

TA METHOD APPLICATION TO THE HIGHEST REFRACTORY
OXIDE SYSTEMS INVESTIGATION

A.V. Shevthenko, L.M. Lopato
Institut of Material Science, Kiev, Akad. Nauk USSR

ABSTRACT

DTA method application to the controlled gaseous environment up to the temperature of 2600°C and TA method application in air using solar heating up to the 3000°C are considered. Phase transformations in the highest refractory systems based on the lanthanide oxides: zirconium, hafnium, yttrium oxides as well as IIA subgroup element oxides have studied by these methods.

INTRODUCTION

TA method has succeeded in phase equilibration investigation in salt and metal systems. The study of oxide systems with the melting temperature above 2000°C using DTA method has not yet found a sufficient development. It is due to the deficiencies of the hardware, to the compatibility problem of the test material and container to the atmosphere reaction. The oxide TA systems radiation heating provides the measurements in air and in different gaseous environments. Since heating occurs only for the sake of radiation on the test object contamination should be excluded.

MEASURING METHOD

The idea of string thermocouple by J.A. Kocherghinsky [1] is used in this investigation. It lies in the base of DTA device design working in the controlled gaseous environment up to the 2600°C. Thermodevice made of tungsten is used with the thermocouples W-Re 20-W, differentially connected. Thermal curves record was performed by M.S. Khurnakov pyrometer, providing a low noise level. The device calibration was done according to the fine metal and oxides melting points. Phase transformation temperatures in the oxide systems were defined during heating, cooling curves were read with the aim of quality picture obtaining owing to the oxide increased tendency to supercooling.

Oxide systems TA is carried out on the solar furnace the capacity of which is 1,5 kW using the pyrometric systems working in

the wave length range from 0,65 to 2,0 μ m as a temperature transducer. The temperature determination method according to the rotating blade [2] and according to partially melted sample surface [3] was used for the oxide TA carrying out. In both cases the cooling curves are recorded after the radiant flux is cut off. Aluminium, yttrium, zirconium, hafnium oxides as well as lanthanide and scandium oxides were used as reference material in the temperature range 2000-2900°C.

EXPERIMENT RESULTS

DTA method application has allowed to investigate high temperature polymorphic transitions of the lanthanide pure oxides and to determine their melting temperatures. More precise melting temperatures and rare earth oxides polymorphic transformations are summarised in Table.

Oxide	Melting temp. °C	Transf. temp. H-X, °C	Transf. melt. A-H, °C	Transf. melt. B-H, °C	Transf. melt. B-A, °C	Transf. melt. C-H, °C	Transf. melt. C-B, °C
La ₂ O ₃	2310	2100	2030				
Ce ₂ O ₃	2240	2170	2120				
Pr ₂ O ₃	2280	2130	2030				
Nd ₂ O ₃	2300	2180	2060				
Sm ₂ O ₃	2310	2225	2070		1870		
Gd ₂ O ₃	2410	2370	2200	2170	2170		
Tb ₂ O ₃	2400	2370	2175	2175	2160		
Dy ₂ O ₃	2400	2385	2190	2200	2190		1980
Y ₂ O ₃	2440		2390			2390	
Ho ₂ O ₃	2400			2200			2190
Er ₂ O ₃	2420					2320	
Tm ₂ O ₃	2410					2350	
Yb ₂ O ₃	2450					2380	
Lu ₂ O ₃	2510						
Sc ₂ O ₃	2470						

HfO₂ influence on the high temperature polymorphic transitions of the lanthanide oxides was studied. At HfO₂ introducing the polymorphic transition temperatures of the lanthanide oxides may reduce (X≠H), (H≠A) and increase (C≠B). The phase field boundaries for the given forms are determined by DTA data which allowed to

define eutectoid range in the investigated systems.

It was found out that HfO_2 maximum permissible solubility in polymorphic forms X, H, A and B reduces and in C-form increases with the lanthanide atomic number growth.

Eutectic melting temperature is raised with the lanthanide atomic number growth. It correlates with the pure oxide melting temperature increase [4].

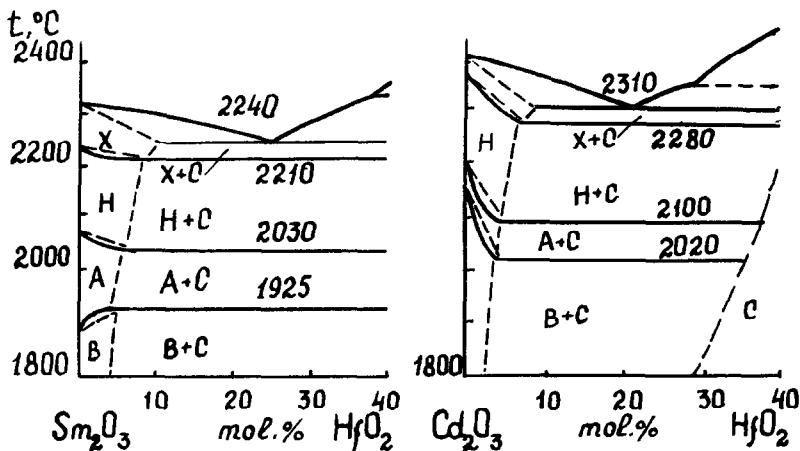


Fig. 1. Phase diagram elements of the lanthanide oxides-hafnium oxide.

High temperature polymorphism ZrO_2 and HfO_2 was analysed and the regularity of rare earth oxide influence on the HfO_2 phase transformation temperature change was studied. It was found out that HfO_2 mon. \rightleftharpoons tetr. transformation in to cubic form occurs at $1830 \pm 20^\circ\text{C}$, the transformation in to cubic form occurs at $2530 \pm 50^\circ\text{C}$. ZrO_2 tetr. \rightleftharpoons cub. transformation occurs at $2330 \pm 30^\circ\text{C}$.

Lanthanide oxides reduce mon. \rightleftharpoons tetr. transition temperature from 1830 to 1750-1780°C along the lanthanide row. Rare earth oxides affect the best cubical modification HfO_2 , extending its concentration and temperature.

At the highest refractory systems study (HfO_2 -CaO, HfO_2 -MgO) eutectic and eutectoid points references (ETA) were established and systems liquidus (TA) was determined [5, 6].

Melting temperature of the congruently melting composition 10HfO_2 equal to $2590 \pm 30^\circ\text{C}$ was defined and for the first time it was found out this composition to undergo some reverse polymorphic transformation rhomb. \rightleftharpoons cub. modification [7] at $2000 \pm 10^\circ\text{C}$.

The analogous transformation temperature of the CaZrO_3 composition was $2000 \pm 10^\circ\text{C}$ as well, melting temperature was $2510 \pm 30^\circ\text{C}$.

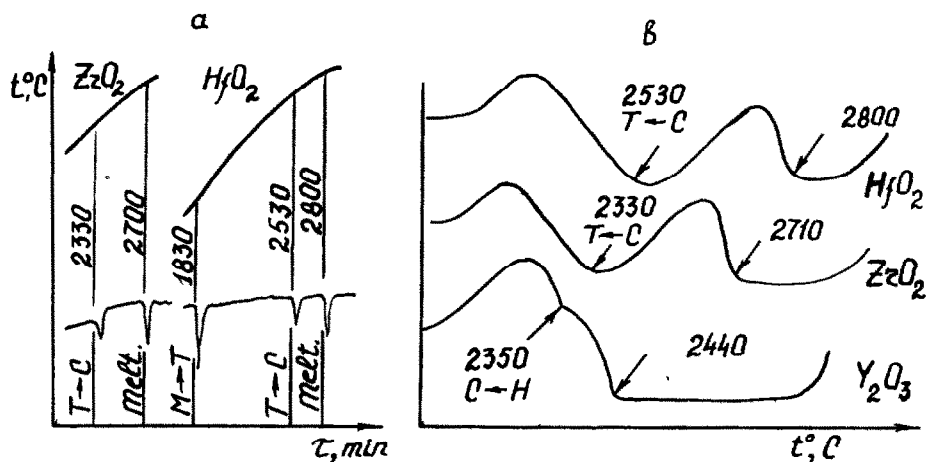


Fig.2(a), (b). DTA curves ZrO_2 and HfO_2 and TA curves HfO_2 , ZrO_2 and Y_2O_3 .

The analogous transformation temperature of the CaZrO_3 composition was $2000 \pm 10^\circ\text{C}$ as well, melting temperature was $2510 \pm 30^\circ\text{C}$.

Phase diagrams of the oxide systems formed by HfO_2 with the rare earth oxides [6] were studied using the above mentioned methods.

CONCLUSION

In this paper we have shown the effectiveness of DTA and TA application to the phase relations in the systems formed of the high refractory oxides.

REFERENCES

- 1 Kotherginsky, Izv. Sib. otd. Acad. Nauk USSR, 9,4 (1974) 52 429
- 2 M. Foex, J.P. Traverse, REV. Int. Haut. Temp. et Refr. 3,4, (1966)
- 3 T. Noguchi, M. Misuno, Bull. Chem. Soc. of Japan, 42,7 (1968) 1583
- 4 A.V. Shevthenko, L.M. Lopato, Inorg. Mat. 18, 11 (1983) 1842
- 5 A.V. Shevthenko, L.M. Lopato, Inorg. Mat. 17, 5 (1981) 838
- 6 L.M. Lopato, A.V. Shevthenko, Inorg. Mat. 17, 1 (1981) 90
- 7 A.V. Shevthenko, Z.A. Zaitzeva, L.M. Lopato, G.I. Gerasimjk, Inorg. Mat. 19, 12, (1983) 1059
- 8 A.V. Shevthenko, L.M. Lopato, Z.A. Zaitzeva, Inorg. Mat. 20, 9, (1984) 1530