

Thermochimica Acta 299 (1997) 123-126

thermochimica acta

Heat capacity of $LaMnO₃$

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Abstract

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The heat capacity of LaMnO_3 was measured by differential scanning calorimetry (DSC) in the (150–760) K range and by altemative current calorimetry (ACC) in the (77-250) K range. The heat capacity curve showed two thermal anomalies due to a structural phase transition at 735 K and a magnetic transition at 140 K. The enthalpy and entropy changes accompanying the magnetic transition are 220 J mol⁻¹ and 1.70 J K⁻¹ mol⁻¹, whereas those for the higher temperature phase transition were estimated to be 3.36 kJ mol⁻¹ and 4.2 J K⁻¹ mol⁻¹. The non-transitional heat capacity of LaMnO₃ was calculated by multiple regression analysis and is given by the following expressions.

$$
C_{p,m} = 2.82 + 0.874T - \frac{6.93 \times 10^4}{T^2} + 398 \times 10^{-7}T^3 - 3.31 \times 10^{-2}T^{3/2} \quad (77 \le T \le 298)
$$

$$
C_{p,m} = 96.73 + 5.36 \times 10^{-2}T - \frac{8.82 \times 10^5}{T^2} \quad (298 \le T \le 760)
$$

Keywords: ACC; DSC; Heat capacity; LaMnO₃; Phase transition

1. Introduction

 $LaMnO₃$ has an orthorhombically distorted perovskite-type structure at room temperature and undergoes a structural phase transition from the ortborhombic to the rhombohedral structure at high temperatures. The transition temperature depends on the oxygen nonstoichiometry [ll. A magnetic transition from antiferromagnetism to paramagnetism is observed near 140 K [2]. Also this transition temperature depend on the oxygen non-stoichiometry. The Gibbs free energy of decomposition or formation of LaMnO_3 has been reported by several authors [3-121. The heat capacity and the derived therrnodynamic properties above room temperature have been estimated by Yokokawa et al. [13,141. NO experimental heat-capacity data has

however been published. We report the results of an experimental study of the heat capacity of LaMnO_3 in the (77-760) K range.

2. Experimental

 $LaMnO₃$ was synthesized by a solid-state reaction method. Starting materials of $La₂O₃$ (Rare Metallic, Japan, of 99.99% purity) and Mn_2O_3 (Kojundo Chem. Lab., Japan, of 99.9% purity) were mixed in an equimolar ratio and pressed into a thin plate-shaped sample. The plate was sintered at 1523 K for 3 days under Ar flow. A single phase was identified by powder X-ray diffractometry and the lattice parameters were $a = 0.5739(1)$ nm, $b = 0.7698(1)$ nm $c = 0.5538(1)$ nm, respectively. These values were in good agreement with that reported by Norby

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paration method [16]. which gave $T_{trans} = 141$ K.

Two types of calorimeter were used for the measurement of the heat capacity of LaMnO_{3} ; the DSC-220C and DSC-200 models of Seiko Electronic, Japan, were used for the $(140-480)$ and $(300-780)$ K temperature regions, respectively. ULVAC-2000 model ACC of Ulvac, Japan, was used in the (77- 250) K temperature range. The accuracy of measurement with DSC was within $\pm 1.5\%$ at high temperatures, which was determined from the results of standard Al_2O_3 . Since data yielded by ACC are relative values of the heat capacity, it is necessary to convert these relative values to the absolute values of heat capacity. A temperature-dependent calibration factor, X , was calculated from the temperature region which overlapped with the regions studied by the DSC method. The detailed procedures of the heat-capacity measurement with DSC and ACC are described in a previous paper [17-19].

3. **Results and discussion**

The heat capacity of LAMnO_3 is shown in Fig. 1 together with the values estimated by Yokokawa et al. [13]. In the heat capacity curve, two thermal anomalies corresponding to phase transitions were observed at 140 and 735 K, respectively. For the lower-tem-

Fig. 1. The heat capacity of LaMnO₃. Solid and dashed lines indicate the present work and that of Yokokawa et al. [13], respectively.

et al. [15], i.e. $a = 0.57046(2)$, $b = 0.77029(4)$ and perature magnetic transition, the transition tempera $c = 0.55353(3)$ nm. The degree of oxygen non-stoi- ture of the present result is 140 K. The present work is chiometry x in LaMnO_{3+x} is below 0.01 in this pre- in good agreement with the result by Matsumoto [2],

> The present work is in good agreement with that of Yokokawa et al. [14] at near room temperature. The agreement is not that good at higher temperatures. Differences between the values of Yokokawa et al. [14] and those of the present work are 3.5 J K^{-1} mol⁻¹ at 600 K and 12.3 J K^{-1} mol⁻¹ at 750 K, respectively. This is due to the fact that the heat capacity reported by Yokokawa et al. [13] was estimated by a modified Neumann-Kopp law. Furthermore, the transition temperature is 735 K in the present work, whereas 675 K is reported by Yokokawa et al. [13]. This difference is considered to be due to differences in oxygen nonstoichiometry of the specimens, as the transition temperature decreases with an increase of oxygen nonstoichiometry.

> For the lower temperature transition, a base line shown as a solid line in Fig. 2 is obtained by multiple regression analysis. From the area between peak and the base line, the enthalpy and entropy changes accompanying the magnetic transition at 140 K are 220 J mol⁻¹ and 1.70 J mol⁻¹ K⁻¹, respectively. Corresponding enthalpy and entropy changes accompanying the structural transition at 735 K were calculated at 3.36 kJ mol⁻¹ and 4.62 J mol⁻¹ K⁻¹, respectively. Two non-transitional heat-capacity curves of LaMnO_3 were estimated by multiple regression analysis as

Fig. 2. The lower peak region of heat capacity of LaMnO₃. Circle and solid line indicate the observed value and the base line, respectively.

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Table 1
Molar heat capacity of the two peak regions of LaMnO₃ Molar heat capacity of the two peak regions of LaMnO, shown in the following, and the smoothed molar heat capacity is tabulated in Table 1:

$$
C_{p,m} = 2.82 + 0.874T - \frac{6.93 \times 10^4}{T^2} + 398 \times 10^{-7}T^3 - 3.31 \times 10^{-2}T^{3/2} (77 \le T \le 298) C_{p,m} = 96.73 + 5.36 \times 10^{-2}T - \frac{8.82 \times 10^5}{T^2} \quad (298 \le T \le 760)
$$

The higher approximation of the Debye function is expressed by an equation such as

$$
C_{\rm v}=3nR\bigg(1-\frac{\theta_{\rm D}^2}{20}\frac{1}{T^2}\bigg)
$$

On the other hand, the temperature-dependence equation of the heat capacity is generally written as follow.

$$
C_{\rm p} \equiv a + bT - \frac{c}{T^2} = a\left(1 - \frac{c}{a}\frac{1}{T^2}\right) + bT
$$

Combining the coefficients of T^2 term from the two foregoing equations, the calculated Debye temperature 427 K was derived, in comparison with 460 K obtained from the heat capacity of $La_{0.8}Ca_{0.2}MnO₃$ from 110 to 300 K by Tamura and Kuriyama [20]. Comparing these two results, the estimated Debye temperature of $LaMnO₃$ (430 K) is a reasonably good derivation.

Acknowledgements

The authors thank Fuji Electrochemical Co., Ltd. for the availability of ACC apparatus. The present work was supported by a Grant-Aid for Scientific Research on Priority Areas "New Development of Rare-earth Complexes" No. 08220233 by the Ministry of Education, Science, Sports and Culture.

References

- [1] A. Wold and R.J. Amott, J. Phys. Chem. Solids, 9 (1959) 176.
- [2] G. Matsumoto, IBM J. Res. Develop., 14 (1970) 258.
- [3] K. Kamata, T. Nakajima, T. Hayashi and T. Nakamura, Mater. Res. Bull., 13 (1978) 49.
- [4] T. Nakamura, G. Petzow and L.J. Gaucker, Mater. Res. Bull., 14 (1979) 649.
- [5] Yu.P. Vorobev, A.A. Novlev, S.A. Leontev, A.N. Men, S.A. Prokudina and Y.S. Rubinchik, Iz. Akad. Nauk SSSR, Neorg. Mater., 15 (1979) 1449.
- [6] O.M. Sreedharan, R. Pankajavalli and J.B. Gnanamoorthy, High Temp. Sci., 16 (1983) 251.
- [7] M.L. Borlera and F. Abbattista, J. Less-Comm., Metals., 92 (1983) 55.
- [8] N. Kamegashira, Y. Miyazaki and Y. Hiyoshi, Mat. Lett., 2 (1984) 194.
- [9] V.A. Cherepanov, L.Y. Barkhatova and A.N. Petrov, J. Phys. Chem. Solids, 55 (1994) 229.
- [10] J.A.M. van Roosmalen and E.H.P. Cordfunke, J. Solid State Chem., 110 (1994) 113.
- [11] R. Hildrum, M. Brustad, W. Changzhen and O. Johnnesen, Mater. Res. Bul]., 29 (1994) 851.
- [12] T. Atsumi, T. Ohgushi and N. Kamegashira, J. Alloys Comp., 238 (1996) 35.
- [13] H. Yokokawa, N. Sakai, T. Kawada and M. Dokiya, J. Am. Ceram. Soc., 73 (1990) 649.
- [14] H. Yokokawa, N. Sakai, T. Kawada and M. Dokiya, J. Electrochem. Soc., 138 (1991) 2719.
- [15] P. Norby, I.G. Krogh Andersen, E. Krogh Andersen and N.H. Andersen, J. Solid State Chem., 119 (1995) 191.
- [16] N. Kamegashira, Y. Miyazaki and H. Yamamoto, Mater. **Chem.** Phys., 11 (1984) 187.
- [17] H. Satoh, M. Horikawa and N. Kamegashira, J. Alloys Comp., 192 (1993) 99.
- [18] H. Satoh, M. Horikawa and N. Kamegashira, Netsu Sokutei, 20 (1993) 193.
- [19] H.W. Xu, J. Iwasaki, T. Shimizu, H. Satoh and N. Kamegashira, J. Alloys Comp., 221 (1995) 274.
- [20] S. Tamura and M. Kuriyama, Phys. Lett., 70A (1979) 469.