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Isothermal in situ reduction kinetics of $CoCl₂-SiO₂$ gels to $Co-SiO₂$ nanocomposites

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Abstract

In situ reduction kinetics of CoCl₂-SiO₂ gel to Co-SiO₂ nanocomposite in the 800–950^oC range has been studied. The presence of nanosized metallic Co in the host matrix has been established from the X-ray diffraction patterns of the reduced gels. The metallic particle size has been found to be 17 and 23 nm when reduced at 850° and 950° C, respectively. Nucleation and growth type of mechanism remain operative during the course of the reduction. The activation energy is found to be in the range of 55-61 kJ/mol. \odot 1999 Published by Elsevier Science B.V. All rights reserved.

Keywords: Nanocomposite; Sol-gel; In situ; Reduction; Kinetics; Mechanism

1. Introduction

The sol-gel technique has emerged as a versatile method for synthesizing a wide range of materials. Recently, preparation of glass-metal nanocomposite through this route has gained considerable momentum on account of the overall versatility and simplicity of the process $[1-6]$. In this method, glass-metal nanocomposites are synthesized by using a sol containing silicon tetraethoxide (TEOS) and a suitable metallic salt in ethyl alcohol, followed by subsequent conversion of sol to gel. The resulting gel is then subjected to reduction treatment at suitable temperatures by passing a mixture of nitrogen and hydrogen gas [2,3] or by generating hydrogen in situ [5,6]. The treatment leads to the evolution of metal islands of diameter

ranging from five to 50 nm within the host $SiO₂$ matrix. The nanocomposites thus prepared exhibit remarkable electrical, magnetic and optical properties which raises the possibility of exploiting these new class of materials as substrates for semiconduction technology. However, the physical properties of these nanocomposites remain functions of shape, size, amount and distribution of the metallic species in the host matrix [7,8]. Again, the shape and size of these metallic islands are influenced, to a great extent, by the process control parameters. Although from the transmission electron micrograph, shape, size and distribution can be measured directly, the amount of metallic species present is assessed on the basis of the assumption that all the metal salt incorporated in the $SiO₂$ matrix is reduced to metal within the stipulated time of reduction. This is probably the main hindrance in preparing nanocomposites with uniform and reproducible physical properties. In reality also, the fore-

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going method does not help one to estimate the actual amount of metallic salt reduced at a particular instant of time during reduction and, therefore, fails to impart any knowledge on the mechanism(s) of reduction. Therefore, it is deeply felt that a study on the kinetics of reduction is essential from the view point of controlling the process parameters effectively to achieve a greater degree of control over the physical properties of the nanocomposites. Keeping the above-mentioned views in mind, the present paper deals with the in situ reduction kinetics of $CoCl₂$ in the $SiO₂$ gel matrix to $Co-SiO₂$ nanocomposites.

2. Experimental procedure

 $SiO₂$ gels containing CoCl₂ and 1.5 times stoichiometric dextrose which when reduced would yield 5 wt% $Co-SiO₂$ nanocomposites were prepared from TEOS, C_2H_5OH and CoCl₂ solution with C_2H_5OH : -TEOS ratio of 4 : 1. The preparation technique has been described elsewhere [6]. The gels were heattreated at 800 $^{\circ}$, 850 $^{\circ}$, 900 $^{\circ}$ and 950 $^{\circ}$ C, respectively, under nitrogen gas atmosphere at a flow rate of 5 cc/s . The heat treatments were carried out in an electrical heating furnace. The desired temperature was maintained by a PID controller with an accuracy of $\pm 1^{\circ}$ C. The gel samples were placed in a porcelain boat, one at a time, and introduced into the constant heating zone of the furnace for heat treatment. The HCl vapour generated during the course of the heat treatment is absorbed in known volume of double-distilled water. The pH values of the resulting HCl solution were recorded as a function of heat-treatment time. The pH values of the solution yield $[H^+]$ in the solution from which the fraction of $CoCl₂$ reduced within the $SiO₂$ gel matrix were calculated.

3. Results and discussion

Figs. 1 and 2 are the representative characteristic Xray diffraction (XRD) patterns of $CoCl₂-SiO₂$ gels, reduced at 850° and 950° C. The interplanar spacings (d_{hkl}) were computed from the XRD patterns. It is observed that the computed (d_{hkl}) values match reasonably well with the ASTM standard (d_{hkl}) values [9] of Co which confirms the presence of metallic Co in

Fig. 1. Characteristic XRD patterns of $CoCl₂-SiO₂$ gels reduced at $850^\circ C$

Fig. 2. Characteristic XRD patterns of $CoCl₂-SiO₂$ gels reduced at 950° C.

the $SiO₂$ gel matrix. The average particle size of the reduced metal is calculated from the XRD pattern with the help of the following equation: $t=0.9\lambda/BCos \theta_B$, where t is the particle size, λ the monochromatic X-ray wavelength, B the width of diffraction curve in radians at an intensity equal to half the maximum intensity and $\theta_{\rm B}$ the Bragg angle in degrees [10].

The average particle size has been found to be 17 and 23 nm, respectively, for temperatures of 850° and 950° C. Therefore, assignment of the term nanocomposite to the reduced gels seems to be reasonable and appropriate. The size of the metallic particles has been found to increase with increase in reduction temperature. This is attributed to the effect of grain coarsening with increase in temperature.

Fig. 3 depicts the fractional conversion (α) vs. time (t) plot of the $Co-SiO₂$ nanocomposites. It is evident from the figure that a maximum reduction occurs within the first 5 min. This can be attributed to the

Fig. 3. Fractional conversion vs. time plot of $Co-SiO₂$ nanocomposites.

in situ generation of H_2 at the reaction sites. At the latter period, the reduction rate falls due to the fact that water vapour is only made available from the polycondensation reaction of $SiO₂$ gels [11]. As the rate of polycondensation reaction is slow, availability of water vapour is restricted, a fact that tends to slow down the generation of H_2 gas via the water-gas reaction. This causes the conversion rate to drop.

4. Identification of mechanism(s)

To ascertain the appropriate mechanism(s) which remain operative during the course of the reduction, the experimental data on time-dependent fractional conversion have been analyzed by taking recourse to the reduced time plots [12]. The analysis is as follows. The general equation for all the mechanism(s) is written as

$$
g(\alpha) = kt \tag{1}
$$

where $g(\alpha)$ is an appropriate function of α , k the specific rate constant and t the time.

For $\alpha = n$,

$$
g(n) = kt_n \tag{2}
$$

Where t_n is the time required for fractional conversion of *n*. Dividing Eq. (1) by Eq. (2) , we obtain

$$
\frac{g(\alpha)}{g(n)} = \frac{t}{t_n} \equiv \theta \tag{3}
$$

where θ is the reduced time. Eq. (3) is independent of k , the reaction rate constant, which is the temperaturedependent term in kinetics. Therefore, for the same mechanism operative over the entire range of temperature, the computed α vs. θ plot from experimental data will coincide with the theoretically computed α vs. θ plot for the mechanism. By comparing the computed α vs. θ plot from experimental data with the theoretically computed α vs. θ plot for the mechanisms, it is possible to find out the mechanism which actually remains operative.

The reduced time plot shown in Fig. 4 reveals that the nucleation and growth(NG2)-type of mechanism is operative. The validity of the above-mentioned mechan-

Fig. 4. Reduced time plot.

Fig. 5. Plots of ln t_{α} vs. $1/T$ at different levels of α .

ism can be established by considering the fact that equilibrium vapour pressure of $CoCl₂$ rises sharply with increase in temperature in the $800-950^{\circ}$ C range [13]. Therefore, in the experimental temperature range, it is highly probable that the in situ generated H_2 will reduce $CoCl₂$ in the vapour state. Since, the melting point [13] of Co is very high $(1480^{\circ}C)$, solid particles will precipitate out from the vapour phase by nucleation and growth process. However, conversion of $CoCl₂$ to metallic Co in the gaseous state will decrease the vapour pressure of CoCl₂. Therefore, with the progress of the reduction, $CoCl₂$ will vaporise continuously to maintain an equilibrium vapour pressure. As the reduction in the gaseous phase is much faster than the gas-solid or gas-liquid reaction, the former, i.e. nucleation and growth of Co from the vapour state, will predominate and control the kinetics of reduction. It is for this reason that the nucleation-and-growth type of mechanism (NG2) is logical and physically viable.

The activation energies at different levels of α for $CoCl₂-SiO₂$ gel reductions has been calculated by the differential method [14]. The slope of the ln t_{α} vs. $1/T$ plots (see Fig. 5) multiplied by the universal gas constant, R, yields the activation energy at different levels of α . The computed activation energy has been found to be in the range of 55–61 KJ/mol.

5. Conclusion

The following conclusions are drawn from the present investigation

- 1. The in situ reduction of $CoCl₂-SiO₂$ gel to Co- $SiO₂$ nanocomposite progresses at a rapid rate and a maximum conversion occurs within the first 5 min of the reaction.
- 2. The particle size of Co in the $SiO₂$ matrix has been found to be 17 and 23 nm for the gels reduced at 850° and 950° C.
- 3. Co $-SiO₂$ nanocomposites exhibit grain coarsening with increase in temperature.
- 4. During the course of reduction, nucleation and growth type of mechanism remain operative.
- 5. The activation energy for the reduction process has been found to be in the range of $55-61$ KJ/mol.

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