

## On the Preparation and Characterization of a New Polymorph of $V_2O_5$

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A new metastable  $\gamma'$  variety of  $V_2O_5$  has been prepared by chemical or electrochemical deintercalation of lithium from a  $\gamma$ - $LiV_2O_5$  bronze. The structure of the new phase has been determined with a Rietveld method and compared to those of the normal  $V_2O_5$  oxide and of  $\gamma$ - $LiV_2O_5$ .  $\gamma'$ - $V_2O_5$  transforms into the normal form of  $V_2O_5$  at 340°C. © 1991 Academic Press, Inc.

### Introduction

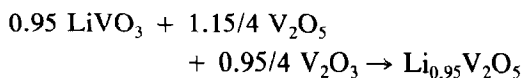
Cycling a  $Li//V_2O_5$  battery in the 3.8- to 2-V range leads to the formation of a new polymorph of  $V_2O_5$  which we have called  $\gamma'$ - $V_2O_5$ . This material is obtained by the deintercalation of lithium from the  $\gamma$ - $Li_xV_2O_5$  bronze formed after the positive electrode is discharged beyond  $x = 1$  (1, 2). During the deintercalation process  $\gamma'$ - $V_2O_5$  is mixed with the normal form of  $V_2O_5$ . Pure  $\gamma'$ - $V_2O_5$  has also been obtained by charging a  $Li//\gamma$ - $LiV_2O_5$  battery (3).

The realization of a positive  $V_2O_5$  or  $\gamma$ - $LiV_2O_5$  electrode requires the addition of some carbon for improving the conductivity and teflon as a binder. In order to obtain a pure product, we propose a chemical method for preparing  $\gamma'$ - $V_2O_5$  based on the oxidation of  $\gamma$ - $LiV_2O_5$  by a solution of bromine in acetonitrile. Such a mild synthesis method leading to a metastable phase at

room temperature excludes single crystal growth. On the other hand, vanadium atoms cannot be precisely located by neutron diffraction. Therefore, the structure of this phase has been refined with a Rietveld treatment of the X-ray diffraction data.

### Experimental Results

$\gamma$ - $LiV_2O_5$  was synthesized at 650°C in an evacuated silica tube according to the reaction (4)



Deinsertion of Li from  $\gamma$ - $LiV_2O_5$  was performed with a solution of bromine in acetonitrile (300 mg of the bronze reacts with 200 ml of a 10 M solution of bromine). After 2 weeks of stirring, we obtained an orange-yellow product which was then washed with pure acetonitrile.

The residual concentration of lithium determined by photoemission with a Per-

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TABLE I  
DETAILS OF DATA COLLECTION AND OF RIETVELD REFINEMENT

Diffractometer	Philips PW 1050
Wavelength	$\lambda$ CuK $\alpha$ ( $K\alpha_2/K\alpha_1 = 0.50$ )
$2\theta$ range; step ( $^\circ$ )	10–110; 0.02
Computer program	DBW.2S (6)
Space group	<i>Pnma</i>
Z	4
a ( $\text{\AA}$ ):	9.9461(3)
b ( $\text{\AA}$ ):	3.5852(1)
c ( $\text{\AA}$ ):	10.0423(4)
Number of reflections	268
Number of refined parameters	37
Scattering factors	V <sup>5+</sup> and O <sup>2-</sup> corrected for anomalous dispersion (7)
Profile function	Pseudo-Voigt (PV = $\eta L + (1 - \eta)G$ ) ( $\eta = 0.682$ ) (13)
Background correction	Polynomial function (degree 5 in $2\theta$ )
Full width at half maximum (H) function	$H^2 = U \text{tg}^2\theta + V \text{tg}\theta + W$ (8)
Maximal final value for shift/e.s.d.	0.03
$R_p = \sum  y_i - y_{ci}  / \sum y_i$	0.0812
$R_{wp} =  \sum w_i (y_i - y_{ci})^2 / \sum w_i y_i^2 ^{1/2}$	0.1108
$R_f = \sum  F_o'' - F_c  / \sum  F_o'' $	0.0738
$R = \sum  F_o'' - F_c  / \sum  F_o'' $	0.0702

kin-Elmer spectrophotometer was smaller than  $x = 0.006$ . The quasi-complete deintercalation of lithium was confirmed by the color of the compound, which is similar to that of normal V<sub>2</sub>O<sub>5</sub>.

Electrochemical preparation of  $\gamma'$ -V<sub>2</sub>O<sub>5</sub> was performed in a coin cell. The anode was a lithium foil and the positive electrode a mixture of  $\gamma$ -LiV<sub>2</sub>O<sub>5</sub> with teflon and graphite. The electrolyte was a solution of LiAsF<sub>6</sub> in PC/EC/DME.

The powder diffraction data were collected on a Philips PW 1050 powder diffractometer using a Bragg-Brentano geometry

with a copper target and a takeoff angle of 6°. In order to minimize any preferential orientation of the crystallites, the upper layer of the sample was sifted. Actually 99% of the total energy is diffracted by a 4.6- $\mu\text{m}$  thick and a 46- $\mu\text{m}$  thick layer for 10° and 120° ( $2\theta$ ), respectively. The powder diffraction pattern was scanned by steps of 0.02° ( $2\theta$ ) from 10° to 110° ( $2\theta$ ) with a constant counting time of 40 sec.

The X-ray diffractograms of the new phase are quite different from that of normal V<sub>2</sub>O<sub>5</sub> and reveal an orthorhombic cell close to that of  $\gamma$ -LiV<sub>2</sub>O<sub>5</sub> (Table I). Therefore, we

TABLE II  
POSITIONAL AND ISOTROPIC THERMAL PARAMETERS FOR  $\gamma'$ -V<sub>2</sub>O<sub>5</sub>  
(SPACE GROUP *Pnma*)

Atom	Position	x	y	z	$B_{\text{iso}}$ ( $\text{\AA}^2$ )
V <sub>1</sub>	4c	0.3758(30)	0.2500	0.5164(20)	1.24(7)
V <sub>2</sub>	4c	0.0699(30)	0.2500	0.6015(20)	1.50(7)
O <sub>1</sub>	4c	0.2521(90)	0.2500	0.6369(80)	3.1(3)
O <sub>2</sub>	4c	0.4997(70)	0.2500	0.7571(80)	1.5(2)
O <sub>3</sub>	4c	0.2934(90)	0.2500	0.3856(90)	4.4(3)
O <sub>4</sub>	4c	0.5676(70)	0.2500	0.4616(80)	1.6(2)
O <sub>5</sub>	4c	-0.0675(80)	0.2500	0.4600(70)	3.0(2)

TABLE III  
INTERATOMIC DISTANCES AND ANGLES IN  $\gamma'$ -V<sub>2</sub>O<sub>5</sub>

V <sub>1</sub>	O <sub>1</sub>	O <sub>3</sub>	O <sub>4</sub>	O <sub>4</sub> <sup>i</sup>	O <sub>4</sub> <sup>ii</sup>	O <sub>2</sub>
O <sub>1</sub>	1.725(7)	2.55(1)	2.72(1)	2.72(1)	2.72(1)	2.74(1)
O <sub>3</sub>	102°56	1.548(7)	2.83(1)	2.73(1)	2.73(1)	4.25(1)
O <sub>4</sub>	151°66	105°77	1.985(6)	2.37(1)	2.37(1)	3.04(1)
O <sub>4</sub> <sup>i</sup>	97°46	104°97	75°30	1.891(6)	3.59(1)	2.91(1)
O <sub>4</sub> <sup>ii</sup>	97°46	104°97	75°30	142°54	1.891(6)	2.91(1)
O <sub>2</sub>	72°49	175°05	79°18	76°05	76°05	2.713(6)
V <sub>2</sub>	O <sub>1</sub>	O <sub>2</sub> <sup>iii</sup>	O <sub>5</sub>	O <sub>5</sub> <sup>iv</sup>	O <sub>5</sub> <sup>v</sup>	O <sub>3</sub>
O <sub>1</sub>	1.846(7)	2.72(1)	3.64(1)	2.74(1)	2.74(1)	2.55(1)
O <sub>2</sub> <sup>iii</sup>	105°08	1.581(6)	2.92(1)	2.79(1)	2.79(1)	4.62(1)
O <sub>5</sub>	144°98	109°93	1.973(6)	2.38(1)	2.38(1)	3.66(1)
O <sub>5</sub> <sup>iv</sup>	94°30	106°66	75°90	1.896(7)	3.59(1)	3.26(1)
O <sub>5</sub> <sup>v</sup>	94°30	106°66	75°90	141°95	1.896(7)	3.26(1)
O <sub>3</sub>	55°38	160°46	89°60	77°38	77°38	3.104(7)

Note. Symmetry code: (i)  $1 - x, \frac{3}{2}, 1 - z$ ; (ii)  $1 - x, -\frac{1}{2}, 1 - z$ ; (iii)  $x - \frac{1}{2}, \frac{3}{2}, \frac{3}{2} - z$ ; (iv)  $-x, \frac{3}{2}, 1 - z$ ; (v)  $-x, -\frac{1}{2}, 1 - z$ .

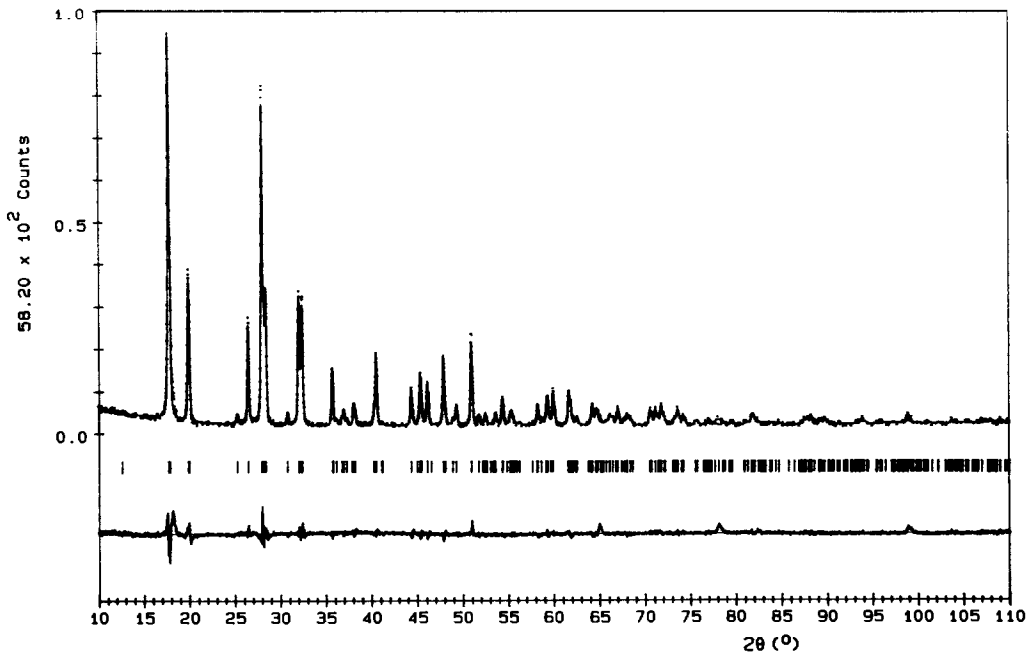


FIG. 1. Final Rietveld difference plots for  $\gamma'$ -V<sub>2</sub>O<sub>5</sub>. In the upper part, the observed data are shown by dots and the calculated pattern by a solid line.

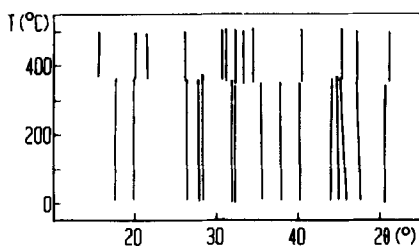


FIG. 2. Evolution of the X-ray diffractogram of  $\gamma'$ - $V_2O_5$  with temperature showing the transformation of  $\gamma'$ - $V_2O_5$  into the normal form.

took as starting positions for vanadium and oxygen atoms those of  $\gamma$ - $LiV_2O_5$ . Experimental conditions and computation details are given in Table I, final atomic positions in Table II and selected bond distances and angles in Table III. The calculated and experimental spectra are compared in Fig. 1. These results correspond to satisfactory crystal structure model indicators ( $R_1 = 0.074$ ,  $R = 0.070$ ) and profile factors ( $R_p = 0.081$ ,  $R_{wp} = 0.111$ ).

It was worthwhile to investigate the thermal behavior of  $\gamma'$ - $V_2O_5$  as a spontaneous transformation into the stable form of  $V_2O_5$  may be expected. Figure 2 gives the thermal evolution of the X-ray diffraction pattern obtained with a Guinier-Simon camera using  $CuK\alpha$  radiation. It shows that  $\gamma'$ - $V_2O_5$  undergoes a transformation into the normal form of  $V_2O_5$  near  $340^\circ C$ . The transformation gives rise to an exothermic peak at the same temperature in DTA experiments performed in dry oxygen atmosphere.

## Discussion

The layer structure of  $\gamma'$ - $V_2O_5$  is based on  $VO_5$  distorted pyramids sharing edges and forming infinite double ribbons parallel to the  $b$ -axis. These ribbons are linked by pyramid corners forming puckered layers perpendicular to the  $c$ -axis (Fig. 3b).

$\gamma'$ - $V_2O_5$  is formed by chemical deinterca-

lation of lithium from  $\gamma$ - $LiV_2O_5$  at room temperature. Under such conditions a solid-state reaction tends to proceed in the easiest way, i.e., by the route involving a minimum of structural change (9). Thus in the oxidation reaction of  $\gamma$ - $LiV_2O_5$  by bromine, lithium atoms which lie between the sheets are deintercalated without any major modification of the way in which  $VO_5$  pyramids are linked to each other. However, interatomic distances undergo significant changes. As lithium is deintercalated, the sheets are slightly unfolded, leading to an increase of the  $a$  parameter while the interlayer dis-

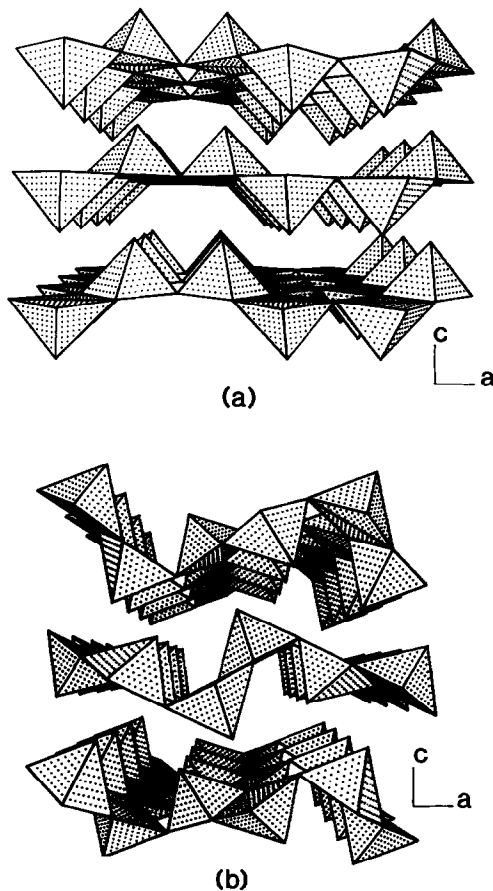


FIG. 3. Structure of the normal (a) and the  $\gamma'$  form (b) of  $V_2O_5$ .

TABLE IV  
SOME CHARACTERISTIC V–O DISTANCES OF THE VO<sub>5</sub> PYRAMIDS IN SOME RELEVANT VANADIUM OXIDES

	Normal V <sub>2</sub> O <sub>5</sub>	γ'-V <sub>2</sub> O <sub>5</sub>	γ-LiV <sub>2</sub> O <sub>5</sub> <sup>a</sup>	δ-LiV <sub>2</sub> O <sub>5</sub> <sup>b</sup>
Average V–O distance in the VO <sub>5</sub> pyramid (Å)	V <sub>2</sub> site	1.84	1.820	
	V <sub>1</sub> site	1.838	1.880	1.850
Shortest V–O distance (Å)	V <sub>2</sub> site	1.58	1.608	
	V <sub>1</sub> site	1.577	1.599	1.610
Difference between the largest and the shortest distance in the pyramid base (Å)	V <sub>2</sub> site	0.13	0.23	
	V <sub>1</sub> site	0.24	0.05	0.08

<sup>a</sup> In the structure of γ, the V<sub>1</sub> sites are occupied by V<sup>4+</sup> ions.

<sup>b</sup> V<sup>5+</sup> and V<sup>4+</sup> ions are randomly distributed in the structure of δ.

tances decrease. Double chains exist in both polymorphs, but VO<sub>5</sub> pyramids alternate up and down individually for γ'-V<sub>2</sub>O<sub>5</sub>, whereas they alternate by pairs for normal V<sub>2</sub>O<sub>5</sub> (Fig. 3a).

The VO<sub>5</sub> pyramids have similar features in normal V<sub>2</sub>O<sub>5</sub>, δ-LiV<sub>2</sub>O<sub>5</sub>, γ-LiV<sub>2</sub>O<sub>5</sub>, and γ'-V<sub>2</sub>O<sub>5</sub>: (i) the shortest V–O distance remains between 1.55 and 1.61 Å for an oxidation state of vanadium between 5 and 4, (ii) the average V–O distance increases as the oxidation state of vanadium decreases from 5 to 4, and (iii) the pyramids tend to become more regular with reduction of the vanadium as shown in Table IV. Such an evolution is normally expected, and one may observe that for δ-LiV<sub>2</sub>O<sub>5</sub> in which V<sup>5+</sup> and V<sup>4+</sup> are randomly distributed, the characteristic distances have values intermediate between those relative to the V<sub>1</sub> and V<sub>2</sub> sites in γ-LiV<sub>2</sub>O<sub>5</sub>.

An interesting feature of the structure of γ'-V<sub>2</sub>O<sub>5</sub> is the existence of two different sites for the vanadium atoms, unlike what is observed in the normal form of V<sub>2</sub>O<sub>5</sub> but like the situation in the γ-LiV<sub>2</sub>O<sub>5</sub> bronze. However, the average V–O distance characteristic of the V<sub>1</sub> site strongly decreases on passing from the γ- to the γ'-phase to

reach a value similar to that observed for VO<sub>5</sub> pyramids in other V<sup>5+</sup> oxides (Table IV). Simultaneously, as shown in Table IV by the dispersion of the V–O distances in the pyramid base, the V<sub>1</sub> site is more distorted in the γ'-V<sub>2</sub>O<sub>5</sub> oxide than the V<sup>4+</sup> (V<sub>2</sub>) site of the γ-LiV<sub>2</sub>O<sub>5</sub> bronze, in agreement with the general tendency pointed out above. However, the difference between the longest and the shortest V–O distance in the pyramid base is smaller for the V<sub>2</sub> site of γ'-V<sub>2</sub>O<sub>5</sub> than for the three other phases mentioned in Table IV, including the V<sub>1</sub> site of γ'-V<sub>2</sub>O<sub>5</sub>.

### Conclusion

Preparation of γ'-V<sub>2</sub>O<sub>5</sub> is a significant example of a topochemical deinsertion leading to a new metastable variety of V<sub>2</sub>O<sub>5</sub> but keeping the memory of the original intercalated product. This new polymorph of V<sub>2</sub>O<sub>5</sub> is kinetically stable up to ≈340°C, the temperature at which it transforms into the thermodynamically stable form of vanadium pentoxide. The electrochemical behavior of γ'-V<sub>2</sub>O<sub>5</sub> is interesting as it explains the enhancement of the cell potential of Li//V<sub>2</sub>O<sub>5</sub> batteries cycled in the 3.8- to 2-V range as reported elsewhere (3, 10).

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