# The Selenide Systems of Niobium and Tantalum

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The selenide systems of niobium and tantalum were investigated under high pressures in the composition limits of  $MX_2$  to  $MX_3$ . Two distinct phases, NbSe<sub>2</sub> and NbSe<sub>3</sub>, were obtained in the niobium system. In the tantalum system, there were three phases, TaSe<sub>2</sub>, "TaSe<sub>3</sub>," which has been thought of as a triselenide, and stoichiometric TaSe<sub>3</sub>. The ratio of Se/Ta in "TaSe<sub>3</sub>" was 2.8. Conversions of the triselenides under high pressure to NbSe<sub>2</sub> or TaSe<sub>2</sub> were observed at the starting composition of Se/M = 2.0 and to "TaSe<sub>3</sub>" at Se/Ta = 2.8.

#### Introduction

Several kinds of phases having various compositions in the binary systems of niobium or tantalum with selenium have been reported (1, 2). The crystal structures can be viewed as metal clusters consisting of metal-metal bonds in the metal-rich compounds such as Nb<sub>2</sub>Se and Nb<sub>5</sub>Se<sub>4</sub> (3). On the other hand, low-dimensional linkage of  $MSe_6$  units can be observed in the chalcogen-rich region. There exist two families of low-dimensional compounds, the  $MX_2$ layered dichalcogenides and the pseudoone-dimensional  $MX_3$  trichalcogenides (4). There are strong ionocovalent or metallic bonds inside the layers or chains. These structural units are linked together with weak interlayer or interchain bonding of the order of van der Waals force. The structural anisotropy results in a very high anisotropy in the electronic, vibrational, and mechanical properties. The most interesting physical aspects of these materials are the observation of phenomena such as Peierls transitions, Kohn anomalies, and charge density waves which are characteristic of low-dimensional compounds (5). From a chemical point of view, intercalations into di- and trichalcogenides have received considerable attention in recent years (6). The compounds can reversibly take various chemical species in and out of their van der Waals gaps. The lithium intercalation has been applied in making the cathode of a secondary battery.

Thorough understanding of these properties necessitates some knowledge of the phase relationships in these systems. Revolinsky *et al.* and Selte *et al.* have already investigated the phase relations between the composition limits  $M_5Se_4$  and  $MSe_3$ (1, 2). However, the information was not sufficient in the vicinity of  $MSe_3$ . Selte *et al.* announced three forms of a niobium selenide with an approximate composition NbSe<sub>4</sub> (namely  $\alpha$ ,  $\beta$ ,  $\gamma$ ) (2). Meerschaut and co-workers proved that three different materials were concerned, the chemical compositions of which were NbSe<sub>3</sub>, Nb<sub>2</sub>Se<sub>9</sub>, and  $I_{0.33}$ NbSe<sub>4</sub> (7-10). In the accompanying manuscript, Kikkawa *et al.* found a modification of TaSe<sub>3</sub> having a crystal structure similar to that of NbSe<sub>3</sub> under high pressure (11). These recent works stimulate some questions on the chemical composition of TaSe<sub>3</sub>, for which characterizations have already been done (12-14).

The present study is concerned with determining the phase relations in selenides of niobium and tantalum under high pressures. The study was thus limited to the composition range from  $MSe_2$  to  $MSe_3$ . The molar ratio of Se/Ta was about 2.8 in "TaSe<sub>3</sub>," which was prepared in evacuated quartz ampoules above 700°C.

## Experimental

The selenides of niobium and tantalum were prepared under high-pressure and temperature conditions using a cubic anviltype device as described in the accompanying article (11). The preparations were made by changing the starting composition of Se/M from 2.0 to 3.2, reaction pressure from 1 to 5 GPa, reaction temperature from 500 to 700°C, and duration from 30 to 240 min. The phases obtained were identified using X-ray powder diffraction. The yield of each phase was calculated using characteristic diffraction lines of the basal reflection for  $MSe_2$  at CuK $\alpha 2\theta = 14.2^\circ$ , reflections indexed as 100 for  $MSe_3$  at

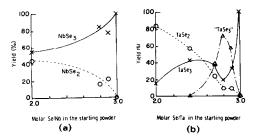


FIG. 1. Relative yields of niobium (a) and tantalum (b) selenides in the starting composition range from Se/M = 2.0 to 3.0.

 $Cu K\alpha 2\theta = 9.3^{\circ}$ , and as 001 for "TaSe<sub>3</sub>" at  $Cu K\alpha 2\theta = 11.4^{\circ}$ . Relative intensities of the diffraction lines were measured. The yield of NbSe<sub>3</sub>, for example, was computed as (intensity of 100 for NbSe<sub>3</sub>)/{(intensity for 100 for NbSe<sub>3</sub>) + (intensity of the basal reflection for NbSe<sub>2</sub>)}. Thermogravimetry was applied to determine the chemical compositions of both triselenides prepared in quartz ampoules and ones under high pressures.

#### Results

## Niobium Selenide System

The niobium selenide system was prepared by changing the starting composition of Se/M from 2.0 to 3.0 at 700°C and 2 GPa for 60 min. Only NbSe<sub>3</sub> was observed at the limit of Se/Nb = 3.0, as shown in Fig. 1a. NbSe<sub>2</sub> appeared and the amount of NbSe<sub>2</sub> increased with the decrease of selenium in the starting material. However, a mixture of NbSe<sub>2</sub> and NbSe<sub>3</sub> was obtained even at Se/Nb = 2.0 with coexistence of unreacted niobium. The reaction system was probably not in equilibrium under the above-mentioned reaction conditions. Thus, the duration was prolonged to 180 min under the same reaction temperature-pressure conditions at Se/Nb = 2.0. The yields of  $NbSe_2$ and NbSe<sub>3</sub>, respectively, increased and decreased with the duration, as represented in Fig. 2a. Three hours of duration was not enough to obtain NbSe<sub>2</sub> as a single phase. Niobium selenides were also prepared by changing the pressure over the range 2-5GPa at Se/Nb = 3.0 at 700°C for 30 min. The products always consisted of a single phase of NbSe<sub>3</sub> in the above pressure range.

## Tantalum Selenide System

The tantalum selenide system was also prepared by changing the starting composition of Se/Ta from 2.0 to 3.2 as in the case of the niobium selenide. TaSe<sub>3</sub> having a

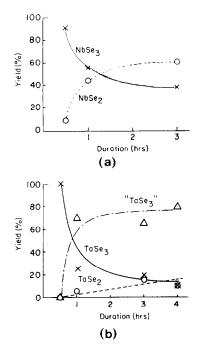


FIG. 2. Duration dependence of relative yields in the systems Se/Nb = 2.0 (a) and Se/Ta = 2.8 (b).

crystal structure similar to NbSe<sub>3</sub> was obtained without coexistence of another phase at Se/Ta = 3.0, as depicted in Fig. 1b. Coexisting selenium was observed with  $TaSe_3$  at Se/Ta = 3.2.  $TaSe_2$  and "TaSe\_3," which have been obtained in evacuated quartz ampoules, appeared in the starting compositional range of less selenium than Se/Ta = 3.0. The phase "TaSe<sub>3</sub>" could be observed at Se/Ta = 2.7, 2.8, and 2.9. The largest amount of "TaSe<sub>3</sub>" was obtained at Se/Ta = 2.8. The relative amount of  $TaSe_2$ rapidly increased with further decrease of selenium in the starting powder. However, TaSe<sub>2</sub> coexisted with TaSe<sub>3</sub> even at Se/Ta = 2.0. The yields were investigated by changing duration at Se/Ta = 2.8 and at Se/Ta = 2.0, respectively, under the reaction conditions of 2 GPa and 700°C. Only TaSe<sub>3</sub> was observed at a duration of 30 min at Se/Ta = 2.8, as shown in Fig. 2b, with slight coexistence of Ta. The amount of TaSe<sub>3</sub> quickly decreased with rapid increase of "TaSe<sub>3</sub>" within 1 hr. TaSe<sub>2</sub> also appeared. The quantities of "TaSe<sub>3</sub>" and TaSe<sub>2</sub> gradually increased and the amount of TaSe<sub>3</sub> decreased with longer duration. Although there were coexisting TaSe<sub>3</sub>, "TaSe<sub>3</sub>," and TaSe<sub>2</sub> in the reaction for 4 hr at 700°C, only TaSe<sub>3</sub> was observed at 500°C at the same duration. The rate of decomposition to "TaSe<sub>3</sub>" and TaSe<sub>2</sub> is probably slow at this temperature. At Se/Ta = 2.0, 60% of TaSe<sub>3</sub> and 40% of TaSe<sub>2</sub> were observed at a duration of 30 min. The amount of TaSe<sub>3</sub> decreased to 10% and the quantity of TaSe<sub>2</sub> increased to 90% within 1 hr. The relative amounts were almost the same even at a duration of 3 hr. This situation was quite similar to the case of the niobium selenide. The phases which appeared were compared under two different pressures, 2 and 5 GPa, at 700°C for a duration of 30 min. Only TaSe<sub>3</sub> was detected at the composition of Se/Ta = 3.0 under both pressures.

## Discussion

In the compositional range from  $MX_2$  to  $MX_3$ , the niobium selenide system had two distinct phases, NbSe2 and NbSe3. The tantalum selenide system, on the other hand, gave an additional phase, "TaSe<sub>3</sub>". The largest amount of this compound was obtained at the composition Se/Ta = 2.8 in the starting material, as shown in Fig. 1b. This fact suggests that the phase "TaSe<sub>3</sub>" has less selenium composition than stoichiometric TaSe<sub>3</sub> which was first prepared under high pressures (11). In order to determine the chemical compositions, TG-DTA was measured on about 20 mg of "TaSe<sub>3</sub>" and NbSe<sub>3</sub> prepared in evacuated quartz tubes. The corresponding diselenides were not observed in the samples by X-ray measurement. The triselenides were rapidly oxidized at around 300°C in air. Large exotherms and quick weight losses were observed. The weight of samples became

constant in the temperature range 600–700°C. X-Ray diffraction showed that the samples taken out from 700°C consisted of Ta<sub>2</sub>O<sub>5</sub> and Nb<sub>2</sub>O<sub>5</sub>. The amounts of Ta or Nb in the triselenides were estimated by weighing as Ta<sub>2</sub>O<sub>5</sub> or Nb<sub>2</sub>O<sub>5</sub>. The results are summarized in Table I along with values for the samples prepared under high pressures. From these results, it was decided that "TaSe<sub>3</sub>" had the chemical composition TaSe<sub>2.8</sub>, and that, on the contrary, NbSe<sub>3</sub> had stoichiometric composition. The triselenides, TaSe<sub>3</sub> and NbSe<sub>3</sub>, prepared under high pressures were stoichiometric, of course.

The stoichiometric TaSe<sub>3</sub> partially decomposed to "TaSe3" and Se due to heating at 250°C for 1 day in an evacuated quartz tube. This fact also shows that "TaSe<sub>3</sub>" has less selenium than stoichiometric TaSe<sub>3</sub>. Moreover, TaSe<sub>3</sub> completely decomposed to TaSe<sub>2</sub> and Se when it was heated at 500°C for 20 hr. The rate of decomposition was depressed with the addition of selenium in the ampoule. It is quite clear that "TaSe<sub>3</sub>" is a less-selenium phase and the compound obtained under high pressures is stoichiometric TaSe<sub>3</sub>, considering the results of the preparations under high pressures, the chemical analysis, and the thermal decompositions.

TABLE I

CHEMICAL COMPOSITIONS OF NIOBIUM TRISELENIDE AND TANTALUM TRISELENIDE

Material	Sample wt (mg)	Metal (wt%)	Se (wt%)	Molar ratio Se/M	Average
"TaSe <sub>a</sub> "*	14.3 15.3	45. <sub>8</sub> 44.4	54. <sub>8</sub> 55. <sub>6</sub>	2.6 2.8	2.8
TaSe3	12.0 17.9	43. <sub>7</sub> 43. <sub>5</sub>	56. <sub>s</sub> 56. <sub>s</sub>	2.9 <sub>2</sub> 2.9 <sub>4</sub>	2.9
NbSe <sub>3</sub> *	13.0	27.5	72. <sub>5</sub>	3.07	3.1
NbSe <sub>3</sub>	13.4 14.3	28.4 27.0	71.8 73.0	2.9 <sub>7</sub> 3.1 <sub>8</sub>	3.1

Note. Materials marked with asterisks were prepared in evacuated tubes.

There are two possibilities for the phase "TaSe<sub>3</sub>" of TaSe<sub>3- $\delta$ </sub> and Ta<sub>1+ $\epsilon$ </sub>Se<sub>3</sub>. Density measurement may give the answer. However, it is very difficult to obtain enough "TaSe<sub>3</sub>" without contamination of TaSe<sub>2</sub> for the measurement. Bjerkelund and Kjekshus reported the observed and calculated densities as 7.913 and 8.080 g cm<sup>-3</sup>, respectively (12). Simple calculation using these values leads to the composition  $TaSe_{3-0.1}$ , which is not so reliable due to the possible contamination. Unit cell variation was also checked according to composition. The lattice parameters a, b, and c were calculated using 200, 020, and 006 diffractions for TaSe<sub>3</sub> on the samples prepared from the starting compositions of Se/Ta = 3.0, 2.9,and 2.8 at 700°C and 2 GPa for 60 min. The variation of these lattice parameters was less than 0.02 Å. This suggests that excess tantalum is not intercalated as the lithium intercalated into NbSe<sub>3</sub> (15). The parameters a and c expanded 1.4 Å and b did not change in the lithium intercalation. Thus, TaSe<sub>3</sub> and "TaSe<sub>3</sub>" seem to have exact compositions without solid solution range. This is probably the reason why TaSe<sub>3</sub>, "TaSe<sub>3</sub>," and TaSe<sub>2</sub> coexisted in the preparation under pressure. The residual resistance ratio for "TaSe<sub>3</sub>" was reported to be about 70 (16). The value is too low to consider the phase as nonstoichiometric. The crystal structure of TaSe<sub>3</sub> is not directly related to that for "TaSea" at the moment even if it is considered to be the same as that for NbSe<sub>3</sub>. Structure analysis is required.

Among the three tantalum selenides, TaSe<sub>3</sub> is probably equilibrated with the highest selenium vapor pressure on decomposition. Therefore, high-pressure technique was quite effective in the preparation of TaSe<sub>3</sub>. Phase relations can be schematically represented as shown in Fig. 3. TaSe<sub>3</sub> has the highest vapor pressure at every temperature. When the vapor pressure is kept at  $P_1$  and there is a temperature gradi-

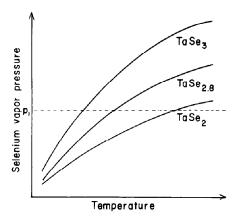


FIG. 3. Schematic pressure and temperature phase diagram of the tantalum selenide system.

ent in the reaction vessel, TaSe<sub>3</sub> crystallizes in the lowest temperature region, TaSe<sub>2.8</sub> in the middle, and TaSe<sub>2</sub> at the highest. This figure leads to the idea that TaSe<sub>3</sub> might be obtained even in a quartz tube if its decomposition to "TaSe<sub>3</sub>" or TaSe<sub>2</sub> is well depressed. The reaction vessel was kept just above the boiling point of selenium. A very small crystal of TaSe<sub>3</sub>, in fact, was prepared in an evacuated quartz tube at 450°C for 1 month. Its crystal structure analysis will be reported elsewhere.

NbSe<sub>3</sub> was obtained at the beginning and then reacted with Nb changing to NbSe, in the preparation under high pressure at the starting composition of Se/Nb = 2.0, as depicted in Fig. 2a. In contrast, NbSe<sub>3</sub> was directly prepared at Se/Nb = 3.0. TaSe<sub>2</sub> was also obtained through TaSe<sub>3</sub>. MoS<sub>2</sub> can be prepared by the thermal decomposition of amorphous MoS<sub>3</sub> but the situation is different from the present case (17). The present reaction is not only a decomposition of triselenide to diselenide but also a recombination of the metal with the decomposed selenium. "TaSe3" was also obtained through TaSe3 at the starting composition of Se/Ta = 2.8, as shown in Fig. 2b.

In conclusion, it was shown that the tantalum selenide which has been reported as  $TaSe_3$  has the actual composition of TaSe<sub>2.8</sub>. The trisulfides of Ta and Nb also have less sulfur composition than stoichiometry. The trichalcogenides prepared under high pressure and in evacuated ampoules were not pressure modifications. They have different chemical compositions. The kinetics under high pressures were observed in the preparation of diselenides through triselenides in both systems of niobium-selenide and tantalum-selenide. TaSe<sub>2.8</sub> was also obtained through the recombination of TaSe<sub>3</sub> with the coexisting TaSe<sub>2</sub> and Ta.

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## References

- 1. K. SELTE, E. BJERKELUND, AND A. KJEKSHUS, J. Less-Common Met. 11, 14 (1966).
- 2. E. REVOLINSKY, B. E. BROWN, D. J. BEERNTSEN, AND C. H. ARMITAGE, J. Less-Common Met. 8, 63 (1965).
- 3. H. F. FRANZEN, Prog. Solid State Chem. 12, 1 (1978).
- F. HULLIGER, "Physics and Chemistry of Materials with Layered Structures," Vol. 5, "Structural Chemistry of Layer Type Phases" (F. Levy, Ed.), Reidel, Dordrecht (1976).
- 5. A. J. WILSON, Phy. Rev. B 19, 6456 (1979).
- 6. M. S. WHITTINGHAM, Prog. Solid State Chem. 12, 41 (1978).
- A. MEERSCHAUT AND J. ROUXEL, J. Less-Common Met. 39, 197 (1975).
- J. L. HODEAU, M. MAREZIO, C. ROUCAU, R. AYROLES, A. MEERSCHAUT, J. ROUXEL, AND P. MONCEAU, J. Phys. C 11, 4117 (1978).
- 9. A. MEERSCHAUT, L. GUÉMAS, R. BERGER, AND J. ROUXEL, Acta Crystallogr. Sect. B 35, 1747 (1979).
- 10. A. MEERSCHAUT, P. PALVADEAU, AND J. ROUXEL, J. Solid State Chem. 20, 21 (1977).

- 11. S. KIKKAWA, N. OGAWA, M. KOIZUMI, AND Y. ONUKI, J. Solid State Chem. 41, 315 (1982).
- 12. E. BJERKELUND AND A. KJEKSHUS, Z. Anorg. Allg. Chem. 328, 235 (1964).
- 13. E. BJERKELUND AND A. KJEKSHUS, Acta Chem. Scand. 19, 701 (1965).
- 14. E. BJERKELUND, J. H. FERMOR, AND A. KJEK-

shus, Acta Chem. Scand. 20, 1836 (1966).

- 15. R. R. CHIANELLI AND M. B. DINES, Inorg. Chem. 14, 2417 (1975).
- 16. М. ҮАМАМОТО, J. Phys. Soc. Japan 45, 431 (1978).
- 17. J. C. WILDERVANCK AND F. JELLINEK, Z. Anorg. Allg. Chem. 328, 309 (1964).