

Structural Investigation of K^+ and Tl^+ β -Aluminas

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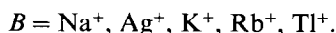
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Structures of K^+ and Tl^+ aluminas were investigated by single-crystal X-ray diffraction. In the final refinements, corresponding to R values of 0.034 and 0.058, respectively, the distribution of conducting ions was found to be similar to that of Na β -alumina. In K^+ β -alumina clear evidence of additional electronic density in the spinel-like blocks is obtained and is interpreted as due to a Frenkel defect.

Introduction

β -Alumina compounds are double oxides with the general formula $nA_2O_3-B_2O$ where



The structure has hexagonal symmetry (space group $P6_3/mmc$). The outstanding feature is the stacking of close-packed spinel blocks separated by mirror planes perpendicular to the c axis and containing B^+ ions (1, 2).

The great interest in these compounds arises from their very high ionic conductivity in the solid state, which allows them to be put among superionic conductors (3). This conductivity is due to the diffusion of B^+ ions in the intermediate conduction planes; major questions to be answered concern the origin of the temperature dependence of their electrical properties and their variation with regard to temperature. An X-ray diffuse scattering investigation of the distribution of conducting ions (4-6) revealed the existence of two-dimensional ordered microdomains, the size of which varies with temperature and cation type

For silver β -alumina, the average distribu-

tion of electron density in the mirror planes, obtained by X-ray diffraction study (7), allowed a semiquantitative analysis of diffuse intensities and gave a description of cation distribution within the microdomains (8).

To account for the variations in the scattered intensity distribution (related to local order changes) which are observed for different types of B^+ ions and preparation conditions (5, 6), it is necessary to have a good knowledge of the mean structures of those different compounds.

This paper reports an X-ray diffraction investigation of the nature and mean occupancy factors of the sites occupied by conducting ions in potassium and thallium β -aluminas.

Experimental

Sodium β -alumina crystals are obtained by the slow cooling of a melt heated by direct induction at high frequency through the oxide (Kyropoulos self-crucible method). Their composition, deduced from neutron activation analysis, is about $11 Al_2O_3-1.3 \pm 0.03 Na_2O$. Replacement of Na^+ by K^+ or Tl^+ is obtained by an exchange reaction in molten salts (9).

β -alumina single crystals grow in the shape of easily cleavable thin platelets. The choice of

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crystals for structural determination was made according to the quality of diffraction spots observed. In both studies, intensities were measured on a $P2_1$ Syntex four-circle diffractometer, between $2\theta = 0$ to 65° , using $\text{MoK}\alpha$ radiation with a graphite monochromator. A standard reflection was measured every 30 reflections. For each crystal, data was collected in a section of reciprocal space where h, k , and l were all >0 , and equivalent reflections were averaged.

Absorption corrections were needed for crystals in platelet form (especially for Ti^+). They were performed with the ψ scan technique using 15 reflections.

Refinement was done according to the ORXFLS III routine (10) with all the measured intensities (even zero values) by using Na β -alumina as starting positional parameters (11).

Refinement of K β -Alumina Structure

The single crystal of K β -alumina had a triangular shape ($350 \mu\text{m}$ for the side and $150 \mu\text{m}$ for thickness). The lattice constants of the hexagonal cell have the following values: $a = 5.602 \text{ \AA}$, $c = 22.734 \text{ \AA}$. Four hundred and ninety-four independent reflections were collected on the four-circle diffractometer. The first cycles of refinement using all the variable parameters—scale factor, positional parameters, occupancy, and individual temperature factors in the intermediate planes—revealed very high terms in the correlation matrix. Therefore we employed the least number of variable parameters in the refinements.

In the first step, the scale factor, general temperature factor, and occupancy coefficient of potassium atoms located in the Beever-Ross (BR) and mid-oxygen (mO) positions were refined (Fig. 1).

The successive cycles indicated that:

1. some K^+ ions were located on the anti-Beever-Ross (aBR) site,
2. some interstitial aluminum was found on the position previously mentioned by Reidinger *et al.* for Na β -alumina (12) (Fig. 2),
3. all atoms were, more or less, shifted out of the theoretical positions BR, mO, and

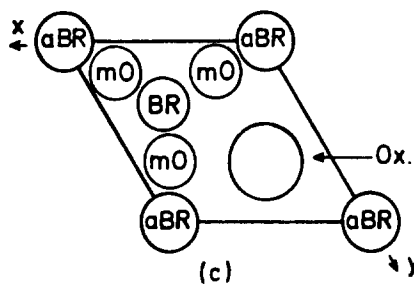


FIG. 1. Simplified site model with the conventional site denomination. BR = Beever-Ross, aBR = anti-Beever-Ross, mO = mid-oxygen.

aBR, and the same thing occurred for O (5) in the plane.

We then froze the scale factor and refined the occupancy coefficients of aluminum atoms in the spinel blocks together with the occupancy factors of potassium atoms, interstitial Al, and the general temperature factor. This showed that the Al_1 position was partly vacant and on the other positions the variations were within the error limits.

In the next step, refinements were attempted alternatively either on individual isotropic temperature factors or on occupancy factors,

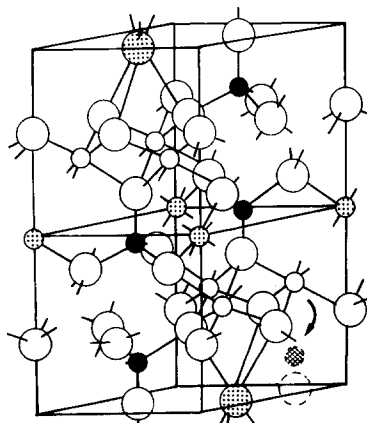
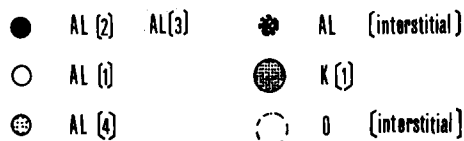


FIG. 2. Frenkel defect on K^+ β -alumina.

TABLE I
POTASSIUM β -ALUMINA: POSITION AND THERMAL PARAMETERS

	Position	Number per unit cells	$x \times 10^4$	$z \times 10^4$	$\beta_{11} \times 10^5$	$\beta_{22} \times 10^5$	$\beta_{33} \times 10^5$	$\beta_{13} \times 10^5$	B
Al (1)	12(<i>k</i>)	11.50 (3)	8322 (1)	1052 (1)	819 (23)	743 (31)	38 (1)	3 (3)	
Al (2)	4(<i>f</i>)	4	3333	245 (1)	676 (32)	β_{11}	34 (1)	0	
Al (3)	4(<i>f</i>)	4	3333	1749 (1)	1034 (36)	β_{11}	31 (2)	0	
Al (4)	2(<i>a</i>)	2	0	0	773 (46)	β_{11}	30 (3)	0	
Al interstitial	12(<i>k</i>)	0.36 (2)	8380 (28)	1750 (10)					0.2 (2)
O (1)	12(<i>k</i>)	12	1571 (2)	496 (1)	880 (52)	718 (74)	38 (3)	-6 (7)	
O (2)	12(<i>k</i>)	12	5027 (2)	1454 (1)	721 (48)	792 (70)	46 (3)	-8 (7)	
O (3)	4(<i>f</i>)	4	6666	552 (2)	758 (79)	β_{11}	37 (5)	0	
O (4)	4(<i>e</i>)	4	0	1411 (2)	670 (74)	β_{11}	41 (5)	0	
O (5)	6(<i>h</i>)	2	3151 (10)	2500	1400 (220)	1060 (454)	32 (8)	0	
K (3)	6(<i>h</i>)	0.28 (1)	9707 (22)	2500					0.8 (2)
K (7)	6(<i>h</i>)	0.75 (1)	8824 (9)	2500	1995 (186)	3218 (365)	29 (8)	0	
K (1)	6(<i>h</i>)	1.56 (1)	6887 (3)	2500	1347 (72)	1352 (68)	48 (3)	0	

TABLE II
THALLIUM β -ALUMINA: POSITION AND THERMAL PARAMETERS

	Position	Number per unit cell	$x \times 10^4$	$z \times 10^4$	$\beta_{11} \times 10^5$	$\beta_{22} \times 10^5$	$\beta_{33} \times 10^6$	$\beta_{13} \times 10^6$
Al (1)	12(<i>k</i>)	12	8324 (3)	1042 (2)	795 (72)	715 (94)	440 (40)	10 (80)
Al (2)	4(<i>f</i>)	4	3333	246 (2)	610 (98)	β_{11}	350 (60)	0
Al (3)	4(<i>f</i>)	4	3333	1748 (2)	845 (103)	β_{11}	250 (60)	0
Al (4)	2(<i>a</i>)	2	0	0	691 (132)	β_{11}	270 (80)	0
O (1)	12(<i>k</i>)	12	1577 (6)	488 (2)	594 (138)	582 (201)	310 (70)	260 (180)
O (2)	12(<i>k</i>)	12	5021 (6)	1437 (2)	649 (137)	437 (184)	440 (80)	50 (180)
O (3)	4(<i>f</i>)	4	6666	548 (4)	628 (216)	β_{11}	250 (140)	0
O (4)	4(<i>e</i>)	4	0	1397 (4)	483 (201)	β_{11}	300 (140)	0
O (5)	6(<i>h</i>)	2	3333	2500	2613 (536)	β_{11}	90 (210)	
Tl (1)	6(<i>h</i>)	1.75 (2)	6807 (32)	2500	2246 (308)	2649 (765)	630 (20)	0
Tl (2)	6(<i>h</i>)	0.51 (2)	9099 (15)	2500	2630 (283)	5367 (746)	300 (90)	0
Tl (3)	2(<i>b</i>)	0.21 (2)	0	2500	3397 (585)	β_{11}	250 (140)	0

the general scale factor being fixed; then complete convergence was reached ($R = 0.040$). The very last attempt was to refine anisotropic temperature factors; a final R value of 0.034 was obtained, corresponding to the values gathered in Table I.

The last refinement cycle dealing with all the variable parameters brought no meaningful difference in the values of structural parameters and confirmed the reliability of the results.

Description of the Structure

(a) *Spinel blocks.* As pointed out by Reindinger, a partial occupation of the interstitial tetrahedral Al site is noticed. For this atom only a very weak thermal coefficient is obtained; thus if the partial occupation of this position is firmly established, the absolute value of the occupancy factor is not known with precision. But we observed that this occupancy factor is nearly identical to the lack of

TABLE III
COMPONENTS OF PRINCIPAL AMPLITUDES OF THERMAL
VIBRATION OF ATOMS IN POTASSIUM β -ALUMINA, AND
IN THALLIUM β -ALUMINA^a

Atoms in the mirrorplane	$(U_{11}^2)^{1/2}$		$(U_{22}^2)^{1/2}$ (Å)
	(Å)	(Å)	
K(1)	0.127	0.127	0.112
K(2)	0.154	0.196	0.087
Tl(1)	0.163	0.177	0.129
Tl(2)	0.177	0.253	0.089
Tl(3)	0.201	0.201	0.082

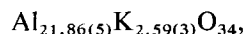
^a U_{11} , U_{22} , U_{33} are the root mean square displacements of the vibration parallel to a^* , b^* , and c^* .

aluminum in the Al(1) site, and this is consistent with the hypothesis of a Frenkel-type defect within the spinel blocks (Fig. 2).

(b) *Mirror planes.* All three possible cationic sites (BR, mO, aBR) are simultaneously occupied (but partly), and the refinement did not reveal a lack of oxygen in the O(5) position.

No pronounced anisotropy is detected for the atoms in the conducting plane (Table III).

The final formula may be written as follows:



i.e., 68.17 positive charges for 68 negative ones.

This formula is not accurate enough to confirm the charge compensation model, by in-

TABLE IV
INTERATOMIC DISTANCES

	Number of bonds	K β -Al ₂ O ₃ (Å)	Tl β -Al ₂ O ₃ (Å)
Octahedra			
Al(1)-O(1)	2	2.023	2.025
-O(2)	2	1.841	1.840
-O(3)	1	1.968	1.965
-O(4)	1	1.823	1.817
Al(4)-O(1)	6	1.896	1.895
Tetrahedra			
Al(2)-O(1)	3	1.802	1.791
-O(3)	1	1.811	1.819
Al(3)-O(2)	3	1.775	1.784
-O(5)	1	1.716	1.723
Al(interstitial)-O(1)	2	1.762	
-O(4)	1	1.751	
-O ^a	1	1.758	
Polyhedron 9-coordinated			
K(1) or Tl(1)-O(2)	2	2.810	2.876
-O(2)	2	2.810	2.876
-O(2)	2	2.986	2.988
-O(5)	3(average of three)	3.192	3.232
Polyhedron 8-coordinated			
K(2) or Tl(2)-O(5)	2(average of three)	2.841	2.895
-O(4)	2	2.727	2.673
-O(2)	4	3.040	3.205
Polyhedron 9-coordinated			
K(3) or Tl(3)-O(4)	2	2.493	2.527
-O(5)	3	3.243	3.231
-O(2)	4	3.492	

^a Interstitial oxygen located in the mid-oxygen position.

terstitial oxygen on mid-oxygen positions, that would be suggested by a Frenkel defect in the blocks.

Three outstanding differences are to be noticed in regard to Dernier and Remeika's (D.R.) paper (13):

1. the presence, in our crystal, of interstitial aluminium,
2. the occupation of the aBR position, as in previously described alumina structures (7, 8, 12, 14),
3. and a noticeable difference in the occupancy rate of K atoms.

	D.R.	This work
BR	1.42 (2)	1.56 (1)
mO	1.22 (2)	0.75 (1)
aBR	0.0	0.28 (1)
	<hr/>	<hr/>
	2.64 (4)	2.59 (3)

The total amount of potassium in both cases is very similar; there is in fact a different distribution between the positions.

The divergence between Remeika's and our results is presumably due to different preparation conditions: our crystals were obtained at higher temperatures than those of Dernier and Remeika, which were prepared by flux method; in our method, the occupation of aBR sites is favored, in which the environment of the potassium ion is less favorable (K (3)-O (4) distances are short), (Table IV).

Refinement of the Tl β -Alumina Structure

The lattice constants are: $a = 5.596 \text{ \AA}$, $c = 22.912 \text{ \AA}$. Four hundred ninety-nine independent reflections were collected. (Tables V and VI). The refinement was performed in the same way as in K^+ β -alumina.

No interstitial aluminum was detected in the spinel blocks, probably because the corresponding electron density was too low in relation to that of Tl atoms.

The final value of the R parameter was 0.058 corresponding to the parameters of Table II.

Discussion

Here again we find simultaneous partial occupation of the three possible cationic sites

in the conducting planes; these positions are all slightly shifted off the theoretical ones.

The main differences with a recent paper of Kodama and Muto (K.M.) (14) are:

1. First, no negative thermal coefficient was noticed when the necessary absorption corrections were correctly made,
2. Also, the strong correlation between the theoretical BR position ($\frac{2}{3} \frac{1}{3} \frac{1}{4}$) and the actual BR position ($\frac{2}{3} + \epsilon$, $\frac{1}{3} - \epsilon$, $\frac{1}{4}$) does not allow their simultaneous occupations as in the K.M. paper,
3. Finally occupancy factors for cationic sites are different from those of K.M.:

	K.M. ¹	This work
BR	1.86	1.76 (2)
mO	0.73	0.51 (2)
aBR	0.07	0.21 (2)
	<hr/>	<hr/>
	2.66	2.48 (6)

The divergence between these results may again be related to:

1. the difference in preparation methods as for K β -alumina,
2. the lack of absorption corrections in K.M.'s work; this led to imprecise values of occupancy and thermal coefficients (especially anisotropic factors).

Conclusion

For crystals obtained in the same way, the occupancy factor of the BR position increases when Tl^+ takes the place of K^+ , and simultaneously the mO site occupancy factor decreases. The aBR site is always weakly occupied. Also the total amount of M^+ in the conducting planes decreases when Tl^+ substitutes for K^+ .

On the other hand, if our results are compared with the parameters of Na β -alumina the shift from the theoretical positions is related to the nature of the conducting ion.

	Theoretical position	Na ⁽¹¹⁾	K	Tl
BR	0.666	0.706	0.689	0.681
mO	0.833	0.873	0.882	0.910
aBR	1.0	—	0.971	1.00

¹ K.M. do not reveal the error for occupancy factors.

TABLE V
OBSERVED AND CALCULATED STRUCTURE FACTORS ($K^+ \beta$ -ALUMINA)

H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)
0	0	2	155.3284	145.4621	0	0	2	78.1069	81.2364	0	3	21	2.8930	1.6350	1	1	26	69.6190	66.3833
0	0	2	162.0249	150.0659	0	2	1	127.8728	129.3522	0	3	22	2.9152	1.6528	1	1	26	69.6947	66.4590
0	0	0	169.3707	158.8050	0	2	3	54.1332	55.0132	0	3	24	34.0000	34.9400	1	1	30	20.5383	20.5383
0	0	0	171.3181	160.8587	0	2	4	37.7980	38.4965	0	5	16	14.9638	15.0352	1	1	30	35.9515	35.9515
0	0	14	215.5000	216.6901	0	2	4	80.9851	80.9851	0	5	19	17.3124	16.4010	1	2	0	33.6736	33.6736
0	0	0	71.3552	72.3572	0	2	5	164.6682	162.6714	0	5	20	27.7821	28.1531	1	2	4	40.2136	41.4940
0	0	18	171.3552	172.3572	0	2	7	150.1135	137.6408	0	5	22	16.6487	15.4866	1	2	4	40.2136	41.4940
0	0	20	202.0161	203.2721	0	2	9	44.9491	44.8116	0	5	23	1.6007	2.0358	1	2	4	39.9360	39.9360
0	0	22	13.7945	10.6544	0	2	9	56.7848	84.0094	0	5	24	4.4821	1.2787	1	2	4	38.5848	37.6181
0	0	24	101.0067	102.4549	0	2	11	170.9991	171.3987	0	3	1	67.2873	64.5791	1	2	7	12.3585	13.5999
0	0	26	140.6547	140.9662	0	2	11	125.9843	125.1828	0	4	1	51.1290	51.3277	1	2	7	12.3585	13.5999
0	0	28	34.4656	33.7599	0	2	12	26.7231	25.1268	0	4	1	67.9942	65.9765	1	2	7	15.4171	15.4171
0	0	30	65.4627	64.6823	0	2	13	28.4854	28.8095	0	4	2	39.3322	40.9356	1	2	9	35.4844	35.7013
0	0	32	101.0067	102.4549	0	2	13	250.9411	250.1827	0	4	3	6.0659	6.0659	1	2	10	75.6543	63.9500
0	0	34	18.9504	17.9067	0	2	16	31.5660	31.9293	0	4	5	54.0222	51.8250	1	2	12	6.3485	11.6328
0	1	1	2.6124	4.902	0	2	16	53.6350	53.9884	0	4	6	104.0267	105.9939	0	6	6	30.6027	32.9555
0	1	2	66.6627	63.6882	0	2	18	88.2756	93.3769	0	4	7	71.2719	72.1181	1	2	14	45.0971	48.2279
0	1	4	24.9703	23.9375	0	2	20	21.7592	20.9502	0	4	9	40.9425	40.4611	1	2	16	10.6876	11.4924
0	1	5	13.1039	12.5669	0	2	21	30.5637	31.6316	0	4	9	60.9425	60.4611	1	2	16	10.6876	11.4924
0	1	6	32.8718	31.0807	0	2	22	58.8469	61.4894	0	4	10	36.2134	36.1351	1	2	17	22.4050	22.4392
0	1	8	97.0817	98.3572	0	2	24	18.6820	19.4845	0	4	11	131.5142	133.5118	0	6	10	10.0000	10.0000
0	1	9	60.6787	59.5299	0	2	24	92.0351	91.6844	0	4	12	159.2534	159.4009	0	6	11	3.5166	4.6216
0	1	10	66.7152	67.4596	0	2	26	65.4970	63.2847	0	4	13	12.1851	12.7214	1	2	18	44.2508	46.0589
0	1	11	16.6631	15.5348	0	2	26	152.6523	157.9907	0	4	13	18.0495	18.9715	1	2	18	39.7916	37.7195
0	1	12	50.4229	52.2091	0	2	28	2.35658	2.2461	0	4	14	168.7603	174.5846	0	6	12	75.2232	76.2072
0	1	13	95.0879	96.1298	0	2	30	2.3648	3.2663	0	4	18	85.0984	85.9358	0	6	14	31.9405	34.1782
0	1	14	39.6377	39.0337	0	2	31	59.0809	58.2030	0	4	19	97.3997	97.0511	1	2	20	6.4880	6.4589
0	1	15	52.3511	53.4948	0	2	32	122.4819	122.7993	0	4	20	17.9750	16.9738	1	2	20	6.4880	6.4589
0	1	17	32.3511	35.4948	0	2	33	5.3194	5.1534	0	4	22	44.0980	46.1726	0	6	16	42.1816	41.2885
0	1	18	50.8456	52.6712	0	3	1	6.9158	7.4844	0	4	23	3.9364	3.6102	1	2	21	16.4871	16.3215
0	1	19	47.4756	47.9667	0	3	1	30.4788	29.0889	0	4	24	7.2783	6.4425	1	2	21	16.4871	16.3215
0	1	21	40.5323	40.9507	0	3	3	30.4788	29.0889	0	4	26	50.0883	47.7312	1	2	23	12.5935	12.5935
0	1	23	17.3166	17.3177	0	3	5	13.4669	12.6601	0	4	28	18.1991	16.6772	1	2	23	12.5935	12.5935
0	1	24	15.3311	14.9851	0	3	7	108.1798	109.1901	0	4	28	14.2667	13.2088	1	2	23	12.5935	12.5935
0	1	26	38.7701	38.9588	0	3	9	6.5988	6.6360	0	4	30	94.3357	91.4303	1	2	25	39.3836	39.2496
0	1	27	9.1022	9.8213	0	3	9	10.9826	10.7071	0	4	30	16.4667	16.8009	1	2	25	39.3836	39.2496
0	1	28	17.6673	18.5970	0	3	10	11.0049	12.1536	0	4	32	23.4642	22.0177	1	2	27	26.8136	26.9075
0	1	30	24.8880	24.7923	0	3	13	13.9115	13.9331	0	4	34	95.8285	96.2667	1	2	29	26.8136	26.9075
0	1	32	14.0800	14.1841	0	3	14	14.7188	15.6329	0	4	34	39.0820	40.5993	1	2	29	26.8136	26.9075
0	1	34	14.2647	14.7046	0	3	14	51.3187	53.0389	0	4	36	34.2011	34.2884	1	2	31	31.4106	31.4106
0	1	34	14.2647	14.7046	0	3	16	81.0771	81.3800	0	4	36	57.1857	57.2074	1	2	31	31.4106	31.4106
0	1	34	14.2647	14.7046	0	3	18	49.0672	43.7485	0	4	38	3.2895	3.2895	1	2	31	31.4106	31.4106
0	1	34	14.2647	14.7046	0	3	20	67.6782	67.4693	0	4	40	31.4756	30.8494	1	2	31	31.4106	31.4106

H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)	H	K	L	Y(OBS)	Y(CALC)
1	1	3	41.056	42.713	1	1	3	13.077	14.072	2	3	15	25.102	20.026	2	3	18	19.349	18.261
1	1	3	27.021	27.021	1	1	3	8.982	9.042	2	3	17	24.564	24.564	2	3	16	19.610	19.610
1	1	3	38.981	38.981	1	1	3	13.077	13.077	2	3	15	25.102	25.102	2	3	17	24.564	24.564
1	1	3	38.981	38.981	1	1	3	13.077	13.077	2	3	15	25.102	25.102	2	3	17	24.564	24.564
1	1	3	20.300	20.300	1	1	3	20.300	20.300	2	3	19	31.032	30.032	2	3	19	31.032	30.032
1	1	3	18.935	18.935	1	1	3	17.259	17.259	2	3	22	31.032	31.032	2	3	22	31.032	31.032
1	1	3	32.790	32.790	1	1	3	19.085	19.085	2	3	23	19.085	19.085	2	3	23	19.085	19.085
1	1	3	12.037	12.037	1	1	3	19.085	19.085	2	3	23	19.085	19.085	2	3	23	19.085	19.085
1	1	3	10.762	10.762	1	1	3	10.762	10.762	2	3	26	16.487	16.487	2	3	26	16.487	16.487
1	1	3	23.625	23.625	1	1	3	6.357	6.357	2	3	26	21.039	21.039	2	3	26	21.039	21.039
1	1	3	82.879	82.879	1	1	3	7.383	7.383	2	3	27	42.317	42.317	2	3	27	42.317	42.317
1	1	3	11.091	11.091	1	1	3	53.280	53.280	2	3	27	42.317	42.317	2	3	27	42.317	42.317
1	1	3	13.140	13.140	1	1	3	11.091	11.091	2	3	28	54.342	54.342	2	3	28	54.342	54.342
1	1	3	90.857	92.588	1	1	3	15.176	13.765	2	3	3	5.093	5.093	2	3	3	5.093	5.093
1	1	3	12.115	12.115	1	1	3	12.115	12.115	2	3	6	67.702	69.370	2	3	6	67.702	69.370
1	1	3	6.1634	6.1634	1	1	3	8.025	8.705	2	3	7	54.963	54.973	2	3	7	54.963	54.973
1	1	3	1.359	1.359	1	1	3	23.893	23.927	2	3	9	23.022	21.842	2	3	9	23.022	21.842
1	1	3	21.020	21.020	1	1	3	16.170	16.170	2	3	10	47.014	46.586	2	3	10	47.014	46.586
1	1	3	50.107	51.705	2	2	2	59.219	60.075	2	2	11	93.014	92.778	2	2	11	93.014	92.778
1	1	3	7.656	7.656	2	2	2	10.494	10.494	2	2	12	12.140	12.683	2	2	12	12.140	12.683
1	1	3	4.331	4.331	2	2	2	8.074	8.074	2	2	14	12.358	12.307	2	2	14	12.358	12.307
1	1	3	11.1616	10.426	2	2	6	82.269	87.007	2	2	15	4.928	5.440	2	2	15	4.928	5.440
1	1	3	35.1953	36.839	2	2	10	23.768	23.768	2	2	17	7.025	8.350	2	2	17	7.025	8.350
1	1	3	11.344	11.344	2	2	14	13.528	13.528	2	2	18	54.525	55.249	2	2	18	54.525	55.249
1	1	3	1.159	1.159	2	2	16	18.496	17.124	2	2	19	30.989	30.956	2	2	19	30.989	30.956
1	1	3	49.783	49.783	2	2	16	66.990	65.800	2	2	20	17.190	17.156	2	2	20	17.190	17.156
1	1	3	5.4234	5.4234	2	2	22	18.675	18.675	2	2	22	32.879	32.879	2	2	22	32.879	32.879
1	1	3	4.984	4.984	2	2	24	90.077	89.235	2	2	23	4.267	4.267	2	2	23	4.267	4.267
1	1	3	5.4258	5.4258	2	2	26	69.268	67.918	2	2	25	47.031	47.031	2	2	25	47.031	47.031
1	1	3	6.272	6.272	2	2	30	19.135	19.135	2	2	25	12.376	13.009	2	2	25	12.376	13.009
1	1	3	2.582	2.582	2	2	30	9.702	9.702	2	2	26	14.248	14.368	2	2	26	14.248	14.368
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	3	17.445	17.089	2	2	3	17.445	17.089
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	5	17.445	17.089	2	2	5	17.445	17.089
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	6	58.285	58.287	2	2	6	58.285	58.287
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	7	6.070	7.076	2	2	7	6.070	7.076
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	9	4.246	4.246	2	2	9	4.246	4.246
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	10	14.005	12.610	2	2	10	14.005	12.610
1	1	3	1.5607	1.5607	2	2	3	1.028	1.028	2	2	11	3.287	1.712	2	2	11	3.287	1.712
1	1	3	20.123	20.123	2	2	6	28.954	28.146	2	2	11	24.100	24.100	2	2	11	24.100	24.100
1	1	3	3.217	3.217	2	2	10	21.862	21.732	2	2	13	4.076	4.076	2	2	13	4.076	4.076
1	1	3	3.217	3.217	2	2	10	8.918	7.582	2	2	13	14.438	14.438	2	2	13	14.438	14.438
1	1	3	1.2509	1.2509	2	2	12	9.108	9.108	2	2	15	17.828	15.811	2	2	15	17.828	15.811
1	1	3	2.488	2.488	2	2	13	21.042	21.042	2	2	16	50.025	50.389	2	2	16	50.025	50.389
1	1	3	2.488	2.488	2	2	13	21.042	21.042	2	2	17	16.389	16.389	2	2	17	16.389	16.389

TABLE VI
OBSERVED AND CALCULATED STRUCTURE FACTORS (Ti+β-ALUMINA)

Table with 16 columns: I, K, L, Y(OBS), Y(CALC), H, K, L, Y(OBS), Y(CALC), H, K, L, Y(OBS), Y(CALC), H, K, L, Y(OBS), Y(CALC). The table contains numerical data for various reflections (I, K, L) and their observed and calculated structure factors.

Both Tl and K β -alumina compounds are similar in nature to Na β -alumina, rather than to Ag β -alumina, in which aBR positions have significant occupation factors.

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