

## Magnetic Properties in the System $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$ and $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$ ( $0 \leq x \leq 1$ )

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The compounds in the systems of  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) and  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) were prepared at an oxygen pressure of 1400 bars. The former had a two-layer hexagonal structure and that of the latter was cubic perovskite type. From the variation of the unit-cell parameters and of the magnetic properties, it is found that the  $\text{Co}^{4+}$  ions change from the low-spin to the high-spin state. In the system of  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$ , the change of magnetic property from ferromagnet to antiferromagnet is related to the spin state of  $\text{Co}^{4+}$  ions located at the octahedral sites.

### Introduction

Since Yakel (1) synthesized  $\text{SrFe}^{4+}\text{O}_3$ , interest in the perovskite oxides with the chemical formula  $A^{2+}B^{4+}\text{O}_3$  ( $A$ , alkaline earth ions;  $B$ , first-row transition metal ions) has been great. Their crystal structure stabilizes the tetravalent state of the first-row transition metal ions such as  $\text{Fe}^{4+}$ ,  $\text{Co}^{4+}$ , and  $\text{Ni}^{4+}$  more easily than any other oxides.

The ideal perovskite structure has the cubic unit cell, in which octahedra share corners and the  $B^{4+}\text{-O-B}^{4+}$  angle is  $180^\circ$ . In the case where the  $A$  cation is large, the stacking sequence of the closed-packed  $\text{AO}_3$  layers is changed from cubic to hexagonal. In the two-layer hexagonal structure  $2H$  (2), octahedra share faces with each other to form linear chains parallel to the hexagonal  $c$ -axis.

Both  $\text{BaMnO}_3$  and  $\text{BaCoO}_3$  have the  $2H$  hexagonal structure and their magnetic properties were investigated by Christensen *et al.* (2) and Krischer *et al.* (3). Both oxides are

antiferromagnetic. The  $\text{Co}^{4+}$  ion was in the low-spin state with  $(d\epsilon)^5(d\gamma)^0$  electron configuration.

The compounds,  $\text{SrMnO}_3$  and  $\text{SrCoO}_3$ , have a cubic perovskite structure. The magnetic properties of these oxides, reported by Takeda *et al.* (4) and Watanabe *et al.* (5), indicated that  $\text{SrMnO}_3$  was an antiferromagnet with a Néel temperature of  $260^\circ\text{K}$ ,  $\text{SrCoO}_3$  was a ferromagnet with a Curie temperature of  $200^\circ\text{K}$ , and the  $\text{Co}^{4+}$  ion was in the low-spin state. In both structures, the  $\text{Co}^{4+}$  ion was in the low-spin state, and  $J_{90}^S$  for  $\text{Co}^{4+}$   $(d\epsilon)^5(d\gamma)^0$  was antiferromagnetic, but  $J_{180}^S$  was ferromagnetic.

In the present study, an attempt was made to synthesize compounds with the chemical compositions of  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  and  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  to study the effect of replacement of  $\text{Co}^{4+}$  ions with  $\text{Mn}^{4+}$  ions on the magnetic property and the spin state of the  $\text{Co}^{4+}$  ion in both of  $2H$  and cubic perovskite structures.

These results will also provide some information on the  $\text{Co}^{4+}\text{-O-Mn}^{4+}$  interaction in both the  $90^\circ$  and  $180^\circ$  cases.

### Experimental

For the preparation of the compounds, mixtures of raw materials,  $\text{BaCO}_3$ ,  $\text{SrCO}_3$ ,  $\text{CoCO}_3\cdot 6\text{H}_2\text{O}$ , and  $\text{MnCO}_3$ , in the desired ratios were fired at  $1050^\circ\text{C}$  for 24 hr in air for the  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) system, and at  $1000\text{--}1350^\circ\text{C}$  in pure oxygen gas for 24 hr for the  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) system. The firing was repeated three times. The oxygen-deficient materials obtained in this way were annealed under oxygen pressures of 1400 bars at  $650^\circ\text{C}$  for the former and at  $400^\circ\text{C}$  for the latter for 48 hr (6).

Phases of the powder samples in the products were identified by X-ray diffraction with filtered  $\text{CuK}\alpha$  radiation. Lattice constants of the compounds were determined using  $\alpha\text{-SiO}_2$  as a standard material.

Magnetic properties were measured by an automatic recording magnetic balance in the temperature range from 77 to  $300^\circ\text{K}$ .

### Results and Discussion

X-ray powder diffraction patterns of all samples of  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) were completely indexed as the  $2H$  structure. The relation between composition and lattice constants is shown in Fig. 1. In this system, a single crystal of  $\text{BaCoO}_3$  ( $x=0$ ) was prepared under high oxygen pressures, and the results of X-ray analysis showed that the space group was  $P6_3/mmc$  with the cell dimensions  $a=5.649$  and  $c=4.760$  Å (7). As seen in Fig. 1, the  $a$ -axis increased linearly with increasing  $x$ . The  $c$ -axis expanded linearly with increasing  $x$  from  $x=0$  to  $0.5$  and from  $x=0.5$  to  $1.0$ . The atomic distance between transition metals along the  $c$ -axis was elongated from  $2.380$  Å for  $\text{BaCoO}_3$  to  $2.405$  Å for  $\text{BaMnO}_3$ . From these results, it appears that the distance of  $\text{Mn}^{4+}\text{-O}$  is longer than that of  $\text{Co}^{4+}\text{-O}$ .

In the system,  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$ , the X-ray patterns of the specimens were indexed as cubic perovskite structure. The relation of cell

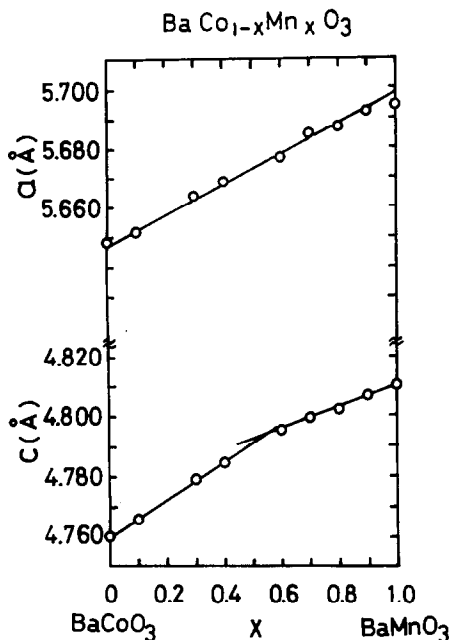


FIG. 1. Cell parameter vs composition in the system  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) with a  $2H$  structure.

parameter vs composition in the whole range is shown in Fig. 2. The lattice parameter of the  $a$ -axis increased linearly with increasing  $x$  from  $x=0$  to  $x=0.3$ , and for  $x=0.3$  to  $x=1.0$ , it decreased monotonously with increasing of  $x$ . The distances between the tetravalent transition-metal and oxygen atoms were calculated using the cell dimension in this solid solution series. The distance of  $1.904$  Å for  $\text{Mn}^{4+}\text{-O}$  was shorter than that of  $1.921$  Å for  $\text{Co}^{4+}\text{-O}$ . This result was inverse in comparison with that of the  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  series with the  $2H$  structure.

The changes of lattice parameter with a break at  $x=0.5$  for  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  and  $x=0.3$  for  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  indicates that the ionic radius of the tetravalent transition metal ions located at octahedral site changed due to the change of spin state from low to high with increasing  $x$ . The results of the structure analysis of the  $\text{BaNiO}_3$  with the  $2H$  structure (8) indicated that the  $c$ -axis is twice the  $\text{B}^{4+}\text{-B}^{4+}$  distance, but the  $a$ -axis corresponds to the length between the chains of octahedron held together by large  $\text{A}^{2+}$  ions. It is expected, therefore, that the change of

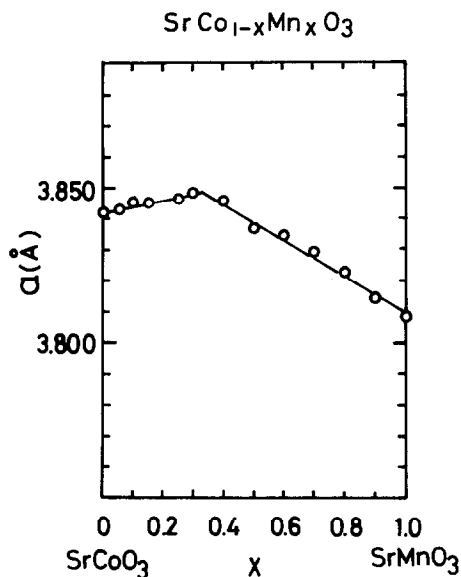


FIG. 2. Cell parameter vs composition in the system  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  ( $0 \leq x \leq 1$ ) with a cubic perovskite structure.

ionic radius of  $B^{4+}$  ions strongly influences the variation in the unit cell parameter for  $c$  but not for  $a$ .

The results of the magnetic susceptibility measurements of the  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  compounds showed that the inverse susceptibility vs temperature curve obeys the Curie-Weiss law above  $77^\circ\text{K}$ . The total spin angular momentum of  $S$  was calculated and is presented in Fig. 3. A full line indicates the observed values, and a dotted line is drawn for the theoretical values calculated in cases of high- and low-spin states of  $\text{Co}^{4+}$  ion located at octahedral site respectively; one is the high-spin state with the  $(d\epsilon)^3(d\gamma)^2$  electron configuration and the other is the low-spin state with the  $(d\epsilon)^5(d\gamma)^0$  electron configuration. The  $\text{Mn}^{4+}$  ion with a  $3d^3$  electron configuration has only one spin state with the  $(d\epsilon)^3(d\gamma)^0$  electron configuration. The observed and calculated values of  $S$  shown in Fig. 3 were nearly equal to each other under the assumption of  $S = \frac{1}{2}$  for  $\text{Co}^{4+}$  and  $S = \frac{3}{2}$  for  $\text{Mn}^{4+}$  in the compositional range of  $0.0 \leq x \leq 0.5$ , and in the range  $0.6 \leq x \leq 1.0$  the observed values of  $S$  lie on the dotted line, calculated assuming that  $S = \frac{5}{2}$  for  $\text{Co}^{4+}$  and  $S = \frac{3}{2}$  for

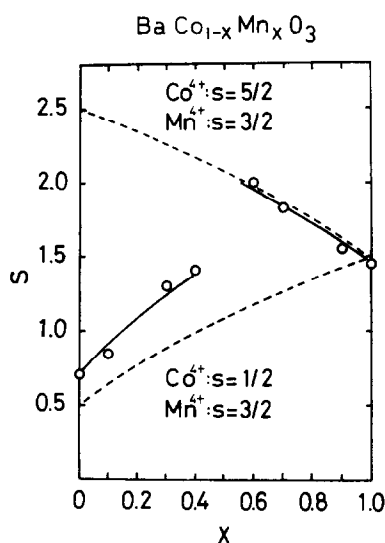


Fig. 3. Total spin angular momenta of  $S$  as a function of composition in the system  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$ .

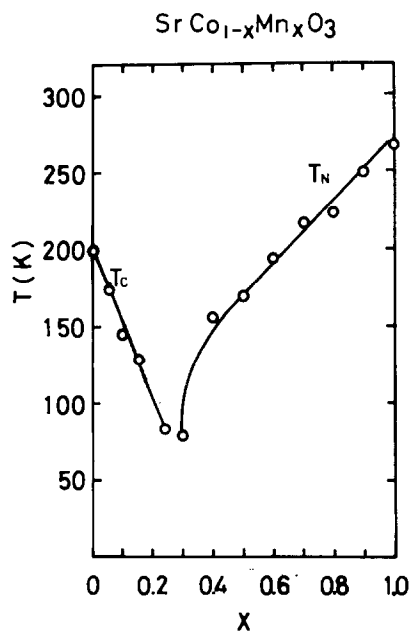


FIG. 4. Curie temperature and Néel temperature as a function of composition in the system  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$ .

$\text{Mn}^{4+}$  ion. From these results, it was concluded that the  $\text{Co}^{4+}$  ion changed its spin state from low to high at  $x = 0.5$ . This conclusion is supported by the result of the X-ray analysis.

In the system,  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$ , it is known that  $\text{SrCoO}_3$  is ferromagnetic and  $\text{SrMnO}_3$  is antiferromagnetic. In Fig. 4, the Curie temperature ( $T_C$ ) and the Néel temperature ( $T_N$ ) are shown as a function of  $x$ .  $\text{SrCoO}_{2.97}$  gave the spontaneous magnetization of 34.0 emu/g at 77°K. From this value, a magnetic moment of  $1.18 \mu_B/\text{mole}$  at 77°K was obtained, corresponding to the moment caused by one unpaired electron. This fact suggests that cobalt ion is in a low-spin state with the  $(d\epsilon)^5(d\gamma)^0$  electron configuration. As seen in Fig. 4, in the compositional range of  $0.0 \leq x < 0.3$ , the products were ferromagnetic and the Curie temperature decreased linearly with increasing  $x$ . Around  $x = 0.3$  the samples were antiferromagnetic and the Néel temperature increased with increasing  $x$ . The Curie temperature and the saturation magnetization at 77°K of  $\text{SrCo}_{0.9}\text{Mn}_{0.1}\text{O}_{2.98}$  were determined to be 140°K and 27.1 emu/g, respectively. The decrease of saturation magnetization with an increase of  $x$  was explained under the assumption of the antiferromagnetic superexchange coupling between  $\text{Co}^{4+}$  and  $\text{Mn}^{4+}$  ion. In the antiferromagnetic region of the solid solution, it is considered that  $J_{180}^s$  of the superexchange interactions of  $\text{Mn}^{4+}\text{--O--Mn}^{4+}$  is antiferromagnetic and dominates the magnetic properties of samples. In particular, it is expected that the change of the magnetic properties from ferromagnetic to antiferromagnetic at  $x = 0.3$  is strongly influenced by the change of lattice constant, as shown in Fig. 2; that is, the  $\text{Co}^{4+}$  ion is in the low-spin state at  $x < 0.3$  but in the high-

spin state at about  $x = 0.3$ . As for the superexchange interaction, it is expected that in the range  $0 \leq x < 0.3$ ,  $J_{180}^s$  for  $\text{Co}^{4+} (d\epsilon)^5(d\gamma)^0$  is strongly ferromagnetic and  $J_{180}^s$  for  $\text{Co}^{4+} (d\epsilon)^5(d\gamma)^0\text{--O--Mn}^{4+} (d\epsilon)^3(d\gamma)^0$  is antiferromagnetic, while, for  $0.3 \leq x \leq 1.0$ , all the  $J_{180}^s$  for  $\text{Mn}^{4+} (d\epsilon)^3(d\gamma)^0$ ,  $\text{Co}^{4+} (d\epsilon)^3(d\gamma)^2$ , and  $\text{Mn}^{4+}\text{--O--Co}^{4+}$  are antiferromagnetic.

It is concluded that tetravalent cobalt ions located at octahedral sites in both solid solution series of  $\text{BaCo}_{1-x}\text{Mn}_x\text{O}_3$  with a  $2H$  structure and  $\text{SrCo}_{1-x}\text{Mn}_x\text{O}_3$  with a cubic perovskite structure change their spin states from low to high at  $x = 0.5$  for the former and at  $x = 0.3$  for the latter. This change of spin state of the  $\text{Co}^{4+}$  ion strongly influences the lattice parameters and magnetic properties of the products in both solid solution series.

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