a Probe of the Microenvironment in Dendritic Polyethers-Transition from an Extended to a Globular Structure. J. Am. Chem. Soc. 1993, 115, 4375–4376.
- Hawker, C. J.; Malmstrom, E. E.; Frank, C. W.; Kampf, J. P. Exact Linear Analogs of Dendritic Polyether Macromolecules: Design, Synthesis and Unique Properties. J. Am. Chem. Soc. 1997, 119, 9903–9907.
- Jansen, J.; de Brabander-van den Berg, E.; Meijer, E. Encapsulation of Guest Molecules into a Dendritic Box. Science 1994, 226, 1226– 1229.
- Jansen, J.; Meijer, E.; de Brabander-van den Berg, E. The Dendritic Box: Shape-Selective Liberation of Encapsulated Guests. J. Am. Chem. Soc. 1995, 117, 4417–4418.
- Kambour, R. P.; Karasz, F. E.; Daane, J. H. Kinetic and Equilibrium Phenomena in the System: Acetone Vapor and Polycarbonate Film. *J. Polym. Sci., Polym. Phys. Ed.* **1966**, *4*, 327–347.
- Knapen, J. W. J.; van der Made, A. W.; de Wilde, J. C.; van Leeuwen, P. W. N. M.; Wijkens, P.; Grove, D. M.; van Koten, G. Homogeneous Catalysts Based on Silane Dendrimers Functionalized with Arylnickel(II) Complexes. *Nature* **1994**, *372*, 659–663.
- Lue L.; Prausnitz J. M. Structure and Thermodynamics of Homogeneous-Dendritic-Polymer Solutions: Computer Simulation, Integral Equation, and Lattice-Cluster Theory. *Macromolecules* **1997**, *30*, 6650–6657.
- Mio, C.; Jayachandran, K. N.; Prausnitz, J. M. Vapor-Liquid Equilibria for Binary Solutions of Some Comb Polymers Based on Poly(Styreneco-Maleic Anhydride) in Acetone, Methanol and Cyclohexane. *Fluid Phase Equilibr.* **1997**, *141*, 165–178.
- Naylor, A. M.; Goddard, W. A., III; Kiefer, G. E.; Tomalia, D. A. Starburst Dendrimers: Molecular-Level Control of Size, Shape, Surface Chemistry, Topology, and Flexibility from Atoms to Macroscopic Matter. J. Am. Chem. Soc. 1989, 111, 2339–2341.
- Nemirovsky, A. M.; Dudowicz, J.; Freed, K. F. Dense Self-Interacting Lattice Trees with Specified Topologies: from Light to Dense Branching. *Phys. Rev. A* **1992**, *45*, 7111–7127.
- Newkome, G. R. Advances in Dendritic Macromolecules; Jai Press Inc.: Greenwich, CT, 1994.
- Newkome, G. R.; Moorefield, C. N.; Vögtle, F. *Dendritic Molecules. Concepts, Syntheses, Perspectives*; VHC: Weinheim, Germany, 1996.
- Overbergh, N.; Berghmans, H.; Smets, G. Crystallization of Isotactic Polystyrene Induced by Organic Vapours. *Polymer* **1975**, *16*, 703– 710.
- Panayiotou, C. P.; Vera, J. H. Thermodynamic of Polymer–Polymer– Solvent and Block Copolymer–Solvent Systems. I. Experimental Systems. *Polym. J.* 1984, 16, 89–102.
- Spindler, R. Personal communication, 1996.
- Tomalia, D. A. Meet the Molecular Superstars. New Sci. 1991, 132, 30–34.
- Tomalia, D. A.; Durst, H. D. Genealogically Directed Synthesis: Starburst/Cascade Dendrimers and Hyperbranched Structures. *Top. Curr. Chem.* 1993, 165, 193–312.

- Vittoria, V. Solvent-Induced Crystallization of Isotactic Polypropylene in Cyclohexane at Different Temperatures. *Polymer* **1991**, *32*, 856– 859.
- Vittoria, V.; Olley, R. H.; Bassett, D. C. Structural Changes in Solvent-Induced Crystallization of Quenched Isotactic Polypropylene. *Colloid Polym. Sci.* **1989**, *267*, 661–667.
- Wells, M.; Crooks, R. M. Interactions between Organized, Surface-Confined Monolayers and Vapor-Phase Probe Molecules. 10. Preparation and Properties of Chemically Sensitive Dendrimers Surfaces. J. Am. Chem. Soc. 1996, 118, 3988–3989.
- Wiener, E. C.; Brechbiel, M. W.; Brothers, H.; Magin, R. L.; Gansow, O. A.; Tomalia, D. A.; Lauterbur, P. C. Dendrimer-Based Metal Chelates: A New Class of Magnetic Resonance Imaging Contrast Agents. *Magn. Reson. Med.* **1994**, *31*, 1–8.

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