

The Crystal Structure of $\text{La}_3\text{Ru}_3\text{O}_{11}$: A New Cubic KSbO_3 Derivative Oxide with No Metal-Metal Bonding

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$\text{La}_3\text{Ru}_3\text{O}_{11}$ was prepared by the reaction of La_2O_3 , RuO_2 , and NaClO_3 in a KCl flux under vacuum at 950°C . The crystal structure of this new cubic KSbO_3 derivative oxide was determined from single-crystal X-ray diffraction data collected on an automated diffractometer with $\text{MoK}\alpha$ radiation. Principal crystallographic data: Cubic, space group $Pn\bar{3}$; $a = 9.451(2) \text{ \AA}$; $V = 844.2 \text{ \AA}^3$; $d_x = 7.049 \text{ g cm}^{-3}$. Final discrepancy indices $R = 0.036$, $R_w = 0.042$. $\text{La}_3\text{Ru}_3\text{O}_{11}$ is isomorphous with $\text{Bi}_3\text{Ru}_3\text{O}_{11}$, but is notably different in showing no direct bonding between ruthenium atoms; the closest Ru-Ru contact in this new oxide is $2.994(1) \text{ \AA}$.

Introduction

Recently two ternary ruthenium oxides were reported which exhibit the first known evidence in oxide systems of metal-to-metal bonding between Ru atoms. $\text{La}_4\text{Ru}_6\text{O}_{19}$ (1) and $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ (2), despite their dissimilar stoichiometries, adopt closely related structures derived from the cubic KSbO_3 structure type. In both compounds the short Ru-Ru distances (2.448 and 2.60 \AA , respectively), as well as the displacement of the paired Ru atoms from the centers of their coordination octahedra toward each other, clearly indicate the presence of a bonding interaction.

In the course of our study of compounds in the lanthanide-ruthenium-oxide system, we have synthesized $\text{La}_3\text{Ru}_3\text{O}_{11}$, a third ternary Ru oxide structurally related to KSbO_3 . Our investigation of the crystal structure of this new compound revealed one surprising difference between it and that of the isomorphous Bi compound: $\text{La}_3\text{Ru}_3\text{O}_{11}$ exhibits a

repulsive rather than a bonding interaction between the ruthenium atoms. In this paper we present the details of the $\text{La}_3\text{Ru}_3\text{O}_{11}$ crystal structure along with a structural comparison of ternary transition metal KSbO_3 derivative oxides.

Experimental

All starting materials were reagent grade chemicals and, with the exception of La_2O_3 , were used without further purification. Lanthanum oxide was roasted in air at 800°C for a few hours, cooled in an inert atmosphere, and weighed quickly to avoid contamination with carbonate or hydroxide.

$\text{La}_3\text{Ru}_3\text{O}_{11}$ was obtained as one product of the reaction of a 1:1 molar mixture of La_2O_3 and RuO_2 with enough NaClO_3 to oxidize Ru from the 4^+ to the 4.5^+ oxidation state, using a large molar excess of KCl as a flux. (Hexagonal $\text{La}_8\text{Ru}_4\text{O}_{21}$, described in a previous paper (3), was another crystalline product obtained). After being sealed under vacuum in a Vycor tube, the reaction mixture was heated

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at 950°C for 1 day, cooled to 700°C over a 12-hr period, then quenched to room temperature. The solidified KCl flux was removed from the products by leaching with water.

La₃Ru₃O₁₁ was obtained as black cuboctahedral crystals up to 2 mm across. A crystal of dimensions 0.15 × 0.15 × 0.15 mm was mounted for X-ray study.

X-ray Data Collection

All data were collected at room temperature on a Syntex P1 automated diffractometer using graphite-monochromated MoK α radiation. The automatic centering and indexing procedures used have been described elsewhere (4). Preliminary photographs revealed *m*3 (cubic) Laue symmetry. The systematic absence (0*kl*, *k* + *l* ≠ 2*n*, etc.) established *Pn*3 (No. 201) as the space group. The principal crystallographic data are as follows: *a* = 9.451 (2) Å; *V* = 844.18 Å³; *d_x* = 7.049 g/cm⁻³ for *Z* = 4 and a formula weight of 895.93.

A total of 630 unique reflections with 5° ≤ 2θ ≤ 70° were collected using the θ-2θ scan technique, variable scan rates from 4.0 to 24.0°/min, and a scan range from 2θ(MoK α ₁) - 0.8° to 2θ(MoK α ₂) + 0.8°. The intensities of three standard reflections measured after every 100 reflections showed no significant variation during data collection. Lorentz and polarization corrections¹ were applied. The crystal was measured with a micrometer eyepiece, and a numerical absorption correction (linear absorption coefficient μ = 202.8 cm⁻¹) was applied to the data. Transmission coefficients ranged from 8.60 to 12.89%.

Solution and Refinement of the Structure

A three-dimensional Patterson function was used to determine the positions of the metal atoms (Ru, La(1) and La(2)). These atoms were used to calculate approximate phases for

¹ Computer programs used on a PDP 11/45 computer at the Molecular Structure Corporation, College Station, Texas, were those of the Enraf-Nonius structure determination package.

TABLE I
POSITIONAL AND THERMAL PARAMETERS^a WITH STANDARD DEVIATIONS IN PARENTHESES

| Atom | x | y | z | β_{11} | β_{22} | β_{33} | β_{12} | β_{13} | β_{23} |
|-------|------------|-------------------|-----------|--------------|--------------|--------------|--------------|--------------|--------------|
| Ru | 0.09160(7) | 0.75 ^b | 0.25 | 0.00065(5) | 0.00075(5) | 0.00073(5) | 0.0 | 0.0 | 0.0000(1) |
| La(1) | 0.5 | 0.5 | 0.5 | 0.00111(9) | β_{11} | β_{11} | -0.0002(2) | β_{12} | β_{12} |
| La(2) | 0.1129(1) | x | x | 0.00112(6) | β_{11} | β_{11} | -0.0004(2) | β_{12} | β_{12} |
| O(1) | 0.6203(7) | 0.25 | 0.25 | 0.0015(6) | 0.0022(6) | 0.0016(6) | 0.0 | 0.0 | 0.0003(11) |
| O(2) | 0.4169(5) | 0.7462(6) | 0.0397(5) | 0.0011(4) | 0.0026(4) | 0.0011(4) | -0.0003(8) | -0.0011(7) | -0.0007(8) |
| O(3) | 0.3559(5) | x | x | 0.0013(9) | β_{11} | β_{11} | -0.0009(21) | β_{12} | β_{12} |

^a The form of the anisotropic thermal parameter is: $\exp[-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + \beta_{12}hk + \beta_{13}hl + \beta_{23}kl)]$.

^b Entries with no *esd*'s are fixed by the symmetry of the space group.

a Fourier map, from which the three oxygen atoms were located. The origin of the unit cell was chosen at the $\bar{3}$ special position, which is $\frac{1}{4}\frac{1}{4}\frac{1}{4}$ from that used by Abraham *et al.* (2) for $\text{Bi}_3\text{Ru}_3\text{O}_{11}$. Three cycles of least-squares refinement of a scale factor, all variable positional parameters, and isotropic temperature factors gave discrepancy indices:

$$R_1 = \sum | |F_o| - |F_c| | / |F_o| = 0.052,$$

$$R_2 = [\sum w (|F_o| - |F_c|)^2 / \sum w |F_o|^2]^{1/2} = 0.061$$

The function minimized during all least-squares cycles was $\sum w (|F_o| - |F_c|)^2$, where the weighting factor $w = 4F_o^2 / \sigma(F_o^2)^2$, with σ determined from counting statistics. Only those 568 reflections with $F_o^2 > 3\sigma(F_o^2)$ were included in the refinements. Scattering factors used were those of Cromer and Waber (5) for neutral atoms, and anomalous dispersion corrections (6) were included for all atoms.

Five subsequent refinement cycles in which the scale factor, all variable atomic positions, anisotropic temperature factors, and an isotropic extinction correction were varied reduced R_1 to 0.036 and R_2 to 0.042. Final

shifts in the parameters were less than 10% of the estimated standard deviations of the individual parameters. The esd of an observation of unit weight was 2.60. The extinction correction was of the form $F_{\text{corr}}^2 = F_o^2 / (1 + sI_o)$, with a final s value of $4.88(6) \times 10^{-7}$. A final Fourier difference map showed no significant residual electron density.

The final atomic parameters listed in Table I were used along with the variance-covariance matrix to calculate the interatomic distances and angles and their *esd*'s reported in Table II. Observed and calculated structure factor amplitudes are listed in Table III.

Results

$\text{La}_3\text{Ru}_3\text{O}_{11}$ crystallizes in a structure derived from that of cubic KSbO_3 , which is also adopted by $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ (2), $\text{Bi}_3\text{GaSb}_2\text{O}_{11}$ (7), and a number of related compounds. The basic structural unit in all KSbO_3 derivatives is the $M_{12}\text{O}_{36}$ three-dimensional network ($M = \text{Ru}, \text{Ga}, \text{Sb}$, and other octahedral ions) consisting of edge-sharing pairs of MO_6 octahedra, further linked through their remaining vertices to other $M_2\text{O}_{10}$ moieties. The resulting framework is quite open and admits a variety of counter ion configurations. In KSbO_3 the interstices of this network are occupied by 12 K^+ ions, although this compound generally shows nonintegral K^+ ion occupancy and is an ionic conductor (8). In $\text{La}_4\text{Ru}_6\text{O}_{19}$ (1) the interstices contain two tetrahedral La_4O groupings, while in $\text{La}_3\text{Ru}_3\text{O}_{11}$ there is a La_{12}O_8 network. The reader is referred to other works (1, 2, 7-9) for drawings and overall description of KSbO_3 -type structures; we will confine structural description to the immediate environments of the metal ions.

The unique Ru atom in $\text{La}_3\text{Ru}_3\text{O}_{11}$ has a distorted octahedral coordination geometry, with Ru-O distances varying from 1.935(3) to 2.007(3) Å, and *cis* O-Ru-O angles ranging from 78.6(2) to 92.4(1)°. Two RuO_6 octahedra share an edge to form the $\text{Ru}_2\text{O}_{10}^{11/3}$ -unit shown in Fig. 1. The group lies on a position

TABLE II

BOND DISTANCES (Å) AND ANGLES (°) WITH STANDARD DEVIATIONS IN PARENTHESES

| Bond Distances | |
|-----------------|----------|
| Ru-O(1)(X2) | 1.935(3) |
| -O(2a)(X2) | 1.990(3) |
| -O(2b)(X2) | 2.007(3) |
| La(1)-O(2)(X6) | 2.551(3) |
| -O(3)(X2) | 2.359(5) |
| La(2)-O(1)(X3) | 2.866(3) |
| -O(2)(X3) | 2.665(3) |
| -O(3)(X3) | 2.334(1) |
| Angles | |
| O(1)-Ru-O(1)' | 78.6(2) |
| O(1)-Ru-O(2a) | 91.1(1) |
| O(1)-Ru-O(2a)' | 92.4(1) |
| O(1)-Ru-O(2b) | 167.5(1) |
| O(1)-Ru-O(2b)' | 88.9(1) |
| O(2a)-Ru-O(2a)' | 175.4(2) |
| O(2a)-Ru-O(2b) | 90.4(2) |
| O(2b)-Ru-O(2b)' | 86.8(2) |

TABLE III

| VALUES OF 18#FOBS AND 18#FCALC | | | | L43 R03 011 | | | | | | | | | | | | | | | |
|--------------------------------|-----|-----|------|-------------|-----|-----|-----|------|-------|-----|-----|-----|------|-------|-----|-----|-----|------|-------|
| H | K | L | FOBS | FCALC | H | K | L | FOBS | FCALC | H | K | L | FOBS | FCALC | H | K | L | FOBS | FCALC |
| 000 | 000 | 000 | 3640 | 3486 | 000 | 000 | 000 | 1674 | 1673 | 000 | 000 | 000 | 1310 | 1315 | 000 | 000 | 000 | 631 | 680 |
| 000 | 000 | 000 | 2467 | 2627 | 000 | 000 | 000 | 2512 | 2612 | 000 | 000 | 000 | 1821 | 1845 | 000 | 000 | 000 | 781 | 729 |
| 000 | 000 | 000 | 941 | 1011 | 000 | 000 | 000 | 246 | 220 | 000 | 000 | 000 | 900 | 951 | 000 | 000 | 000 | 1300 | 1336 |
| 000 | 000 | 000 | 1841 | 1968 | 000 | 000 | 000 | 4235 | 4499 | 000 | 000 | 000 | 1068 | 1088 | 000 | 000 | 000 | 651 | 657 |
| 000 | 000 | 000 | 2098 | 2156 | 000 | 000 | 000 | 1067 | 1056 | 000 | 000 | 000 | 1224 | 1224 | 000 | 000 | 000 | 769 | 742 |
| 000 | 000 | 000 | 264 | 191 | 000 | 000 | 000 | 450 | 419 | 000 | 000 | 000 | 570 | 610 | 000 | 000 | 000 | 360 | 309 |
| 000 | 000 | 000 | 561 | 580 | 000 | 000 | 000 | 1172 | 1083 | 000 | 000 | 000 | 2411 | 2449 | 000 | 000 | 000 | 889 | 859 |
| 000 | 000 | 000 | 33 | 384 | 000 | 000 | 000 | 2717 | 2754 | 000 | 000 | 000 | 1321 | 1273 | 000 | 000 | 000 | 651 | 642 |
| 000 | 000 | 000 | 1896 | 1868 | 000 | 000 | 000 | 1505 | 1530 | 000 | 000 | 000 | 1224 | 1164 | 000 | 000 | 000 | 876 | 823 |
| 000 | 000 | 000 | 1187 | 1154 | 000 | 000 | 000 | 826 | 830 | 000 | 000 | 000 | 362 | 295 | 000 | 000 | 000 | 685 | 639 |
| 000 | 000 | 000 | 2364 | 2426 | 000 | 000 | 000 | 842 | 829 | 000 | 000 | 000 | 457 | 417 | 000 | 000 | 000 | 454 | 450 |
| 000 | 000 | 000 | 1896 | 1868 | 000 | 000 | 000 | 1172 | 1162 | 000 | 000 | 000 | 2691 | 2743 | 000 | 000 | 000 | 1044 | 1043 |
| 000 | 000 | 000 | 1156 | 1203 | 000 | 000 | 000 | 965 | 925 | 000 | 000 | 000 | 1407 | 1362 | 000 | 000 | 000 | 193 | 194 |
| 000 | 000 | 000 | 1094 | 1146 | 000 | 000 | 000 | 625 | 637 | 000 | 000 | 000 | 2215 | 2212 | 000 | 000 | 000 | 1345 | 1369 |
| 000 | 000 | 000 | 1063 | 974 | 000 | 000 | 000 | 1137 | 1152 | 000 | 000 | 000 | 1861 | 1924 | 000 | 000 | 000 | 1820 | 1820 |
| 000 | 000 | 000 | 2019 | 2441 | 000 | 000 | 000 | 885 | 831 | 000 | 000 | 000 | 878 | 1023 | 000 | 000 | 000 | 1561 | 1565 |
| 000 | 000 | 000 | 1584 | 1615 | 000 | 000 | 000 | 402 | 299 | 000 | 000 | 000 | 1075 | 1062 | 000 | 000 | 000 | 1077 | 1037 |
| 000 | 000 | 000 | 914 | 916 | 000 | 000 | 000 | 1174 | 1122 | 000 | 000 | 000 | 1621 | 1689 | 000 | 000 | 000 | 1251 | 1191 |
| 000 | 000 | 000 | 733 | 758 | 000 | 000 | 000 | 1674 | 1726 | 000 | 000 | 000 | 1197 | 1197 | 000 | 000 | 000 | 575 | 599 |
| 000 | 000 | 000 | 1685 | 1700 | 000 | 000 | 000 | 1081 | 1039 | 000 | 000 | 000 | 667 | 841 | 000 | 000 | 000 | 267 | 231 |
| 000 | 000 | 000 | 519 | 586 | 000 | 000 | 000 | 1451 | 1503 | 000 | 000 | 000 | 3469 | 3553 | 000 | 000 | 000 | 656 | 624 |
| 000 | 000 | 000 | 513 | 439 | 000 | 000 | 000 | 452 | 424 | 000 | 000 | 000 | 230 | 200 | 000 | 000 | 000 | 861 | 350 |
| 000 | 000 | 000 | 1971 | 1905 | 000 | 000 | 000 | 1644 | 1567 | 000 | 000 | 000 | 1794 | 1729 | 000 | 000 | 000 | 983 | 988 |
| 000 | 000 | 000 | 3399 | 3355 | 000 | 000 | 000 | 1105 | 1130 | 000 | 000 | 000 | 620 | 590 | 000 | 000 | 000 | 1035 | 1057 |
| 000 | 000 | 000 | 1023 | 1011 | 000 | 000 | 000 | 1761 | 1766 | 000 | 000 | 000 | 1472 | 1425 | 000 | 000 | 000 | 1077 | 1082 |
| 000 | 000 | 000 | 839 | 839 | 000 | 000 | 000 | 688 | 116 | 000 | 000 | 000 | 1391 | 1395 | 000 | 000 | 000 | 314 | 296 |
| 000 | 000 | 000 | 244 | 201 | 000 | 000 | 000 | 2618 | 2634 | 000 | 000 | 000 | 1300 | 1301 | 000 | 000 | 000 | 256 | 214 |
| 000 | 000 | 000 | 2301 | 2343 | 000 | 000 | 000 | 419 | 445 | 000 | 000 | 000 | 327 | 333 | 000 | 000 | 000 | 926 | 914 |
| 000 | 000 | 000 | 1142 | 1142 | 000 | 000 | 000 | 1139 | 1123 | 000 | 000 | 000 | 1729 | 1809 | 000 | 000 | 000 | 869 | 869 |
| 000 | 000 | 000 | 1789 | 1783 | 000 | 000 | 000 | 781 | 763 | 000 | 000 | 000 | 277 | 272 | 000 | 000 | 000 | 423 | 301 |
| 000 | 000 | 000 | 1202 | 1210 | 000 | 000 | 000 | 1859 | 1835 | 000 | 000 | 000 | 2712 | 2635 | 000 | 000 | 000 | 1152 | 1127 |
| 000 | 000 | 000 | 3627 | 3627 | 000 | 000 | 000 | 1148 | 1156 | 000 | 000 | 000 | 311 | 315 | 000 | 000 | 000 | 427 | 427 |
| 000 | 000 | 000 | 1042 | 1043 | 000 | 000 | 000 | 409 | 355 | 000 | 000 | 000 | 404 | 348 | 000 | 000 | 000 | 356 | 306 |
| 000 | 000 | 000 | 2027 | 2016 | 000 | 000 | 000 | 1874 | 1919 | 000 | 000 | 000 | 1163 | 1187 | 000 | 000 | 000 | 421 | 421 |
| 000 | 000 | 000 | 265 | 153 | 000 | 000 | 000 | 905 | 863 | 000 | 000 | 000 | 340 | 305 | 000 | 000 | 000 | 285 | 208 |
| 000 | 000 | 000 | 3129 | 3129 | 000 | 000 | 000 | 1549 | 1549 | 000 | 000 | 000 | 1524 | 1482 | 000 | 000 | 000 | 805 | 802 |
| 000 | 000 | 000 | 2742 | 2799 | 000 | 000 | 000 | 2455 | 2404 | 000 | 000 | 000 | 1608 | 1526 | 000 | 000 | 000 | 1805 | 1916 |
| 000 | 000 | 000 | 710 | 665 | 000 | 000 | 000 | 936 | 979 | 000 | 000 | 000 | 478 | 478 | 000 | 000 | 000 | 978 | 1000 |
| 000 | 000 | 000 | 3964 | 3964 | 000 | 000 | 000 | 366 | 366 | 000 | 000 | 000 | 2407 | 2407 | 000 | 000 | 000 | 2347 | 2347 |
| 000 | 000 | 000 | 1314 | 1314 | 000 | 000 | 000 | 457 | 413 | 000 | 000 | 000 | 654 | 654 | 000 | 000 | 000 | 1662 | 1679 |
| 000 | 000 | 000 | 1288 | 1293 | 000 | 000 | 000 | 2350 | 2198 | 000 | 000 | 000 | 919 | 913 | 000 | 000 | 000 | 559 | 549 |
| 000 | 000 | 000 | 466 | 435 | 000 | 000 | 000 | 2128 | 2176 | 000 | 000 | 000 | 746 | 703 | 000 | 000 | 000 | 746 | 703 |
| 000 | 000 | 000 | 2409 | 2531 | 000 | 000 | 000 | 289 | 851 | 000 | 000 | 000 | 263 | 251 | 000 | 000 | 000 | 792 | 793 |
| 000 | 000 | 000 | 204 | 119 | 000 | 000 | 000 | 1176 | 1147 | 000 | 000 | 000 | 481 | 483 | 000 | 000 | 000 | 400 | 405 |
| 000 | 000 | 000 | 3759 | 3739 | 000 | 000 | 000 | 1783 | 1751 | 000 | 000 | 000 | 775 | 775 | 000 | 000 | 000 | 1043 | 712 |
| 000 | 000 | 000 | 240 | 2385 | 000 | 000 | 000 | 1712 | 1712 | 000 | 000 | 000 | 1712 | 1712 | 000 | 000 | 000 | 1712 | 1712 |
| 000 | 000 | 000 | 1156 | 1111 | 000 | 000 | 000 | 1045 | 1030 | 000 | 000 | 000 | 179 | 156 | 000 | 000 | 000 | 759 | 1066 |

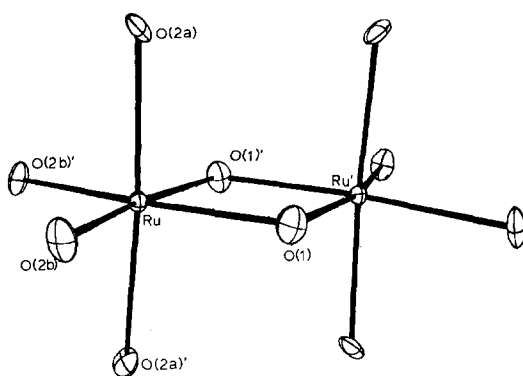


FIG. 1. The $\text{Ru}_2\text{O}_{10}^{1/3-}$ unit.

of 222 symmetry, with one crystallographic two-fold axis passing through both metal atoms. The two ruthenium atoms of this group are displaced from ideal octahedral geometry away from each other, resulting in an acute $\text{O}(1)\text{-Ru-O}(1)'$ angle of 78.6° , and the closest approach between Ru atoms is $2.994(1) \text{ \AA}$.

Two crystallographically independent lanthanum atoms occupy the cavities in the $\text{Ru}_{12}\text{O}_{36}$ network; their coordination environments are shown in Fig. 2. La(1) lies on a site of 23 symmetry and is coordinated by an approximate cube of eight oxygen atoms; six of these are at La–O distances of $2.551(3) \text{ \AA}$ and two more at $2.359(3) \text{ \AA}$. Nine-coordinate La(2) lies on a threefold axis, with La–O distances ranging from $2.334(1)$ to $2.866(3) \text{ \AA}$. The O–La–O angles are not

particularly unusual and were omitted from Table II in the interest of brevity.

Discussion

In $\text{La}_4\text{Re}_6\text{O}_{19}$ (9) the short Re–Re contact (2.415 \AA) has been interpreted as denoting a double bond (10). The somewhat longer Ru–Ru distances in $\text{La}_4\text{Ru}_6\text{O}_{19}$ (1) and $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ (2) (2.488 and 2.60 \AA , respectively) betoken a substantial bonding interaction, while $\text{Ba}_{0.5}\text{IrO}_3$ (11) but slight, if any, metal–metal bonding ($d_{\text{Ir-Ir}} = 2.96 \text{ \AA}$). The trend toward decreasing metal–metal interaction along this series of KSbO_3 -derived oxides is readily understandable as the result of the increase in nuclear charge going from $\text{Re}^{4.33+}$ to $\text{Ru}^{4.33+}$ to Ir^{5+} . However, it is not so easy to explain why, among the three oxides $\text{La}_4\text{Ru}_6\text{O}_{19}$, $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ and $\text{La}_3\text{Ru}_3\text{O}_{11}$, all with Ru in the 4.33 formal oxidation state, two compounds should show Ru–Ru bonding while the third does not. Particularly puzzling is the comparison of $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ with $\text{La}_3\text{Ru}_3\text{O}_{11}$, which raises the question of why the isomorphous replacement of Bi^{3+} with La^{3+} should result in such a dramatic change in Ru–Ru separation, from 2.60 to 2.994 \AA .

Since neither Bi^{3+} nor La^{3+} is likely to participate in covalent interactions with the ruthenium atoms, the only factors likely to affect the structures are some difference in the over-

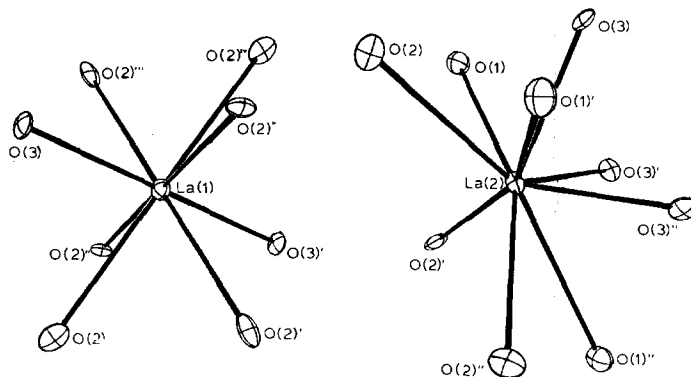


FIG. 2. The La(1) and La(2) coordination environments.

all size or some particular coordination geometry of the tripositive ions. However, a general dilation of the crystal lattice due to ionic size effects does not seem sufficient to explain the drastic increase in Ru–Ru distance observed. The unit cell volume of $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ is fully 10% greater than that of $\text{La}_4\text{Ru}_6\text{O}_{19}$, yet its Ru–Ru distance is only 0.11 Å longer. By contrast, the 5% cell volume increase from $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ to $\text{La}_3\text{Ru}_3\text{O}_{11}$ results in a 0.39 Å increase in $d_{\text{Ru–Ru}}$. Aside from general ionic size differences, the coordination geometries of La^{3+} and Bi^{3+} in the two compounds are roughly similar; if anything, the Bi^{3+} coordination is more regular than that of La^{3+} . Thus the structural differences between $\text{La}_3\text{Ru}_3\text{O}_{11}$ and $\text{Bi}_3\text{Ru}_3\text{O}_{11}$ are apparently not attributable to distortions due to a $\text{Bi}^{3+} 6s^2$ lone pair effect.

In summary, there does not seem to be any single, simple reason why Ru–Ru bonding should be absent in $\text{La}_3\text{Ru}_3\text{O}_{11}$. We therefore suggest that those Ru–Ru bonds that are found in ruthenium oxides with Ru having oxidation numbers >4 are only barely favored energetically, so that even so subtle a change as the replacement of Bi^{3+} by La^{3+} can dis-

favor them. Perhaps this low stability is the reason why so few examples of metal–metal bonding have been found in ruthenium oxide systems.

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