

## Effect of Rare Earth Ions on the Phase Transition of Na<sub>2</sub>SO<sub>4</sub> Crystals

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The  $V \rightleftharpoons I$  phase transition of Na<sub>2</sub>SO<sub>4</sub> crystals was investigated on a sample of pure Na<sub>2</sub>SO<sub>4</sub> and on rare-earth-ion ( $Ln^{3+} = La^{3+}, Eu^{3+}, Tm^{3+}$ )-doped Na<sub>2</sub>SO<sub>4</sub> samples in various ambient gases (O<sub>2</sub>, N<sub>2</sub>, NH<sub>3</sub>) with high temperature X-ray diffraction and differential thermal analysis. On heating in N<sub>2</sub> flow, the initiating temperature for the  $V \rightarrow I$  transition was lowered by doping with  $Ln^{3+}$  ion and the doping effect was enhanced by an increase in the ionic size ratio  $r_{Ln^{3+}}/r_{Na^+}$ . The low temperature form of the solid solution (LSS) Na<sub>2</sub>SO<sub>4</sub> and rare earth sulfate, which was a by-product in the preparation of the  $Ln^{3+}$ -doped samples, transformed to a high temperature form (HSS) after the  $V \rightarrow I$  transition, and the initiating temperature for the LSS  $\rightarrow$  HSS transition was highest in the Eu<sup>3+</sup>-doped sample ( $r_{Ln^{3+}}/r_{Na^+} \approx 1$ ). On the other hand, on cooling in N<sub>2</sub> flow, the HSS was stable until room temperature in the La<sup>3+</sup>- or Eu<sup>3+</sup>-doped sample ( $r_{Ln^{3+}}/r_{Na^+} \geq 1$ ), but transformed to LSS in the Tm<sup>3+</sup>-doped sample ( $r_{Ln^{3+}}/r_{Na^+} < 1$ ). The results obtained in other ambient gas flows (O<sub>2</sub>, NH<sub>3</sub>) were considerably affected by the sorption of ambient gas into Na<sub>2</sub>SO<sub>4</sub> crystals and/or the reaction between ambient gas and Na<sub>2</sub>SO<sub>4</sub> and were different from those obtained in N<sub>2</sub> flow. © 1991 Academic Press, Inc.

Na<sub>2</sub>SO<sub>4</sub> crystals are known to exhibit five polymorphisms (phases I, II, III, IV, and V). Many workers (1-4) have found that the phase transition is affected by the coexistence of a foreign sulfate and/or an ambient gas. Eysel *et al.* (1) found that divalent ion-doped Na<sub>2</sub>SO<sub>4</sub> was stable in phase I even at room temperature. Saito *et al.* (2) reported that the III  $\rightarrow$  I transition of Y-doped Na<sub>2</sub>SO<sub>4</sub> depended on the amount doped of Y<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>. In alkali metal ion- or alkaline earth ion-doped Na<sub>2</sub>SO<sub>4</sub>, we (3, 4) have found that the  $V \rightarrow I$  and  $I \rightarrow III$  transitions depend on the crystal structure of the dopant or product (compound or solid solution) made from Na<sub>2</sub>SO<sub>4</sub> but the III  $\rightarrow$  V transition is mainly affected by the ambient gas (especially water vapor).

In this study, the effect of ambient gas on

the  $V \rightleftharpoons I$  transition of Na<sub>2</sub>SO<sub>4</sub> crystals for a pure Na<sub>2</sub>SO<sub>4</sub> sample and rare earth ion ( $Ln^{3+}$ )-doped Na<sub>2</sub>SO<sub>4</sub> samples was investigated with the high temperature X-ray diffraction (high temp-XRD) and differential thermal analysis (DTA).

### Experimental

The pure and doped Na<sub>2</sub>SO<sub>4</sub> samples used were prepared by evaporating to dryness aqueous solutions of Na<sub>2</sub>SO<sub>4</sub> to which were added given amounts (0 to 20 mol%) of Ln<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> ( $Ln = La, Eu, Tm$ ). This was followed by vacuum drying at 130°C for 2 hr. N<sub>2</sub> and O<sub>2</sub> were obtained by evaporation of pure liquid of N<sub>2</sub> and O<sub>2</sub>, respectively, followed by a further purification by passage through a glass tube containing molecular

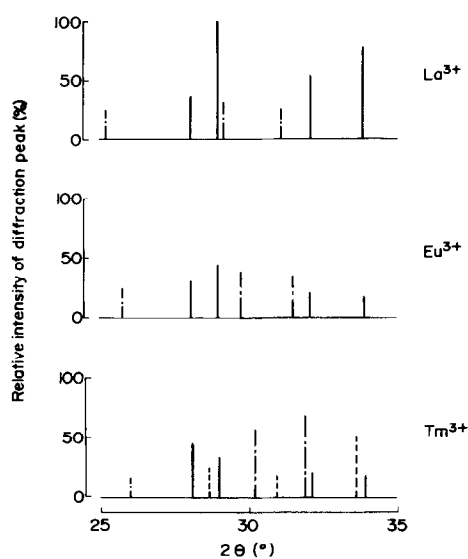


FIG. 1. The XRD pattern of  $\text{Ln}^{3+}$  (10 mol%)-doped  $\text{Na}_2\text{SO}_4$  samples before heating; (—) phase V of  $\text{Na}_2\text{SO}_4$ , (---)  $\text{Ln}_2(\text{SO}_4)_3$ , (-·-) LSS.

sieve 5A at 0°C.  $\text{NH}_3$  was obtained by drying a cylinder of highly pure  $\text{NH}_3$  through a soda lime tube at 25°C.

The progress of the phase transition reaction was measured by an apparatus which enabled a simultaneous measurement of high temp-XRD and DTA (Rigaku Denki Company, Limited: Geigerflex Type 2035) as described in the previous paper (3). The temperature was varied at the rate 5°C/min.

## Results

Figure 1 shows the XRD patterns of the  $\text{Na}_2\text{SO}_4$  samples doped with 10 mol% of  $\text{Ln}_2(\text{SO}_4)_3$  before heating. All samples consisted of the phase V of  $\text{Na}_2\text{SO}_4$  and the low temperature form of solid solution (LSS) between  $\text{Na}_2\text{SO}_4$  and rare earth sulfate. The  $\text{Tm}^{3+}$ -doped  $\text{Na}_2\text{SO}_4$  sample alone, furthermore, contained  $\text{Tm}_2(\text{SO}_4)_3$  crystals. The diffraction peak height of  $\text{Na}_2\text{SO}_4$  crystals in the  $\text{La}^{3+}$ -doped  $\text{Na}_2\text{SO}_4$  sample was much higher than that in the  $\text{Eu}^{3+}$ - or  $\text{Tm}^{3+}$ -doped  $\text{Na}_2\text{SO}_4$  sample.

The V → I transition of  $\text{Na}_2\text{SO}_4$  was mea-

sured on the  $\text{La}^{3+}$ -doped  $\text{Na}_2\text{SO}_4$  sample in  $\text{N}_2$  flow by means of the high temp-XRD and DTA (Fig. 2). The diffraction peak of the phase V became low, while that of the phase I grew at 188°C with a simultaneous endothermic effect on the DTA curve (A in Fig. 2). Then, the diffraction peaks of both phase I and the LSS began to decrease, while that of the high temperature form of solid solution (HSS), the crystal system of which is similar to that of phase I  $\text{Na}_2\text{SO}_4$ , began increasing at 222°C with a simultaneous endothermic effect on the DTA curve (B in Fig. 2).

Similar examinations were carried out on a pure  $\text{Na}_2\text{SO}_4$  sample and the  $\text{La}^{3+}$ -,  $\text{Eu}^{3+}$ -, or  $\text{Tm}^{3+}$ -doped  $\text{Na}_2\text{SO}_4$  samples in a gas flow of  $\text{N}_2$ ,  $\text{O}_2$ , or  $\text{NH}_3$ . Figure 3 shows the effects of doping with  $\text{Ln}^{3+}$  ion and ambient gas on the initiation temperature for the transitions of both the V → I of  $\text{Na}_2\text{SO}_4$  crystals and the LSS → HSS. The initiating temperature for the V → I transition was lowered with an increase in the cation size of dopant in any ambient gas, and the initiating temperature in  $\text{N}_2$  flow was remarkably

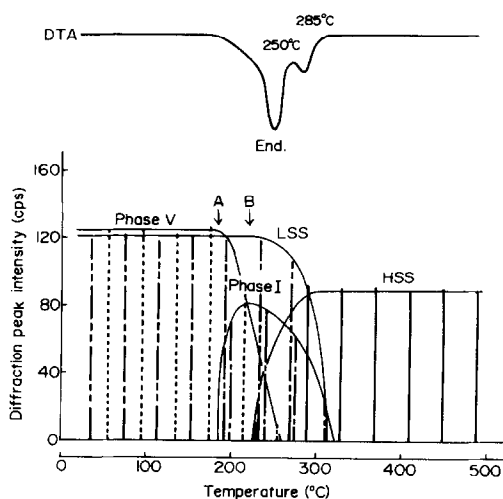


FIG. 2. High temp-XRD and DTA traces on the  $\text{La}^{3+}$  (10 mol%)-doped  $\text{Na}_2\text{SO}_4$  sample in  $\text{N}_2$  flow at heating rate of 5°C/min, (A) Initiating temperature of the V → I transition, (B) Initiating temperature of the LSS → HSS transition.

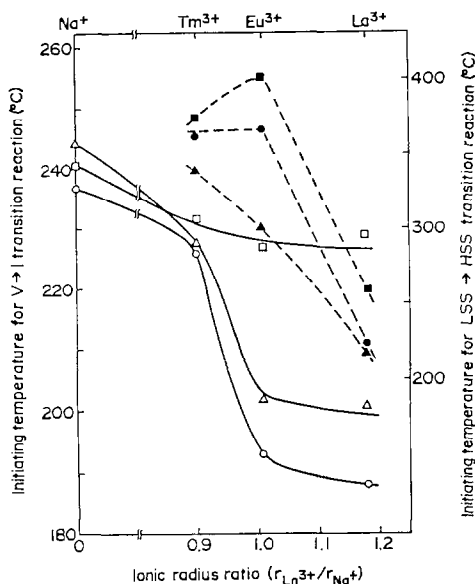


FIG. 3. The V → I transition (solid line) of Na<sub>2</sub>SO<sub>4</sub> crystals and the LSS → HSS transition (dotted line) of pure and Ln<sup>3+</sup> (10 mol%)-doped Na<sub>2</sub>SO<sub>4</sub> samples, in (○,●) N<sub>2</sub>, (△,▲) O<sub>2</sub>, (□,■) NH<sub>3</sub> (flow rate: 100 ml/min).

lower in any sample than that in O<sub>2</sub> flow or NH<sub>3</sub> flow. The initiating temperature for the LSS → HSS transition in N<sub>2</sub> flow became lower in the order Eu<sup>3+</sup>-doped ≈ Tm<sup>3+</sup>-doped ≫ La<sup>3+</sup>-doped, and it was considerably higher or lower than that in O<sub>2</sub> or NH<sub>3</sub> flow, respectively, in any Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample.

Figure 4 shows the effect of amount doped with Ln<sup>3+</sup> on the V → I transition for the Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples in N<sub>2</sub> flow. The transition was further promoted with an increase in the doping amount of La<sup>3+</sup> or Eu<sup>3+</sup> ( $r_{Ln^{3+}}/r_{Na^+} \geq 1$ ), but it was hardly affected by doping with Tm<sup>3+</sup> ( $r_{Ln^{3+}}/r_{Na^+} < 1$ ).

On cooling in the case of the pure Na<sub>2</sub>SO<sub>4</sub> sample, the initiating temperature for the I → III transition became lower in the order N<sub>2</sub> flow > NH<sub>3</sub> flow > O<sub>2</sub> flow, contrary to the V → I transition. However, no III → V transition was observed in any ambient gas. On the other hand, in the case of the Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples, the HSS produced on heating in any ambient gas was stable

until room temperature in the La<sup>3+</sup>- or Eu<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample but it transformed to the LSS in the Tm<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample.

**Discussion**

The crystallographic patterns of the Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples before heating (Fig. 1) are thought to be closely related to the solubilities of Na<sub>2</sub>SO<sub>4</sub>, Ln<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>, and LSS in water under the preparation conditions of Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples.

The solubilities of Ln<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> and LSS are known to be much less than that of Na<sub>2</sub>SO<sub>4</sub> (5). It is also known that the solubility of Ln<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> decreases in order La<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> > Eu<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> ≫ Tm<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> and the solubility of LSS decreases in the order (Tm,Na)SO<sub>4</sub> > (Eu,Na)SO<sub>4</sub> ≫ (La,Na)SO<sub>4</sub> (5). From these facts and the results shown in Fig. 1, Na<sub>2</sub>SO<sub>4</sub> would precipitate on the produced (La,Na)SO<sub>4</sub> particles in an La<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample, and it would coprecipitate with (Eu,Na)SO<sub>4</sub> on (Eu,Na)SO<sub>4</sub> particles which had precipitated partially in the Eu<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples. However in a Tm<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample, Na<sub>2</sub>SO<sub>4</sub> and (Tm,Na)SO<sub>4</sub> would coprecipitate on Tm<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> particles.

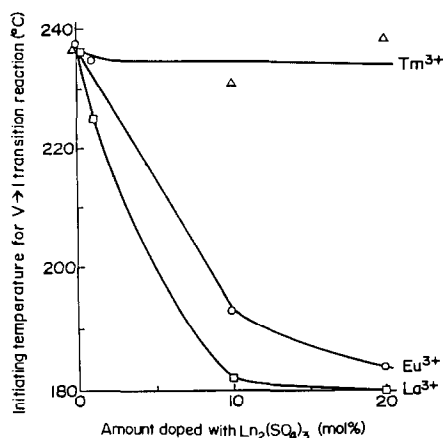


FIG. 4. The effect of amount doped with Ln<sup>3+</sup> on the V → I transition of Na<sub>2</sub>SO<sub>4</sub> crystals on the Ln<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples in N<sub>2</sub> flow.

Saito *et al.* reported that the III  $\rightleftharpoons$  I transition of Na<sub>2</sub>SO<sub>4</sub> proceeds with the rotation of the SO<sub>4</sub><sup>2-</sup> tetrahedron, and this transition model was supported by the results of Mehrotra *et al.* (6) and Amirthalingam *et al.* (7). Therefore the results obtained in N<sub>2</sub> flow (Fig. 3) suggest that the rotation of the SO<sub>4</sub><sup>2-</sup> tetrahedron in Na<sub>2</sub>SO<sub>4</sub> crystals becomes easy owing to the Na vacancy formed in Na<sub>2</sub>SO<sub>4</sub> crystals due to the diffusion of Na<sup>+</sup> ions into the LSS, resulting in promotion of the V  $\rightarrow$  I transition. The diffusion of Na<sup>+</sup> ions into the LSS in the La<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample ( $r_{\text{Ln}^{3+}}/r_{\text{Na}^{+}} > 1$ ) would most easily occur among the samples used here (Figs. 3 and 4), resulting in promotion of the transition from LSS to HSS, which corresponds to the solid solution between the phase I of Na<sub>2</sub>SO<sub>4</sub> and a small amount of foreign cation sulfate as pointed out by Eysel *et al.* (8).

We have found that Na<sub>2</sub>SO<sub>4</sub> crystals rapidly sorb O<sub>2</sub> gas (9). It is well known that the chemical affinity between rare earth and oxygen is strong. In O<sub>2</sub> flow, the SO<sub>4</sub><sup>2-</sup> tetrahedron of Na<sub>2</sub>SO<sub>4</sub> crystals is thought to become considerably difficult to rotate due to the sorption of O<sub>2</sub> gas into Na<sub>2</sub>SO<sub>4</sub> crystals, resulting in inhibition of the V  $\rightarrow$  I transition in all Na<sub>2</sub>SO<sub>4</sub> samples. In the Eu<sup>3+</sup>- and Tm<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> samples ( $r_{\text{Ln}^{3+}}/r_{\text{Na}^{+}} \leq 1$ ), the diffusion of Na<sup>+</sup> into LSS and the diffusion of Ln<sup>3+</sup> into Na<sub>2</sub>SO<sub>4</sub> occur simultaneously and latter is promoted by oxygen sorbed into Na<sub>2</sub>SO<sub>4</sub> due to the strong chemical affinity between rare earth and oxygen, resulting in the promotion of the LSS  $\rightarrow$  HSS transition. In the La<sup>3+</sup>-doped Na<sub>2</sub>SO<sub>4</sub> sample ( $r_{\text{Ln}^{3+}}/r_{\text{Na}^{+}} > 1$ ), the diffusion of Ln<sup>3+</sup> into Na<sub>2</sub>SO<sub>4</sub> cannot be expected, in which case the LSS  $\rightarrow$  HSS transition could not be promoted even if in O<sub>2</sub> flow. In NH<sub>3</sub> flow, we think that the NH<sub>3</sub> complex formed on the crystal surface of Na<sub>2</sub>SO<sub>4</sub> suppressed not only the rotation of the SO<sub>4</sub><sup>2-</sup> tetrahedron of Na<sub>2</sub>SO<sub>4</sub> crystals but also the diffu-

sion of Na<sup>+</sup> ion and/or Ln<sup>3+</sup> ion, resulting in a rise in the initiation temperature of both the V  $\rightarrow$  I transition and the LSS  $\rightarrow$  HSS transition. It could be thought that the I  $\rightarrow$  III transition in pure Na<sub>2</sub>SO<sub>4</sub> sample on cooling was also suppressed by the existences of oxygen sorbing into Na<sub>2</sub>SO<sub>4</sub> crystals and of the NH<sub>3</sub> complex on the crystal surface of Na<sub>2</sub>SO<sub>4</sub> as well as the V  $\rightarrow$  I transition on heating.

## Conclusion

The phase transition reaction of Na<sub>2</sub>SO<sub>4</sub> crystals was promoted by the enhanced rotation of the SO<sub>4</sub><sup>2-</sup> tetrahedron caused by Na vacancy in Na<sub>2</sub>SO<sub>4</sub> crystals. However, the promotive effect of Na vacancies was suppressed by not only the Ln<sup>3+</sup> ion diffusing into Na<sub>2</sub>SO<sub>4</sub> crystals and but also by O<sub>2</sub> sorbing into Na<sub>2</sub>SO<sub>4</sub> crystals and the NH<sub>3</sub> complex on the crystal surface of Na<sub>2</sub>SO<sub>4</sub>.

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