

Structure of Sodium Tetradeuteroborate, NaBD₄

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The structure of NaBD₄ was determined from powder neutron data and refined using Rietveld's method. The compound has a sodium chloride-type structure, in space group $F\bar{4}3m$, $a = 6.137(7)$ Å, B-D, 1.160(7); angle D-B-D, 109.5(2)° at 295 K, with D atoms tetrahedrally oriented about B (at 1/2,0,0, etc.), and along all cube diagonals. This gives a random distribution of BD₄ tetrahedra in two different configurations. © 1985 Academic Press, Inc.

Introduction

Recent structural work in this laboratory has been on Cu(I)-BH₄ complexes. In reviewing BH₄ compounds the somewhat anomalous structure of NaBH₄ was found where the tetrahedral BH₄ anion crystallized at a site with octahedral symmetry. Consequently, it was decided to check this structure using neutron diffraction.

In 1947, Soldate (1) analyzed X-ray powder data for sodium tetrahydrogen borate, NaBH₄, at room temperature. The compound crystallized in a face-centered cubic cell, $a = 6.151(9)$ kX, $Z = 4$. Intensity calculations involving only sodium and boron atoms favored a sodium chloride structure with BH₄ octahedral sites (*oct*); over a zinc blende one with BH₄ in tetrahedral holes (*tet*). For model *oct*, two possible arrange-

ments were proposed for the BH₄ tetrahedra which were centered at 1/2,0,0; 0,1/2,0; 0,0,1/2; 1/2,1/2,1/2. They were the disordered model (*dis*) with the hydrogens on all eight cube diagonals which gave two randomly tetrahedral distributed BH₄⁻ groups and an ordered model (*ord*), where there was no disorder with only one set of tetrahedra, and the hydrogens on four of the cube diagonals. In his work, Soldate did not reach any conclusions about the location of the hydrogens. Abrahams and Kainajs (2) confirmed the cell, $a = 6.1635(5)$ Å at 298 K, and found a phase change at 190 K. The new phase at 78 K had a body-centered tetragonal cell, $a = 4.354(5)$ Å, $c = 5.907(5)$ Å. The λ or transition point was, according to Stockmayer and Stephenson (3), the result of an order-disorder transition involving different orientations of BH₄⁻.

However, Abrahams and Kalnajs (2) found the predicted low-temperature structure to be correct.

Peterson (4) collected single-crystal neutron data on potassium tetrahydrogen borate. His solution was based on a sodium chloride-type (*oct*) structure in *Fm3m*, with half hydrogen atoms (*dis*) along cube diagonals. This is analogous to Soldate's *oct dis* model, except a more symmetrical space group was being selected. B-H, 1.260 Å, and H-B-H, 109.50° were found.

This current work clarifies the situation using neutron powder data collected on isostructural NaBD₄. The deuteride was selected in order to avoid the large back-

ground that is produced by the large spin incoherent neutron scattering of ¹H. Naturally occurring boron containing ¹⁰B was used. To reduce the effect of absorption, the material was packed lightly into a small radius (*R_s*) tube ($\mu R_s = 0.56$ measured). This gave limited intensities, and, because of high thermal parameters, little significant data at larger 2θ angles.

NaBD₄ [Batch No. 1468431 from E. Merck, Darmstadt, West Germany; fine powder, guaranteed deuteration at 98%] (0.5 g) was tampered under dry helium into a 5-mm-diameter vanadium can, and enclosed with an indium O-ring seal. The can was rotated at the center of a horizontal

TABLE I
REFINEMENT OF VARIOUS MODELS FOR NaBD₄

Model	<i>R_p</i>	<i>R_w</i>	<i>R_B</i>	<i>G₀F</i>	Reason for rejection
<i>oct dis F43m</i>	2.6	3.4	1.5	1.8	Accepted
<i>oct dis F43m</i> Isotope	2.6	3.3	1.8	1.7	Refined H and D occupation; found H insignificant
<i>oct ord F43m</i>	3.2	4.3	1.5	2.9	Temp. factor for D too high [17(1) Å ²]
<i>oct dis Fm3m</i>	Same as <i>oct dis F43m</i>			<i>F43m</i>	
<i>oct dis F23</i>	Same as <i>oct dis F43m</i>			<i>F43m</i>	
<i>oct anion Fm3m</i>	3.3	4.4	1.6	3.1	Temp. factors too high [31(2) Å ²]
<i>tet dis F43m</i>	3.3	4.5	4.5	3.1	Refinement did not converge, temp. factors too high [18(4) Å ²]
<i>tet ord F43m</i>	3.2	4.2	4.4	2.8	Refinement did not converge, temp. factors too high [50(5) Å ²]
<i>R_p</i> , the pattern <i>R</i> factor					$= 100[\sum y_i - y_{ci} /\sum y_i]$
<i>R_{wp}</i> , weighted pattern <i>R</i> factor					$= 100[\sum w(y_i - y_{ci})^2/\sum w y_i^2]^{1/2}$
<i>R_{Bragg}</i> , Bragg <i>R</i> factor					$= 100\sum I_{ohkl} - I_{chkl} /\sum I_{ohkl}$
<i>y_i</i> , observed intensity at a 2θ value, <i>y_{ci}</i> , calculated intensity at same value					
<i>I_{hkl}</i> , intensity for a reflection <i>hkl</i>					
<i>R_w</i> , for integrated intensity					$= 100[\sum w_i(I_{oi} - I_{ci})/\sum w_i I_{oi}]$

Note. Weight used in Rietveld refinement = $\frac{1}{\sigma^2(y_i)}$ if background refined.

neutron diffractometer ($\lambda = 1.0887(5)$ Å, copper monochromator about 111) with a ^3He gas proportional counter, attached to hole 4H1 on the Australian Atomic Energy Commission's HIFAR (DIDO Type) 10-MW reactor (flux at specimen 6.7×10^9 neutrons $\text{m}^{-2} \text{sec}^{-1}$). Data were obtained using a digital electronic data acquisition over the complete spectrum.

Attempts were made to fit the data to the various models already proposed. A further arrangement, model *anion*, was considered with the BD_4^- group taken as a complete scattering entity and refined as such. Calculations on the various models are summarized in Table I. The Rietveld method (5) was used for refinement with scattering lengths for BD_4^- group, D, B, Na, and H at 32.04, 6.672, 5.35, 3.63, and -3.741 fm $^{-1}$, respectively. Initially, Soldate's original model was refined. These refined parameters were then used to test all other models. Refinement was continued for up to 15 cy-

cles after which no significant further refinement was possible. Data had been collected from 5.0 to 110.0° in 2θ , but the only significant intensities were from 15.10 to 21.70° and 27.20 to 52.90°. In profile refinement, absorption correction was incorporated into the Wiles and Young program (5) using the algorithm of Rouse *et al.* (6). Background was interpolated between 12 fixed points (F_B) or refined (R_B) using a three-parameter algorithm [$b_0 + b_1(2\theta) + b_{-1}(2\theta)^{-1}$].

In the Wiles and Young program, the positions used were Na(0,0,0); B(either *oct* 1/2,1/2,1/2 or *tet* 1/4,1/4,1/4), and D(x,x,x and/not $-x,-x,-x$) with relative occupancy factor to give total atom ratios of 1:1:4. All x parameters were constrained to move together. In the case of two related D atoms, their temperature factors were also constrained to move together. Refinement for the model [*oct dis F23*] in noncentrosymmetric space group *F23*, gave a simi-

TABLE II
DATA FOR *oct dis F43m* MODEL

(<i>hkl</i>)	2θ	Integrated observed intensity	Mult	Integrated calculated intensity ^a
111	17.683(3)	38,991 ± 259	8	38,700
002	20.456(4)	23,823 ± 230	6	22,939
022	29.093(19)	9,115 ± 160	12	10,819
113	34.21(1)	4,198 ± 150	24	3,951
222	35.78(1)	3,504 ± 174	8	3,845
004	41.55(3)	2,260 ± 128	6	2,447
133	45.44(4)	1,726 ± 170	24	1,436
024	46.73(3)	1,915 ± 150	24	2,194
224	51.524(13)	5,268 ± 160	24	3,868

$\lambda = 1.0887(5)$ Å; $\mu R_s = 0.56$; $a = 6.137(7)$ Å includes error in λ

Atomic parameters				
Atom	x/a	y/b	z/c	Å ²
Na	0.0	0.0	0.0	1.8(2)
B	0.5	0.5	0.5	1.8(2)
D(1) ^b	0.3901(6)	0.3901(6)	0.3901(6)	4.1(3)
D(2) ^b	0.6099(6)	0.6099(6)	0.6099(6)	4.1(3)

^a Estimated using F_{obs} and F_{calc} in LINUS refinement.

^b Half weight.

lar result to *oct dis F43m*. This had deuterium atoms tetrahedrally orientated about B, and along the eight cube diagonals, to produce a random distribution of tetrahedra in two different static configurations, *F43m*. For all models, the halfwidth parameters were determined independently from single Gaussian fits and then fixed. In order to verify the results for the original model, the integrated intensities were refined using the full matrix least-squares program LINUS (7). The calculations were in excellent agreement with the fixed background results, $R = 0.083$, $R_w = 0.075$, $N_o = 9$, $N_v = 2$.

Tables I and II show that model *oct dis F43m* fits the data. Other unrelated models did not successfully refine.

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