VACUUM BALANCE STUDIES OF PHOSPHATE-BONDED OXIDE CERAMICS

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ABSTRACT

Methods of phosphate bonding of oxide ceramics are summarlsed. Applications of vacuum balance techniques are described to study the sintering of lime, magnesia, alumina and kaolin in the presence of phosphoric acid and sodium polymetaphosphate additives. Changes in surface area, crystallite and aggregate sizes durlng sintering have been determined by gravlmetric B.E.T. nitrogen gas sorption techniques, supported by X-ray diffractometry, optical and electronmicroscopy.

Phosphoric acid has comparatively little influence on lime and magnesia srntering. There is no enhancement of slntering and even slight activation at temperatures between 800-1000 $^{\circ}$ C, when chemical reactions with the additive are completed. In contrast, sodium polymetaphosphate additive enhances sintering of llme and magnesia at temperatures between 5OO-1OOO °C. When kaolin is calcined wlth sodium polymetaphosphate, the activated alumina in the intermediate metakaollnite readily forms aluminlum phosphate to accelerate slnterlng.

INTRODUCTION

The mechanical strength and hardness of ceramics are important industrially. High-melting metal oxldes, such as llme, magnesia and purified alumlna are obtalned generally from natural sources in finely-divlded form and require sintering to produce strong dense bodies. Sintering is promoted by additives(1).

Kingery(2) has summarised the earlier literature on phosphate bonding and has studied the reactions and properties of cold-setting phosphoric acid bonds with metal oxides and hydroxides. His experimental data indicates that the formation of acid phosphates is responsible for bond formation at room temperature. With the more *refractory* and inert oxides, e.g., alumina, heat is required to achieve reaction and hence the setting of the cement. Fischer(3) has compared the effect of temperature on bond strength for alumina using different sources of phosphate radical and the two precipitation cements ethyl silicate and alumlnium monoacetate. For the Dhosphate reaction bonds, strength increases and porosity decreases with increasing acidity of the phosphate. Further research (to 1977) on phosphate bonding has been reviewed by Cassidy(41. Phosphoric acid and monoammonium phosphate react too quickly with basic aggregates and so alkali polyphosphates are preferred. Sodium hexametaphosphate

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 $(NaPO₃)₆$, forms rubberlike polymers and yields high strength mortars with fireclay agqregates. These materials are used in hlgh alumina refractory mortars and ramming mixes and are partlcularly useful as bonds for basic aggregates.(5). Foessel and Treffner {6), who reviewed the earlier literature on the use of sodium molyphosphates in basic refractories, showed that very high hot strength (1500-2500 psi) could be achieved at 2700 $^{\circ}$ F (about 1500 $^{\circ}$ C) in magnesite and magnesite chrome refractories under certain conditions. The most important criterion was the CaO: $(P_2O_5+ S_1O_2)$ ratio, for limes. Venables and Treffner(7) reported that the bonding phase was $CaNaPO_{4}$, sodium rhenanite.

The effect of natural impurities and additives such as chloride, sulphate, phosphate and iron oxide on lime slnterlng has been investigated earlier (1967) by the author(1). Thus in the present research, vacuum balance techniques are applied further to study the sintering of lime, magnesia, alumina and kaolin In the presence of phosphoric acid and sodium polymetaphosphate additives. Changes in surface area, crystalllte and aggregate sizes during slntering have been determined by gravimetric B.E.T. nitrogen gas sorption techniques, supported by X-ray diffractometry, optical and electron-mlcroscopy.

EXPERIMENTAL

Materials

Samples of calcium hydroxide and magnesium hydroxide (British Drug Houses) were intimately mixed with known amounts of phosmhorlc acid or sodium polymetaphosphate. The M.H3PO4 produced partial *dissolution,* after which the solutions were carefully evaporated and the residues dried at $140-150$ °C(1).

Procedure

Separate portions of the calcium and magnesium hydroxides, with or without the additives, were calcined for various times at each of a number of fixed temperatures in alr in an electrical furnace.

The specific surfaces, S, of the cooled samples were determined by a gravlmetric B.E.T. method(8), using nitrogen (or occasionally oxygen) gas sorption at -183 ^o or -196 ^oC recorded on an electrical sorption balance constructed by the author (quarter mg sensitivity using 5-10 g samples) or on vacuum microbalances (pg to mg sensitivity using 0.25 g samples or less), viz., Sartorlus or CI Microforce Mark 2B. The adsorption Isotherms also indicated any porosity present (from hysteresis) and pore size ranges. Average crystallite size ranges deduced from the specific surfaces of the less porous materials were compared wlth aggregate sizes observed by optical- and electronmicroscopy (Phillips TEM 300).

To determine the nature of any chemical reactions between the llme, magnesia, alumina or clay and the additives, mixtures were made of stoicheiometric amounts of oxide and additive. These mixtures were calcined at the temperatures and times employed for the stud,es of bonding and sintering. The components in the products were identified by X-ray diffractometry.

RESULTS AND DISCUSSION

Physical nature of activated llme and magnesia

The surface activities of lime and magnesia produced by thermal decomposition of the hydroxldes are compared in relation to the changes in crystal structure during preparation at several temperatures up to 1000 $^{\circ}$ C. cf. Fig. 1-5. The magnesia samples have correspondingly much larger surface areas than the lime samples. This is due mainly to the greater volume decreases associated with the crystal changes and to the diminished sintering of the magnesia at the lower decomposition temperatures. Thus magnesium hydroxide almost completely decomposes within 5 h in air above 400 $^{\circ}$ C compared with temperatures of above 450 $^{\circ}$ C which are required for calcium hydroxide. Earlier research(9) has shown that the specific surface of the magnesia produced at 400 $^{\circ}$ C is 180 $m^{2}\sigma^{-1}$ compared with $28 \text{ m}^2 \text{g}^{-1}$ for lime at 450 $^\circ$ C, representing the maximum surface activities of magnesia and llme produced by thermal decomposition in air.

However, the magnesia and llme samples produced at the lowest temperatures have had little opportunity to sinter and have a low degree of crystallinity, especially the magnesia whlch has an appreciably dilated crystal lattice (2.3 vol-%) and shows some mesoporoslty (Fig. i). The nitrogen adsorption isotherm at -196 $^{\circ}$ C for the 400 $^{\circ}$ C sample in Fig. 1 was recorded on the CI Microforce balance. A good B.E.T. plot is given at lower relative pressures (0.05-0.3) indicating S = 175 m^2q^{-1} in good agreement with 180 m^2q^{-1} from the nitrogen adsorption isotherm at -183 $^{\circ}$ C recorded earlier on the older electrical sorption balance(9). At higher relative pressures (0.40-0.96) there is some hysteresis over practically the whole of the mesopore range (pore size diameters of 2-50 nm). This isotherm is almost identical with that for a magnesia sample produced by calcining the hydroxide for a shorter time $(\frac{1}{2}$ hr) at 500 $^{\circ}$ C. Longer calcination at 500 $^{\circ}$ C to give a total time of 5 h enables the oxide to sinter and the surface area decreases to 126 m^2q^{-1} , while the unit cell dilation volume reduces to only 0.9 %. At higher temperatures the dilation diminishes still further and the mesoporosity becomes very small above 700 $^{\circ}$ C, as indicated in Fig. 1, where there is comparatively little hysteresis in the nitrogen isotherm of the 700 $^{\circ}$ C Sample.

Effect of additives on magnesia and lime sintering

Phosphoric acid hes comparatively little influence on the sintering of magnesia.

Fig. 1. Magnesium hydroxide calcined in air at 400 $^{\circ}$ C and 700 $^{\circ}$ C for 5 hours. Adsorption isotherms for nitrogen at -196 °C.

There is no enhancement of sinterlng and even slight activation at temperatures between 800 $^{\circ}$ and 1000 $^{\circ}$ C when chemical reactions with the additive are completed. Thus the curves in Fig. 2 indicate that small amounts of additive (up to 0.5 %) cause some activatlon, but larger amounts give less actlvatlon. This is similar to what the author has found previously for lime(l). The Magnesium and calcium hydrogen phosphates initially formed with the phosphoric acid additive decompose at higher temperatures, first to pyrophosphates which then react wlth excess of magnesia and lime to give the normal phosphates,

ig. 2. Magnesium hydroxide ($+H_3PO_4$) calcined at 1000 ${}^{O}C$ in air.

Fig. 3. Magnesium hydroxide (+NaPO3) calcined at 1000 $^{\circ}$ C in air.

 $Mg_4(PO_4)_2$ and $Ca_3(PO_4)_2$, as the final products. Since these phosphates are hlgh-melting, they are not very effective in promotlng slntering of the maqnesla and llme.

The curves in Fig. 3, 4 and 5 indlcate that sodium molymetaphosphate has much more influence on the slnterlng of magnesla and lime. Fig. 3 shows the effect of the additive with time of sintering of magnesia at $1000\ ^{\circ}\textrm{C}$, while Fig. 4 compares the effect of the additive at different temperatures for equal calclnation times of 5 h. Althouqh the llmes have correspondingly smaller surface areas and larger crystallite sizes than the magnesias, their slnterlna is still enhanced by the sodlum polymetanhosphate as shown in Fig. 5. Hence sodium polymetaphosphate considerably enhances slnterlng of hoth magnesia and lime at temperatures between 500° and 1000° C.

However, optical- and electron-microscopic observations show that at 500 $^{\circ}$ C , the increased aggregation of the llme and magnesla crystallltes is accompanied by very little crystal growth, the sodium polymetaphosphate evidently forming

Fig. 4. Magnesium hydroxide calcined for 5 hours at different temperatures.

Fig. 5. Calcium hydroxide calcined for 5 hours at different temperatures

a matrix around the oxlde crystall±tes. Extensive surface and crystal lattice diffusion of the sodium polymetaphosphate is possible, since 500 \degree C is well above its Tammann temperature (equlvalent to ½ m.p. in K) and indeed near to its m.p. of about 650 $^{\circ}$ C. At higher temperatures, especially between 800 $^{\circ}$ and 1000 \degree C, sodium polymetaphosphate reacts chemically with lime and magnesia and becomes even more effective in enhancing slntering, slnce it forms lower-melting NaCaPO4 and NaMgPO4 compared with the Ca3(PO4)2 and Mg3(PO4)2 formed with the phosphorlc acid addltlve.

Phosphate bonding of llme and magnesla with sodium polymetaphosphate

To understand more fully the actlon of the sodium polymetaphosphate addltive, stoichelometric amounts of lime and magnesia were reacted with $(NaP03)_6$ at 500^{-6} and 1000 $^{\circ}$ C according to the equations:-CaO + NaPO3 = NaCaPO4 and MgO - NaPO3 = NaCaPO4. X-ray examinatlon showed that there was no appreciable formatlon of NaCaPO₄ or NaMaPO₄ at 500 $^{\circ}$ C even after 20 h calcination, but the reactions were practically comolete after 20 h at $1000 {\rm ^oC.}$ The NaMgPO $_{\rm V}$ is molten when formed above 800 $^{\circ}$ C and crystallises from the melt on cooling. The X-ray data showed that $6-\text{NaCaPO}_+$ was formed, orthorhomblc with a = 5.20 A, b = 9.32 A and $c = 6.84$ A in good agreement with A.S.T.M. data. The NaMgPO₁, was found also to be orthorhombic with a = 4.76 A, b = 11.17 A and c = 5.94 A.

The magnesia formed and slntered wlthout additive was still quite finelydivided even when calcined at 1000 $^{\circ}$ C in air for 20 h; its specific surface was about 20 $m^2 \sigma^{-1}$ with an average crystallite size of about 0.1 µm cf. Fig. 6(a), but the sodium magneslum phosphate crystalllslng from the melt had a considerably different morphology and larger crystal size, cf. Fig. 6(b). The magnesia mainly consisted of aggregates of submlcron-sized crystallltes, but the phosphate bonding and crystalllsatlon led to material mainly above micron size. The crystal slze of lime sintered without additive at 1000 $^{\circ}$ C is already above micron size, but nevertheless the phosphate bonding with sodium polymetaphosphate considerably increases most of the crystal sizes to above 10 um. The effectiveness of the sodium polymetaphosphate on lime sintering at lower temperatures may be assessed from the data in Fig. 5, to calculate the multiple decreases in the number of crystallites, $(S_1/S)^3$ where S and S_1 are the surface areas of lime sintered with or without additive; at 600 $^\circ$ and 800 $^\circ$ C, the multiple decreases are 15 and 200fold.

Phosphate bonding of alumina and kaolin with sodium polymetaphosphate

Optical micrographs showed little change in crystal size for $b-AL_2O_3$ calcined in air for 20 h at 1000 $^{\circ}$ C with or without 5 % NaPO3 additive. The size remained mainly between 20-100 um averaging about 70 cm . In longer calcination (20-120 h) with 10-25 % NaPO3, the material formed lumps of up to 10 mm size less easily

Fig. 6. Transmission electron-microqraphs of (a) MgO (b~ MmO + NaPO3 (i:i mol ratio) after 20 hours sintering at 1000 °C in air.

crushed when more NaPO3 was present. Evidently the NaPO3 was forming a matrix around the Al_2O_3 crystals. Thus when equimolecular proportions of Al_2O_3 and $NaPO_3$ were calcined for 20 h, practically all of the material formed one hard mass, with the surface containing small glass nodules of 4-10 µm. However, no appreciable amounts of $A1PO_u$ were shown on the X-ray traces, not even when more active γ -Al₂O3 was used. But when Kaolin and NaPO3 were calcined at 1000 $^{\circ}$ C, a glass product containing AlPO. was obtained; dehydration of the kaolinite to metakaollnite provided very active alumina and silica, so that alumlnium phosphate was formed more readily to accelerate sintering.

REFERENCES

1 D.R. Glasson, J. appl. Chem., Lond., 17 (1967) 91-96 2 W.D. Kingery, J. Amer. ceram. Soc., 33 (1950) 239-242 3 K. Fischer, Proc. Brlt. ceram. Soc., 12 (1969) 51-64 4 J.E. Cassidy, Amer. ceram. Soc. Bull., 56 (1977) 640-645 5 R.W. Limes and R.O. Russell, U.S. Patent, 3, 304, 187, March i, 1965 6 A.H. Foessel and W.S. Treffner, Amer. ceram. Soc. Bull., 49 (1970) 652-657 7 C.L. Venables and W. S. Treffner, Amer. ceram. SOC. Bull., 49 (1970) 660-665 8 D.R. Glasson, J. *Chem.* Soc., 1956, pp. 1506-1510 9 D.R. Glasson, J. appl. Chem., Lond., 13 {1963) 111-119