# Synthesis of Metal Molybdenum Tellurides of Composition $Mo_{6-x}M_xTe_8$ (M = Ru or Rh)<sup>†</sup>

## Frank J. Berry\* and Christopher D. Gibbs

School of Chemistry, University of Birmingham, P.O. Box 363, Birmingham B15 2TT, UK

Monophasic compounds of compositions  $Mo_{6-x}Ru_xTe_8$  (x=0.5, 1.0 or 1.5) and  $Mo_{5.5}Rh_{0.5}Te_8$  have been synthesised. The preparation of the pure compounds depends critically on the thermal treatments used. The maximum value of x in compounds of the type  $Mo_{6-x}Ru_xTe_8$  is ca. 1.75. The compounds have been characterised by X-ray powder diffraction and the lattice parameters shown to vary with ruthenium content.

Metal molybdenum chalcogenides of composition Mo<sub>6</sub>M<sub>x</sub>X<sub>8</sub>, where M represents a wide variety of metallic elements and X is sulphur, selenium or tellurium have been known for some time to adopt structures containing the Mo<sub>6</sub>X<sub>8</sub> unit <sup>1</sup> (Fig. 1). In these materials, which are commonly known as the Chevrel phase compounds, the ternary metal occupies positions in channels created by stacking of the  $Mo_6X_8$  units.<sup>2-7</sup> Since the ternary molybdenum chalcogenides were first reported the number of characterised Chevrel phase sulphides and selenides has grown rapidly with alkali-, alkaline earth-, p block-, transition- and lanthanide-metals all being reported as ternary elements.1 Pseudo-ternary compounds containing two chalcogenides such as  $Mo_6PbS_{8-x}Se_x$  (0 < x < 8) or two ternary metals such as  $Mo_6So_{1-x}Zo_xS_8$  (x=0.2 or 0.4) have also been synthesised and characterised. 8.9 The partial substitution of molybdenum atoms by selected second- or third-row transition metals has also been described although only two compounds, Mo<sub>4</sub>Ru<sub>2</sub>Se<sub>8</sub> and Mo<sub>1.5</sub>Re<sub>4.5</sub>Se<sub>8</sub>, have been the subject of detailed structural examination. <sup>10</sup> In contrast, the number of tellurium-containing analogues of the Chevrel phase compounds is sparse. The lattice parameters of compounds described as  $Mo_6M_xTe_8$  (M = Fe, Ni or Cu, 0 < x < 4; M = Co, 0 < x < 0.66) have been reported 11 and a compound of composition Mo<sub>6</sub>Ni<sub>0.85</sub>Te<sub>8</sub> described.<sup>12</sup> The detailed preparation and characterisation of compounds of the type  $Mo_{6-x}M_xTe_8$  has not been described. <sup>13,14</sup>

The considerable interest in the Chevral phase compounds prior to the discovery of high-temperature oxide superconductors derives  $^1$  from their superconducting properties ( $T_c$  ca. 13 K) with large critical magnetic fields [ $B_c$  ca. 650 kG (0.065 T) at 2 K]. Some of the compounds have been found to be both superconducting and antiferromagnetically ordered  $^{15}$  and some have been evaluated as components in solid-state batteries. $^{16}$ 

In this paper we report on the preparation of metal molybdenum tellurides of the type  $Mo_{6-x}M_xTe_8$  (M = Ru or Rh) and their characterisation by X-ray powder diffraction.

### **Experimental**

Compounds of the type  $Mo_{6-x}M_xTe_8$  (M = Ru or Rh) were prepared by heating mixtures of the appropriate molar ratios of the high purity elements (99.9%+) in evacuated sealed quartz ampoules according to the conditions described in Table 1. It should be noted that the times of heating include ca. 15 min

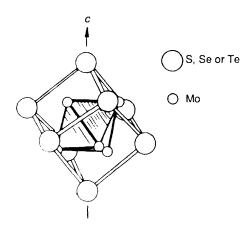


Fig. 1 The  $Mo_6X_8$  (X = S or Se) unit

**Table 1** Conditions required for the preparation of monophasic compounds of compositions  $Mo_6Te_8$  and  $Mo_{6-x}M_xTe_8$  (M = Ru or Rh)

Compound	Conditions
$Mo_6Te_8$	1150 °C (2 d)
$Mo_{5.5}Ru_{0.5}Te_8$	1200 °C (1 d)
Mo <sub>5</sub> RuTe <sub>8</sub>	1150 °C (4 d)
$Mo_{4.5}Ru_{1.5}Te_8$	1200 °C (4 d), furnace cooled, ground, 1200 °C (45 min)
$\mathrm{Mo}_{5.5}\mathrm{Rh}_{0.5}\mathrm{Te}_{8}$	1150 °C (2 d)

during which the furnace was heating up to the selected temperature.

X-Ray powder diffraction patterns were recorded with a Philips PW1050/70 diffractometer using Cu-Kα radiation.

## **Results and Discussion**

Monophasic compounds of compositions  $Mo_6Te_8$  and  $Mo_{6-x}M_xTe_8$  (M=Ru or Rh) were prepared according to the conditions described in Table 1. Conditions other than those presented in Table 1 were found by X-ray powder diffraction to give multiphasic products. The X-ray powder diffraction pattern recorded from  $Mo_6Te_8$  compared well with that reported previously.  $^{17,18}$  The X-ray powder diffraction patterns recorded from the new compounds were indexed by comparison with  $Mo_6Te_8$ .

Attempts to prepare Mo<sub>6</sub>Te<sub>8</sub> under conditions different from

<sup>†</sup> Supplementary data available (No. SUP 56802, 5 pp.): X-ray powder diffraction data for  $\mathrm{Mo_{6-x}Ru_xTe_8}$  (x=0.5, 1.0 or 1.5). See Instructions for Authors, J. Chem. Soc., Dalton Trans., 1991, Issue 1, pp. xviii–xxii.

Table 2 Lattice parameters and unit-cell volumes, with estimated standard deviations in parentheses, of compounds of the type  $Mo_{6-x}M_xTe_8$  (x=0, 0.5, 1.0 or 1.5) and the  $Mo_{6-x}Ru_xTe_8$ -type phases corresponding to nominal compositions  $Mo_4Ru_2Te_8$  and  $Mo_3Ru_3Te_8$ 

Compound	$a_{ m hex}/{ m \AA}$	$c_{ m hex}/{ m \AA}$	$a_{ m rhom}/{ m \AA}$	$\alpha_{\rm rhom}/^{\circ}$	$U_{\sf hex}/{ m \AA}^3$
$Mo_6Te_8$	10.195(3)	11.663(5)	7.054(2)	92.54(3)	1049.8(1.1)
$Mo_{5.5}Ru_{0.5}Te_8$	10.221(6)	11.615(8)	7.058(5)	92.79(5)	1050.8(2.0)
$Mo_{5.5}Rh_{0.5}Te_8$	10.229(5)	11.563(6)	7.052(4)	92.98(4)	1047.8(1.2)
Mo <sub>5</sub> RuTe <sub>8</sub>	10.234(5)	11.488(7)	7.041(4)	93.23(4)	1042.0(1.6)
$Mo_{4.5}Ru_{1.5}Te_8$	10.259(6)	11.391(7)	7.036(4)	93.62(4)	1038.3(1.8)
$Mo_{6-x}Ru_xTe_8$ -type	10.274(3)	11.355(3)	7.036(2)	93.78(2)	1038.3(5)
(nominally Mo <sub>4</sub> Ru <sub>2</sub> Te <sub>8</sub> )					
$Mo_{6-x}Ru_xTe_8$ -type	10.275(6)	11.351(6)	7.036(4)	93.80(4)	1037.8(1.8)
(nominally Mo <sub>3</sub> Ru <sub>3</sub> Te <sub>6</sub> )					

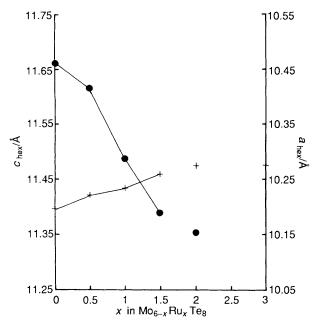


Fig. 2 Variation of hexagonal (hex) lattice parameters for compounds of the type  $Mo_{6-x}Ru_xTe_8$  as a function of x: a, +; c,  $\blacksquare$ 

those described in Table 1 gave materials which were shown by X-ray powder diffraction to contain  $Mo_6Te_8$  and  $\beta$ -MoTe<sub>2</sub>. Attempts to prepare compounds of the type  $Mo_{6-x}Ru_xTe_8$  ( $x=0.5,\ 1.0$  or 1.5) were frequently found to give products containing  $Mo_{6-x}Ru_xTe_8$  with either a  $\beta$ -MoTe<sub>2</sub> or RuTe<sub>2</sub> impurity phase depending on the reaction conditions. Reactions performed for longer periods of time than those specified in Table 1 tended to give the  $\beta$ -MoTe<sub>2</sub> impurity whilst reactions performed at lower temperatures and/or for shorter periods of time tended to give RuTe<sub>2</sub>. Presumably the impure products also contained small amounts of elemental molybdenum and ruthenium which were below the limits of detection by X-ray powder diffraction.

Attempts to prepare  $Mo_4Ru_2Te_8$  were unsuccessful. X-ray powder diffraction showed the products to contain a  $Mo_{6-x}Ru_xTe_8$ -type phase together with  $RuTe_2$  impurity. The attempted preparation of  $Mo_3Ru_3Te_8$  gave a  $Mo_{6-x}Ru_xTe_8$ -type phase together with  $RuTe_2$  and elemental ruthenium according to the reaction  $3Mo + 3Ru + 8Te \rightarrow aMo_{6-x}Ru_xTe_8 + bRuTe_2 + cRu$  in which x is less than three. The results, taken together, suggest that an upper limit exists beyond which ruthenium cannot be accommodated in the  $Mo_6Te_8$  phase.

The refined lattice parameters for  $Mo_6Te_8$ , the pure compounds  $Mo_{6-x}Ru_xTe_8$  (x=0.5, 1.0 or 1.5) and the  $Mo_{6-x}Ru_xTe_8$ -type phases corresponding to the nominal compositions  $Mo_4Ru_2Te_8$  and  $Mo_3Ru_3Te_8$  are presented in Table 2 and the trends described diagramatically in Fig. 2.

The lattice parameters of Mo<sub>6</sub>Te<sub>8</sub> agree well with those reported previously <sup>12</sup>  $(a_{\text{rhom}} = 7.050 \text{ Å}, \alpha_{\text{rhom}} = 92.54^{\circ})$ . The lattice parameters of the pure compounds  $Mo_{6-x}Ru_xTe_8$  (x = 0-1.5) show steady trends with composition as might be expected for a solid solution. The greatest variation involves the decrease in  $c_{\text{hex}}$  with increasing ruthenium content. The similarity between the lattice parameters of the Mo<sub>6-x</sub>Ru<sub>x</sub>Te<sub>8</sub>type phases of nominal compositions Mo<sub>4</sub>Ru<sub>2</sub>Te<sub>8</sub> and Mo<sub>3</sub>Ru<sub>3</sub>Te<sub>8</sub> suggests the presence of a comparable ruthenium content in both compounds. It also appears that the ruthenium content in these Mo<sub>6-x</sub>Ru<sub>x</sub>Te<sub>8</sub>-type compounds is less than the nominal value since the  $c_{\text{hex}}$  lattice parameters are smaller than would be expected by extrapolation of the lattice parameters of the pure compounds of composition  $Mo_{6-x}Ru_xTe_8$  (x = 0.5, 1.0 or 1.5). The result is consistent with the detection by X-ray powder diffraction of both ruthenium ditelluride and elemental ruthenium in the reaction products of the attempted preparation of Mo<sub>3</sub>Ru<sub>3</sub>Te<sub>8</sub>. The ruthenium content in Mo<sub>6-x</sub>Ru<sub>x</sub>Te<sub>8</sub>-type phases of nominal compositions Mo<sub>4</sub>Ru<sub>2</sub>Te<sub>8</sub> and Mo<sub>3</sub>Ru<sub>3</sub>Te<sub>8</sub> was estimated by extrapolating their  $c_{\text{hex}}$  lattice parameters to the approximately straight line connecting the  $c_{
m hex}$  lattice parameters of the three pure Mo<sub>6-x</sub>Ru<sub>x</sub>Te<sub>8</sub> compounds. The procedure gives a maximum value of x of ca. 1.75, corresponding to the formulation  $Mo_{4.25}Ru_{1.75}Te_8$ . Hence the value of x in  $Mo_{6-x}Ru_xTe_8$ appears to vary from 0 to 1.75 and corresponds to a maximum of 3.5 additional electrons that can be introduced into the Mo<sub>6</sub>Te<sub>8</sub>-type structure by the incorporation of ruthenium atoms. This is less than the 4.6 additional electrons which are reportedly 10,19 accommodated into the Mo<sub>6</sub>Se<sub>8</sub>-type structure when compounds of the types Mo<sub>1.5</sub>Re<sub>4.5</sub>Se<sub>8</sub> and Mo<sub>3.7</sub>Ru<sub>2.3</sub>Se<sub>8</sub> are prepared.

The compound  $Mo_{5.5}Rh_{0.5}Te_8$  was prepared by heating appropriate molar ratios of molybdenum, rhodium and tellurium for 2 d at 1150 °C. The lattice parameters are included in Table 2. Attempts to prepare compounds of compositions  $Mo_5RhTe_8$  and  $Mo_{4.66}Rh_{1.33}Te_8$  were unsuccessful, the products being invariably composed of mixtures of a  $Mo_{6-x}Rh_xTe_8$  phase and  $RhTe_2$ .

Attempts to prepare compounds of the type  $Mo_{6-x}M_xTe_8$  (M = Nb, Ta, W, Re, Os or Ir) gave multiphasic products which were generally dominated by the presence of the metal telluride.

In conclusion, we would emphasise that the preparation of pure compounds of the type  $Mo_{6-x}M_xTe_8$  where M is ruthenium or rhodium is a non-trivial matter. Our results do not confirm those described earlier <sup>13,14</sup> which, without details of preparation conditions and comprehensive X-ray diffraction data, have suggested that the solubility of ruthenium in  $Mo_{6-x}Ru_xTe_8$  exceeds the limit reported here. We would make clear that we have found no generality in the heating conditions required to synthesise monophasic compounds of the type  $Mo_{6-x}M_xTe_8$  and that most routes to prepare these types of compounds lead to the formation of impurity metal and metal telluride phases.

Finally, we would mention that low-temperature specific heat measurements indicate that the compound of composition Mo<sub>5</sub>RuTe<sub>8</sub> shows the onset of superconductivity at 2.21 K. A detailed account of these measurements will be published elsewhere.

#### Acknowledgements

We thank the SERC for support.

#### References

- 1 K. Yvon, in Current Topics in Materials Science, ed. E. Kaldis, North-Holland, Amsterdam, 1979, vol. 3, p. 53.
- 2 R. Chevrel, J. Prigent and M. Sergent, J. Solid State Chem., 1971, 3,
- 3 O. Bars, D. Grandjean and J. Guillevic, J. Solid State Chem., 1973, 6,
- 4 O. Bars, D. Grandjean and J. Guillevic, J. Solid State Chem., 1973, 6, 335.
- 5 O. Bars, D. Grandjean and J. Guillevic, J. Solid State Chem., 1973, 7, 158.
- 6 E. Corenzwit, P. D. Dernier, M. Marezio, B. T. Matthias and J. P. Remeika, Mat. Res. Bull., 1973, 8, 657.

- 7 R. Chevrel and M. Sergent, J. Solid State Chem., 1973, 6, 433.
- 8 R. Chevrel, O. Fischer and M. Sergent, Mat. Res. Bull., 1975, 10, 1169.
- 9 J. W. Downey and F. Y. Fradin, Mat. Res. Bull., 1979, 14, 1525.
- 10 H. D. Flack, W. Hönle and K. Yvon, J. Solid State Chem., 1983, 49,
- 11 R. Chevrel and M. Sergent, in Superconductivity in Ternary Compounds, eds. O. Fischer and M. B. Maple, Springer-Verlag, Berlin, Heidelberg, 1982, vol. 1, p. 30.

  12 W. Hönle and K. Yvon, *J. Solid State Chem.*, 1987, **70**, 235.
- 13 O. Fischer, A. Perrin and M. Sergent, Mat. Res. Bull., 1978, 13, 259
- 14 R. Chevrel, O. Fischer, A. Perrin and M. Sergent, J. Solid State Chem., 1980, 33, 43.
- 15 O. Fischer and M. Ishikawa, Solid State Commun., 1977, 24, 747.
- 16 R. Kanno, Y. Takeda, M. Ohya and O. Yamamoto, Mat. Res. Bull., 1987, **22**, 1283,
- 17 Joint Committee on Powder Diffraction Standard (J.C.P.D.S.) Index Card No. 23-1257.
- 18 R. Chevrel, J. Prigent and M. Sergent, Mat. Res. Bull., 1974, 9, 1487.
- 19 N. V. Vante, W. Jaegermann, H. Tributsch, W. Hönle and K. Yvon, J. Am. Chem. Soc., 1987, 109, 3251.

Received 14th June 1990; Paper 0/02669G