

## Preparation, Structure, and Magnetic Properties of Isostructural $\text{La}_3\text{MAIS}_7$ and $\text{La}_3\text{MFeS}_7$ ( $M = \text{Mg, Mn, Fe, Co, Ni, or Zn}$ )\*

K. S. NANJUNDASWAMY AND J. GOPALAKRISHNAN

*Solid State and Structural Chemistry Unit, Indian Institute of Science, Bangalore 560012, India*

Received April 6, 1983

A series of quaternary metal sulfides of the general formula  $\text{La}_3\text{MM}'\text{S}_7$  ( $M = \text{Mn, Fe, Co}$ ;  $M' = \text{Al}$  and  $M = \text{Mg, Mn, Fe, Co, Ni}$ ;  $M' = \text{Fe}$ ) consisting of linear chains of face shared  $\text{MS}_6$  octahedra and isolated  $M'\text{S}_4$  tetrahedra has been prepared and studied. The aluminium compounds  $\text{La}_3\text{MAIS}_7$  ( $M = \text{Mn, Fe, Co}$ ) exhibit linear chain antiferromagnetism. Magnetic behavior of other  $\text{La}_3\text{MFeS}_7$  sulfides has been examined in detail. The magnetic susceptibility of  $\text{La}_3\text{MgFeS}_7$  shows that tetrahedral site  $\text{Fe}^{3+}$  undergoes a transition from  $S = \frac{5}{2}$  to  $S = 2$  spin state around 150 K.

### Introduction

Flahaut and co-workers (1-3) reported a family of metal sulfides of the general formula  $\text{Ln}_3\text{MM}'\text{S}_7$  ( $\text{Ln} = \text{rare earth}$ ,  $M = \text{a divalent metal such as Mg, Mn, etc.}$ , and  $M' = \text{Al or Ga}$ ). These sulfides crystallize in a hexagonal structure (space group  $P6_3$ ) consisting of chains of  $\text{MS}_6$  octahedra sharing faces along the  $c$  axis and isolated  $M'\text{S}_4$  tetrahedra (Fig. 1). The structure appears to be stabilized by the tetrahedral site  $M'$  atoms that form strong covalent bonds with sulfur (1). It is apparent from Fig. 1 that magnetic interaction among the  $M$  ions in the octahedral chains can occur in only one dimension. With a view to investigate such a one-dimensional behavior, we have carried out a magnetic susceptibility study of sulfides of the formula  $\text{La}_3\text{MAIS}_7$  where  $M$

$= \text{Mn, Fe, or Co}$ . The study has indeed revealed linear chain antiferromagnetism in these novel sulfides.

Another interesting aspect of  $\text{La}_3\text{MM}'\text{S}_7$  sulfides that we have investigated relates to the replacement of Al or Ga at the tetrahedral ( $M'$ ) sites by a paramagnetic ion. The obvious choice for a paramagnetic  $M'$  would be  $\text{Fe}^{3+}$  since this ion is known to exist in tetrahedral sulfur coordination in several sulfides such as  $\text{AFeS}_2$  ( $A = \text{K, Rb, Cs}$ ) (4),  $\text{BaFe}_2\text{S}_4$  (5),  $\text{Ba}_5\text{Fe}_4\text{S}_{11}$  (6), and  $\text{Na}_3\text{FeS}_3$  (7). We have successfully prepared a series of sulfides of the general formula  $\text{La}_3\text{MFeS}_7$  with  $M = \text{Mg, Mn, Fe, Co, Ni, and Zn}$ . These sulfides are isostructural with the  $\text{La}_3\text{MAIS}_7$  series, but show distinctly different magnetic properties. In particular,  $\text{La}_3\text{MgFeS}_7$  shows a transition involving the spin state of the  $\text{Fe}^{3+}$  ion from  $S = \frac{5}{2}$  to  $S = 2$  state which is likely to arise from an electron transfer from sulfur ( $3p^6$ ) to iron ( $3d^5$ ).

\* Contribution 207 from the Solid State and Structural Chemistry Unit.

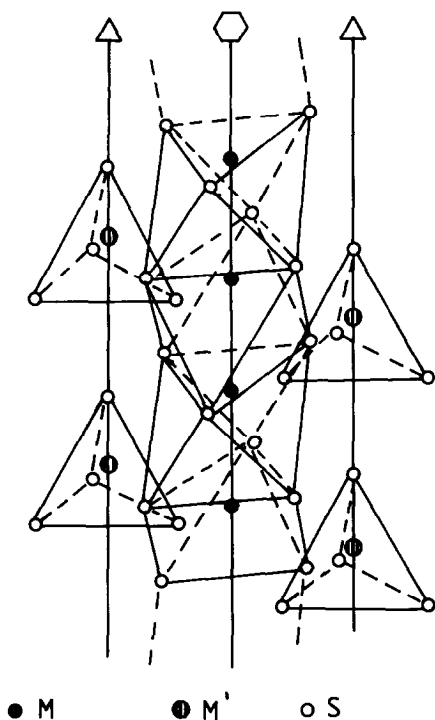


FIG. 1. Structure of  $\text{La}_3\text{MM}'\text{S}_7$ . Only  $\text{MS}_6$  octahedra forming chains and isolated  $\text{M}'\text{S}_4$  tetrahedra are shown (after Ref. (1)).

### Experimental

$\text{La}_3\text{MFeS}_7$  ( $M = \text{Mg}, \text{Mn}, \text{Fe}, \text{Co}, \text{Ni},$  or  $\text{Zn}$ ) were prepared by sulfidation of oxide mixtures using  $\text{CS}_2$  vapors at  $1100^\circ\text{C}$ . The oxide mixtures were prepared by mixing aqueous solutions of the nitrates of lanthanum, iron, and  $M$  in the required stoichiometry, concentrating to a small volume and decomposing it around  $600^\circ\text{C}$ . Graphite boats were used for sulfidation. Sulfide formation was found to be complete in about 6 days with two grindings in between. The sulfur content of the samples was determined gravimetrically as  $\text{BaSO}_4$  after oxidizing the sulfide with  $\text{Br}_2$  in  $\text{CCl}_4$  and  $\text{HNO}_3$ . The procedure for the preparation of  $\text{La}_3\text{MAIS}_7$  ( $M = \text{Mn}, \text{Fe}, \text{Co}$ ) was similar except that slightly higher temperature ( $1150$ – $1200^\circ\text{C}$ ) was required for sulfidation.

Experimental procedures for recording X-ray powder diffraction patterns and measuring magnetic and electrical properties were the same as reported by us in a recent paper (8).

### Results and Discussion

#### $\text{La}_3\text{MAIS}_7$

To characterize the magnetic behavior of paramagnetic  $M^{2+}$  ions at the octahedral sites of the  $\text{Ln}_3\text{MAIS}_7$  structure (1), we have measured the susceptibilities of  $\text{La}_3\text{MAIS}_7$  with  $M = \text{Mn}, \text{Fe},$  or  $\text{Co}$  and the  $\chi_m - T$  plots are shown in Fig. 2. All three sulfides show a broad maximum around  $115$ – $130$  K indicating an antiferromagnetic ordering. The susceptibility increases sharply below  $80$  K. Magnetism in these sulfides consisting of linear chains of face-sharing  $\text{MS}_6$  octahedra along the hexagonal  $c$  axis is comparable to that of  $\text{Ba}_2\text{MS}_3$  (9). The linear chain antiferromagnetism of the latter series of sulfides has been attributed to a short-range magnetic ordering in the chains formed by corner-sharing  $\text{MS}_4$  tetrahedra. The temperature dependence of the susceptibility of these sulfides has been analyzed on the basis of a reduced-spin model proposed by Emori *et al.* (10) for one-dimensional systems with  $S \geq 1$ . The susceptibility according to this model is given by

$$\chi = \frac{Ng^2\beta S(S+1)}{3kT} e^{J'/kT}$$

where  $J' = \frac{1}{3}JS(S+1)$  and  $N, g,$  and  $\beta$  have the usual significance. A plot of  $\log[3k\chi_m T/N\beta^2]$  against  $T^{-1}$  would be linear above the ordering temperature, the slope and the intercept giving  $J$  and  $\mu_{\text{eff}}$  values. Such plots for  $\text{La}_3\text{MAIS}_7$  are shown in Fig. 3 and the relevant data from the plots in Table I. The  $\mu_{\text{eff}}$  and  $J/K$  values are comparable to those of  $\text{Ba}_2\text{MS}_3$  indicating that  $M^{2+}$  in  $\text{La}_3\text{MAIS}_7$  are in the high spin state and that these sulfides show linear chain antiferromagnetism

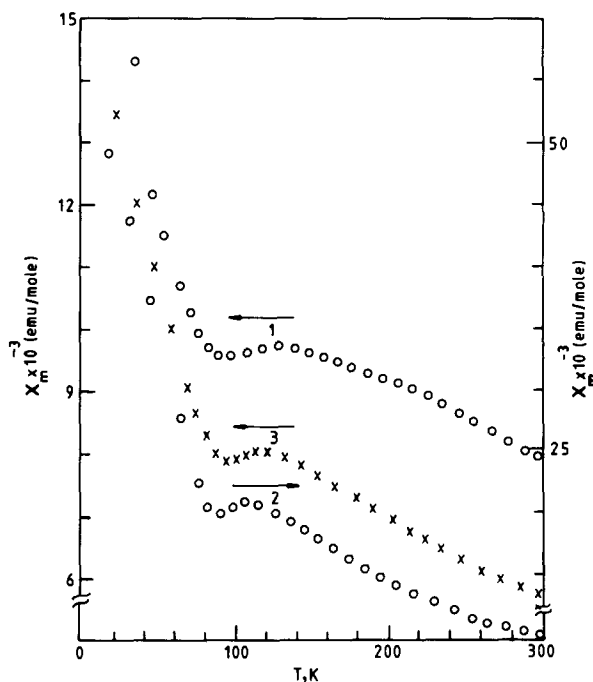


FIG. 2.  $\chi_m - T$  plots of  $\text{La}_3\text{MAIS}_7$ : 1,  $M = \text{Mn}$ ; 2,  $M = \text{Fe}$ ; and 3,  $M = \text{Co}$ .

due to short-range ordering in the  $\text{MS}_6$  octahedral chains. The sharp increase of susceptibility at low temperatures is seen in other linear chain antiferromagnetic systems such as  $(\text{NH}_4)_2\text{MnF}_5$  and has been attributed by Emori *et al.* (10) to interchain interaction and end effects in the chain.

### $\text{La}_3\text{MFeS}_7$

We have prepared a new series of  $\text{La}_3\text{MM}'\text{S}_7$  sulfides with  $M' = \text{Fe}$  and  $M = \text{Mg}, \text{Fe}, \text{Co}, \text{Ni},$  or  $\text{Zn}$  by reaction of the

corresponding oxide mixtures with  $\text{CS}_2$  vapors around  $1100^\circ\text{C}$ . Chemical analysis of sulfur and X-ray diffraction data (Table II) reveal that stoichiometric  $\text{La}_3\text{MFeS}_7$  are formed for all the  $M$  investigated. The Mn, Fe, Co, and Ni sulfides are isostructural crystallizing in the hexagonal structure of  $\text{La}_3\text{MAIS}_7$  (1-3). The  $c$  parameter remains nearly constant but the cell volume shows a slight decrease along the series Mn-Ni as expected. The  $M = \text{Mg}$  and  $\text{Zn}$  compounds show a distortion to orthorhombic symme-

TABLE I  
MAGNETIC DATA ON  $\text{La}_3\text{MAIS}_7$  ( $M = \text{Mn}, \text{Fe}, \text{Co}$ )

Compound	$T_N$ (K)	$\theta$ (K)	$\mu_{\text{eff}}$ (BM) from $\chi_m^{-1}$ vs $T$ plot	$\mu_{\text{eff}}$ (BM) = $[g^2 S(S+1)]^{1/2}$	$J/K$	$M-M$ distance (Å)
$\text{La}_3\text{MnAIS}_7$	115	-515	6.80	5.72	-14	3.01
$\text{La}_3\text{FeAIS}_7$	110	-48	5.65	5.62	-6.2	3.01
$\text{La}_3\text{CoAIS}_7$	130	-260	4.90	4.30	-21	3.01

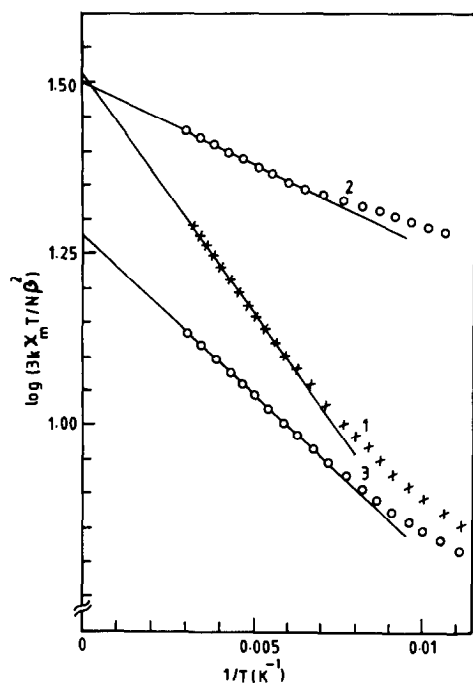


FIG. 3. Analysis of magnetic susceptibility data of  $\text{La}_3\text{MAIS}_7$  according to the reduced-spin model: 1,  $M = \text{Mn}$ ; 2,  $M = \text{Fe}$ ; and 3,  $M = \text{Co}$ .

try with  $a_o \approx a_h$ ,  $b_o \approx \sqrt{3} a_h$ , and  $c_o = c_h$ , where  $a_h$  and  $c_h$  denote the hexagonal cell parameters of  $\text{La}_3\text{MM}'\text{S}_7$  cell. A comparison of the unit cell parameters of  $\text{La}_3\text{MFeS}_7$  series with those of the corresponding  $\text{La}_3$

$\text{MAIS}_7$  (2) reveals that the  $c/a$  ratio is significantly smaller in the iron series ( $c$  axis particularly smaller). Flahaut and Laruelle (1) have remarked that the  $c$  parameter is principally determined by the tetrahedral site  $M'$  atom. The distinct shortening of the  $c$  parameter in the  $\text{La}_3\text{MFeS}_7$  series as compared to the aluminium series may be taken as an indication of strong covalent Fe-S bonds in the  $\text{FeS}_4$  tetrahedra.

We have refined the atomic position parameters of  $\text{La}_3\text{MnFeS}_7$  in the space group  $P6_3$  using X-ray powder diffraction intensities and the program POWDER (11). The observed and calculated intensities are given in Table III and the corresponding atomic positions in Table IV. The refinement shows that  $\text{La}_3\text{MnFeS}_7$  is indeed isostructural with the  $\text{La}_3\text{MAIS}_7$  series. The bond lengths indicate that the coordination of Mn is octahedral with two sets of Mn-S bonds (2.81 and 2.77 Å). The  $\text{FeS}_4$  tetrahedron is distorted with one short (2.11 Å) and three long (2.22–2.23 Å) bonds. These bond lengths are believed to be accurate only within  $\pm 0.01$  Å since the values are obtained from a refinement of limited number of powder diffraction intensity data. The average Fe-S distance is usually around 2.233 Å in tetrahedrally coordinated  $\text{Fe}^{3+}$  sulfides (12). A similarly distorted

TABLE II  
UNIT CELL PARAMETERS AND SULPHUR ANALYSIS DATA OF  $\text{La}_3\text{MFeS}_7$

Compound	$a$ (Å)	$c$ (Å)	$c/a$	Unit cell volume $V$ (Å <sup>3</sup> )	S (%)	
					Exp.	Calc.
$\text{La}_3\text{MnFeS}_7$	10.323(5)	5.803(3)	0.5620	803.4	29.95	29.85
$\text{La}_3\text{Fe}_2\text{S}_7$	10.321(6)	5.804(4)	0.5623	803.2	29.85	29.80
$\text{La}_3\text{CoFeS}_7$	10.321(5)	5.804(4)	0.5622	802.8	30.60	29.70
$\text{La}_3\text{NiFeS}_7$	10.322(6)	5.802(3)	0.5621	803.0	28.25	29.70
$\text{La}_3\text{MgFeS}_7$	$a = 10.297(7)$ $b = 18.020(1)$	5.807(5)	0.5640	808.0	31.06	31.11
$\text{La}_3\text{ZnFeS}_7$	$a = 10.286(8)$ $b = 18.050(1)$	5.802(5)	0.5640	808.0	29.30	29.44

TABLE III  
 X-RAY POWDER DIFFRACTION DATA OF  $\text{La}_3\text{MnFeS}_7$ 

<i>hkl</i>	<i>d</i> <sub>obs</sub> (Å)	<i>d</i> <sub>calc</sub> (Å) <sup>a</sup>	<i>I</i> <sub>obs</sub>	<i>I</i> <sub>calc</sub> <sup>b</sup>
101	4.87	4.86	22.00	21.80
200	4.48	4.47	21.30	20.44
201	3.54	3.54	100.00	102.54
210	3.380	3.380	35.90	35.24
211	2.921	2.920	14.60	10.96
002	2.894	2.901	17.30	17.38
301	2.652	2.651	26.40	24.50
220	2.585	2.581	50.10	49.88
112	2.529	2.529	31.50	31.60
310	2.480	2.479	49.80	49.92
212	2.196	2.201	31.60	31.61
321	1.933	1.934	21.20	24.31
222		1.928		
312	1.880	1.884	31.60	30.14
103		1.890		
411	1.848	1.849	23.60	24.26
402	1.771	1.770	18.60	18.43
203		1.775		
501	1.709	1.709	8.00	9.25
331	1.650	1.649	12.20	11.37
412	1.618	1.619	15.40	14.47
421		1.622		
303		1.622		

<sup>a</sup> The unit cell parameters are given in Table II.

<sup>b</sup> The atomic position parameters are given in Table IV.

$\text{AlS}_4$  tetrahedron with three long and one short Al–S bonds is found in the corresponding aluminium sulfides (1).

The inverse magnetic susceptibility–temperature plots for  $\text{La}_3\text{MFeS}_7$  sulfides are shown in Fig. 4. We see two linear regions with a break or plateau in the region 150–180 K in most cases. The magnetic behavior of  $\text{La}_3\text{MgFeS}_7$  is expected to be the simplest arising from tetrahedral site  $\text{Fe}^{3+}$ . Indeed, the high temperature linear region of the  $X_m^{-1} - T$  plot of  $\text{La}_3\text{MgFeS}_7$  gives a  $\mu_{\text{eff}}$  of 5.85 B.M. ( $\theta = -50$  K) which is consistent with high spin ( $S = \frac{5}{2}$ )  $\text{Fe}^{3+}$ . The moment in the low-temperature region is, however, 4.90 B.M. ( $\theta = -10$  K), which indicates a  $S = 2$  state for iron. It is difficult to understand the origin of lower moment

of iron in terms of the structure of  $\text{La}_3\text{MFeS}_7$  because there seems to be no possibility of direct Fe–Fe interaction as the  $\text{FeS}_4$  tetrahedra are isolated.  $\text{Ba}_3\text{FeS}_5$  in which the formal valency of iron is 4+ possesses similar isolated  $\text{FeS}_4$  tetrahedra and exhibits a  $S = 2$  spin state for iron (13). Occurrence of  $S = 2$  iron in this sulfide has been explained as arising from coupling of a hole in the S:  $3p^6$  band with Fe:  $3d^5$  configuration resulting in a net spin of  $S = 2$  at iron (14). It is likely that in  $\text{La}_3\text{MgFeS}_7$  the  $\text{Fe}^{3+}: 3d^5$  state overlaps with the  $\text{S}^{2-}: 3p^6$  band states at lower temperatures resulting in creation of holes in the S( $3p$ ) band and a  $S = 2$  state for iron.  $\text{Na}_3\text{FeS}_3$  is an example of Fe(III) sulfide consisting of a pair of  $\text{FeS}_4$  tetrahedra sharing a common edge ( $\text{Fe}_2\text{S}_6$ )<sup>6-</sup> wherein iron is in the  $S = 2$  state (7).  $\text{La}_3\text{ZnFeS}_7$  gives a  $\mu_{\text{eff}}$  of 4.60 B.M. and the value does not change significantly at low temperatures. It appears that in this compound iron is present in the  $S = 2$  state even at room temperature.

 TABLE IV  
 CRYSTALLOGRAPHIC DATA OF  $\text{La}_3\text{MnFeS}_7$   
 (COMPOSITION:  $\text{La}_3\text{MnFeS}_7$ ; SPACE GROUP:  $P6_3$   
 ( $Z = 2$ ))

Atom	Position	Atomic coordinates		
		<i>x</i>	<i>y</i>	<i>z</i>
La	6c	0.361(1)	0.128(2)	0.250(3)
Fe	2b	$\frac{1}{3}$	$\frac{2}{3}$	0.164(4)
Mn	2a	0	0	0.015(8)
S(I)	6c	0.263(6)	0.091(7)	0.758(5)
S(II)	6c	0.525(8)	0.421(5)	0.523(4)
S(III)	2b	$\frac{1}{3}$	$\frac{2}{3}$	0.528(4)
Residual factor $R = 3.5\%$				
Selected bond distances (Å)				
		Fe–S		Mn–S
		2.11	2.81(3×)	
		2.22(2×)	2.77(3×)	
		2.23		
Average		2.19	2.79	

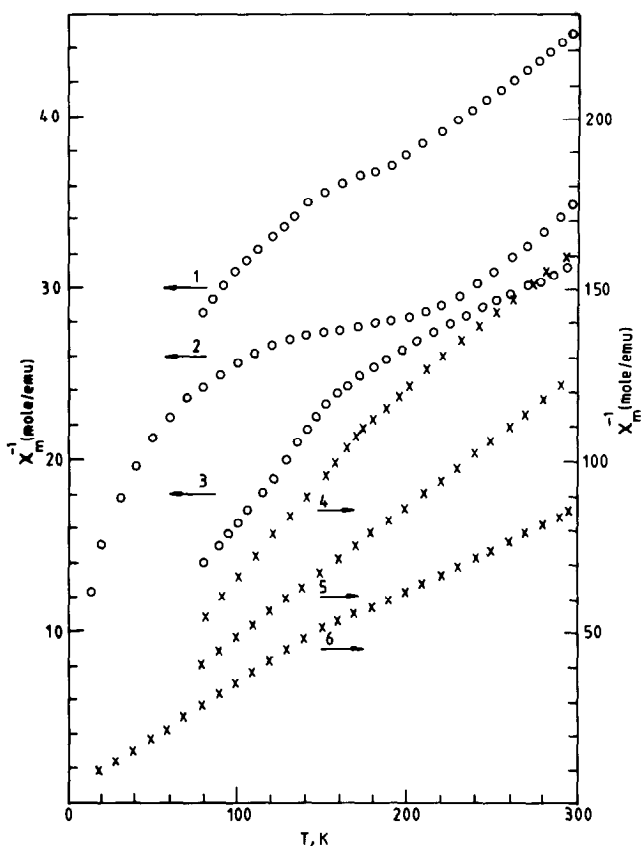


FIG. 4.  $\chi_m^{-1} - T$  plots of  $La_3MFeS_7$ : 1,  $M = Mn$ ; 2,  $M = Fe$ ; 3,  $M = Co$ ; 4,  $M = Ni$ ; 5,  $M = Zn$ ; and 6,  $M = Mg$ .

The  $\mu_{\text{eff}}$  and  $\theta$  values of the other  $La_3MFeS_7$  ( $M = Mn, Fe, Co, Ni$ ) are given in Table V. The high temperature moments are higher than the moments at low temperatures in all the cases except the nickel compound. The high temperature moments are also higher than the spin-only values calculated assuming that  $M^{2+}$  and  $Fe^{3+}$  ions are in the high spin state. It is likely that there is some interaction between the octahedral site  $M$  and tetrahedral site  $Fe$  in these phases. The magnetic behavior of  $La_3NiFeS_7$  deserves special mention. Both the high temperature and low temperature moments are lower than the spin-only moment of  $Fe^{3+}$ ; it appears that there is no contribution from  $Ni^{2+}$  to the magnetic sus-

ceptibility. This is consistent with the presence of itinerant electron conduction in this compound arising from the face-shared  $NiS_6$  octahedral chains.

$La_3MFeS_7$  sulfides are  $p$ -type semiconductors except for the nickel compound (Table V and Fig. 5). Both the activation energy for electrical conduction and the resistivity, measured using sintered pellets, decreases in the order  $Mg \sim Zn > Mn > Fe \sim Co > Ni$ ; resistivity of the nickel compound is the lowest ( $\sim 10^{-2}$  ohm-cm) and is nearly temperature-independent in the region 300–100 K. It is reasonable to assume that variation of electrical conductivity in the series arises from a variation of  $M-S-M$  covalent interaction in the octahedral

TABLE V  
MAGNETIC AND ELECTRICAL PROPERTIES OF  $\text{La}_3\text{MFeS}_7$  ( $M = \text{Mg, Mn, Fe, Co, Ni, Zn}$ )

Compound	$\mu_{\text{eff}}$ (BM)		Spin-only moment	$\theta$ (K)		$\rho$ at 300 K (ohm-cm)	$E_a$ (eV)
	Exp.			Low temp. range	High temp. range		
	Low temp. range	High temp. range		Low temp. range	High temp. range		
$\text{La}_3\text{MgFeS}_7$	4.90 (20–180 K)	5.85 (180–300 K)	5.92 ( $\text{Fe}^{2+}$ hs)	–10	–50	$3 \times 10^3$	0.30
$\text{La}_3\text{MnFeS}_7$	9.00 (80–150 K)	10.30 (180–300 K)	6.17 ( $\text{Mn}^{2+}$ ls) 8.37 ( $\text{Mn}^{2+}$ hs)	–210	–300	$5 \times 10^2$	0.15
$\text{La}_3\text{Fe}_2\text{S}_7$	6.24 (20–70 K)	10.40 (225–300 K)	5.92 ( $\text{Fe}^{2+}$ ls) 7.67 ( $\text{Fe}^{2+}$ hs)	–50	–140	$2 \times 10^2$	0.11
$\text{La}_3\text{CoFeS}_7$	7.88 (80–160 K)	12.80 (180–300 K)	6.12 ( $\text{Co}^{2+}$ ls) 8.35 ( $\text{Co}^{2+}$ hs)	–30	–300	$1 \times 10^2$	0.10
$\text{La}_3\text{NiFeS}_7$	3.65 (80–180 K)	4.44 (180–300 K)	6.38 —	–15	–100	$7 \times 10^{-2}$	—
$\text{La}_3\text{ZnFeS}_7$	—	4.60 (80–300 K)	5.92 —	—	–20	$2 \times 10^3$	0.26

chains since the discrete  $\text{FeS}_4$  tetrahedra are common to all the phases. The increasing conductivity in the Mn, Fe, Co, and Ni series with the nickel compound showing

almost itinerant conduction is consistent with increasing  $M$ – $S$ – $M$  covalent interaction across the series as it occurs in the  $MS$  and  $MS_2$  ( $M = \text{Mn, Fe, Co, Ni, Cu}$ ) sulfides (15).

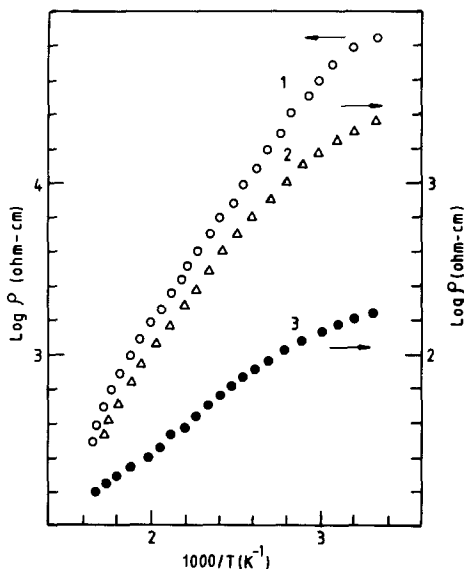


FIG. 5. Temperature dependence of electrical resistivity of some  $\text{La}_3\text{MM}'\text{S}_7$ : 1,  $\text{La}_3\text{FeAlS}_7$ ; 2,  $\text{La}_3\text{ZnFeS}_7$ ; and 3,  $\text{La}_3\text{Fe}_2\text{S}_7$ .

### Acknowledgment

Grateful thanks of the authors are due to Professor C. N. R. Rao and Professor J. B. Goodenough for valuable suggestions and discussion.

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