

## Studies of the reduction and reoxidation processes of Nd<sub>4</sub>PdO<sub>7</sub>

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### Abstract

It has been demonstrated that Nd<sub>4</sub>PdO<sub>7</sub>, when exposed to a gas mixture of CO and He, fully decomposes to Nd<sub>2</sub>O<sub>3</sub> and nano-sized Pd-particles, which are found on the surfaces of the micron-sized Nd<sub>2</sub>O<sub>3</sub> particles. This material, known to exhibit catalytic activity for oxidation of CO and hydrocarbons, and reduction of NO, can be reoxidised to Nd<sub>4</sub>PdO<sub>7</sub> by heat treatment at 1085°C in air for a short period of time, i.e. when utilised as catalyst this material can easily be regenerated if it has become aged. The reduced material has been studied in scanning and transmission electron microscopes (SEM and TEM) equipped with element analysis facilities, and characterised by X-ray powder diffraction (XRPD) studies. A wet-chemistry method for extraction of the nanoparticles formed has been developed, based on selective dissolution of Nd<sub>2</sub>O<sub>3</sub>, and the bare Pd-particles have been characterised by XRPD, SEM and TEM studies. The reduction and reoxidation of Nd<sub>4</sub>PdO<sub>7</sub> material was monitored in a thermogravimetric setup. Nd<sub>4</sub>PdO<sub>7</sub> decomposed in air to Nd<sub>2</sub>O<sub>3</sub> and Pd at 1185°C, which is ≈50°C higher than the previously reported decomposition temperature. © 1998 Elsevier Science B.V.

*Keywords:* Nd–Pd oxide; Thermal analysis; Reduction; Oxidation; Nanoparticles

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### 1. Introduction

In a previous article [1], a La<sub>4</sub>PdO<sub>7</sub>-based catalyst was reported to acquire a reproducible catalytic three-way activity (i.e. simultaneous oxidation of CO and hydrocarbons and reduction of NO) after several light-off runs in a simulated car exhaust gas, and the catalytic effect was attributed to the formation of nano-sized Pd-particles on the surface of La<sub>2</sub>O<sub>3</sub>. It was also found that this reduction process was reversible by heat treatment in air at elevated temperatures, which reformed La<sub>4</sub>PdO<sub>7</sub>. Utilising this property, catalysts based on this material could be easily regenerated by heat treatment in air followed by reactivation.

A possible drawback of the La<sub>4</sub>PdO<sub>7</sub> material could be the moisture sensitivity of La<sub>2</sub>O<sub>3</sub>, since our previous studies have shown that the formation of La(OH)<sub>3</sub> cracks the material and embed the catalytically active Pd-particles. We believed that this could severely affect the reversibility as well as the activity and lifetime of the catalyst.

As Nd<sub>2</sub>O<sub>3</sub> is known to be much less sensitive to moisture, we prepared the neodymium analogue to La<sub>4</sub>PdO<sub>7</sub>, i.e. Nd<sub>4</sub>PdO<sub>7</sub>, and in this article we discuss the chemical and morphological changes involved in the reduction and reoxidation processes of Nd<sub>4</sub>PdO<sub>7</sub>. Monophasic Nd<sub>4</sub>PdO<sub>7</sub> has thus been reduced in a flow of helium and carbon monoxide and then reoxidised in air. These processes were monitored in a thermogravimetric setup, and the products were characterised by their X-ray powder

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diffraction (XRPD) patterns. Analytical transmission and scanning electron microscopes (ATEM and SEM) equipped with elemental analysis facilities were used to examine the morphology and composition of selected samples, and a wet-chemistry method for extraction and characterisation of the formed nano-sized Pd-particles has been developed. Preliminary studies of the catalytic activity of the reduced  $\text{Nd}_4\text{PdO}_7$  material showed that it can simultaneously oxidise CO and hydrocarbons and reduce NO when exposed to slightly reducing car exhaust conditions [2]. More detailed studies of the catalytic activity of the reduced  $\text{Nd}_4\text{PdO}_7$  material are presently being performed.

## 2. Experimental

### 2.1. Materials

The following starting materials were used:  $\text{Nd}_2\text{O}_3$  (99.9%, Johnson Matthey), anhydrous  $\text{PdCl}_2$  (purum, Fluka), HCl (fuming 37% p.a., Merck), The gases He, CO and compressed air used for the reduction and reoxidation of  $\text{Nd}_4\text{PdO}_7$  were supplied by AGA Gas AB.

### 2.2. Preparation of $\text{Nd}_4\text{PdO}_7$ and reduced material

The  $\text{Nd}_4\text{PdO}_7$  starting material was prepared as follows: (i) neodymium oxide was calcined overnight at  $1000^\circ\text{C}$  and dissolved in fuming hydrochloric acid together with palladium chloride in an Nd/Pd molar ratio of 4 : 1, and the solution was boiled to dryness; (ii) the residue was slowly heated to  $550^\circ\text{C}$ , held at this temperature for 2 h and then ground in a mortar; and (iii) the temperature was then slowly raised to  $1000^\circ\text{C}$ , and the sample was heat treated at this temperature for 7 days.

$\text{Nd}_4\text{PdO}_7$  was then reduced in a tube furnace at  $750^\circ\text{C}$  with a flowing gas mixture of 16 ml CO and 1000 ml He per minute. An alumina container with ca. 150 mg  $\text{Nd}_4\text{PdO}_7$  was placed in the furnace and the temperature was raised to  $750^\circ\text{C}$  during 25 min. After 25 min of isothermal heat treatment the furnace was cooled to room temperature and the sample was transferred to a desiccator.

### 2.3. Thermogravimetry

The reduction and reoxidation of  $\text{Nd}_4\text{PdO}_7$  material was studied in thermogravimetric (TG) setups (Setaram TAG24 and Perkin–Elmer TGA7). Typically, 50 mg samples were used, and heating and cooling rates of  $10^\circ/\text{min}$ . A CO/He gas mixture was used for the reduction, and the oxidation was performed in air.

### 2.4. Characterisation

All samples were characterised by their XRPD patterns obtained in a Guinier–Hägg camera, using  $\text{CuK}\alpha_1$  radiation and Si as internal standard. The photographs obtained were evaluated in a computerised scanner system [3], and the records obtained for  $\text{Nd}_2\text{O}_3$  and Pd were matched with tabulated JCPDS data. We have recently solved the structure of  $\text{Nd}_4\text{PdO}_7$  [4], and the diffraction patterns of the starting and reoxidised material have been indexed with use of this unit cell. The morphologies of selected samples were studied in a scanning electron microscope (SEM, JEOL 880) and in an analytical transmission microscope (ATEM, JEOL 2000FX), both furnished with energy-dispersive spectrometers (EDS, Link AN-10 000) for elemental analysis.

## 3. Results and discussion

### 3.1. Thermogravimetry

The TG curve of  $\text{Nd}_4\text{PdO}_7$  heated to  $750^\circ\text{C}$  in a CO/He gas mixture is shown in Fig. 1. The reduction starts at  $\approx 380^\circ\text{C}$  and is finished at ca.  $600^\circ\text{C}$  with a total mass loss of 2.2%, which is in fair agreement with the calculated value, 2.0%. The DTG (%/min) curve given in the Fig. 1 suggests that the reduction occurs in one step. The reduced sample was reheated in air to  $1085^\circ\text{C}$  and held at this temperature for half an hour; it was then heated to  $1350^\circ\text{C}$  and held at this temperature for half an hour before quenching to room temperature in argon atmosphere. The TG and DTG curves thus obtained are shown in Fig. 2. The oxidation process starts ca.  $260^\circ\text{C}$  and is finished at  $800^\circ\text{C}$ . This process clearly occurs in at least two steps, the first one occurring at temperatures ranging from 260 to

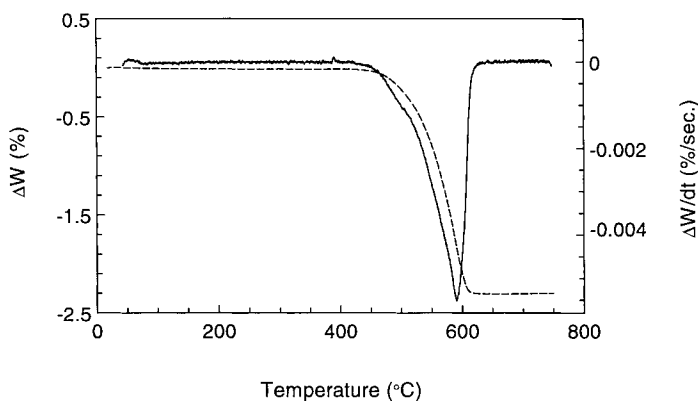


Fig. 1. The TG (· · ·) and DTG curves of  $\text{Nd}_4\text{PdO}_7$  heated to  $750^\circ\text{C}$  in a CO/He gas mixture, using a heating rate of  $10^\circ\text{C}/\text{min}$ .

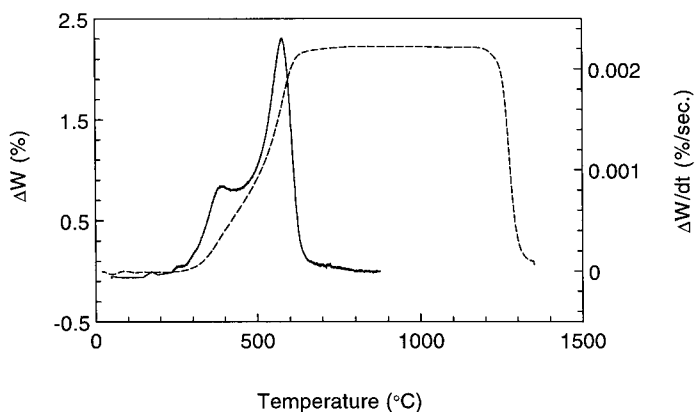


Fig. 2. The TG (· · ·) and DTG curves of  $\text{Nd}_4\text{PdO}_7$  heated in air to  $1085^\circ\text{C}$  and held at this temperature for 1 h, and then heated to  $1350^\circ\text{C}$  and held at this temperature for 0.5 h.

$450^\circ\text{C}$  and the second one from  $450$  to  $800^\circ\text{C}$ . A total mass increase of 2.2% is observed, in good agreement with the mass-loss measurements, but still somewhat larger than the calculated value, 2.05%. Prolonged heat treatment at  $1085^\circ\text{C}$  did not give rise to any further mass increase. The phase analysis studies described below showed that  $\text{Nd}_4\text{PdO}_7$  is rapidly formed at this temperature when reduced material is heat treated in air. Around  $1185^\circ\text{C}$ ,  $\text{Nd}_4\text{PdO}_7$  starts to decompose into neodymia and Pd, and a total mass loss of 2.1% is found (see Fig. 2). McDaniel and Schneider [5] have reported the decomposition temperature in air of  $\text{Nd}_4\text{PdO}_7$  to be  $1135^\circ\text{C}$ , i.e. substantially lower than our observation.

### 3.2. Phase analysis

The XRPD data of the starting material (Fig. 3(a)) was matched with the diffraction pattern of  $\text{Nd}_4\text{PdO}_7$  given in Ref. [4], and the material was found to contain, besides  $\text{Nd}_4\text{PdO}_7$ , some  $\text{Nd}_2\text{O}_3$  (1.9%) and  $\text{Nd}_2\text{Pd}_2\text{O}_5$  (2.2%), the percentages being Rietveld-refined phase fractions. The reduced material gave rise to reflections characteristic of hexagonal  $\text{Nd}_2\text{O}_3$  (Fig. 3(b)), and a single very weak peak indicating the presence of palladium. Line-broadening analysis [6] of the strongest  $\text{Nd}_2\text{O}_3$  reflection indicated that the average crystallite size of  $\text{Nd}_2\text{O}_3$  should be ca.  $0.1\ \mu\text{m}$ , which is five-to-ten times smaller than that observed in

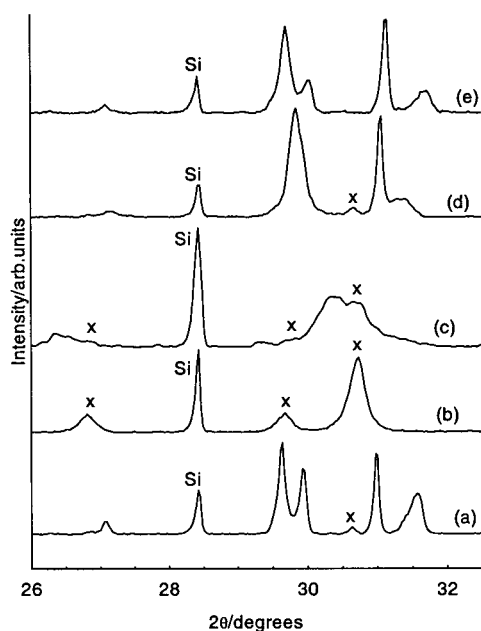


Fig. 3. X-ray diffraction patterns of (a)  $\text{Nd}_4\text{PdO}_7$ , (b) reduced material (reflections denoted by  $x$  are ascribed to  $\text{Nd}_2\text{O}_3$ ), (c) reduced material heated in air for 10 min at  $670^\circ\text{C}$ , (d) for 10 min at  $1085^\circ\text{C}$ , and (e) 720 min at  $1085^\circ\text{C}$ .

the SEM.  $\text{Nd}_4\text{PdO}_7$  did not exhibit any line broadening. Oxidised samples yielded diffraction patterns similar to that of  $\text{Nd}_4\text{PdO}_7$ .

In order to study the oxidation mechanism in greater detail, reduced samples were heat treated in the Perkin–Elmer TG-unit, as described below, using a heating rate of  $200^\circ/\text{min}$  to  $670^\circ\text{C}$ , and were held for 10 min at this temperature. The X-ray powder diffractogram (see Fig. 3(c)) showed that this sample contained  $\text{Nd}_2\text{O}_3$ , but it also exhibited a set of reflections at somewhat lower  $2\theta$ -values than those of  $\text{Nd}_2\text{O}_3$ . At least six of the original  $\text{Nd}_2\text{O}_3$  reflections

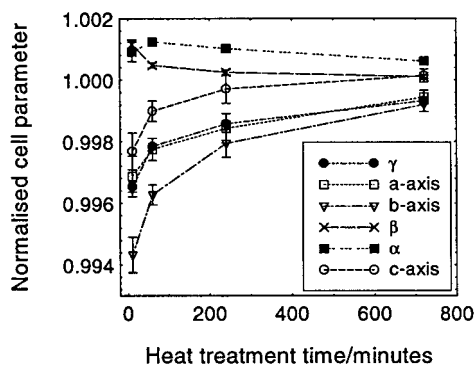


Fig. 4. Unit cell parameters of  $\text{Nd}_4\text{PdO}_7$  obtained from reduced material reheated in air at  $1085^\circ\text{C}$  for 10, 60, 240 and 720 h, respectively, plotted vs. time. The parameters are normalised to those of the starting material.

had counterparts of comparable intensity at lower  $2\theta$ -values. These observations suggest that  $\text{Nd}_4\text{PdO}_7$  is formed via some intermediate compound on the oxidation of  $\text{Nd}_2\text{O}_3$  and Pd (see also above and below).

The X-ray diffraction patterns of the reduced material, heat treated at  $1085^\circ\text{C}$  in air, for periods ranging from 10 min to 12 h revealed that the oxidation process at this temperature is very rapid, as an almost monophasic  $\text{Nd}_4\text{PdO}_7$  sample was obtained after only 10 min. Thus only a minor amount of  $\text{Nd}_2\text{O}_3$  was found in this sample, as seen in Fig. 3(d), but no  $\text{Nd}_2\text{O}_3$  remained after 12 h of heat treatment (see Fig. 3(e)). The lattice parameters of  $\text{Nd}_4\text{PdO}_7$  formed after 10 min, 1 h, 4 h and 10 h of heat treatment and those of the starting material are very similar (see Table 1). A small increase in the  $a$ -,  $b$ -,  $c$ - and  $\gamma$ -parameters can be discerned with increasing heating time, though, whereas there is no significant variation in the  $\alpha$ - and  $\beta$ -angles, as seen in Fig. 4 where the cell parameters, normalised to those of the starting material, are plotted as a function of heating time.

Table 1

Unit cell parameters of reduced  $\text{Nd}_4\text{PdO}_7$  reoxidised at  $1085^\circ\text{C}$  for 10 min (labelled O10), 60 min (O60), 240 min (O240) and 720 min (O720) and those of the starting material (OS)

Sample	$a$ (Å)	$b$ (Å)	$c$ (Å)	$\alpha$	$\beta$	$\gamma$
O10	15.918(10)	7.150(4)	6.898(4)	96.40(3)	131.78(2)	121.03(2)
O60	15.932(5)	7.164(2)	6.907(2)	96.43(1)	131.69(1)	121.19(1)
O240	15.943(7)	7.176(3)	6.912(3)	96.41(1)	131.66(1)	121.28(1)
O720	15.959(2)	7.185(1)	6.915(1)	96.37(1)	131.638(6)	121.365(8)
OS	15.968(3)	7.191(1)	6.914(1)	96.311(7)	131.627(5)	121.452(5)

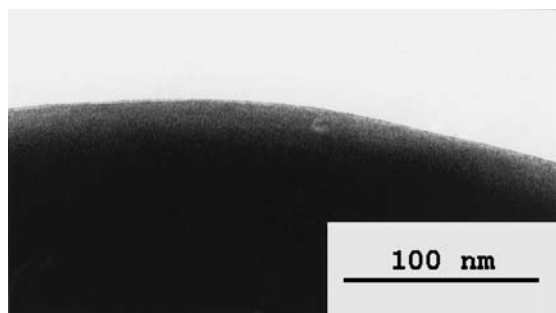


Fig. 5. An ATEM micrograph of an  $\text{Nd}_4\text{PdO}_7$  crystal showing a characteristically smooth surface.

The XRPD pattern of the sample heat treated at  $1350^\circ\text{C}$  and quenched to room temperature contained reflections which all could be ascribed to  $\text{Nd}_2\text{O}_3$  and Pd.

### 3.3. ATEM studies

The starting material consisted of micron-sized particles with smooth and clean surfaces, having Pd and Nd contents of typically 20 and 80%, respectively, in agreement with the  $\text{Nd}_4\text{PdO}_7$  formula. A micrograph of a characteristic  $\text{Nd}_4\text{PdO}_7$  crystal is reproduced in Fig. 5.

The reduced material showed crystalline grains in the same size range as the starting material, and elemental area analysis yielded Nd : Pd ratios of 4 : 1. Closer inspection of these grains revealed that semi-spherical nanoparticles in the 5–20 nm size range were distributed over almost all surfaces (see Fig. 6), and no grain without such particles could be found. These nanoparticles were typically separated

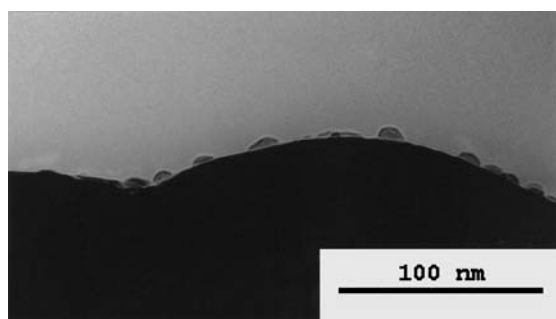


Fig. 6. An ATEM micrograph of an  $\text{Nd}_2\text{O}_3$  crystal with semi-spherical nano-sized Pd-particles on its surface.

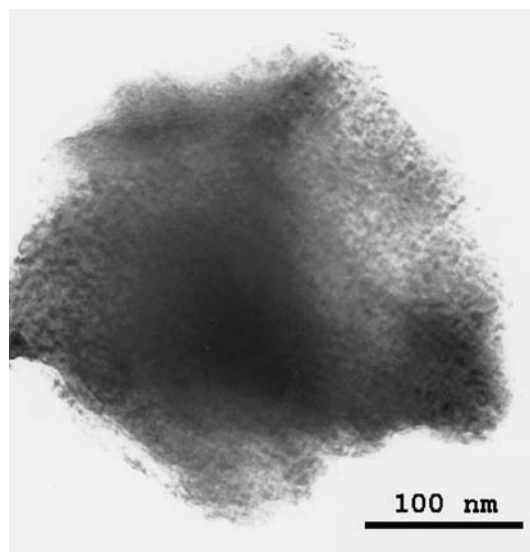


Fig. 7. An ATEM micrograph of a fragment of an unknown phase formed after oxidation of reduced material for 10 min in air at  $670^\circ\text{C}$ .

by 20 to 100 nm, and spot EDS analysis of such a nanoparticle gave rise to a higher Pd signal than the area EDS analysis mentioned above. These results, together with the findings from the XRPD analysis, suggest that the micron-sized grains are crystallites of  $\text{Nd}_2\text{O}_3$  and that the nanoparticles are Pd or PdO. Pd is the most likely candidate, according to the X-ray findings and taking in account the reducing conditions used in the preparation of the material.

The material oxidised at  $670^\circ\text{C}$  for 10 min showed two different types of grains. Some were very similar to the reduced material, having the same size, the same type of semi-spherical nanoparticles on their surfaces and 80/20 Nd/Pd overall composition, whereas other grains showed a totally different morphology as evidenced from Fig. 7. It can be seen that these grains are smaller, exhibit a rougher surface and are also more transparent than the others. Nevertheless, they have Nd/Pd ratios of 80/20 and were found to be crystalline, i.e. their electron diffraction patterns could be recorded.

The materials oxidised at  $1085^\circ\text{C}$  for 10 min and 12 h both totally lacked the semi-spherical nanoparticles and showed a morphology very similar to that of  $\text{Nd}_4\text{PdO}_7$ , i.e. dark smooth micron-sized grains with 80/20 Nd/Pd compositions. On some grain surfaces,

brighter appearing fragments containing 100% Nd were found and believed to be  $\text{Nd}_2\text{O}_3$  or  $\text{Nd}(\text{OH})_3$ .

ATEM studies of the sample heat treated at  $1350^\circ\text{C}$  in air revealed the presence of micron-sized grains of  $\text{Nd}_2\text{O}_3$  and Pd, showing that nano-sized Pd-particles are not produced by decomposition of  $\text{Nd}_4\text{PdO}_7$  in air at elevated temperatures.

### 3.4. Extraction of nano-sized Pd-particles

To be able to extract and characterise the nano-sized particles formed and to verify that they are metallic, a wet-chemistry extraction method was developed. It is based on the findings that PdO and palladium metal are insoluble in oxygen free 1 M HCl, whereas  $\text{Nd}_2\text{O}_3$  and  $\text{Nd}_4\text{PdO}_7$  are soluble. By exposing the reduced starting material to this solvent, all  $\text{Nd}_2\text{O}_3$  and unreacted  $\text{Nd}_4\text{PdO}_7$  (if present) will be dissolved, while PdO (if present) and Pd will remain unaffected. In order to verify that the liquid did not contain any  $\text{Pd}^{2+}$ -ions, a simple gravimetric method was applied. A series of standard solutions containing various amounts of  $\text{Pd}^{2+}$ -ions were thus prepared and  $1\text{ cm}^3$  samples of these solutions were evaporated to dryness in an oven at  $120^\circ\text{C}$ . The residue was dissolved in  $1\text{ cm}^3$  oxygen free 1M HCl (prepared as described below) and this solution was mixed with equal amounts of fuming HCl and 0.13 M  $\text{Na}_2\text{S}$ . A brownish precipitate of PdS could easily be detected, by visual inspection, down to a concentration of  $61.7\ \mu\text{M}$   $\text{Pd}^{2+}$ -ions, which corresponds to 6.6 ppm of  $\text{Pd}^{2+}$ -ions.

The reduced samples were tested in a similar way: (i) oxygen free 1 M HCl was prepared by boiling ca.  $50\text{ cm}^3$  of fuming HCl till the azeotrope was reached, and then for an additional 10 min, whereupon  $5.0\text{ cm}^3$  of this solution was diluted with 25 ml of deionised water which previously had been deoxygenised by bubbling nitrogen through it for 35 min; (ii) 4.85 mg of the reduced sample was mixed with  $1.0\text{ cm}^3$  of the diluted acid in a test tube, which was then flushed with nitrogen, sealed and ultrasonically treated for 1 min. The solution was centrifuged after standing for 30 min, and the supernatant liquid was transferred to another test tube. No brownish precipitate of PdS was found when fuming HCl and  $\text{Na}_2\text{S}$  were added to this liquid.

The crystalline nature, morphology and composition of the undissolved black material were revealed

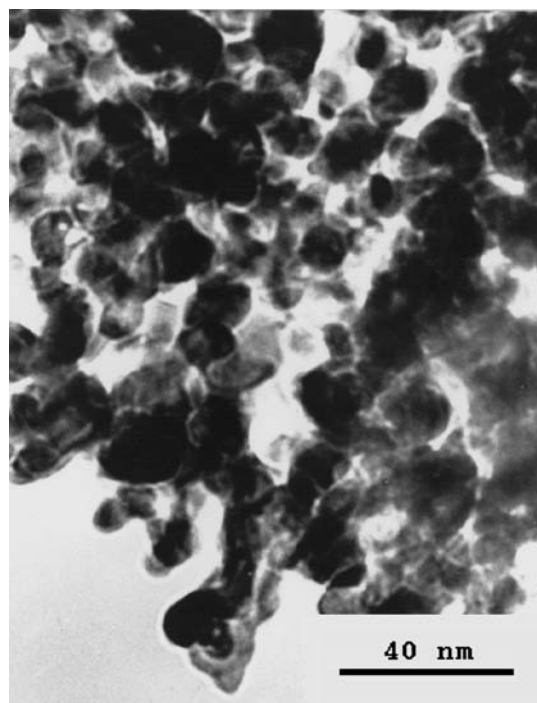


Fig. 8. An ATEM micrograph of the extracted nano-sized Pd-particles.

by ATEM/EDS, SEM/EDS and XRPD studies. The material was found to consist of almost spherical particles of sizes in the 5–25 nm range (see Fig. 8). No other metal besides Pd could be detected, but the sample was found to contain minor amounts of chlorine probably originating from the acid. The X-ray powder pattern contained broad reflections which could all be assigned to Pd. No PdO could thus be found.

Our studies thus prove that the reduced material does not contain any  $\text{Nd}_4\text{PdO}_7$  or PdO.

Line-broadening calculations yielded a crystallite size of  $24.4 \pm 2.1$  nm, according to the Sherrer half-width formula based on the average broadening of three reflections [6]. However, an average particle diameter of  $10.7 \pm 3.1$  nm was found by measuring the diameters of 51 particles depicted in an ATEM micrograph. It is interesting to note that the Pd-particles obtained are almost spherical, whereas in micrographs, such as that given in Fig. 6, they appear to be semi-spherical. We have seldom observed any truly spherical particles on the surfaces of  $\text{Nd}_2\text{O}_3$ . This

might imply that approximately one half of the Pd particle is buried in the  $\text{Nd}_2\text{O}_3$  particle and, if so, the nano-sized particles ought to be well anchored to their hosts. Normally noble-metal-based car exhaust catalysts are prepared by depositing the noble metal onto a washcoat by a dip-coating procedure. After having been used for some time, larger aggregates of the noble metals can be found, implying that the efficiency of the catalysts has decreased, i.e. the catalyst has aged. The tendency to form larger aggregates of Pd-particles in our material is expected to be less, because the nano-sized Pd-particles are much better anchored to their hosts than in a catalyst prepared via the dip coating procedure. If larger aggregates are formed our catalyst can easily be regenerated using the oxidation–reduction procedure outlined above.

#### 4. Concluding remarks

It has been demonstrated that  $\text{Nd}_4\text{PdO}_7$  grains of micrometer size fully decompose to  $\text{Nd}_2\text{O}_3$  and nano-sized Pd-particles when exposed to a gas mixture of carbon monoxide and helium at  $750^\circ\text{C}$ , and that the Pd-particles are found at the surfaces of the  $\text{Nd}_2\text{O}_3$  particles. The  $\text{Nd}_2\text{O}_3$  grains formed are of the same size as those of the starting material,  $\text{Nd}_4\text{PdO}_7$ . A wet-chemistry method for extraction of the nanoparticles by selective dissolution of  $\text{Nd}_2\text{O}_3$  has been developed and used to characterise these particles and to prove that decomposition of  $\text{Nd}_4\text{PdO}_7$  yields  $\text{Nd}_2\text{O}_3$  and Pd. Heat treatment of the reduced material in air for 10 min at  $670^\circ\text{C}$  gave indications of formation of

some metastable Nd–Pd compound, possibly having the morphology shown in Fig. 6. When the reduced material is heat treated at  $1085^\circ\text{C}$ , a material with the same morphology, composition and X-ray diffraction pattern as  $\text{Nd}_4\text{PdO}_7$  is obtained only after 10 min. This material is totally free of the Pd nanoparticles found in the reduced material.  $\text{Nd}_4\text{PdO}_7$  was found to be stable in air up to  $1185^\circ\text{C}$ , which is  $\approx 50^\circ\text{C}$  higher than the previously reported temperature. The catalytic activity of the reduced  $\text{Nd}_4\text{PdO}_7$  material for oxidation of CO and hydrocarbons and reduction of NO is presently being studied in some detail.

#### Acknowledgements

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