

**Preliminary communication**

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**Chromium, molybdenum, and tungsten organometallic polymeric networks with aryldiisocyanide ligands**

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**Abstract**

Organometallic complex precursors of Cr, Mo, and W react readily with various aryldiisocyanide ligands to form insoluble polymers with the general formula  $[M((CN)_2Ar)_x \cdot yH_2O]_n$ , ( $M = W$ ,  $x = 3.30 \pm 0.45$ ;  $M = Cr, Mo$ ,  $x = 2.00 \pm 0.5$ ,  $y = 0-3$ ), in which the metal is in a zero-valent oxidation state. Various oxidized polymers of this type are also described.

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We have been involved in the design and preparation of organometallic polymeric materials with stereochemically rigid aryldiisocyanide [1] ligands exhibiting ordered networks and having a defined microstructure.  $Rh^I$  [2-6],  $Ir^I$  [5],  $Pd^{0,II}$  [7,8], and  $Pt^{0,II}$  [7,8] aryldiisocyanide polymers have all been prepared by this approach. In the framework of new materials research, catalytic [7-10] and electronic [2,11,12] features of these materials have been studied. Chromium(0,II,III), molybdenum(0,II), and tungsten(0,II) homoleptic monoisocyanides of the designated oxidation states are known [13]. These metals therefore appear to be good candidates for the preparation of mixed-oxidation state polymers, and it is in this context that we report the preparation of the first Cr, Mo, and W aryldiisocyanide macromolecular networks.

$[Cr_2(OAc)_4 \cdot 2H_2O]$ ,  $[Mo_2(OAc)_4]$ , and  $W_2(dmhp)_4$  ( $dmhp = 2,4$ -dimethyl-6-hydroxypyrimidine) were treated with various aryldiisocyanides. The chromium and molybdenum acetate as well as  $W_2(dmhp)_4$  precursors in ethanol yielded red-brown fine powders having  $\nu(CN)$  infrared absorption bands at  $\sim 2100$ ,  $\sim 1960$  (br)  $cm^{-1}$  for the  $Cr^0$  product (1) and  $\sim 2119$ ,  $\sim 1960$  (br)  $cm^{-1}$ , for both the  $Mo^0$  (2) and  $W^0$  (3) polymers. Acetonitrile, however, is a better solvent for both the molybdenum and chromium acetate precursors and the aryldiisocyanide ligands. Brown-powdered insoluble products with similar  $\nu(CN)$  infrared bands were isolated for the  $Mo^0$  (4) polymers from this solvent. The reaction with the  $W_2(dmhp)_4$  dimers in methylene chloride gave a dark brown-black powdered polymeric product (5) having  $\nu(CN)$  absorptions at  $\sim 2119$ ,  $\sim 1960$  (br)  $cm^{-1}$ .

The products show strong infrared bands at 1596 and 1576  $\text{cm}^{-1}$  with a shoulder at 1660  $\text{cm}^{-1}$ . These bands do not correspond to those of the precursor ligands, as would probably be the case if incomplete substitution by the aryldiisocyanides had occurred. We believe that the observed bands may be attributed to either two or four electron bridging isocyanides [8,14,15] or carbene-like metal bonds formed from isocyano functionalities with these-electron rich metals [16,17].

A ligand-to-metal ratio of three would be expected if all the aryldiisocyanides were terminally bound, as in the case of an infinite polymer. Bridging aryldiisocyanides in infinite polymers would reduce this ratio. Small oligomers would be expected to give a ligand-to-metal ratio significantly larger than that found. Thus our results appear to be consistent with the presence of extended polymeric networks. The infrared bands in the region  $\sim 2119$  to  $\sim 1960$   $\text{cm}^{-1}$  are reminiscent of that of Cr, Mo, and W hexakis(monoisocyanide) complexes in which there is a significant deviation from  $O_h$  symmetry [18,19]. The  $\Delta\nu(\text{CN})$  found in this region for the Cr, Mo, and W polymers is however larger than that found for similar hexakis(monoisocyanide) complexes ( $\Delta\nu(\text{CN})$  40–60  $\text{cm}^{-1}$ ). These differences may arise from the symmetry changes imposed around the metal atoms due to the bridging or carbene-like aryldiisocyanide ligands mentioned above.

Microanalytical data and thermal gravimetric analysis have indicated that the materials should be formulated as  $[\text{M}((\text{CN})_2\text{Ar})_x \cdot y\text{H}_2\text{O}]_n$  ( $\text{M} = \text{W}$ ,  $x = 3.30 \pm 0.45$ ;  $\text{M} = \text{Mo}$ ,  $\text{Cr}$ ,  $x = 2.00 \pm 0.5$ ;  $y \approx 3$ ). The new materials immediately undergo oxidation even when only traces of oxygen are present. These oxidation changes can be monitored using infrared spectroscopy; disappearance of the  $\sim 2100$  and  $\sim 1960$   $\text{cm}^{-1}$  bands in the case of  $\text{Cr}^0$  (1) networks is accompanied by the appearance of  $\nu(\text{CN})$  bands at  $\sim 2053$  and  $\sim 2119$   $\text{cm}^{-1}$ . For the  $\text{Mo}^0$  (2, 4) and  $\text{W}^0$  (3, 5) polymers the  $\sim 1960$   $\text{cm}^{-1}$  band disappears. In the case of the tungsten polymers a new band at  $\sim 2085$   $\text{cm}^{-1}$  grows up. With the molybdenum materials a band at 2119  $\text{cm}^{-1}$  is observed (Fig. 1).

The zero-valent polymers also underwent facile oxidations with  $\text{I}_2$  at room temperature to give  $[\text{M}((\text{CN})_2\text{Ar})_x\text{I}] \cdot y\text{H}_2\text{O}]_n$ ,  $\text{M} = \text{W}$ ,  $x = 2.9 \pm 0.32$ ,  $y = 2.82$ ;  $\text{M} = \text{Mo}$ ,  $x = 2.2 \pm 0.33$ ,  $y = 3.69$ . Infrared data (Fig. 1) for the products of the reaction with iodine of the  $\text{Mo}^0$  (2, 4) and  $\text{W}^0$  (5) polymers were similar to those of the air oxidations, but there is faster and cleaner formation of the +2 oxidation product. Reactions with  $\text{M}(\text{CNR})_6$ , ( $\text{M} = \text{Mo}$ ,  $\text{W}$ ;  $\text{R} = \text{Ph}$ ) to yield  $[\text{M}(\text{CNR})_6\text{I}]$  complexes having the metal in a +2 oxidation state are known [20].

The UV-VIS diffuse reflectance spectra recorded for  $\text{Mo}^0$  (4) and  $\text{W}^0$  (5) polymers show strong very broad extended absorption bands throughout the visible range, while the  $\text{W}^0$  (3), polymer reveals a distinct but broadened absorption maxima at  $\sim 510$  nm, as do the  $\text{Cr}^0$  (1) and  $\text{Mo}^0$  (2) polymers. The differences between the UV-VIS absorptions and those for monomeric hexakis(arylisocyanide)- $\text{Cr}^0$ ,  $\text{Mo}^0$ , and  $\text{W}^0$  complexes [21], may be due to the presence of somewhat extended  $p$ - $\pi$  orbitals and electron delocalization between metal sites in the polymers. The  $\text{Cr}^0$  (1),  $\text{Mo}^0$  (2), and  $\text{W}^0$  (3), aryldiisocyanide compounds reveal visible absorptions which are red shifted compared to those of the monomeric hexakis(aryliisocyanide) analogues but do not extend throughout the visible range. The broad extended absorption bands observed for  $\text{Mo}^0$  (4) and  $\text{W}^0$  (5) polymers may then be due to mixtures of these polymers with various network sizes. An additional explanation for the absorbance maxima for the polymers  $\text{Mo}^0$  (2) and  $\text{W}^0$

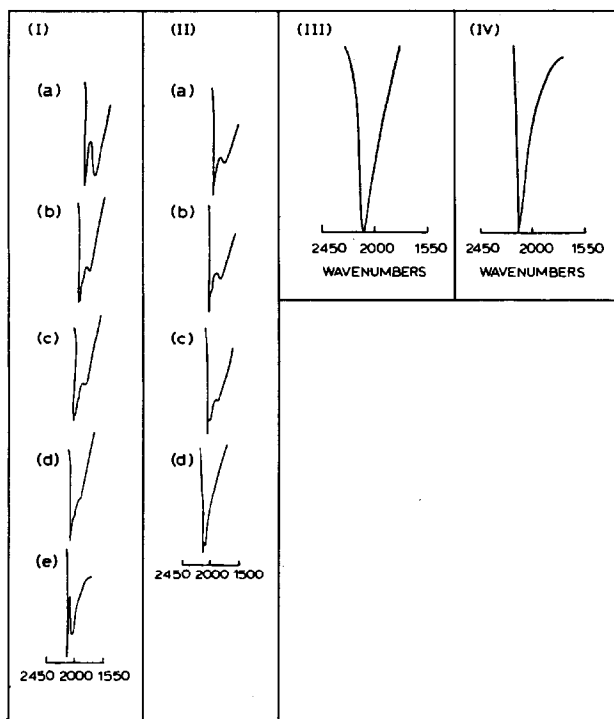


Fig. 1. Oxidations of metal aryldiisocyanide polymers, aryldiisocyanide = 4,4'-diisocyanobiphenyl. (I) M = Cr, (a) unoxidized polymer, (b) 1 day air exposure, (c) 2 days air exposure, (d) 4 days air exposure, (e) 8 days air exposure; (II) M = W, (a) unoxidized polymer, (b) 3-5 min air exposure, (c) 20 min air exposure, (d) several days air exposure. With 1 equiv.  $I_2$  suspension in  $CH_2Cl_2$  stirring overnight: (III) M = W, (IV) M = Mo.

(3) may be that there is some direct metal-metal interaction held together by bridging aryldiisocyanides.

The thermal decomposition of these aryldiisocyanide polymers occurs in two stages. For the  $W^0$  (5) 4,4'-diisocyanobiphenyl polymers there is a weight loss of ~6% in the range 50-150 °C, corresponding to water release (~2.8 molecules per tungsten). An initial decomposition at ~250 °C corresponds to ~14% (by weight). A second decomposition occurs at ~600 °C corresponding to ~63% (by weight of the same polymer). We have noted that the sum of these weight losses corresponds to the expected weight percentage of the organic ligands in these coordination polymers.

Since no single crystals were available, powder X-ray diffraction was used to investigate structural parameters of these polymers. Only one strong sharp reflection was obtained, corresponding to a distance of ~10.75 Å for the  $W^0$  (5) polymers. In the hopes of obtaining more ordered materials, we prepared polymers by slow precipitation in highly dilute solutions using longer reaction times. However, their powder X-ray diffraction patterns suggest that the polymers are rather disordered making structure determination by this technique impracticable.

An alternative approach to the preparation of mixed oxidation polymers via copolymerization of two soluble precursor complexes having the metal in two

different oxidation states, has also been investigated by use of  $\text{Mo}_2(\text{OAc})_4$  and  $[\text{Mo}(\text{CO})_4\text{Cl}_2]_2$  starting complexes and 4,4'-diisocyanobiphenyl ligands. Infrared and XPS data of the products indicate the formation of a rust-brown mixed oxidation state polymer (6) containing both  $\text{Mo}^0$  and  $\text{Mo}^{+2}$  metal atoms.

Electrical conductivities measured (ac, 1 kHz) on pressed powder pellets (10 ton/cm<sup>2</sup>) of the unoxidized  $\text{W}^0$  (5) polymers are of the order of  $10^{-9} \text{ ohm}^{-1} \text{ cm}^{-1}$ . None of the previously prepared mixed oxidation polymers containing W or Mo of the type described gave materials with appreciably improved electrical conductivities. We believe the very disordered nature of the polymers may very well be a major factor preventing the attainment of enhanced electrical conductivities for these materials and so our efforts are being concentrated on using the copolymerization technique of soluble metal complex precursors in two different oxidation states with more ordered polymeric systems in the hopes of obtaining new materials with optimum electronic properties.

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