

THERMAL DECOMPOSITION OF LEAD TITANYL OXALATE TETRAHYDRATE

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ABSTRACT

The thermal behaviour of $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$ (PTO) has been investigated, employing TG, quantitative DTA, infrared spectroscopy and (high temperature) X-ray powder diffraction.

The decomposition involves four main steps. The first is the dehydration of the tetrahydrate (30–180°C), followed by a small endothermic (270–310°C) and a large exothermic decomposition of the oxalate. The main (exothermic) oxalate decomposition (310–390°C) results in a stable oxide-carbonate $\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5}$. In the last step a phase transition, release of CO_2 and ordering of the crystalline cubic PbTiO_3 lattice can be detected (460–530°C).

It can be argued that for thermodynamic reasons the presence of lead-oxo-carbonates in the oxide-carbonate intermediate is not possible.

No differences could be found in thermal behaviour of two crystallographically different synthetic forms of PTO, of which one has an orthorhombic lattice.

INTRODUCTION

As is the case with many fine-grained pure ceramic powders, lead titanate is often synthesized via decomposition of an oxalate complex precursor. The present investigation was undertaken as a part of a larger study of influences of synthesis and thermolysis circumstances on decomposition of i.a. ammonium and lead titanyl oxalate complexes.

When the study of the behaviour of lead titanyl oxalate tetrahydrate, $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$, (PTO), was started, the proposals of earlier investigators for the mechanism of the thermal decomposition of the analogous barium titanyl oxalate (BTO) were highly contradictory^{1,2}. It was interesting to see whether the decomposition scheme for the lead compound had a resemblance to any of the proposed schemes or should have a different explanation. There was a possibility that decomposition could proceed via an oxide-carbonate intermediate, as was found for ammonium titanyl oxalate³. During the experiments for this investigation such an intermediate was indeed considered necessary to explain the data. Recently, Gopalakrishnamurthy et al.⁴ published a new report on BTO decomposition which

strongly supports the mechanism described here. During decomposition of BTO and PTO there is an intermediate, $\text{BaTiO}_{2.5}(\text{CO}_3)_{0.5}$ or $\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5}$, which in the last step gives off carbon dioxide. There are differences, however, e.g., the presence of two carbonate intermediates in the BTO scheme and the true ionic character of the carbonate ion in $\text{BaTiO}_{2.5}(\text{CO}_3)_{0.5}$.

EXPERIMENTAL

For the synthesis of pure PTO a method was chosen, viz., precipitation from a 0.1 M ammonium titanyl oxalate solution by dropwise addition of a 0.2 M $\text{Pb}(\text{NO}_3)_2$ (Baker A.R.) solution. The ammonium titanyl oxalate was synthesized from TiCl_4 , oxalic acid and ammonia⁵. Both solutions (usually 150 and nearly 75 ml, respectively) had a pH 0.5 (HNO_3) and after adding the reagent the titanyl oxalate was slightly in excess. The precipitate was washed twice with 100 ml of 0.5 M HNO_3 , twice with 100 ml of water, and was air-dried. The method did not always result in good products, probably due to a thermodynamical preference for lead oxalate and titanium hydroxide precipitation⁶.

The product was analysed by the method of EDTA/ H_2O_2 addition (for Pb+Ti) or EDTA/lactic acid addition (for Pb) and back-titration of the excess EDTA with a standard lead solution⁷. Oxalate was determined in H_2SO_4 solution of the sample (containing PbSO_4) by permanganate titration.

X-ray powder patterns were taken with a Philips diffraction spectrometer PW-1310, with a Nonius Guinier-De Wolff camera and with a Nonius Guinier-Lenné high-temperature camera, all with $\text{CuK}\alpha_1$ radiation, $\lambda = 1.5405 \text{ \AA}$.

Thermal analyses were carried out with a DuPont 950/900 thermal analyzer, using TG and a quantitative DTA cell, with 10-mg samples, and with a Stanton TR-01 thermo-recording balance. The latter was applied for the use of greater quantities of material (more than 100 mg), for analyses of intermediate products at different decomposition stages. The heating rate was $10^\circ\text{C min}^{-1}$ and the thermal analyses were usually carried out in stationary air at ambient pressure. Infrared spectra were taken with a Perkin-Elmer 257 spectrometer (using KBr pellets).

RESULTS

When the stoichiometry was correct, a typical result of chemical analysis was: Pb: 39.32, Ti: 9.38, C_2O_4 : 33.60 weight %; calculated values for $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$: Pb: 39.91, Ti: 9.23, C_2O_4 : 33.91, H_2O : 13.88%.

Even with ideal stoichiometry, the product could be either X-ray amorphous or have one of two possible structures, of which the powder patterns are given in Table 1. The first one has been found, by computer analysis of the powder diffraction data⁸, to be orthorhombic with cell parameters $a = 11.555 \text{ \AA}$, $b = 16.274 \text{ \AA}$ and $c = 6.361 \text{ \AA}$. For the density was found 2.80 g cm^{-3} , which corresponds with four units per cell (theoretical density: 2.883 g cm^{-3}). The four units probably consist of

one tetramer, because the titanyl oxalate anion in the ammonium compound is a cyclic tetramer⁵, or possibly they consist of two dimeric units.

TABLE 1

X-RAY POWDER DIFFRACTION PATTERNS OF TWO FORMS OF $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$, $\text{CuK}\alpha_1$ RADIATION, $\lambda = 1.5405 \text{ \AA}$, GUINIER-DE WOLFF CAMERA

A. Orthorhombic						B. Unit cell unknown			
$a = 11.555 \text{ \AA}$, $b = 16.274 \text{ \AA}$, $c = 6.361 \text{ \AA}$, $Z = 4$, $d_x = 2.883 \text{ g cm}^{-3}$									
$d(\text{Å})$	I/I_1	hkl	$d(\text{Å})$	I/I_1	hkl	$d(\text{Å})$	I/I_1	$d(\text{Å})$	I/I_1
8.14	70	020	2.669	3	132	9.59	60	3.52	5
6.65	4	120	2.636	40	222	7.81	20	3.46	20
5.78	80	200	2.590	<1	251	7.63	10	3.36	7
5.57	35	101	2.560	80	341	7.37	1	3.32	10
5.27	<1	111	2.506	10	042	6.59	100	3.29	35
5.01	50	021	2.495	10	061	6.24	10	3.26	1
4.71	75	220	2.478	15	232	6.07	10	3.24	5
4.60	75	121	2.439	25	161	5.94	5	3.20	<1
4.14	<1	211	2.355	30	440	5.65	75	3.17	25
4.07	7	040	2.315	25	351	5.48	60	3.11	20
3.89	3	131	2.299	15	242	4.91	20	3.04	3
3.84	30	140	2.291	10	261	4.83	1	2.981	15
3.79	15	221	2.232	5	152	4.79	20	2.948	3
3.43	15	041	2.218	50	360	4.75	20	2.916	15
3.33	45	240	2.209	15	441	4.62	5	2.878	10
3.29	70	141	2.172	35	501	4.53	5	2.820	10
3.18	20	002	2.138	15	402	4.45	15	2.750	3
3.12	2	012	2.120	15	003, 412	4.37	10	2.726	2
3.05	70	321	2.099	100	521, 342	4.24	55	2.676	10
2.962	25	022	2.086	<1	103	4.19	45	2.641	10
2.948	10	241	2.068	40	113, 422	4.13	3	2.629	15
2.889	90	400	2.046	3	451	4.05	10	2.587	10
2.811	25	151	2.032	15	162	3.95	5	2.547	5
2.797	25	340	2.020	2	123	3.90	15	2.529	10
2.746	20	212	2.009	35	540	3.76	2	2.499	5
2.722	65	420	2.003	2	180	3.75	1	2.475	5
2.712	20	060	1.989	<1	432	3.58	40	2.393	10

The other compound, which was the one formed in most cases, was used for this thermal investigation. Cell parameters could not yet be found, because of the more diffuse powder diffraction lines. It has been observed, however, that TG and DTA curves for both samples were identical. A third compound, described by, e.g., Strizhkov et al.⁹ as $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot \text{H}_2\text{C}_2\text{O}_4 \cdot 4\text{H}_2\text{O}$, synthesized by using an extra equivalent of oxalic acid in the ammonium titanyl oxalate solution, could only once be isolated in a yield which was too small to substantiate this composition. However, the powder diffraction pattern was the same as that described by those authors.

Figure 1 gives the TG and DTA curves for PTO in air. Dehydration occurs in one step between 30 and 180°C (I). The water-free product is X-ray amorphous. As indicated in Table 2 the sample contained a slight excess of adsorbed water.

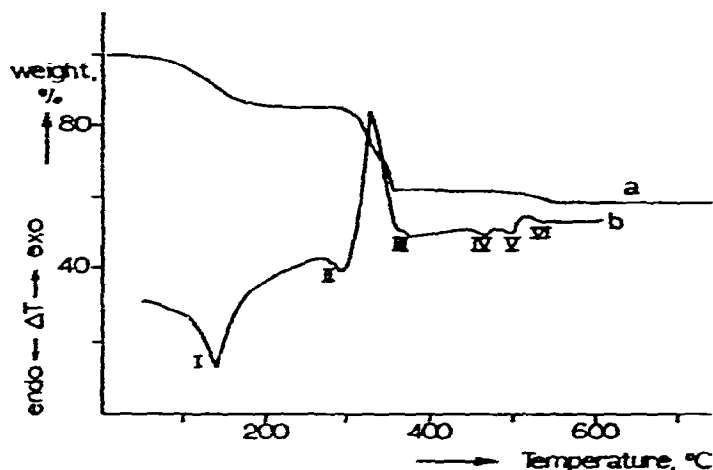


Fig. 1. TG and DTA curves of decomposition of $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$ in static air, atmospheric pressure. (a) TG, sample weight 10 mg (DuPont) or 130 mg (Stanton); (b) DTA, sample weight 10 mg (DuPont quantitative DTA accessory).

TABLE 2

DECOMPOSITION OF PTO IN STATIC AIR ATMOSPHERE, HEATING RATE $10^\circ\text{C min}^{-1}$

Decomposition step	Temp. range ($^\circ\text{C}$)	Peak temp. ($^\circ\text{C}$)	Weight loss (%)	Calc. weight loss	
				Compound	Percentage
I. Dehydration	30–180	138	14.5	$\text{PbTiOox}_2 \cdot 4.2\text{H}_2\text{O}$	
II. Endothermal oxalate decomposition	270–310	292	23.0	PbTiOox_2	14.47
III. Exothermal oxalate decomposition	310–390	328		$\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5} \cdot 0.1\text{C}$	23.11
IV. Phase transition	460–470	464			
V.VI. Carbonate decomposition and forming of PbTiO_3 lattice	490–530	495 511	4.3	$\text{PbTiO}_3 \cdot 0.1\text{C}$	4.21
Final carbon oxidation	400–900		0.2	PbTiO_3	0.23

The oxalate decomposition involves two steps (Fig. 1: II, III), one endothermic peak immediately followed by a large exothermic peak. This latter is absent when the compound is decomposed in nitrogen. Therefore, it is most probably caused by an overwhelming exothermic effect of CO oxidation in this temperature region (310–340°C), which is possibly catalyzed by small quantities of reactive PbTiO_3 or PbTiO_3 precursor¹⁰. As a small percentage of carbon is always found in the residue at the end the small exothermic effect of the Boudouard disproportionation of CO is masked by the other effects. The weight loss after the endothermic part was approximately 24% which is in agreement with the formula $\text{PbTiO}_{1.5}(\text{C}_2\text{O}_4)(\text{CO}_3)_{0.5}$ (75.95 weight % of original quantity) and a loss of 1 CO and $\frac{1}{2}\text{ CO}_2$ per mole. However, DTA and TG are not simultaneous so the initial temperatures of the reactions may be different and there is no definite proof that such an intermediate really exists. After the exothermic part the weight loss is 37.5% which is in agreement with $\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5}$, an oxide-carbonate. The CO_2 content of this compound was confirmed by chemical analysis. The composition turned out to be independent of the apparatus and sample weight, in contrast to what was found for $\text{TiO}_{2-x}(\text{CO}_3)_x$ during decomposition of the ammonium compound³.

Infrared spectra of PTO and of samples taken at different decomposition stages are shown in Fig. 2. This clearly shows that during decomposition of the oxalate typical oxalate vibration frequencies are gradually replaced by carbonate bands which are typical for an oxide-carbonate or an oxide which strongly adsorbs carbon dioxide (see, e.g., Turcotte et al.¹¹, who describe $\text{La}_2\text{O}_2(\text{CO}_3)$). The small differences with i.r.-spectra of lead-oxo-carbonates¹², however, do not exclude the possibility of a $\text{TiO}_2 \cdot (\text{PbO} \cdot \text{PbCO}_3)_{0.5}$ compound.

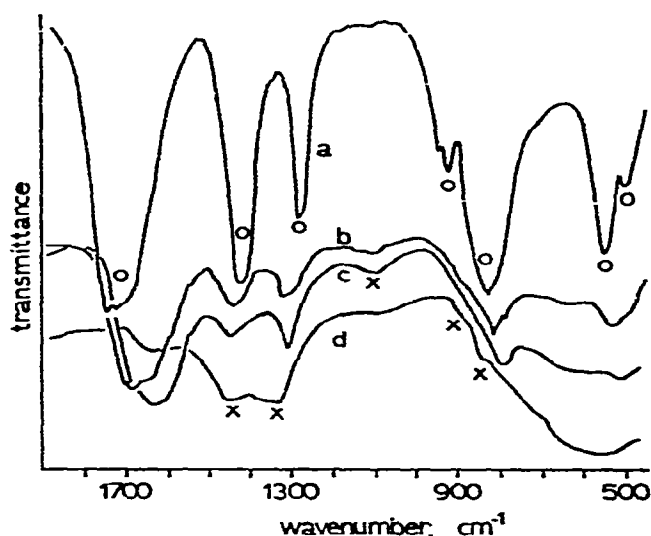


Fig. 2. Infrared spectra of $\text{PbTiO}(\text{C}_2\text{O}_4)_2 \cdot 4\text{H}_2\text{O}$ and decomposition products. \circ = oxalate, \times = carbonate. (a) = PTO; (b) = fraction of starting weight 0.77; (c) = 0.69; (d) = 0.615.

From an analysis of DTA curves of PbCO_3 , TiO_2 - PbCO_3 powder mixtures and a $\text{TiO}_2 \cdot x\text{H}_2\text{O}$ - PbCO_3 coprecipitate it was found that PbCO_3 always decomposes into the oxide before it reacts with TiO_2 . This is in agreement with results of Bergstein¹³, who reported PbTiO_3 is only formed above 500°C after PbCO_3 decomposition when PbCO_3 and anatase are the starting materials. For the above reasons it is unlikely that lead-oxo-carbonate is present in the intermediate.

The following step (Fig. 1: IV, V, VI) is the decomposition of the carbonate and formation of the PbTiO_3 (cubic phase) lattice. Actually IV (at 460°C) is a reversible phase-transition of X-ray amorphous PbTiO_3 (still containing carbonate and a small amount of carbon) to the cubic perovskite lattice. The reversible character only occurs once during cooling, later on the formed PbTiO_3 has its normal tetragonal/cubic perovskite transition at 490°C . The formation of cubic phase at lower temperatures is even more remarkable with high-temperature X-ray powder diffraction. At very slow heating rates the diffraction lines of the cubic phase appear after the X-ray amorphous state between 410 and 460°C . After cooling to 300°C cubic and tetragonal phase lines appear together during reheating until transition to cubic phase at about 460°C . This temperature changes to 490°C after another cooling and heating cycle. These results correspond with the transition which usually occurs at 490°C (Curie temperature) but which is lower during decomposition of PTO due to the highly defective lattice of the product. (See also for dependence of Curie temperature on substitution of dopes, e.g., Shirasaki¹⁴.)

The endothermic effect followed by an exothermic effect (V, VI) is probably caused by the (endothermic) carbonate decomposition, partly overlapped by exothermic lattice ordering.

The product then still contains about 0.4 weight % carbon which can be removed by prolonged heating at a higher temperature, preferable below 850°C to prevent PbO vaporization¹⁵.

Microscopic investigation has shown that the original PTO particles keep their macroscopic appearance (brick-like aggregates of about $5 \mu\text{m}$), during decomposition, which means that neither dehydration nor degassing cause fragmentation. Synthesis of PTO with small particles will therefore result in PbTiO_3 with small particles.

DISCUSSION

As mentioned for BTO by Gopalakrishnamurthy et al.⁴ the first endothermic peak during BTO decomposition in the temperature range of 200 - 250°C is ascribed to the loss of half a mole of carbon monoxide per mole of BTO. For PTO it is more reasonable to think of a process which involves more than only half a mole of CO . Some reasons are: the temperature region for PTO is higher, the endothermic effect is relatively much more pronounced and contains a slight shoulder in the peak and the total weight loss seems more than 17% for the formation of $\text{PbTiO}(\text{C}_2\text{O}_4)_{1.5}(\text{CO}_3)_{0.5}$. In the case of PTO it is possible that one mole of CO and half a mole of CO_2 evolve (total weight loss 23.5%), leading to $\text{PbTiO}_{1.5}(\text{C}_2\text{O}_4)(\text{CO}_3)_{0.5}$, with the i.r. spectrum

b in Fig. 2, but as stated before this is not necessarily a definite intermediate compound.

From this hypothetical state it would take a further loss of one mole of CO and one mole of CO₂ to get the stable oxide-carbonate PbTiO_{2.5}(CO₃)_{0.5}. This formula is preferred to PbTiO₃·0.5CO₂ because of the stable character of this product, which is not influenced by sample weight (10–130 mg), heating rate or type of apparatus.

As stated before, the formation of a separate lead-oxo-carbonate is not very probable. To support this a thermodynamic calculation of the free enthalpies of formation of PbTiO₃ from the oxides and from PbCO₃/TiO₂ or 0.5(PbO·PbCO₂)/TiO₂ was performed.

TABLE 3

FREE REACTION ENTHALPIES ΔG (kcal mol⁻¹) FOR SOME OXIDE/CARBONATE EQUILIBRIA

Reaction	Temperature (K)						
	300	400	500	600	700	800	900
(a) PbO (yellow) + TiO ₂ (anatase) ⇌ PbTiO ₃	-6.15	-7.10	-8.07	-9.06	-10.30	-11.60	-12.97
(b) PbCO ₃ + TiO ₂ ⇌ PbTiO ₃ + CO ₂	3.33	-1.46	-6.36	-11.35	-16.46	-21.68	-27.02
(c) ½(PbO·PbCO ₃) + TiO ₂ ⇌ PbTiO ₃ + ½CO ₂	0.20	-2.67	-5.60	-8.60	-11.67	-14.83	-18.10
(d) PbCO ₃ ⇌ PbO + CO ₂	9.48	5.64	1.71	-2.29	-6.35	-10.40	-14.62

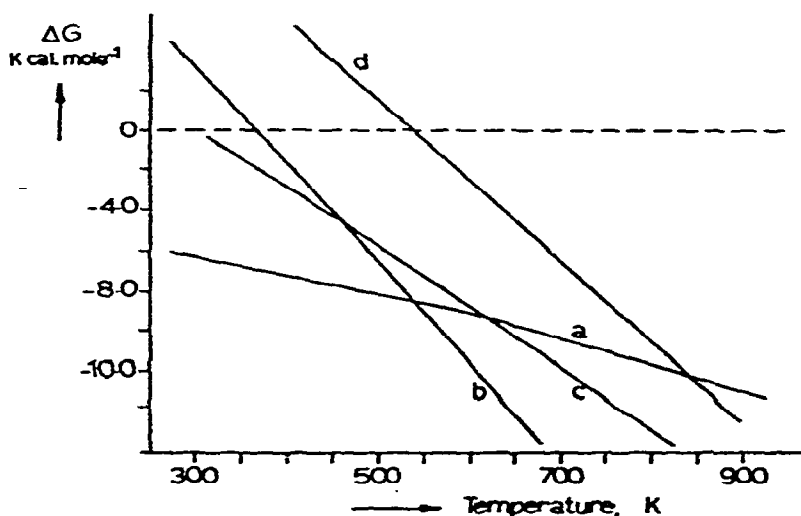


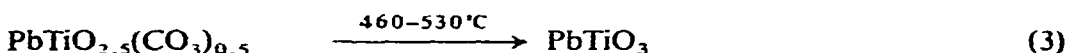
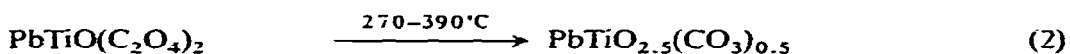
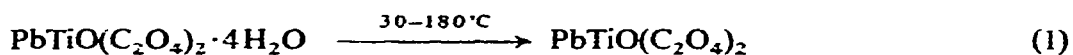
Fig. 3. Free reaction enthalpies ΔG as a function of temperature for oxide/carbonate equilibria (cf. data Table 3).

Table 3 and Fig. 3 have been calculated using the values of $\Delta H_{\text{PbTiO}_3}^0 = -273.7 \text{ kcal mol}^{-1}$ and $\Delta G_{\text{PbTiO}_3}^0 = -256.1 \text{ kcal mol}^{-1}$, taken from Zharkova¹⁶ and using the usual enthalpy-temperature relationship formulas. The same calculation has been performed with the values $\Delta H_{\text{PbTiO}_3}^0 = -277.9$ and $\Delta G_{\text{PbTiO}_3}^0 = -257.3 \text{ kcal mol}^{-1}$, taken from Schwitzgebel¹⁷, but as the results were not significantly different they have not been included. Other ΔH , ΔG , c_p or p data were from^{18,19} (TiO_2),²⁰ (PbO) and²¹ (PbCO_3 , $\text{PbO} \cdot \text{PbCO}_3$).

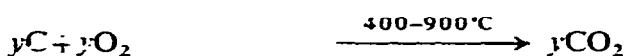
Figure 3 clearly shows (cross points a-b and a-c) that the lowest temperature at which PbTiO_3 is formed just as easily (thermodynamically) from $\text{PbCO}_3/\text{TiO}_2$ as from PbO/TiO_2 lies at about 540 K (270°C) and for $\text{PbO} \cdot \text{PbCO}_3$ at about 615 K (345°C). Above these temperatures the carbonates will decompose to PbO . However, the rate of formation of PbTiO_3 from the oxides only becomes significant at still higher temperatures.

This means that there is no real possibility for a compound $\text{TiO}_2 (\text{PbO} \cdot \text{PbCO}_3)_{0.5}$ to give PbTiO_3 in the temperature range of 460–530°C, in contrast to $\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5}$.

The results are in agreement with the following decomposition scheme:



and slowly



with $y = 0.1$

In the decomposition of BTO another intermediate is found by Gopalakrishnamurthy et al.⁴ prior to the final oxide-carbonate, viz., $\text{BaTiO}_{2.5}(\text{CO}_3)_{0.5} \cdot \frac{1}{2}\text{CO}_2$, which loses $\frac{1}{2}\text{CO}_2$ between 450 and 600°C, without any pronounced heat effect. Such an intermediate is not detected in the decomposition of PTO.

The presence of a 2350 cm^{-1} vibration band in the i.r. spectra of the intermediate in the BTO decomposition and preceding oxalate-carbonate stages and the absence thereof in the last oxide-carbonate is used as an argument for CO_2 adsorbed or retained by that intermediate. This is a free carbon dioxide band but often occurs in the spectra of many carbonates and carbonate complexes as well, and can thus not be used as a diagnostic band for adsorbed carbon dioxide.

In the case of PTO the band (outside the frequency range shown in Fig. 2) also appears in all oxalate-carbonate stages and remains in the oxide-carbonate, in contrast to what is observed for the analogous Ba compound. The infrared spectrum of the latter compound is also different, mainly in the nearly symmetrical 1450 cm^{-1} band and in the sharpness of other bands. analogous to the spectrum of an "ionic" carbonate such as PbCO_3 .

The exact nature of the oxide-carbonate in the PTO scheme is not known; for reasons mentioned before adsorbed or entrapped CO_2 is not considered probable. We feel that a PbTiO_3 lattice in which some oxygen ions are replaced by carbonate ions is a more probable structure.

Note added in proof

The nature of the intermediate $\text{PbTiO}_{2.5}(\text{CO}_3)_{0.5}$ was recently confirmed by a combined EGA-TG experiment on a sample of PTO which had been heated in air at 320°C during 48 h. The weight loss during the carbonate decomposition step in a gasflow of helium corresponded with 0.497 mole of CO_2 which was evolved exclusively. The experiment was kindly performed by Mr. A. Broersma, Dept. of Inorganic Chemistry, State University of Utrecht, in a combination of a Cahn RH electrobalance, a Leybold-Heraeus Topatron-B gas analysis instrument and a DuPont 900 thermal analyzer for temperature programming.

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REFERENCES

- 1 B. V. Strizhkov, A. V. Lapitskii and L. G. Vlasov, *Zh. Prikl. Khim.*, 33 (1960) 2009.
- 2 P. K. Gallagher and J. Thomson, Jr., *J. Am. Ceram. Soc.*, 48 (1965) 644.
- 3 G. M. H. van de Velde and P. J. D. Oranje, *Therm. Anal. Proc. Int. Conf., 4th 1974, Budapest*, Vol. 1, p. 851.
- 4 H. S. Gopalakrishnamurthy, M. Subba Rao and T. R. Narayanan Kutty, *J. Inorg. Nucl. Chem.*, 37 (1975) 891.
- 5 G. M. H. van de Velde, S. Harkema and P. J. Gellings, *Inorg. Chim. Acta*, 11 (1974) 243.
- 6 G. M. H. van de Velde and P. J. Gellings, to be published.
- 7 M. M. A. Perik and P. J. D. Oranje, *Anal. Chim. Acta*, 73 (1974) 402.
- 8 J. W. Visser, *J. Appl. Crystallogr.*, 2 (1969) 89.
- 9 B. V. Strizhkov, A. V. Lapitskii, Yu. P. Simanov and L. G. Vlasov, *Zh. Neorg. Khim.*, 7 (1962) 2181.
- 10 D. W. Johnson, Jr. and P. K. Gallagher, *Thermochim. Acta*, 7 (1973) 303.
- 11 R. P. Turcotte, J. O. Sawyer and L. Eyring, *Inorg. Chem.*, 8 (1969) 238.
- 12 S. D. Ross and J. Goldsmith, *Spectrochim. Acta*, 20 (1964) 781.
- 13 A. Bergstein, *Chem. Listy*, 50 (1956) 3.
- 14 S. I. Shirasaki and K. Takahashi, *J. Am. Ceram. Soc.*, 56 (1973) 430.
- 15 K. H. Hårdtl and H. Rau, *Solid State Commun.*, 7 (1969) 41.
- 16 L. A. Zharkova, *Zh. Fiz. Khim.*, 36 (1962) 1819.

- 17 K. Schwitzgebel, P. S. Lowell, F. B. Parsons and K. J. Sladek, *J. Chem. Eng. Data*, 16 (1971) 418.
- 18 O. Kubaschewski and E. L. Evans, *Metallurgical Thermochemistry*, Pergamon, London, 3rd ed., 1958.
- 19 M. Kh. Karapet'yants and M. L. Karapet'yants, *Thermodynamic Constants of Inorganic and Organic Compounds*, Ann Arbor-Humphrey Science Publishers, Ann Arbor, 1970.
- 20 C. B. Alcock and T. N. Belford, *Trans. Faraday Soc.*, 60 (1964) 822.
- 21 M. Centnerswer, G. Falk and A. Awerbusch, *Z. Phys. Chem.*, 115 (1925) 29.