

BRIEF COMMUNICATIONS

Chemical Preparation and X-Ray Structure Determination of $K_{0.30}NbF_3$

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The preparation and crystal structure of $K_{0.30}NbF_3$ single crystals are described. The compound is orthorhombic with $a = 7.540(3)$, $b = 13.06(2)$, $c = 7.750(3)$ Å, space group $C22_1$. The structure refines to $R = 0.044$ with 455 reflections. The framework derives from the Magneli hexagonal tungsten bronze and is similar to the $K_{0.25}VF_3$ orthorhombic phase.

Introduction

Nb_6F_{15} is the only known niobium cluster fluoride (1). The niobium cluster chloride $K_4Nb_6Cl_{18}$ was obtained by reduction of Nb_3Cl_8 in LiCl, KCl, Nb mixtures (2) and by electrochemical reduction of a LiCl, KCl, eutectic melt containing Nb_3Cl_8 and niobium anode (3). We have tried to apply these two syntheses to prepare potassium niobium cluster fluoride compounds: first, a direct reaction between molten K_2NbF_7 and niobium metal; second, electrochemical reduction of a LiF, KF eutectic melt with a soluble niobium anode. Reactions in LiF, KF, or K_2NbF_7 molten fluorides between Nb^0 and fluorides lead to the production of bronze materials. The electrochemical reduction leads to a compound of ideal formula $KNbF_3$, probably with a small quantity of oxygen on the fluorine site: $KNbF_{3-x}O_x$.

Experimental

Pure K_2NbF_7 (0.760 g) is mixed with 0.116 g of niobium metal (puratronic JMC) and 0.012 g of graphite ($5K_2NbF_7 + 2.5 Nb^0 + 2C$) in a dry box under argon atmosphere. This mixture is evacuated and sealed in a platinum tube and then heated for a week at 850°C. Thereafter, the sample is cooled rapidly down to 500°C, then to 200°C. Two forms of crystals grew on graphite aggregates: thick blue-black crystals with a hexagonal section, and blue transparent hexagonal needles. Both correspond to the same material (verified by X-ray analysis). The needles result from the rapid growth of seeds fixed on graphite aggregates when the temperature of the sample is lowered from 850 to 500°C (Fig. 1).

An investigation with a precession camera ($MoK\alpha = 0.711$ Å) of a thick crystal (or a needle) gives a pseudo-hexagonal cell: $a =$

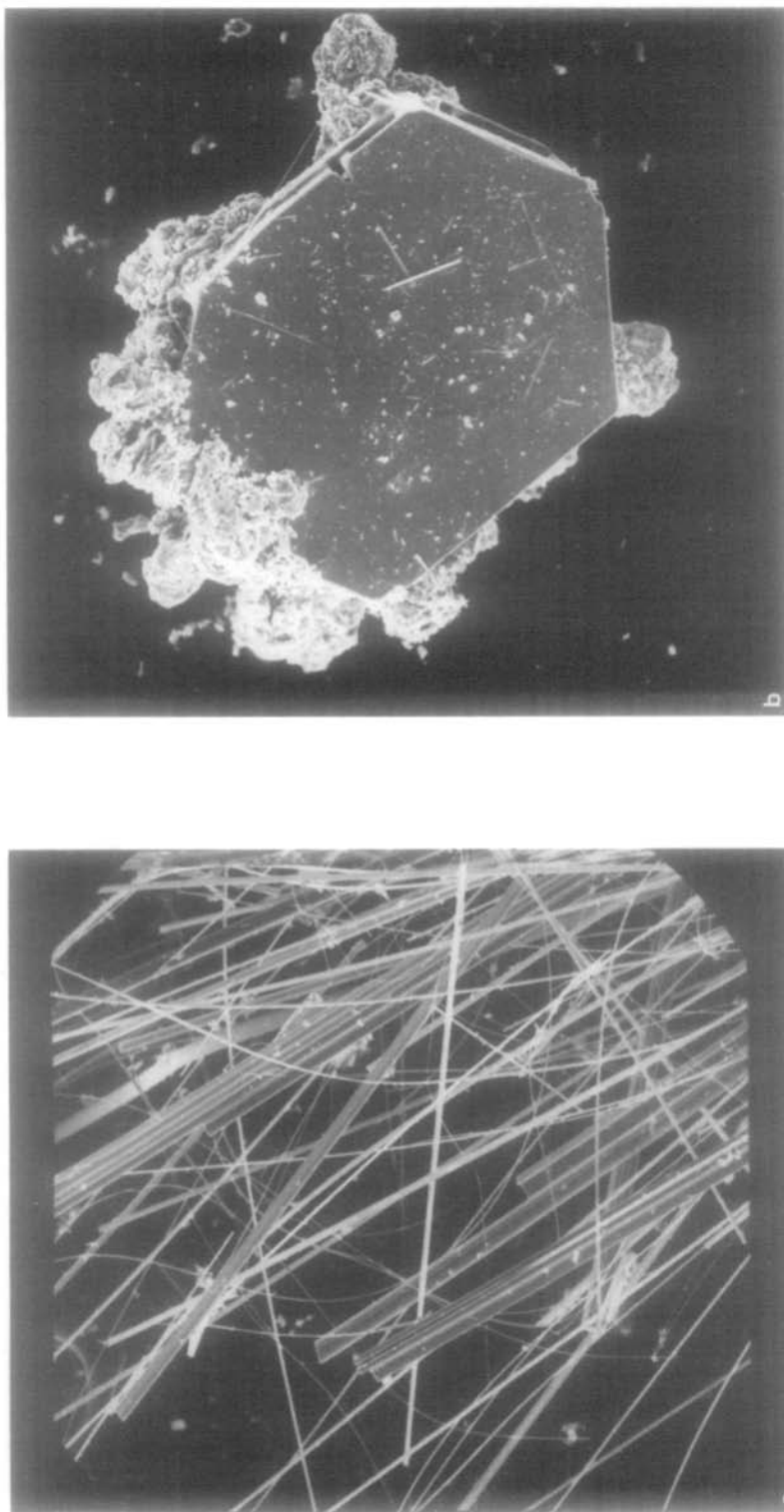


FIG. 1. Crystals of the $K_{0.30}NbF_5$ orthorhombic phase grown with needle or prismatic shape. Images obtained with a scanning electron microscope. (a) Scale: 1 cm/41.6 μm . (b) Scale: 1 cm/62.5 μm .

7.54 Å, $c = 3.87$ Å. Numerous additional weak reflections appear after long exposures, leading to an orthohexagonal cell with a doubling of c . The true cell is orthorhombic with $a = 7.540(3)$ Å, $b = 13.06(2)$ Å, $c = 7.750(3)$ Å, $V = 763.2$ Å³. The space group is $C222_1$, $Z = 12$, $d_x = 4.20$.

A crystal of prismatic shape and approximate dimensions $0.26 \times 0.16 \times 0.16$ mm³ was used for data collection. Five hundred ninety independent reflections were collected in ω scan with an automatic four-circle diffractometer, Philips PW 1100, using $\text{MoK}\alpha$ radiation (0.7107 Å). The θ range was 3 to 30°, the scan speed $0.02^\circ/\text{sec}$, and the scan width 1.40° . The background was measured during 10 sec at the starting and final position of each scan. No absorption correction was made ($\mu\text{MoK}\alpha = 48.6$ cm⁻¹).

The crystal structure was solved and refined using the SDP system (Version RSX 11M, September 1977; Enraf-Nonius) with Patterson and Fourier techniques. Full matrix refinement in the space group $C222_1$ gave $R = \Sigma|F_0 - |F_c||\Sigma F_0 = 0.044$ with 455 independent reflections, satisfying the condition $F_0^2 > 4 \sigma(F_0^2)$, $\sigma(F_0^2) =$ counting statistics. Sixteen reflections with $|F_0| - |F_c| > 25$ on a scale ranging from 0 to 1735 were also omitted. The final difference Fourier map was featureless (± 0.5 e Å⁻³). A list of observed and calculated structure factors will be sent on request (R.M.).

Results and Discussion

The formula established by this structural determination is $\text{K}_{3.64}\text{Nb}_{12}\text{F}_{36}$ or $12\text{K}_{0.30}\text{NbF}_3$. Atomic, isotropic, and anisotropic thermal parameters of $\text{K}_{0.30}\text{NbF}_3$ are given in Table I. The structure is derived from the Magneli hexagonal tungsten bronze type. The interatomic distances and angles of the two NbF_6 octahedra are given in Table II together with the K-F distances.

TABLE I
ATOMIC, ISOTROPIC,^a AND ANISOTROPIC^b THERMAL PARAMETERS OF $\text{K}_{0.30}\text{NbF}_3$

Atom	Position in $C222_1$	x	y	z	B_{eq}	β_{11}	β_{22}	β_{33}	β_{12}	β_{13}	β_{23}
Nb1	4a	-0.0038(4)	0.0000(0)	0.0000(0)	1.93(3)	0.01054(18)	0.00205(5)	0.0082(2)	0.0000(0)	0.0000(0)	-0.0008(5)
Nb2	8c	0.2515(3)	0.2518(2)	0.0013(5)	1.91(2)	0.00728(9)	0.00313(4)	0.0081(1)	-0.0024(1)	-0.0003(4)	-0.0005(4)
K	4b (3.64)	0.0000(0)	0.497(2)	0.2500(0)	16.8(9)	0.039(3)	0.0116(9)	0.14(1)	0.0000(0)	0.01(2)	0.0000(0)
F1	8c	0.5020(15)	0.2095(4)	0.473(1)	2.01(20)	0.0048(6)	0.0025(3)	0.0135(15)	-0.0051(17)	-0.012(3)	0.002(1)
F2	8c	0.6850(13)	0.3964(6)	-0.042(1)	2.17(25)	0.0137(15)	0.0025(4)	0.0071(15)	-0.0008(14)	-0.001(2)	0.001(1)
F3	8c	0.6872(7)	0.3947(4)	0.494(2)	1.08(13)	0.0022(6)	0.0009(2)	0.0088(9)	0.0016(7)	0.003(3)	-0.001(2)
F4	8c	0.7715(11)	0.2625(9)	0.256(2)	3.40(26)	0.0127(15)	0.0098(8)	0.0026(7)	-0.0015(19)	-0.005(3)	-0.005(3)
F5	4b	0.0000(0)	-0.0119(11)	0.250(0)	4.12(39)	0.0417(36)	0.0039(7)	0.0007(9)	0.0000(0)	-0.008(6)	0.000(0)

$${}^a B_{\text{eq}} = \frac{1}{3} \sum_{ij} a_i a_j \beta_{ij}$$

$${}^b T = \exp - (h^2 \beta_{11} + k^2 \beta_{22} + l^2 \beta_{33} + hk \beta_{12} + hl \beta_{13} + kl \beta_{23}). \text{ Standard deviations are in parentheses.}$$

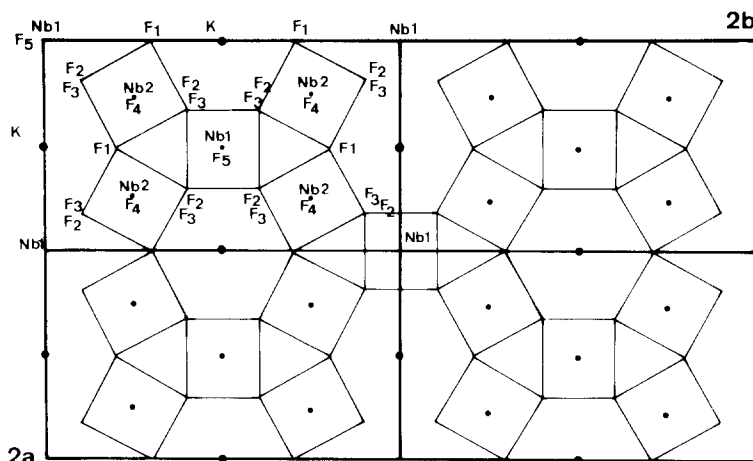


FIG. 2. Projection on the (a,b) plane of the $K_{0.30}NbF_3$ simplified structure. Four cells are drawn to illustrate the channels containing the potassium atoms.

Numerous phases of vanadium fluorobronzes have been pointed out in the literature (4–6). Orthorhombic phases of K_xVF_3 , Rb_xVF_3 , Tl_xVF_3 are described by Hong *et al.* (5). Very precise studies by high-resolu-

tion electron microscopy made by Langley *et al.* (7) and Rieck *et al.* (8, 9) provide excellent support for the comparison between K_xVF_3 , α_{11} , orthorhombic phase, and $K_{0.30}NbF_3$ and an understanding of the possible detailed structure. In Fig. 2, we can see the pseudo-hexagonal arrangement of slightly distorted NbF_6 octahedra with channels containing 3.64 potassium per cell. This model is similar to that given by Rieck *et al.* (9) for $K_{0.25}VF_3$.

TABLE II

INTERATOMIC DISTANCES (Å) IN $K_{0.30}NbF_3$:
SELECTED ANGLES (°)

Nb1–F2	1.993(10)	(×2)	F2–Nb2–F3	169.9(5)
Nb1–F3	1.953(6)	(×2)	F2–Nb2–F4	94.8(4)
Nb1–F5	1.947(1)	(×2)	F2–Nb2–F4	83.9(4)
F2–Nb1–F2	88.6(6)		F3–Nb2–F4	95.2(5)
F2–Nb1–F3	169.2(4)	(×2)	F3–Nb2–F4	86.1(5)
F2–Nb1–F3	91.9(2)	(×2)	F4–Nb2–F4	178.5(1)
F2–Nb1–F5	95.6(4)	(×2)	K–F1	3.27 (2) (×2)
F2–Nb1–F5	83.2(4)	(×2)	K–F1	3.45 (2) (×2)
F3–Nb1–F3	89.7(4)		K–F2	3.54 (1) (×2)
F3–Nb1–F5	95.2(5)	(×2)	K–F2	3.19 (1) (×2)
F3–Nb1–F5	86.0(5)	(×2)	K–F3	3.31 (1) (×2)
F5–Nb1–F5	178.3(2)		K–F3	3.39 (1) (×2)
Nb2–F1	1.992(10)		K–F4	3.52 (2) (×4)
Nb2–F1	1.952(9)		K–F5	3.778(1) (×2)
Nb2–F2	2.026(9)			
Nb2–F3	1.924(7)			
Nb2–F4	2.011(12)			
Nb2–F4	1.898(13)			
F1–Nb2–F1	178.2(2)			
F1–Nb2–F2	91.2(3)			
F1–Nb2–F3	89.2(2)			
F1–Nb2–F4	89.2(3)			
F1–Nb2–F4	90.1(3)			
F1–Nb2–F2	87.1(4)			
F1–Nb2–F3	92.5(2)			
F1–Nb2–F4	90.2(4)			
F1–Nb2–F4	90.4(4)			

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