

## Synthesis and Structures of $\text{Cs}_3\text{TaSe}_4$ and $\text{Cs}_3\text{NbSe}_4$

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A study of the alkali metal tetraselenometalate system has afforded two new isostructural phases,  $\text{Cs}_3\text{TaSe}_4$  and  $\text{Cs}_3\text{NbSe}_4$ . The structures of these phases have been determined by X-ray and neutron powder diffraction techniques.  $\text{Cs}_3\text{TaSe}_4$  crystallizes in the orthorhombic system ( $D_{2h}^6\text{-Pnma}$ ,  $a = 10.277(1)$ ,  $b = 11.918(1)$ ,  $c = 10.238(1)$  Å) with four formula units in the cell. The structure contains discrete  $\text{TaSe}_4^{3-}$  ions (Ta-Se, 2.369(4), 2.374(7), 2.397(6) Å) and  $\text{Cs}^+$  cations. Out to 4 Å the Cs cations are surrounded by five or seven Se atoms. © 1988 Academic Press, Inc.

### Introduction

The tetrachalcogenometalates ( $\text{MQ}_4^{3-}$ ,  $M = \text{V, Nb, or Ta}$ ,  $Q = \text{S or Se}$ ) have been prepared either in the solid state by direct combination of the constituent elements or for  $\text{VS}_4^{3-}$  by the reaction of  $\text{H}_2\text{S}$  with an aqueous solution of the corresponding oxometalate (1). When the oxometalates in question do not exist in solution, only the solid-state reactions have been effective.  $[\text{NH}_4]_3[\text{VS}_4]$  has been prepared from the reaction of  $\text{H}_2\text{S}$  with  $\text{NH}_4\text{VO}_3$  in aqueous  $\text{NH}_4\text{OH}$  solution (2). However, the synthesis of  $\text{NbQ}_4^{3-}$  and  $\text{TaQ}_4^{3-}$  ( $Q = \text{S or Se}$ ) in solution has not been reported. The corresponding oxometalates do not exist even though the chalcogenometalates have been prepared from solid-state reactions (i.e.,  $\text{Tl}_3\text{MQ}_4$  (3),  $M = \text{V, Nb, Ta}$ ;  $Q = \text{S, Se}$  and  $\text{Cu}_3\text{MQ}_4$  (4),  $M = \text{V, Nb, Ta}$ ;  $Q = \text{S, Se, Te}$ ). Attempts to prepare  $\text{MS}_4^{3-}$  ( $M = \text{Nb, Ta}$ ) from  $M(\text{OEt})_5$  and  $(\text{Me}_3\text{Si})_2\text{S}$  in acetonitrile solution did not lead to the desired products but to the polynuclear cage anions  $\text{M}_6\text{S}_{17}^{4-}$  (5).

With the aim of preparing potentially soluble  $\text{MQ}_4^{3-}$  salts we have devised a route for the preparation of  $\text{Cs}_3\text{NbSe}_4$  and  $\text{Cs}_3\text{TaSe}_4$  by solid-state reactions. Here we report the synthesis of these new compounds as well as their structures as determined from neutron diffraction powder data.

### Experimental Section

**Synthesis.** The approximate 8 g of material used in the neutron diffraction experiment was composed of nine separate samples prepared by an identical route. Ta powder (0.181 g (1.00 mmol); 99.98%, AESAR) and Se powder (0.316 g (4.00 mmol); 99.99%, ALFA) were placed in a fused silica tube that had been flame dried under vacuum. Inside a glove box under an Ar atmosphere, 0.399 g (3.00 mmol) of liquid Cs (99.5% AESAR) was added very slowly by pipet to the Ta/Se mixture. (Caution! This reaction is very exothermic.) The tube was then sealed under vacuum ( $\sim 10^{-4}$  Torr). It was next heated in a furnace at 800°C for 8 days to afford  $\text{Cs}_3\text{TaSe}_4$  as a

yellow-green powder (about 95% yield). The color of the product implied the existence of the  $\text{TaSe}_4^{3-}$  ion (3). A chemical analysis (Dornis und Kolbe, Mülheim, FRG) corresponded to the stoichiometry  $\text{Cs}_{3.03}\text{Ta}_{1.00}\text{Se}_{4.06}$  and hence to  $\text{Cs}_3\text{TaSe}_4$ . X-ray powder diffraction patterns were recorded for each sample with  $\text{CuK}\alpha$  radiation on an Enraf-Nonius Model FR552 Guinier camera. The patterns are consistent except for very weak extra peaks in two of the samples. From a comparison of line positions and relative intensities, the nine samples were judged to be essentially identical and thus were combined.

$\text{Cs}_3\text{NbSe}_4$  was prepared by an identical procedure and the red-brown color of the product implied the existence of the  $\text{NbSe}_4^{3-}$  ion (3). The X-ray powder pattern of  $\text{Cs}_3\text{NbSe}_4$  has similar peak positions but different relative intensities compared to the  $\text{Cs}_3\text{TaSe}_4$  powder pattern.

*Structural study.* At the first stage of the structural investigation, the X-ray powder diffraction pattern was indexed on a tetragonal system with the program VISSER (6). Weighted least-squares analysis (7) of 30 reflections in the range  $10^\circ < 2\theta < 50^\circ$  yielded unit cell parameters,  $a = 10.262(3)$ ,  $c = 11.892(7)$  Å. However, the X-ray powder pattern contains additional lines that are very close to the indexed lines. This implies the possibility of lower symmetry with similar cell constants. The possibly isostructural compound  $\text{K}_3\text{VS}_4$  (8) crystallizes in space group  $Pnma$  of the orthorhombic system with  $a = 9.143(4)$ ,  $b = 10.59(1)$ ,  $c = 9.107(4)$  Å. In  $\text{K}_3\text{VS}_4$ ,  $a/b$  and  $c/b$  are 0.863 and 0.860, respectively. As a trial, we used these ratios to assign the cell parameters  $a = 10.26$  Å and  $c = 10.23$  Å in comparison to the longest axis  $b = 11.892$  Å in  $\text{Cs}_3\text{TaSe}_4$  and used as a starting point the positional parameters for  $\text{K}_3\text{VS}_4$ . The simulated X-ray diffraction pattern thus generated by the program LAZY PULVERIX (9) was in good agreement with the Guinier pattern.

Neutron diffraction data for  $\text{Cs}_3\text{TaSe}_4$  powder were collected at room temperature on the general purpose powder diffractometer (GPPD) of the IPNS facility at Argonne National Laboratory (10). A general discussion of techniques involved in acquisition and analysis of data on this facility was given previously (11). Data were collected from the  $2\theta = 148^\circ$ ,  $90^\circ$ , and  $60^\circ$  detector banks, but only back scattering ( $2\theta = 148^\circ$ ) data were used in the profile refinement so as to achieve the highest possible resolution (12). Data reduction and least-squares refinement were carried out with the use of the programs of the IPNS Rietveld analysis package (13). In addition to the structural variables, a scale factor, five background parameters, and one of the peak broadening parameters ( $\sigma_1$ ) were refined. After several cycles of successful refinement, the difference curve showed a significantly wavy background, probably from amorphous scattering of one or more impurity phases in the sample. Since the function used to calculate background in the IPNS Rietveld code is monotonic and slowly varying, it is incapable of handling such residual scattering. Richardson and Faber (14) developed a Fourier-filtering technique that facilitates the removal from the measured data of amorphous scattering contributions and yields information on such impurities. The residual scattering from  $\text{Cs}_3\text{TaSe}_4$  that was not accounted for by the Rietveld refinement was Fourier transformed to generate a real-space correlation function similar to a radial distribution function. The identifiable interatomic spacings corresponded to 2.54 Å (Ta-Se?) and 3.81 Å (Cs-Se?). Thus the amorphous impurities could be binary tantalum and cesium selenides. From the inverse Fourier transformation of the correlation function with use of a value of  $r_{\text{max}} = 10.0$  Å, a Fourier-filtered fit of the residual amorphous scattering was calculated and then subtracted from the observed intensity data. These resulting data were used in the final least-squares refinement. This final re-

$\text{Cs}_3\text{TaSe}_4$  (GPPD,  $2\theta=148^\circ$ ): Orthorhombic  $\text{Pnma}$

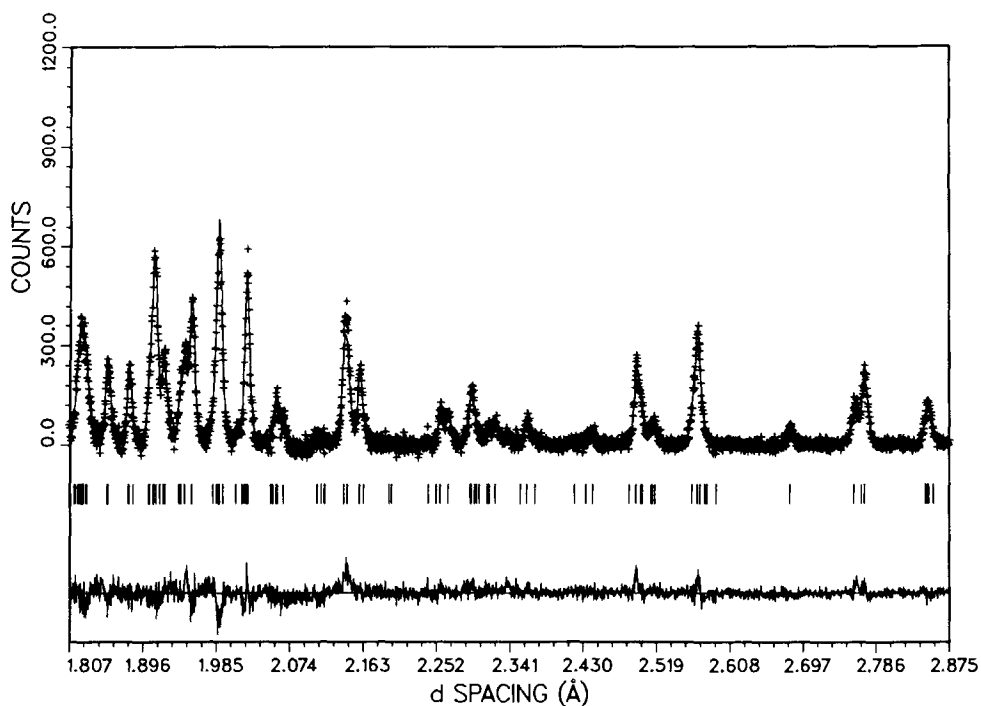
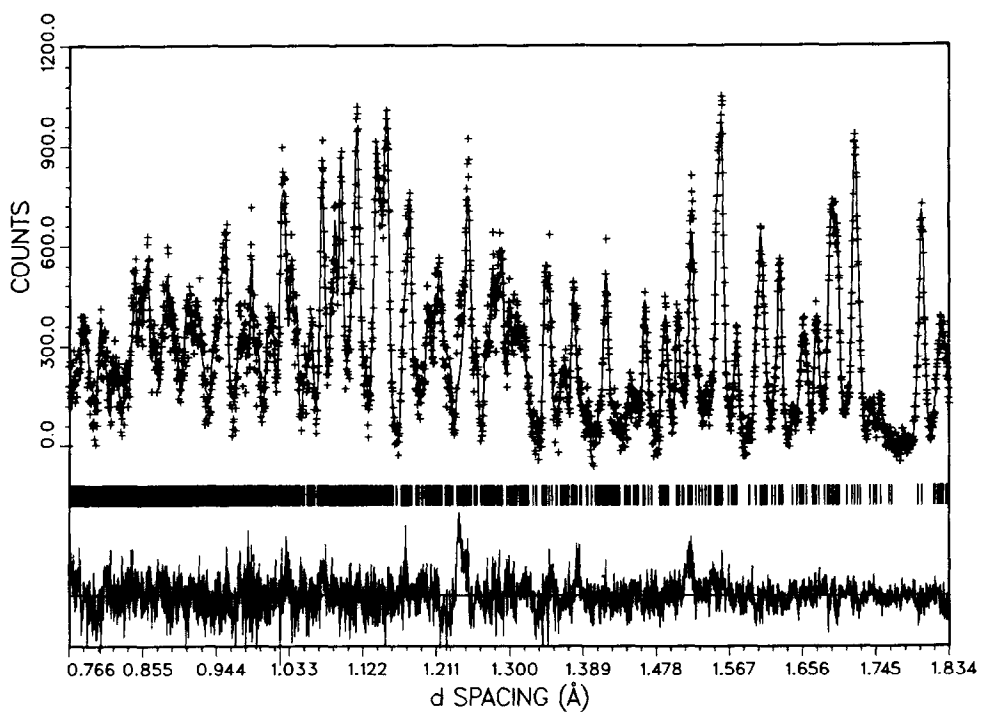


FIG. 1. Final Rietveld profile fit for  $\text{Cs}_3\text{TaSe}_4$  after Fourier filtering. +, Data point; —, calculated profile; vertical lines below the profile indicate the positions of all allowed reflections. A difference plot appears at the bottom of each frame.

TABLE I  
 POSITIONAL PARAMETERS<sup>a</sup> FOR Cs<sub>3</sub>TaSe<sub>4</sub>

Atom	Wyckoff notation	Site symmetry	x	y	z	<i>B</i> <sub>eq</sub> (Å <sup>2</sup> ) <sup>b</sup>
Cs(1)	4c	m	0.1487(7)	1/4	0.4063(7)	3.6(3)
Cs(2)	8d	1	0.0425(5)	0.9599(4)	0.2095(5)	2.0(1)
Ta	4c	m	0.2212(4)	1/4	0.9823(5)	0.9(1)
Se(1)	4c	m	0.0054(4)	1/4	0.0711(5)	1.3(1)
Se(2)	4c	m	0.2051(5)	1/4	0.7510(4)	2.0(3)
Se(3)	8d	1	0.3319(3)	0.0851(3)	0.0483(3)	2.5(1)

<sup>a</sup> Refined cell parameters are  $a = 10.2772(2)$  Å,  $b = 11.9181(3)$  Å,  $c = 10.2380(2)$  Å, which we round to the nearest thousandth.

$${}^b B_{eq} = (8\pi^2/3) \sum_i \sum_j U_{ij} a_i^* a_j^* \cdot a_j$$

finement included 4399 data points (representing 5-μsec-wide time channels) and 1492 independent reflections ( $0.7650$  Å <  $d$  <  $2.8716$  Å). The 52 parameters refined included a scale factor, cell parameters, atomic positional parameters, anisotropic thermal parameters, background parameters, and one of the peak broadening parameters ( $\sigma_1$ ). The final agreement indices ( $I_0$ ) are  $R_p = 0.026$ ,  $R_{wp} = 0.038$ , and  $R_B = 0.085$ . The statistically expected value of  $R_E$  is 0.030. Figure 1 shows the final Rietveld profile fit for Cs<sub>3</sub>TaSe<sub>4</sub>. Atomic parameters and anisotropic thermal parameters are given in Table I and Table II, respectively.

Although the Fourier-filtering technique has reduced the contribution from amor-

phous scattering, there are still a few small peaks (e.g.,  $d = 2.337$  Å) that are not explained by the current cell. But such impurity peaks are of low intensity, and overall the agreement between the observed and calculated pattern of Cs<sub>3</sub>TaSe<sub>4</sub> is very satisfactory.

The X-ray diffraction pattern of Cs<sub>3</sub>NbSe<sub>4</sub> is in very good agreement with the pattern simulated from the Cs<sub>3</sub>TaSe<sub>4</sub> structure with Nb at the Ta position. We thus conclude that Cs<sub>3</sub>NbSe<sub>4</sub> is isostructural with Cs<sub>3</sub>TaSe<sub>4</sub>.

### Discussion

The structure of Cs<sub>3</sub>TaSe<sub>4</sub> consists of the packing of discrete Cs<sup>+</sup> and TaSe<sub>4</sub><sup>3-</sup> ions. A

 TABLE II  
 ANISOTROPIC THERMAL PARAMETERS FOR Cs<sub>3</sub>TaSe<sub>4</sub><sup>a</sup>

Atom	<i>U</i> <sub>11</sub>	<i>U</i> <sub>22</sub>	<i>U</i> <sub>33</sub>	<i>U</i> <sub>12</sub>	<i>U</i> <sub>13</sub>	<i>U</i> <sub>23</sub>
Cs(1)	0.0119(44)	0.1066(76)	0.0181(50)	0	0.0110(31)	0
Cs(2)	0.0242(31)	0.0260(30)	0.0260(34)	0.0064(27)	-0.0085(25)	0.0098(22)
Ta	0.0047(26)	0.0248(30)	0.0055(30)	0	-0.0042(20)	0
Se(1)	0.0064(24)	0.0212(29)	0.0217(29)	0	0.0085(22)	0
Se(2)	0.0163(27)	0.0417(38)	0.0183(30)	0	-0.0065(22)	0
Se(3)	0.0226(21)	0.0427(23)	0.0293(22)	0.0035(19)	0.0014(19)	0.0109(18)

<sup>a</sup> ORTEP II, Type 8.

view of the unit cell of the compound is given in Fig. 2. Distances and angles are given in Table III. The TaSe<sub>4</sub><sup>3-</sup> ion has crystallographically imposed *m* symmetry but shows approximate *43m* molecular symmetry. Tetrahedrally coordinated Ta is not usual. The distances between Ta and Se atoms range from 2.369(4) to 2.397(6) Å. These distances are comparable with those in the solid-state compounds Tl<sub>3</sub>TaSe<sub>4</sub> (3) and Cu<sub>3</sub>TaSe<sub>4</sub> (4) (2.305–2.456 Å). However, as expected, they are shorter than in Ta<sub>2</sub>NiSe<sub>5</sub> (15) (2.523(2) to 2.678(2) Å) where the Ta atom is octahedrally coordinated and where there is no discrete Ta entity. The average Ta–Se distance of 2.377 Å in Cs<sub>3</sub>TaSe<sub>4</sub> is comparable to the average W–Se distance (2.317 Å) in [NH<sub>4</sub>]<sub>2</sub>[WSe<sub>4</sub>] (16) when account is taken of the differing covalent radii of Ta and W. The Se–Ta–Se angles in Cs<sub>3</sub>TaSe<sub>4</sub> range from 108.3(2)° to 112.1(2)°, close to the tetrahedral value.

If one takes a cutoff of 4 Å then atom Cs(1) is coordinated to five Se atoms while atom Cs(2) is coordinated to seven. These Cs(1)Se<sub>5</sub> and Cs(2)Se<sub>7</sub> polyhedra are not easily described in terms of classic polyhedra. The Cs–Se distances vary from 3.576(8) to 3.841(6) Å and these distances

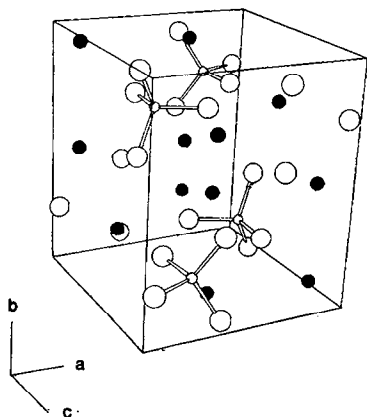


FIG. 2. View of the unit cell of Cs<sub>3</sub>TaSe<sub>4</sub>. Small open circles are Ta atoms, small filled circles are Cs atoms, and large open circles are Se atoms.

TABLE III  
SELECTED DISTANCES (Å) AND ANGLES (DEG.)  
FOR Cs<sub>3</sub>TaSe<sub>4</sub>

Ta–2Se(3)	2.369(4)	Cs(2)–Se(2)	3.629(6)
Ta–Se(2)	2.374(7)	Cs(2)–Se(3)	3.714(6)
Ta–Se(1)	2.397(6)	Cs(2)–Se(3)	3.739(6)
Cs(1)–Se(2)	3.576(8)	Cs(2)–Se(1)	3.756(5)
Cs(1)–Se(1)	3.674(7)	Cs(2)–Se(1)	3.841(6)
Cs(1)–Se(1)	3.735(9)	Se(3)–Ta–Se(3)	112.12(23)
Cs(1)–2Se(3)	3.831(7)	2Se(3)–Ta–Se(2)	108.53(17)
Cs(2)–Se(2)	3.592(6)	2Se(3)–Ta–Se(1)	109.64(17)
Cs(2)–Se(3)	3.614(5)	Se(2)–Ta–Se(1)	108.28(24)

are consistent with those observed in other cesium selenide structures (i.e., 3.545(7) to 3.930(8) Å in Cs<sub>2</sub>Se<sub>5</sub> (17)). Although the TaSe<sub>4</sub><sup>3-</sup> anions have been found in Cu<sub>3</sub>TaSe<sub>4</sub> and Tl<sub>3</sub>TaSe<sub>4</sub> there appear to be interactions among the cations and selenium atoms in these materials. Thus, the distances of 2.41 Å for Cu···Se (4) and 3.22 Å for Tl···Se (3) are significantly shorter than the sums of the ionic radii (18), 2.58 Å for Cu–Se and 3.57 Å for Tl–Se. Attempts to solubilize the anion in Cu<sub>3</sub>TaSe<sub>4</sub> have not been successful (19). In Cs<sub>3</sub>TaSe<sub>4</sub>, the average Cs–Se distance, 3.71 Å, is slightly longer than the sum of ionic radii (3.65 Å) (18).

Prior to the present structural determination an attempt was made to prepare other TaSe<sub>4</sub><sup>3-</sup> salts by solution methods. A sample of Cs<sub>3</sub>TaSe<sub>4</sub> was dissolved at room temperature in acetonitrile in the presence of 18-crown-6 ether and bis(triphenylphosphine)iminium chloride. This yielded not salts of TaSe<sub>4</sub><sup>3-</sup> but rather new salts of the unliganded Se<sub>4</sub><sup>2-</sup> and Se<sub>5</sub><sup>2-</sup> ions (20). Now that the presence of the TaSe<sub>4</sub><sup>3-</sup> ions in Cs<sub>3</sub>TaSe<sub>4</sub> is established, further attempts at dissolution to yield new salts of this ion are underway.

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