

Thermochimica Acta 318 (1998) 125-129

thermochimica acta

An investigation of microsilica by thermoanalytical methods

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Received 15 September 1997; accepted 21 January 1998

Abstract

It has been shown that the Meraker-type microsilica with $SiO₂$ -content of ca. 98.5wt% behaves differently from other types of microsilica investigated earlier at our Institute since it does not follow the normal transformation sequence:

microsilica \rightarrow cristobalite \rightarrow tridymite

when it is heated at ca. 1000° C. Due to the higher purity of this type of microsilica, it transforms directly either to quartz or cristobalite. However, if small amounts of Li₂O were added to the microsilica, the following transitions could be observed by using DSC:

low – tridymite $\stackrel{394 \text{ K}}{\rightarrow}$ high – tridymite $\stackrel{444 \text{ K}}{\rightarrow}$ upper high – tridymite

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Keywords: DSC; Microsilica; $SiO₂$ -modifications

1. Introduction

Microsilica: Microsilica is the name given to a product obtained through cleaning the fumes from electric arc furnaces. These furnaces usually contain quartz, coal and wood chips and are heated to 2000° C. The fumes go through a unique condensing and filtering process, and the product obtained is a very finely divided silica. The mean particle size is $0.15 \,\mu m$ and the surface area is 22 m^2/g . The powder obtained is an amorphous non-crystalline or glassy material with an $SiO₂$ -content usually in the 94–99 wt% range.

The $SiO₂$ -system: It is well known that silica has a multitude of polymorphs which undergo a number of both displacive and reconstructive transformations, the most important of which are summarized in Fig. 1.

It is generally believed that this is the most accepted diagram for the $SiO₂$ -system and, therefore, it can be found in many textbooks in ceramics describing the phase relations in the system. For instance, Sosman [1] in his book gives a broad presentation of the problem of stabilities of the different SiO_2 -phases.

In our investigation of microsilica powder from Fiskaa Verk, Norway, some years ago, the so-called FV-type MS, we observed that when the powder was heated in the $850-950^{\circ}$ C range between three and seven days, the powder transformed from the amorphous state to cristobalite and then to tridymite in accordance with the Fenner [2] and Sosman [1] diagrams.

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Fig. 1. The polymorphs of $SiO₂$.

Fig. 2 presents a DSC-trace from an FV-MS sample heated at 940° C for 28 h. We obtain a peak at 388.5 K, corresponding to the low-tridymite \rightarrow high-tridymite, and another peak at 472.5 K, corresponding to the $low-cristobalite \rightarrow high-cristobalite transitions.$

However, when we started to study a new type of microsilica from Meråker Smelteverk, Norway, we were not able to find any transition corresponding to the tridymite transitions by using the same experimental techniques as in case of previous, DSC and X-ray diffraction (XRD) studies [3]. Instead, we observed that this type of microsilica transformed

on heating into quartz and/or cristobalite. We concluded that this was due to the higher purity (98.4%) of the Meråker-type microsilica as compared to the FVtype (96.4%). We have summarized the results from these comprehensive studies in a free energy or stability diagram shown in Fig. 3.

2. Experimental

2.1. Chemicals

The chemical analysis of Meråker microsilica is given in Table 1 and compared with the corresponding analysis of the Fiskaa-Verk-type microsilica.

The $Li₂CO₃$ used in the present study was $Li₂CO₃$ p.a. min. 99% from Merck, Germany.

Fig. 2. DSC-diagram (heating rate $20^{\circ}/\text{min}$) for microsilica from Fiskaa Verk. Sample heated at 940°C for 28 h.

Fig. 3. Free energy or stability diagram for Meråker microsilica, as proposed by the authors.

2.2. Preparation of samples

Four samples of the following compositions 0.15, 0.3, 0.6 and 0.9 wt% $Li₂O$ were prepared by mixing $Li₂CO₃$ and microsilica in 100% alcohol. The powder was dried at 100° C and sintered at 800° C for 2–3 h. The powder was then pressed into tablets. The tablets were heated at 900° C for periods from 1 to 4 weeks and at 1100° C for 24 and 48 h.

2.3. Method

X-ray investigations were carried out by use of a Phillips PW 1730/10 instrument using CuK_{α} -radiation.

The calorimeter was a DSC-2 instrument from Perkin-Elmer. The calibration of the instrument followed the procedure given by ICTAC using metallic indium, tin and zinc as standards. Platinum crucibles were used as containers.

3. Results and discussion

Calorimetry: Fig. 4 shows a DSC-diagram for a sample containing 0.9% Li₂O and heated for 48 h at

 1100° C. The two peaks in the diagram at 394 and 444 K correspond to the transitions:

low-tridymite
$$
\stackrel{394 \text{ K}}{\rightarrow}
$$
 high-tridymite
 $\stackrel{444 \text{ K}}{\rightarrow}$ upper high-tridymite

In Fig. 5 are plotted the measured enthalpies of the two transitions as a function of $%Li₂O$ added to microsilica as well as the measured enthalpy of the transition:

low-cristobalite $\stackrel{538}{\rightarrow}$ high-cristobalite

All samples investigated are heated for 48 h at 1100° C.

For samples heated at 900° C, only one transition was found at 394 K, corresponding to the low-tridymite \rightarrow high-tridymite transition. In Table 2 we give the measured enthalpies for samples containing 0.9% $Li₂O$ and heated between 1 and 4 weeks.

Our values for the enthalpies of transition:

low-tridymite(
$$
\alpha
$$
) \rightarrow high-tridymite(β),
 $\Delta H_{\text{tr}} = 225 \text{ J/mol at } 394 \text{ K}$

and

Fig. 4. DSC-diagram (heating rate 20° C/min) for microsilica containing 0.9% Li₂O. Sample heated for 48 h at 1100°C.

Fig. 5. Measured enthalpies for the three transitions: low-cristobalite->high-cristobalite; low-tridymite->high-tridymite; and hightridymite->upper high-tridymite. All samples are heated for 48 h at 1100°C.

Table 2 Enthalpy of the low-tridymite \rightarrow high-tridymite transition, peak temperature 394 K

Heating time at 900° C	ΔH (J/mol)
1 week	140
2 weeks	164
3 weeks	182
4 weeks	232

Table 3

Microsilica: phase identification by X-ray and DSC. Q, quartz; C, cristobalite; and T, tridymite

high-tridymite(β) \rightarrow upper high-tridymite(γ), $\Delta H_{tr} = 125$ J/mol at 444 K

should be compared with the values often referred to in the literature by Mosesman and Pitzer [4]; ΔH_{tr} =293±84 J/mol at 390 K for the α to β transition and ΔH_{tr} =167±42 J/mol at 436 K for the β to γ transition.

Knacke, Kubaschewski and Hesselmann [5] report the following values for the enthalpies of transitions; 209 J/mole at 390 K and 188 J/mole at 500 K.

X-ray investigations: In Table 3, are summarized the results from the X-ray diffraction investigations together with the results from the DSC-experiments.

These samples were heated at 900° C for 1, 2 and 4 weeks and had different contents of $Li₂O$.

4. Conclusion

From the results obtained, we can conclude that the purity of the starting material, the microsilica, as well as the amount of impurities added together with the heating time and temperature, are important parameters both, for the kinetics and thermodynamics of the phase transitions in the $SiO₂$ -system. The sequence of the transitions at 900° C follow the scheme:

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microsilica \rightarrow quartz \rightarrow cristobalite \rightarrow tridymite
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The last step will depend on the amount of $Li₂O$ added to the microsilica.

At 1100° C the sequence of the transitions will be:

microsilica \rightarrow cristobalite \rightarrow tridymite

Acknowledgements

Financial support from Borgestads Legat IV is gratefully acknowledged.

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